



*International School
on
Aperiodic Crystals*

**Examples of incommensurately
modulated structures and
composites studied using
transmission electron
microscopy**

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Purpose of this lecture

At the end of this lecture you should be able to:

- Understand the TEM paragraph in papers about IMS and CS
- Be able to make solid comments about conclusions claimed from TEM by different sources (collaborators, papers,...) by knowing some possible pitfalls
- Decide whether it would be useful to do TEM on your own IMS or CS
- Make basic interpretations of TEM data on your own materials by yourself

Outline of the lecture



The three most frequently used techniques in case of IMS/CS:

- Electron diffraction
- HAADF-STEM
- HRTEM



ED: the main differences with XRD



- Shorter wavelength, almost flat sections through reciprocal space
- A nanometer size particle is enough
- Multiple diffraction
- Can be combined with direct imaging and compositional analysis

Appearance very much like single crystal XR patterns



Incommensurately modulated structure example



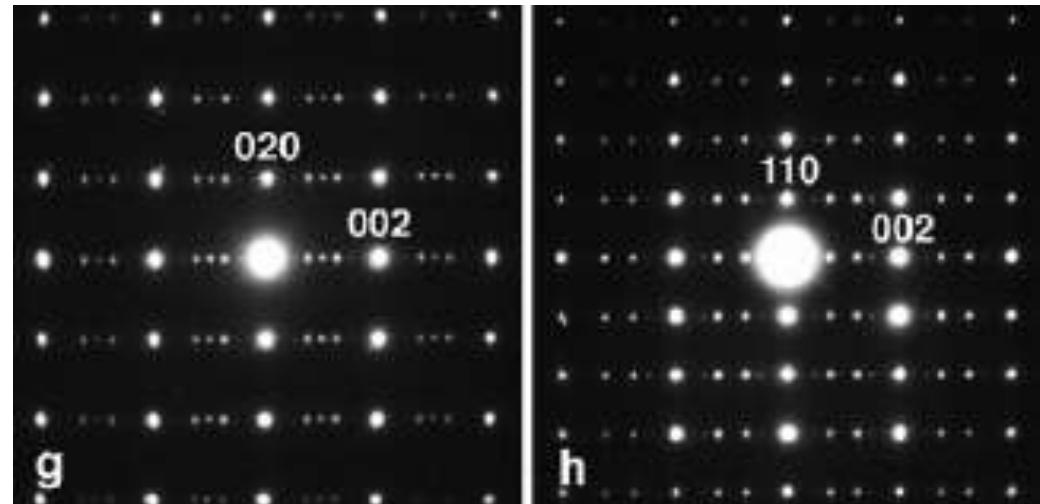
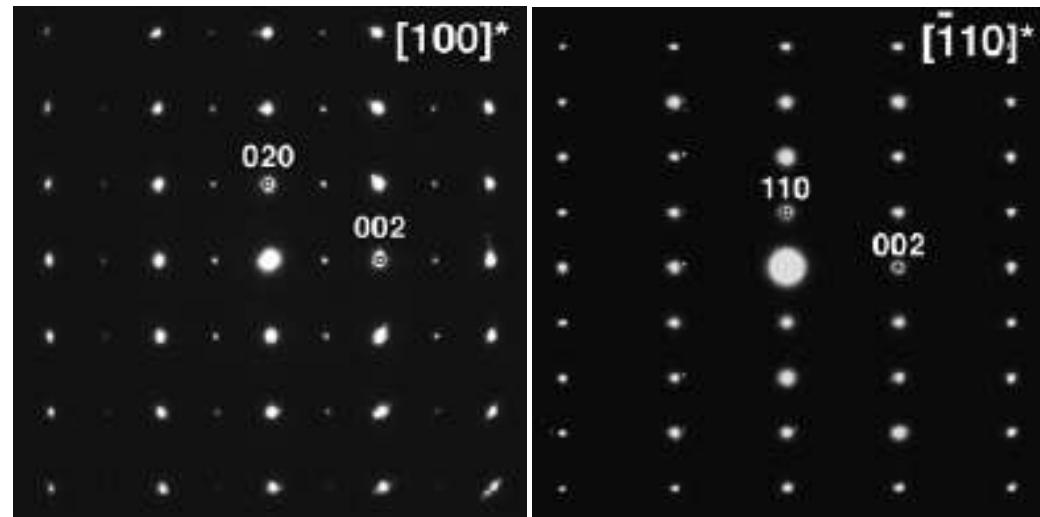
Related commensurate structure



Abakumov et al., Chemistry of Materials, 17, 5, 2005, 1123-1134



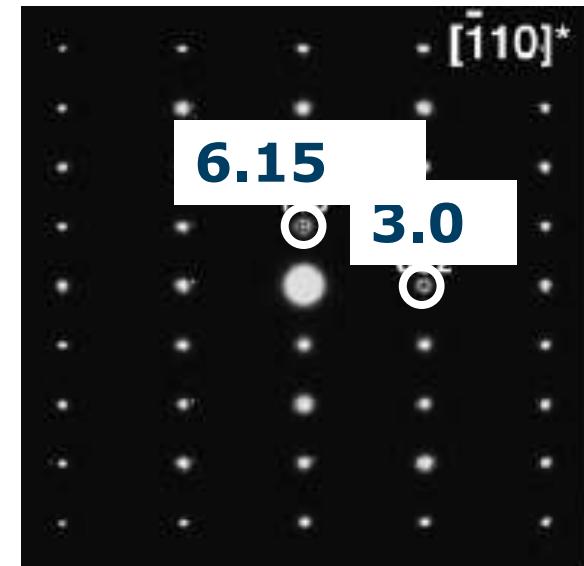
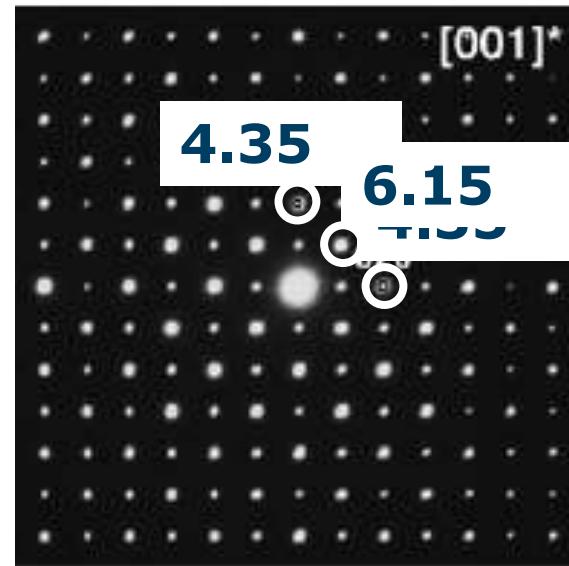
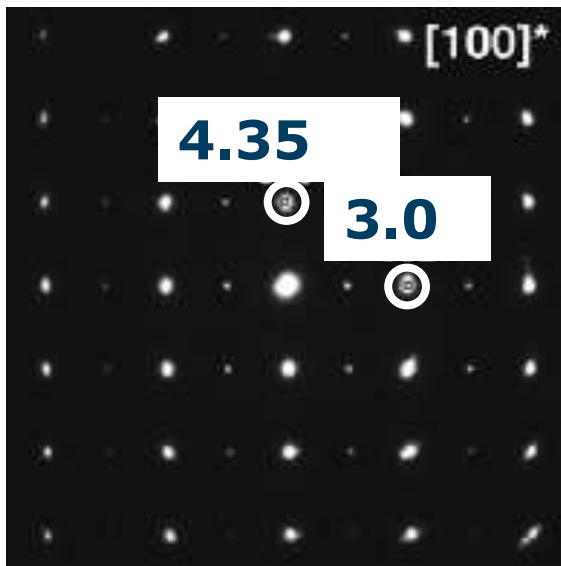
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IMS: indexing the subcell reflections



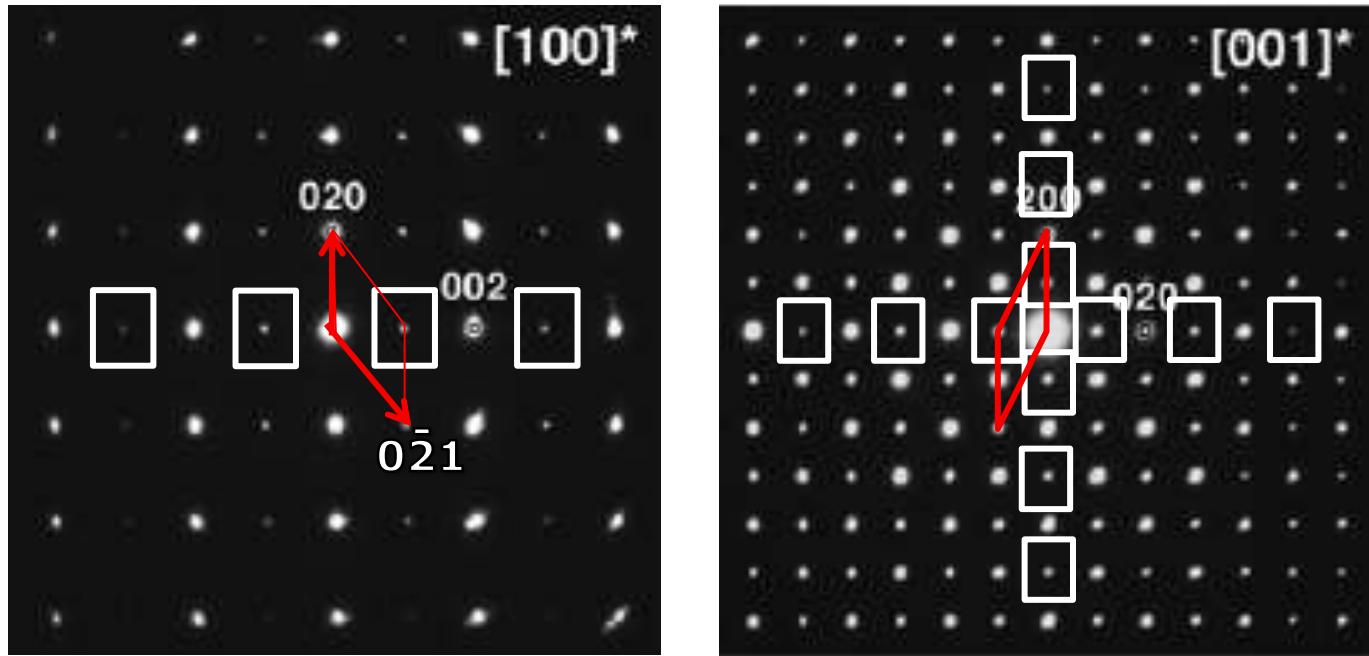
- Related commensurate structure Sb_2MnO_4
- $\text{P}4_2/\text{mbc}$; $a=b=8.7 \text{ \AA}$, $c= 6.0 \text{ \AA}$
- Measure R , calculate d with $R.d=C$ (know instrument constant)
- Index each reflection according to list of d-values for each reflection (cf. XRD)



All distances in Å

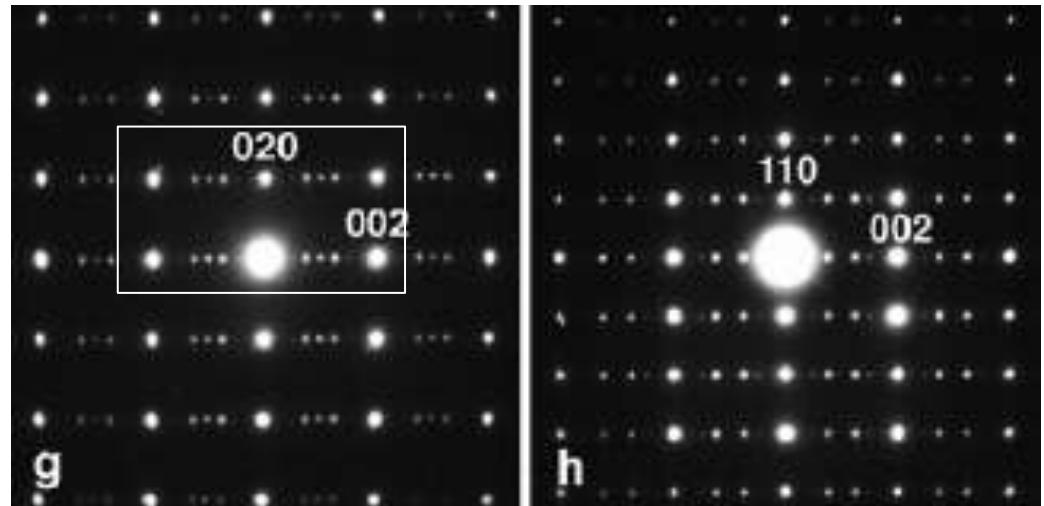
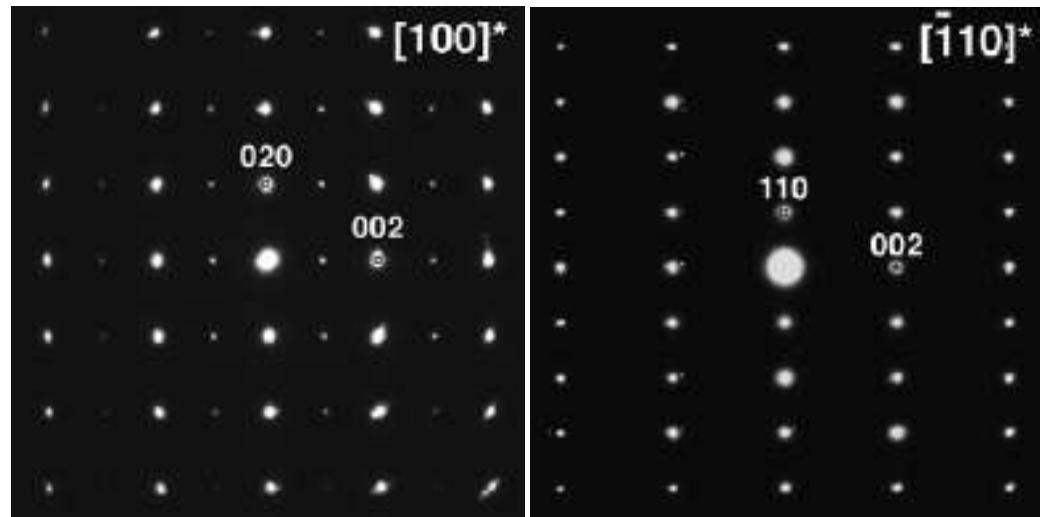
Dangers of double diffraction!

- $P4_2/mbc$
- Reflection conditions: $0kl$: $k=2n$; $hh\bar{l}$: $l=2n$; $00l$: $l=2n$; $h00$: $h=2n$
- In contradiction: $001, 003, \dots, 100, 300, \dots, 010, 030, \dots$ all caused by double diffraction



- Tilt around the row: forbidden reflections will disappear

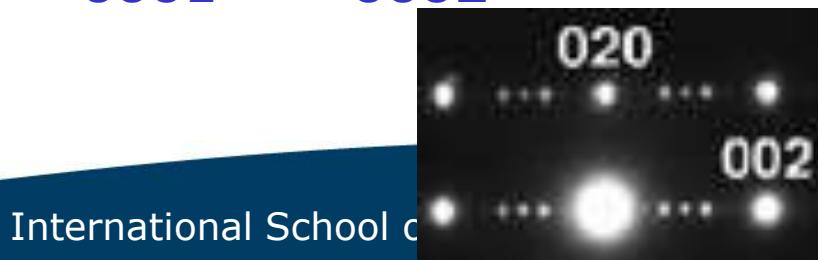
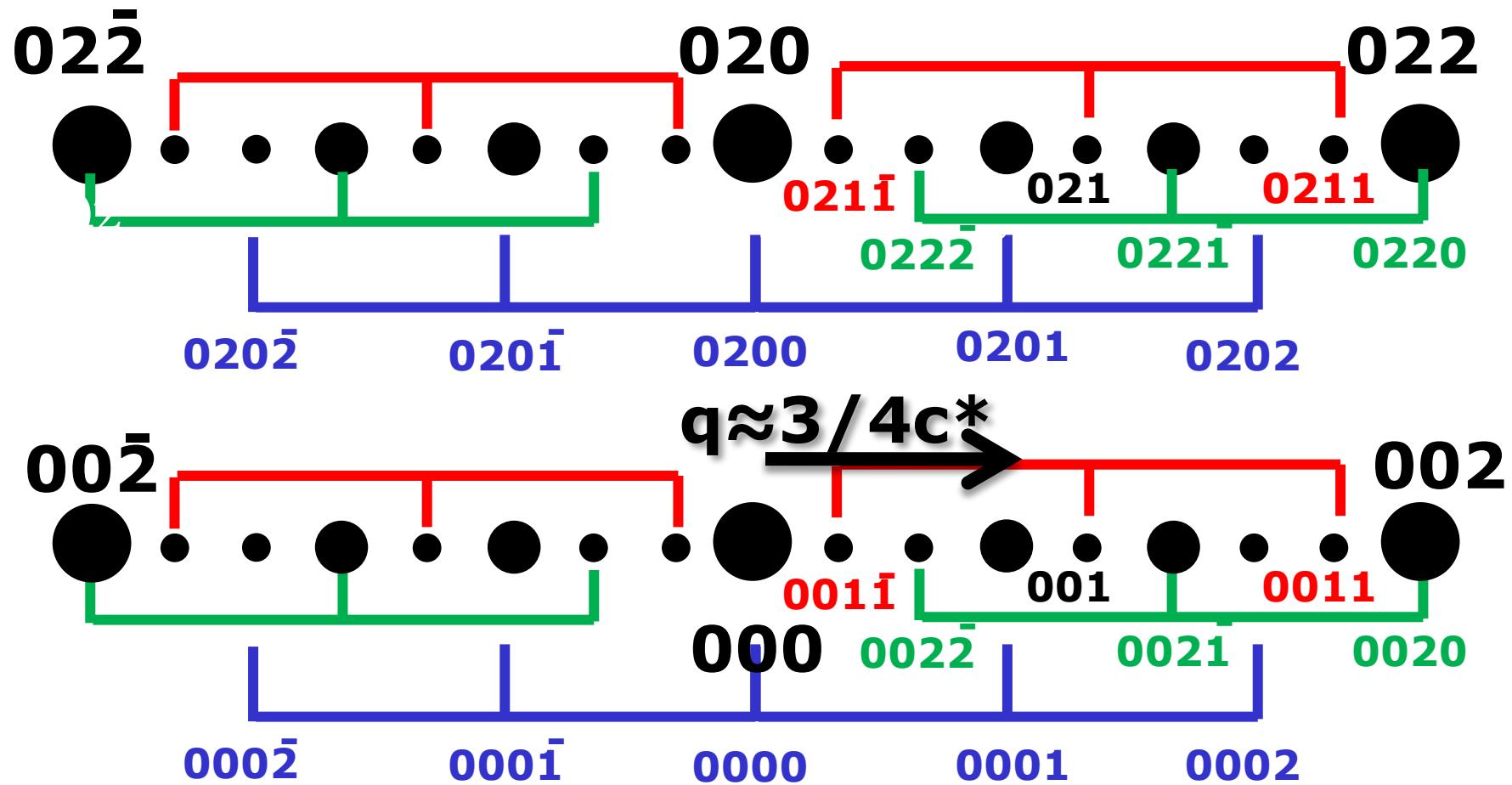
Subcell vs. IMS



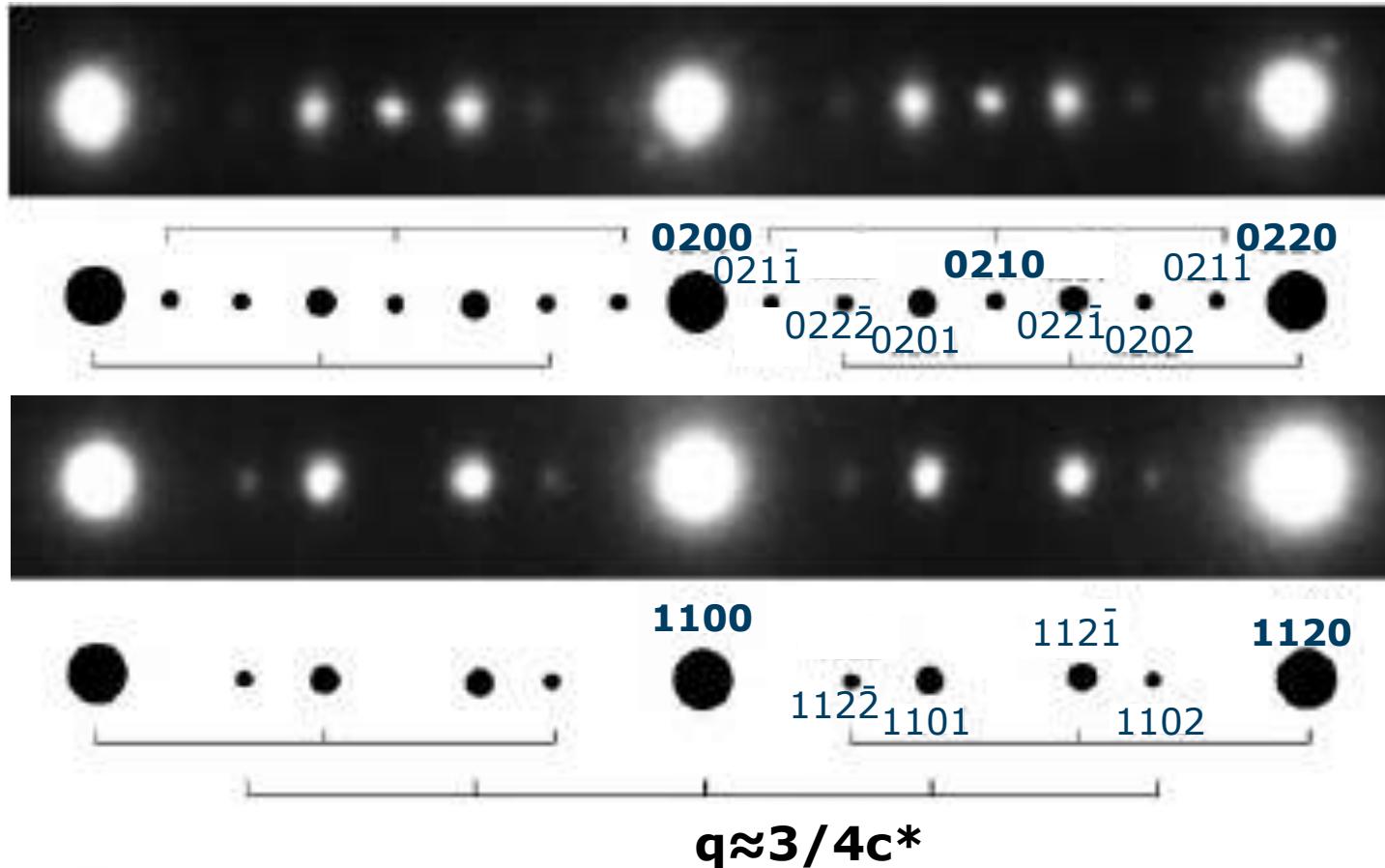
Abakumov et al., Chemistry of Materials, 17, 5, 2005, 1123-1134



Indexing the IMS

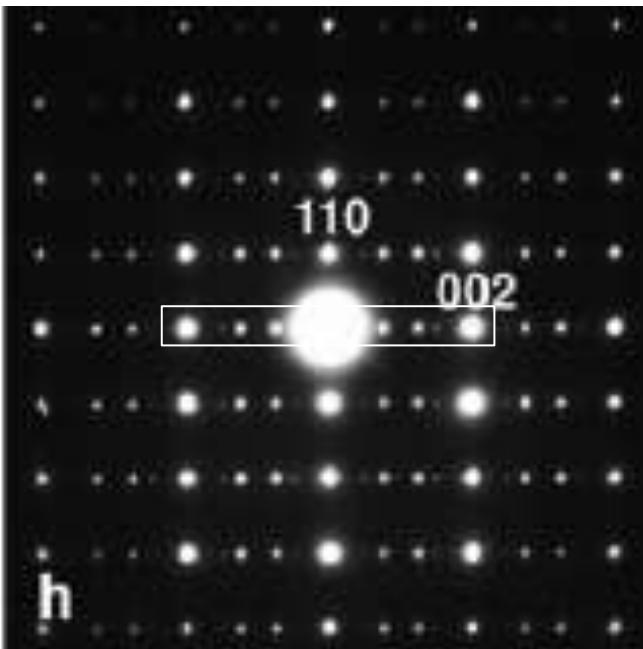
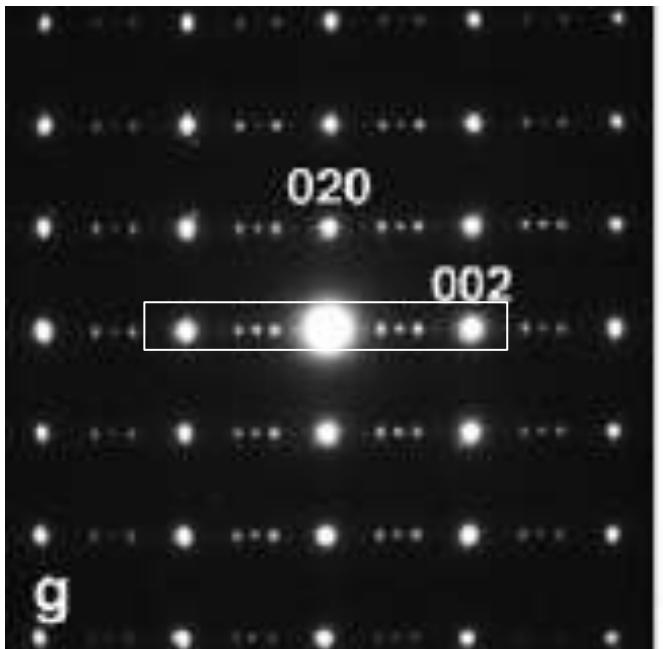


Reflection conditions for m



For this material: no special conditions for m: if subcell reflection is seen, also $hklm$ with $m=1$ and $m=2$ is seen.

Reflection conditions

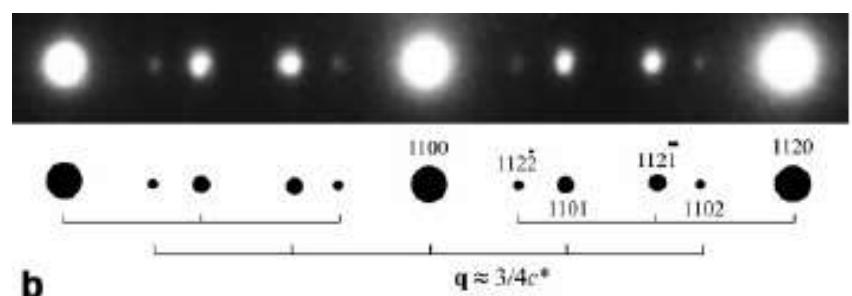
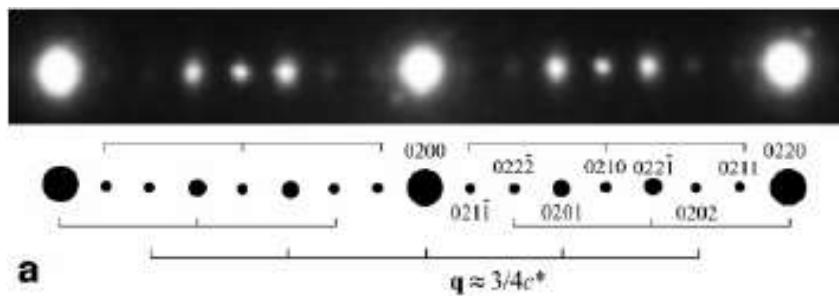


$$\left. \begin{array}{l} P4_2/mbc \\ 0kl: k=2n \\ hhl: l=2n \end{array} \right\}$$

$0klm: k=2n$

$hhlm: l=2n$

$P4_2/mbc(00\gamma)$

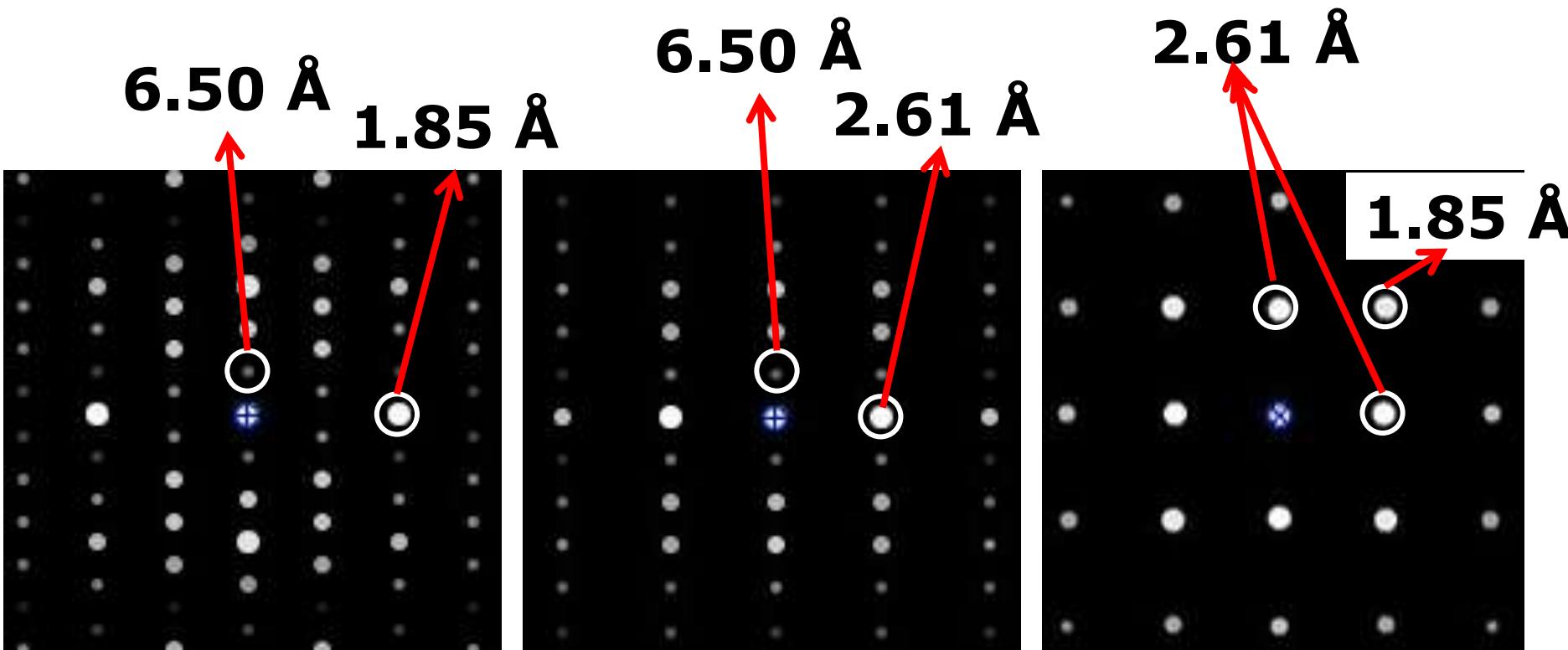


Exercise 1-1: IMS: index the non-modulated phase

- Index the commensurate structure LaSrCuO_4 , this will help you with indexing the next, incommensurate one.

Given data:

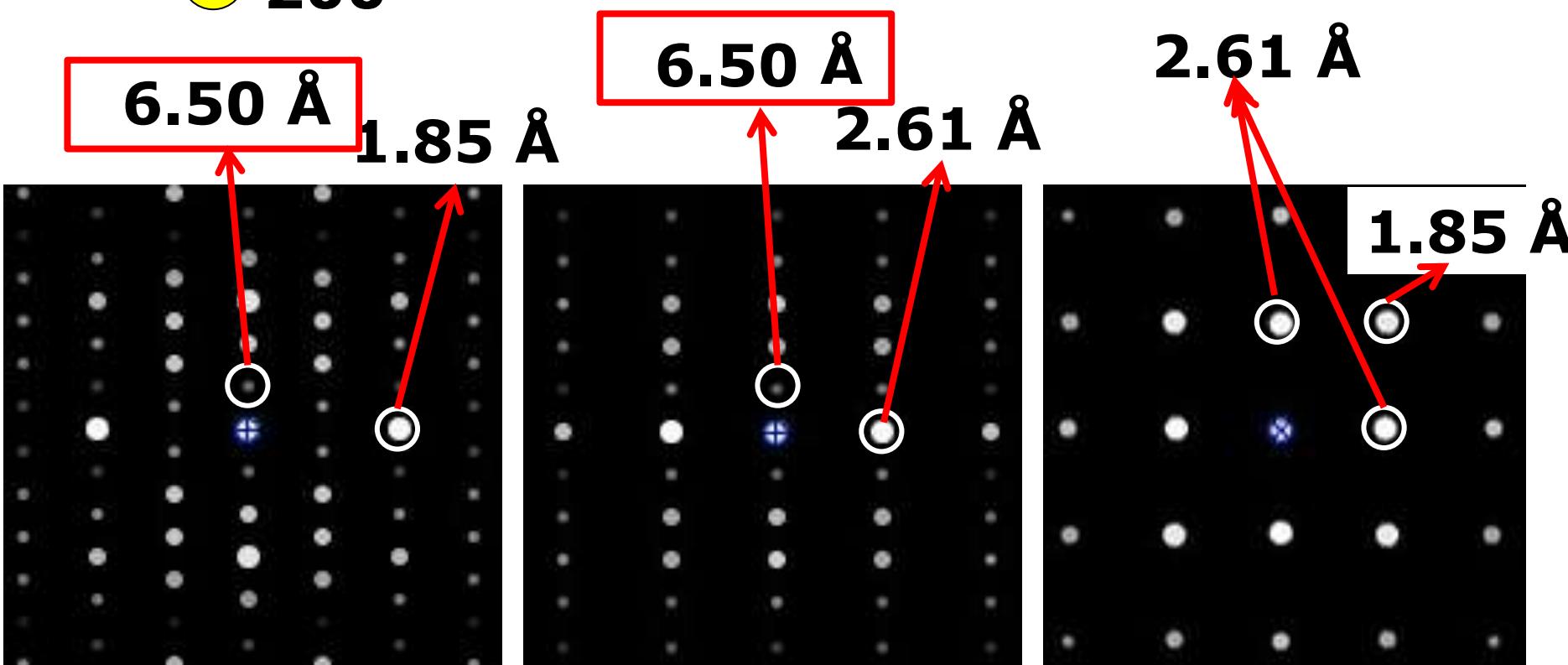
cell parameters of LaSrCuO_4 : $a=b=3.7 \text{ \AA}$, $c=13 \text{ \AA}$.



Given data:

cell parameters of LaSrCuO_4 : $a=b=3.7 \text{ \AA}$, $c=13 \text{ \AA}$.

- 001
- 002
- 200



(Simulated ED patterns)

Given data:

cell parameters of LaSrCuO_4 : $a=b=3.7 \text{ \AA}$, $c=13 \text{ \AA}$.

001

002

200

6.50 Å

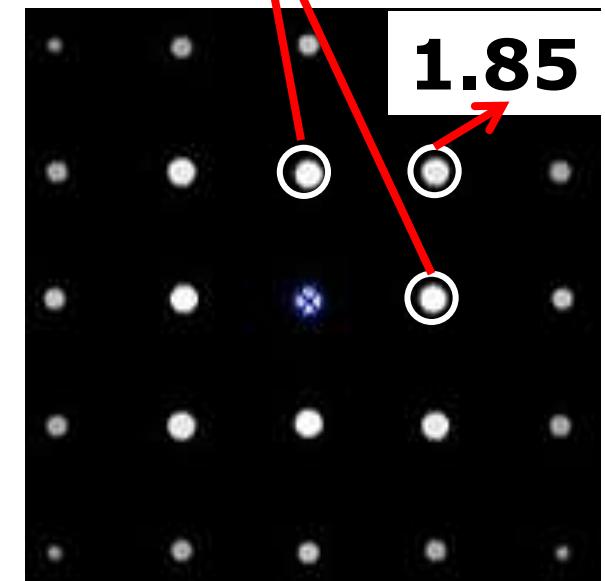
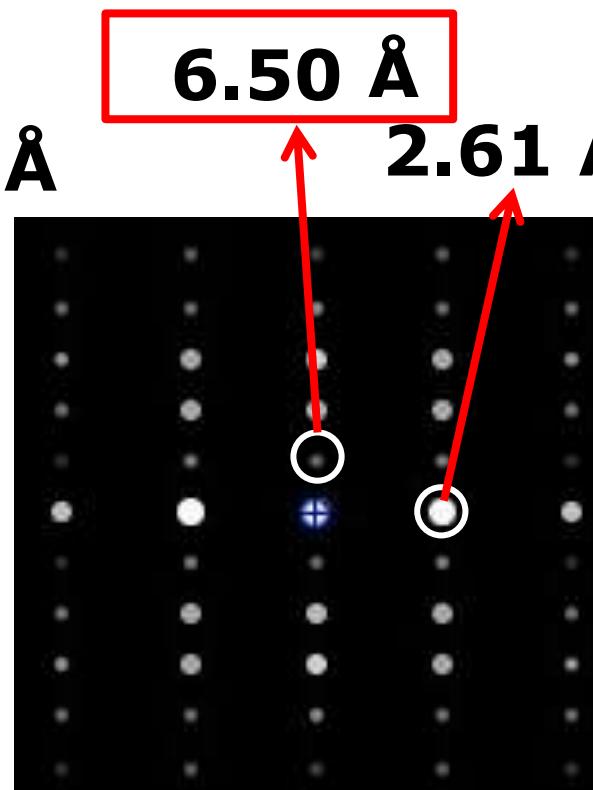
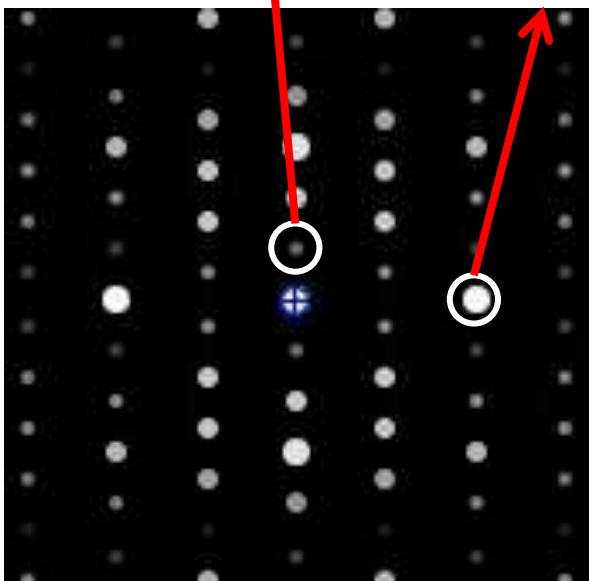
6.50 Å

2.61 Å

1.85 Å

2.61 Å

1.85 Å



(Simulated ED patterns)



Given data:

cell parameters of LaSrCuO_4 : $a=b=3.7 \text{ \AA}$, $c=13 \text{ \AA}$.

010

020

110

002

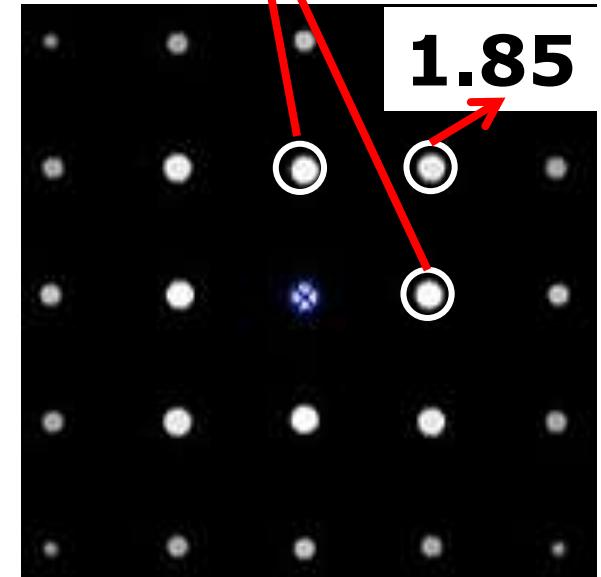
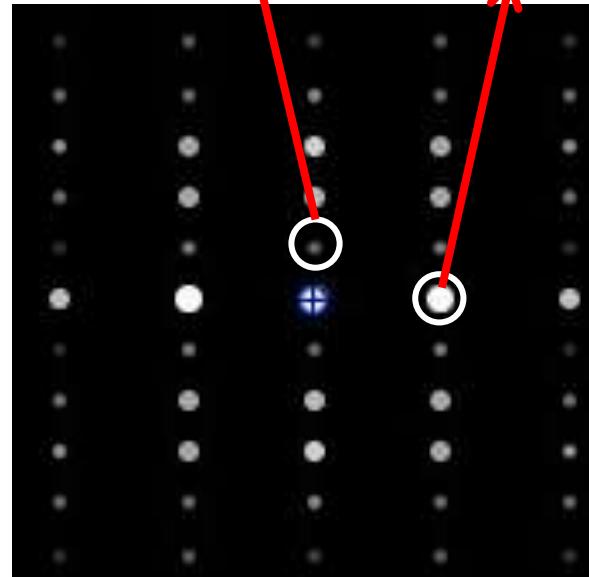
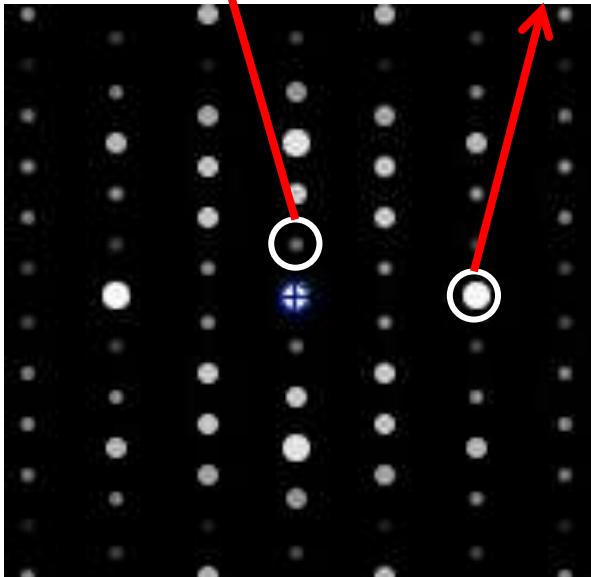
002

2.61 Å

1.85 Å

2.61 Å

1.85 Å

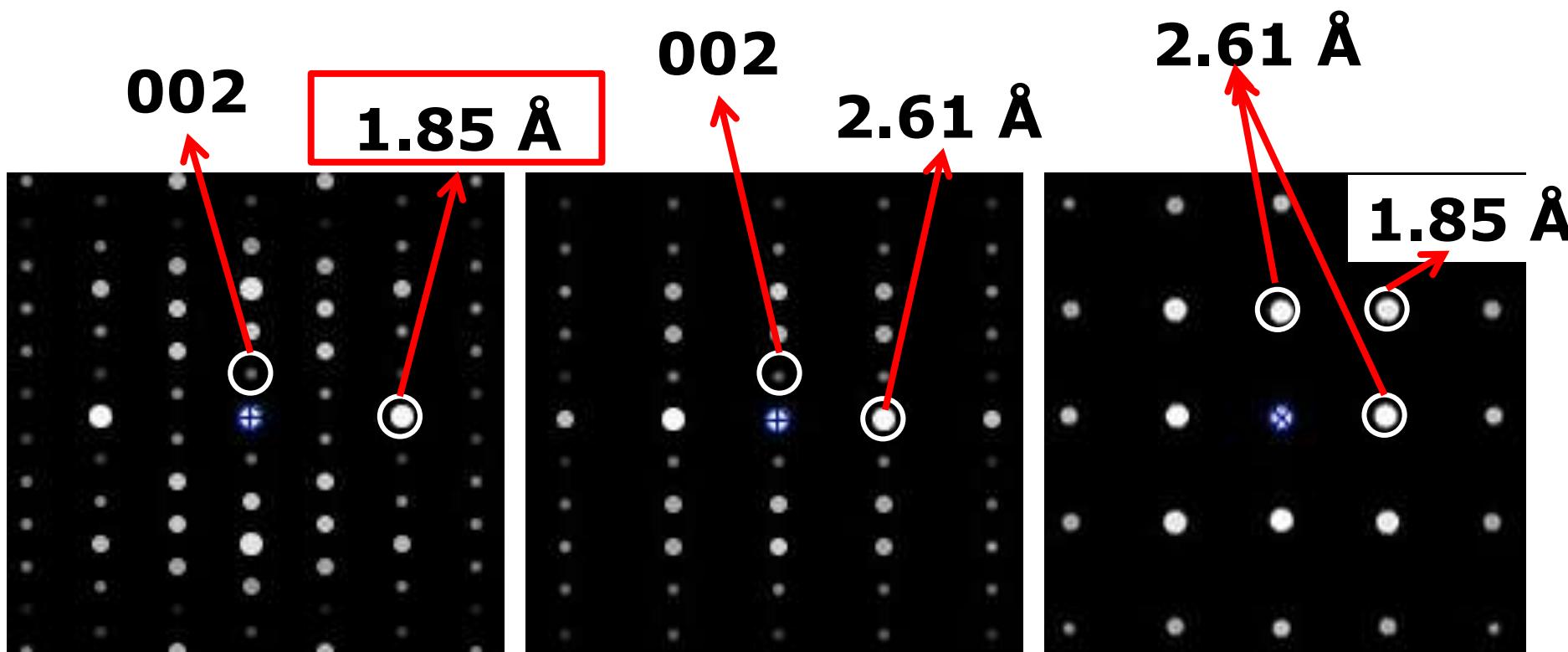


(Simulated ED patterns)

Given data:

cell parameters of LaSrCuO_4 : $a=b=3.7 \text{ \AA}$, $c=13 \text{ \AA}$.

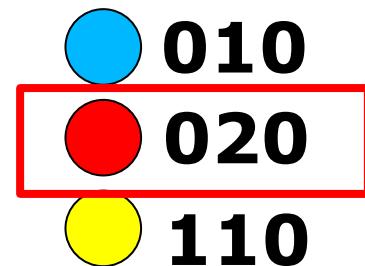
- 010
- 020
- 110



(Simulated ED patterns)

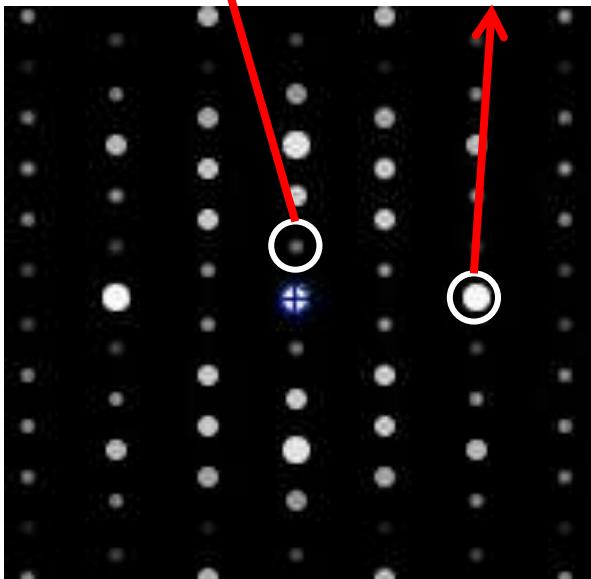
Given data:

cell parameters of LaSrCuO_4 : $a=b=3.7 \text{ \AA}$, $c=13 \text{ \AA}$.



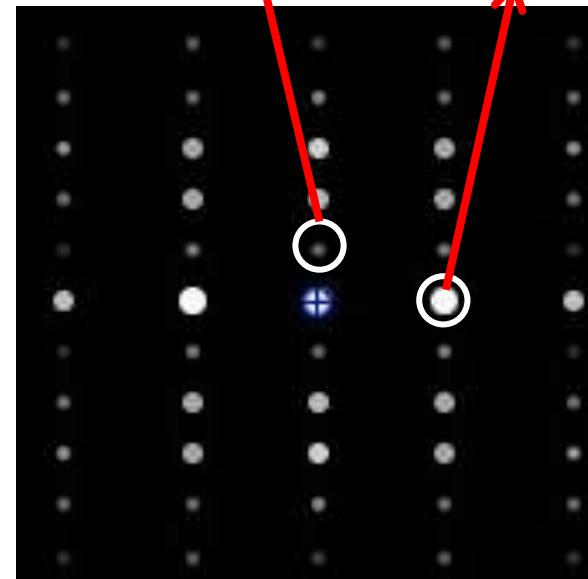
002

020



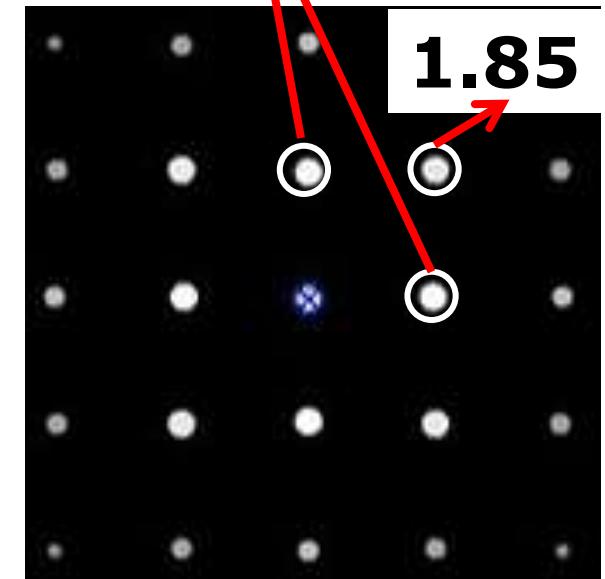
002

2.61 Å



2.61 Å

1.85 Å



(Simulated ED patterns)



Given data:

cell parameters of LaSrCuO_4 : $a=b=3.7 \text{ \AA}$, $c=13 \text{ \AA}$.

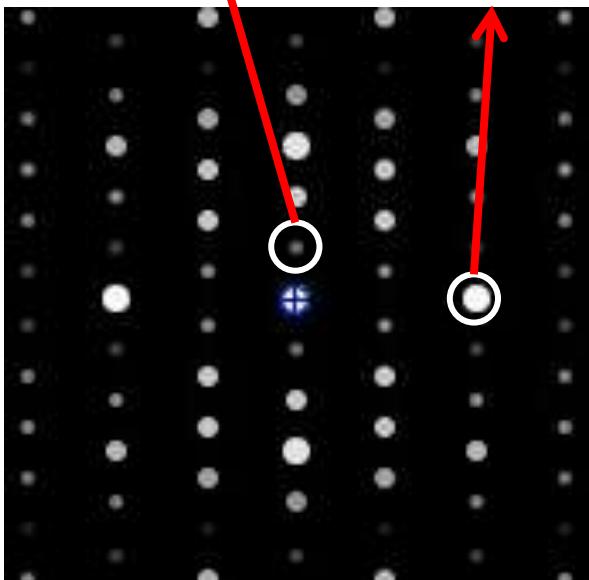
100

200

020

002

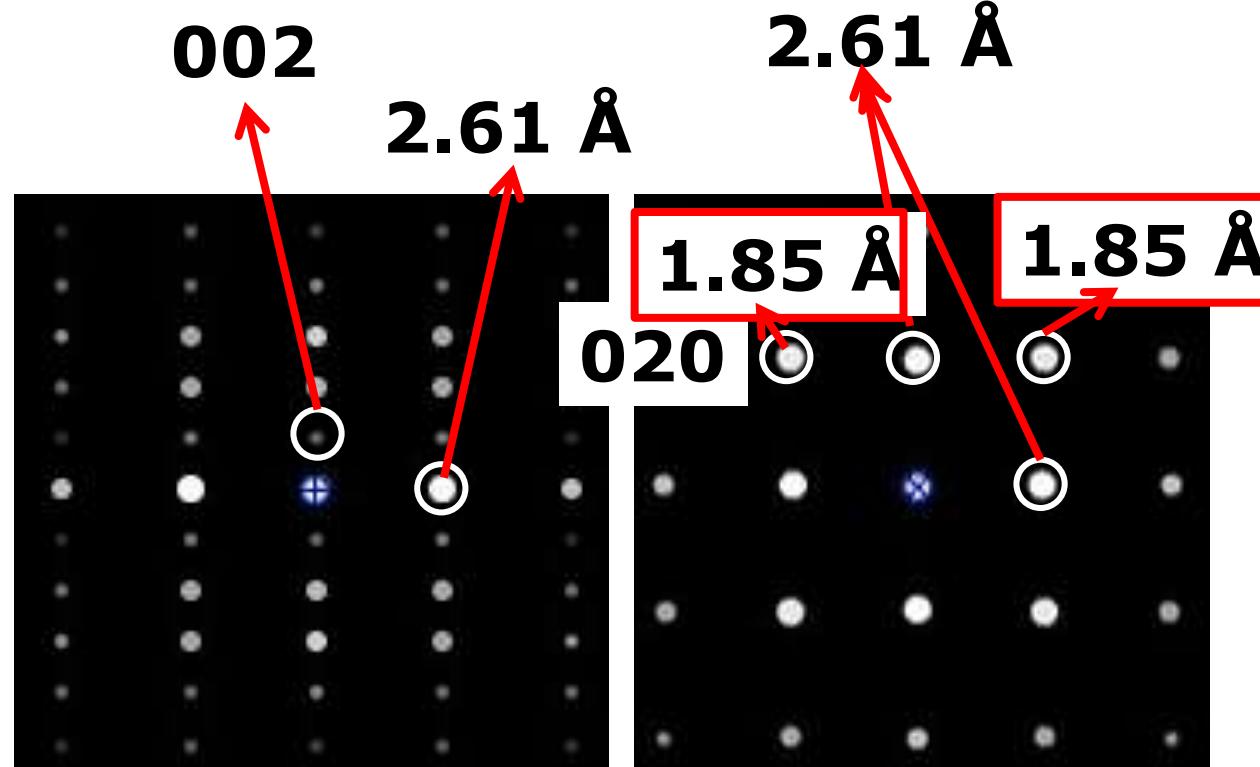
020



002

2.61 Å

2.61 Å

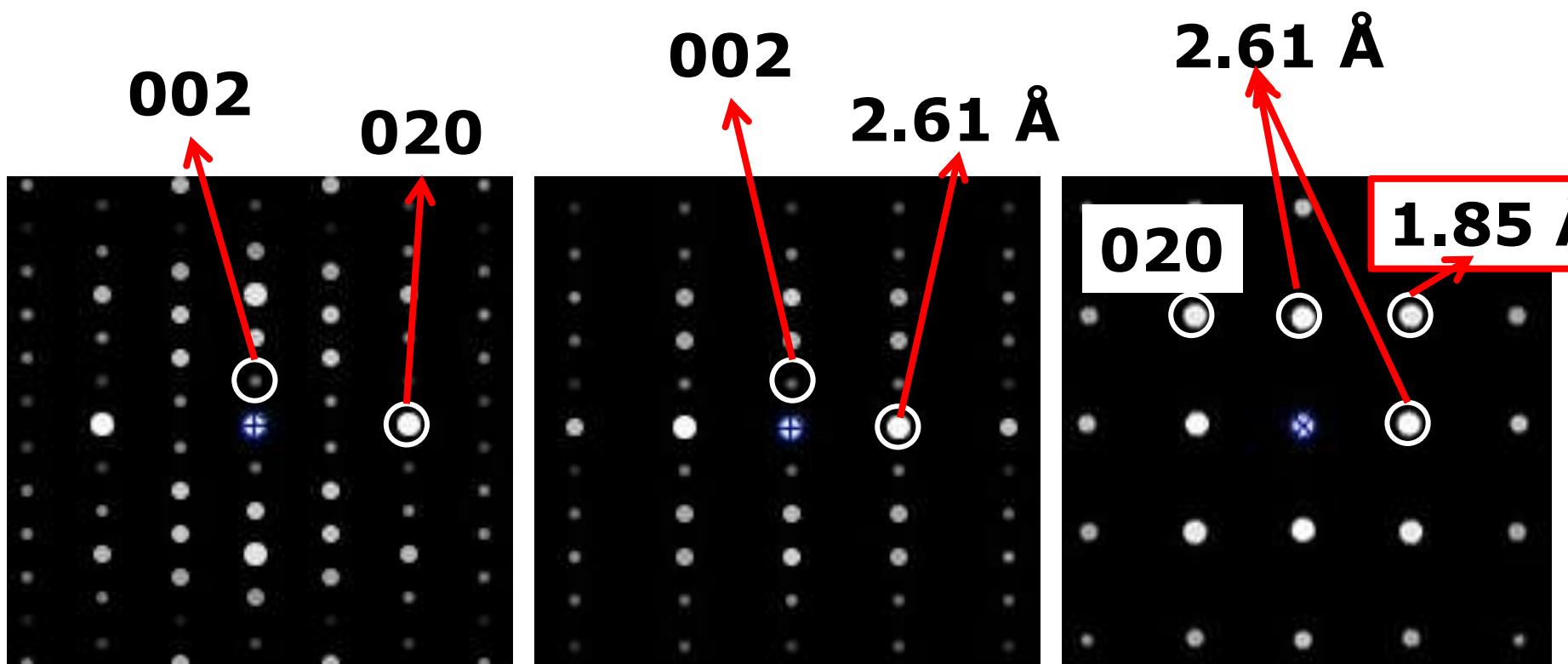
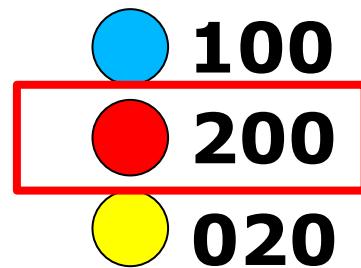


(Simulated ED patterns)



Given data:

cell parameters of LaSrCuO_4 : $a=b=3.7 \text{ \AA}$, $c=13 \text{ \AA}$.



(Simulated ED patterns)

Given data:

cell parameters of LaSrCuO₄: a=b=3.7 Å, c=13 Å.

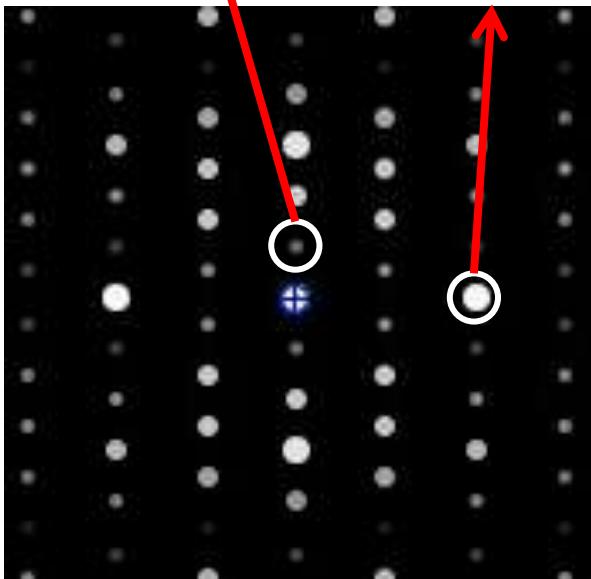
110

220

½ ½ 0

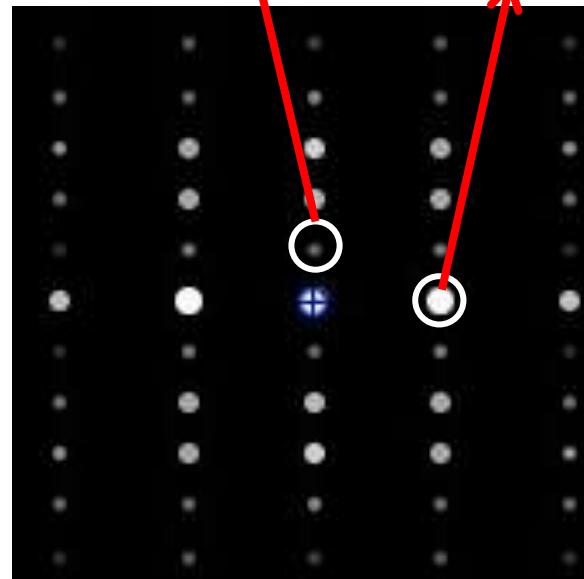
002

020

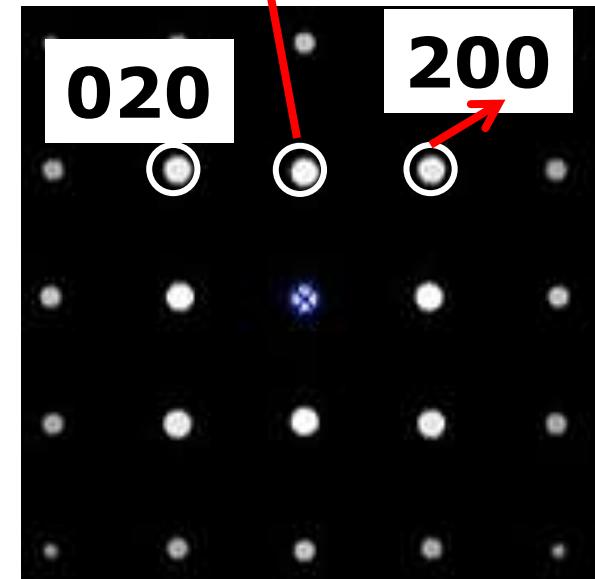


002

2.61 Å



2.61 Å



(Simulated ED patterns)

Given data:

cell parameters of LaSrCuO₄: a=b=3.7 Å, c=13 Å.

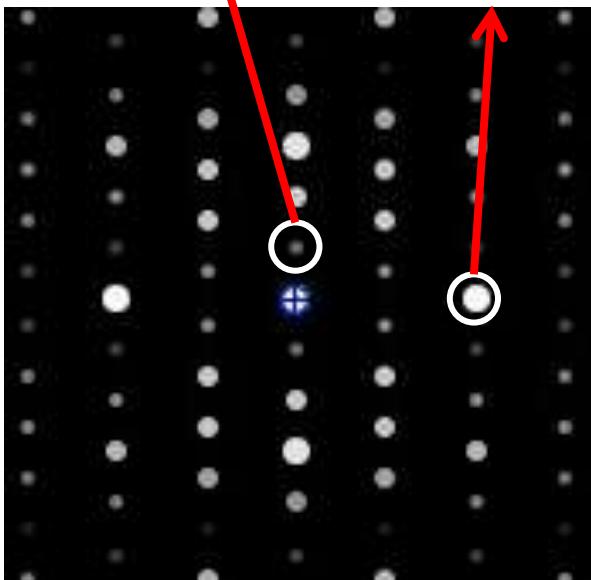
110

220

½ ½ 0

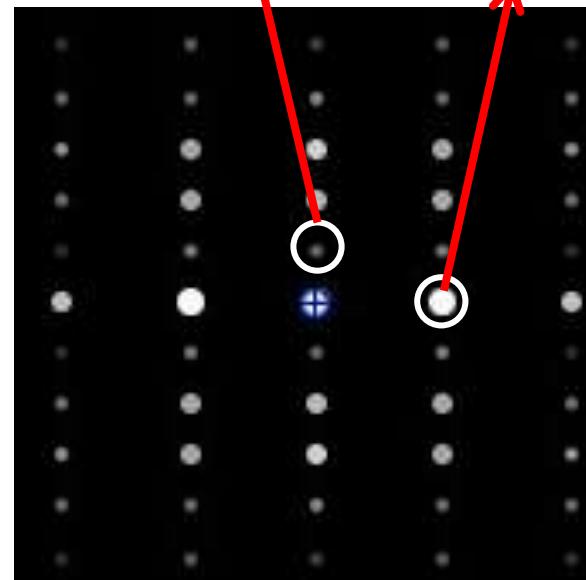
002

020

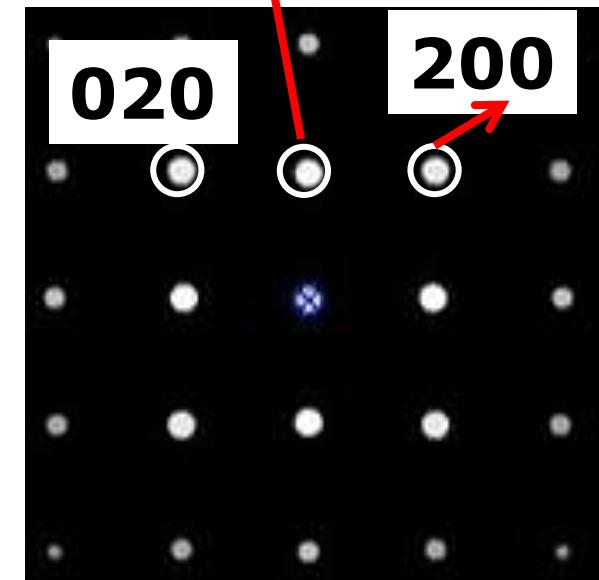


002

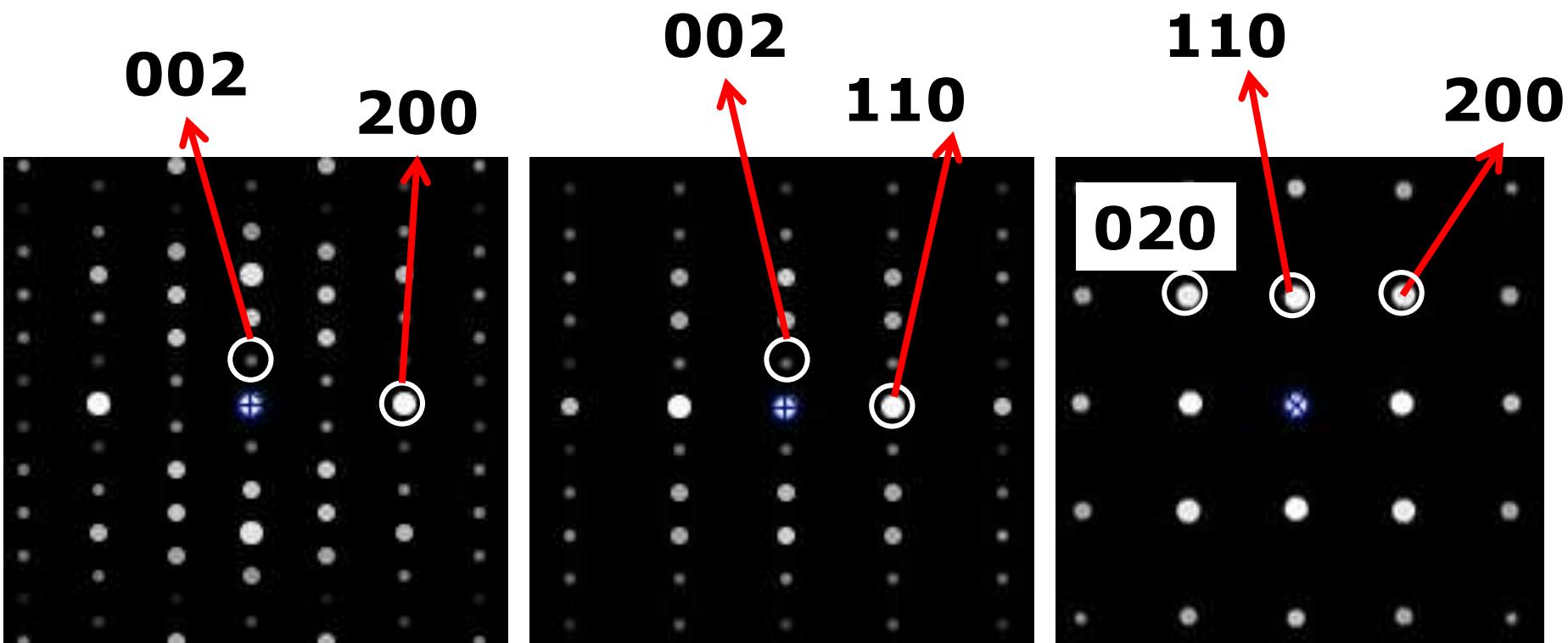
2.61 Å



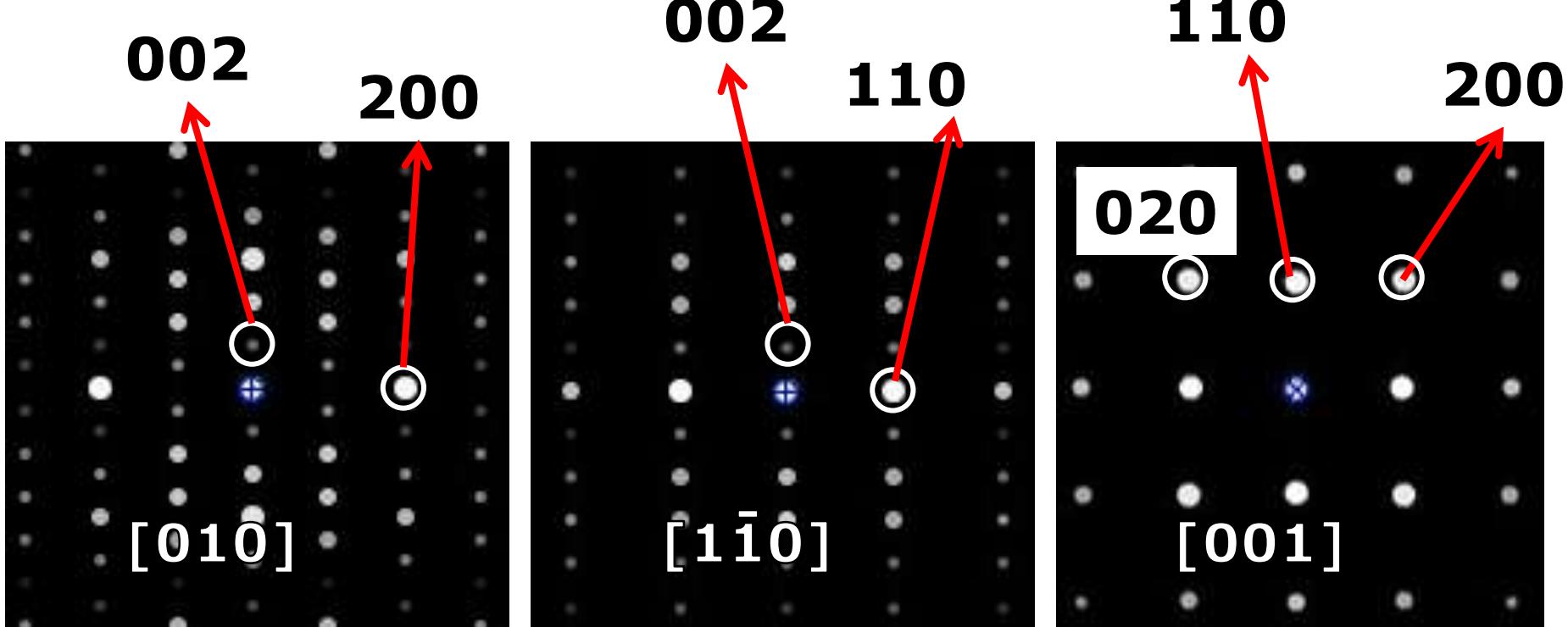
2.61 Å



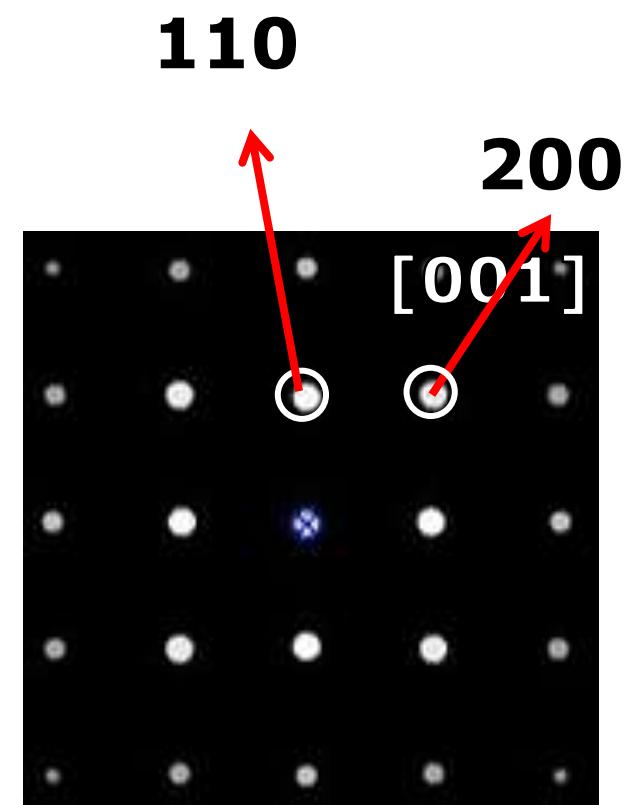
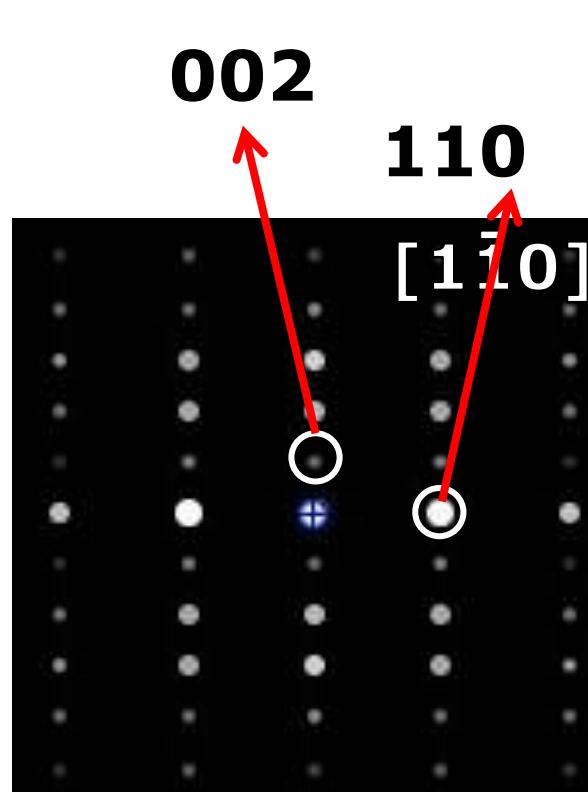
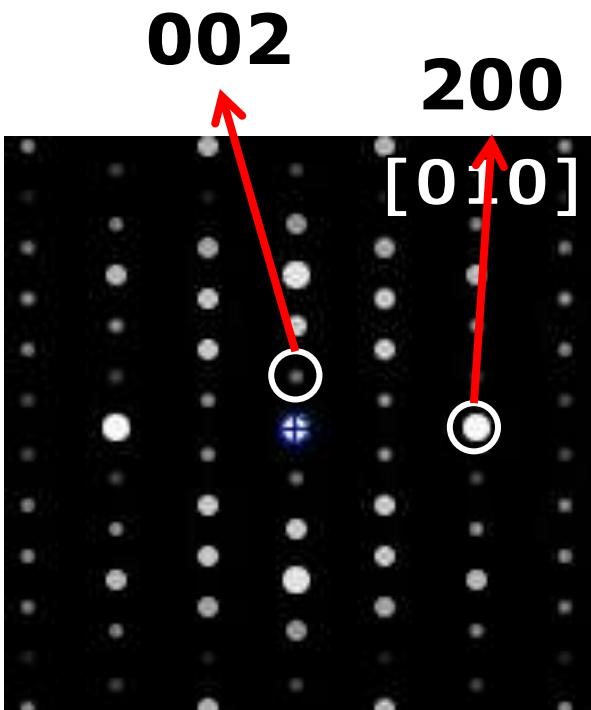
Indexed solution



Zone indices



Exercise 1-2: IMS: determine the reflection condition $h0l$



$$h0l: h+l=2n$$



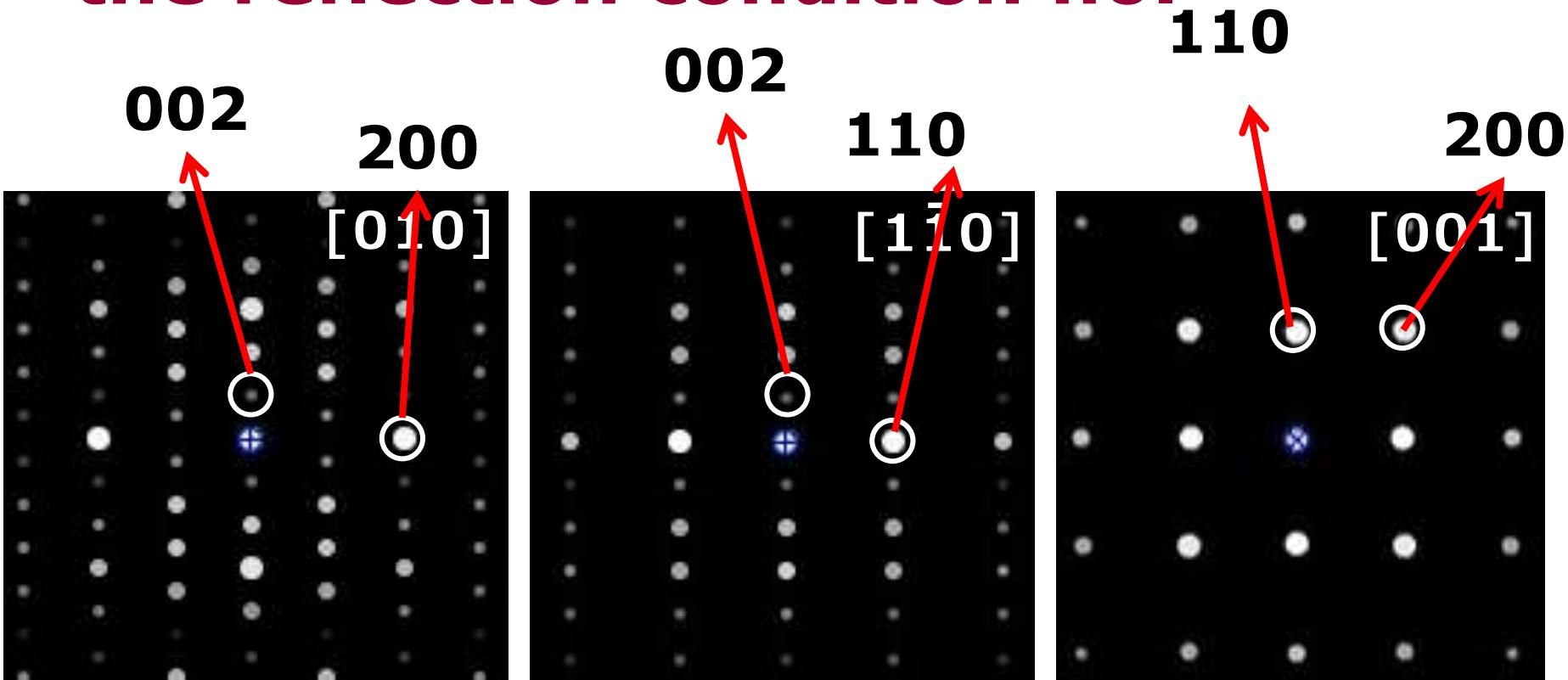
$$h = 2n$$



$$l = 2n$$

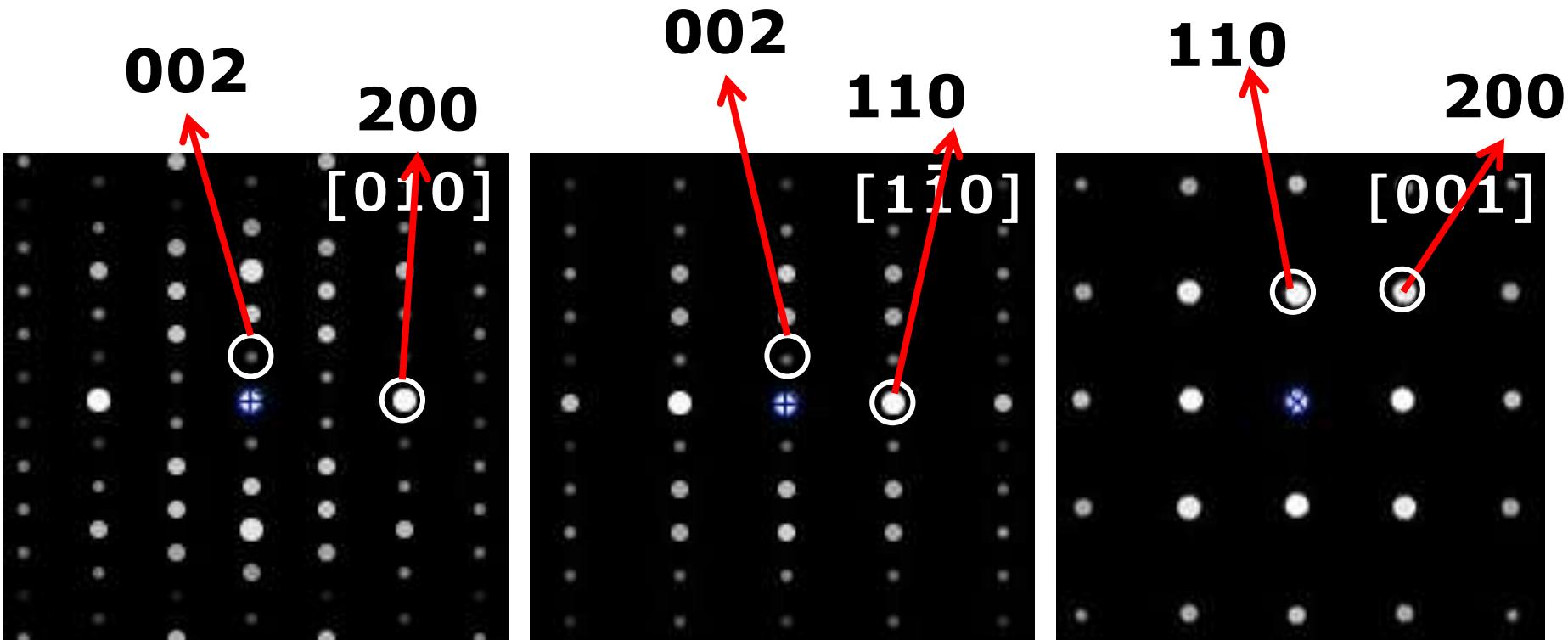


Exercise 1-2: IMS: determine the reflection condition $h0l$



$h0l: h+l=2n$ ●

Exercise 1-2: IMS: determine the reflection condition hhl



$h0l: h+l=2n$

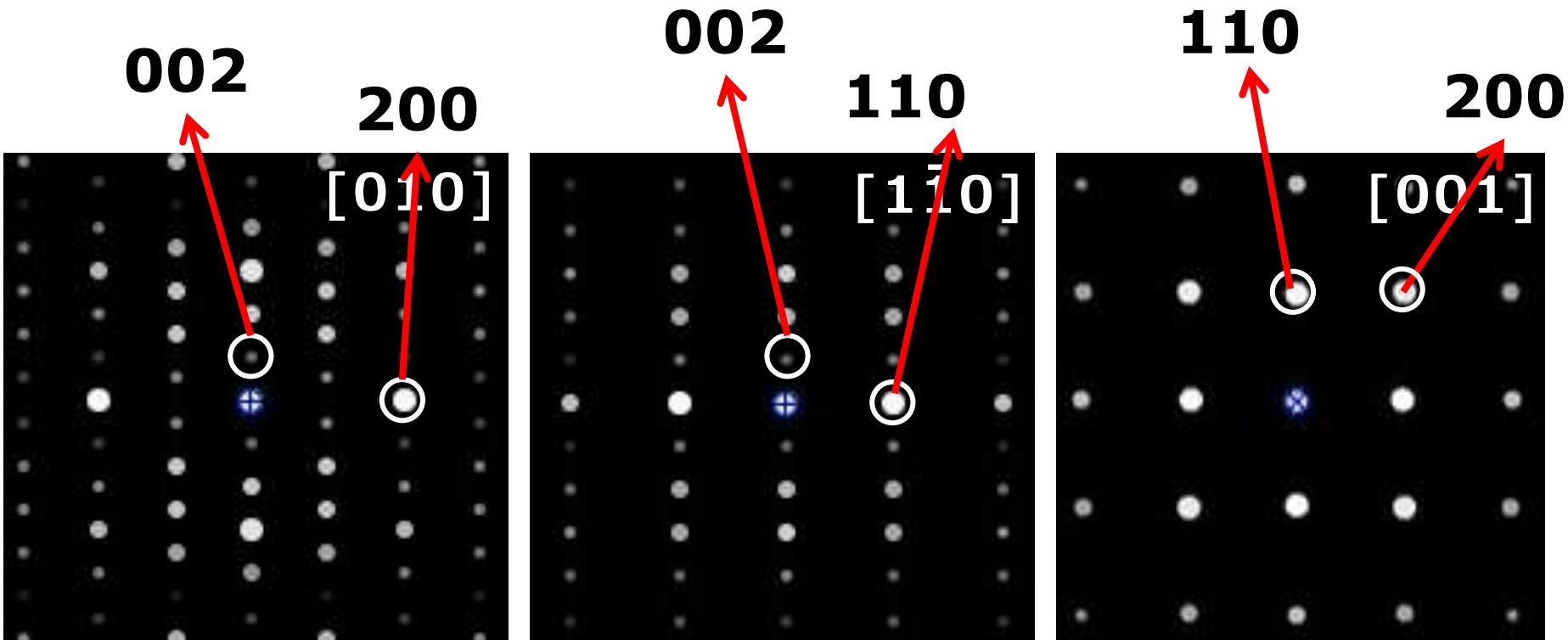
$hhl: h+l=2n$

$h = 2n$

$l = 2n$



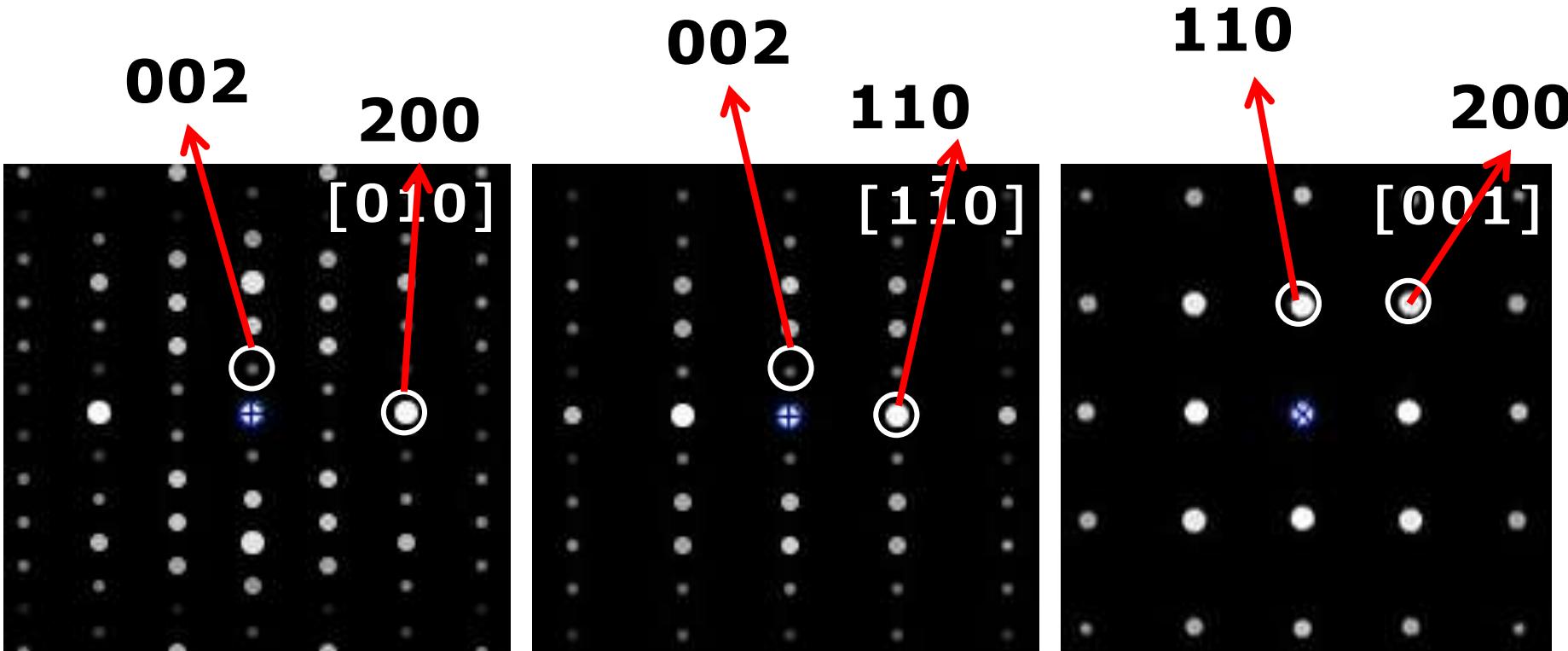
Exercise 1-2: IMS: determine the reflection condition hhl



h0l: $h+l=2n$ **hh_l:**

$$l = 2n$$

Exercise 1-2: IMS: determine the reflection condition $hk0$



$h0l: h+l=2n$

$hh\bar{l}: l = 2n$

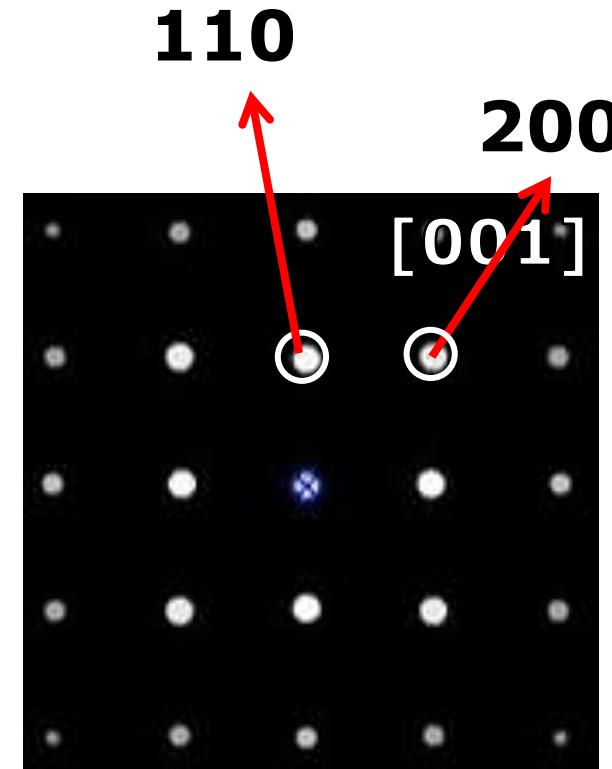
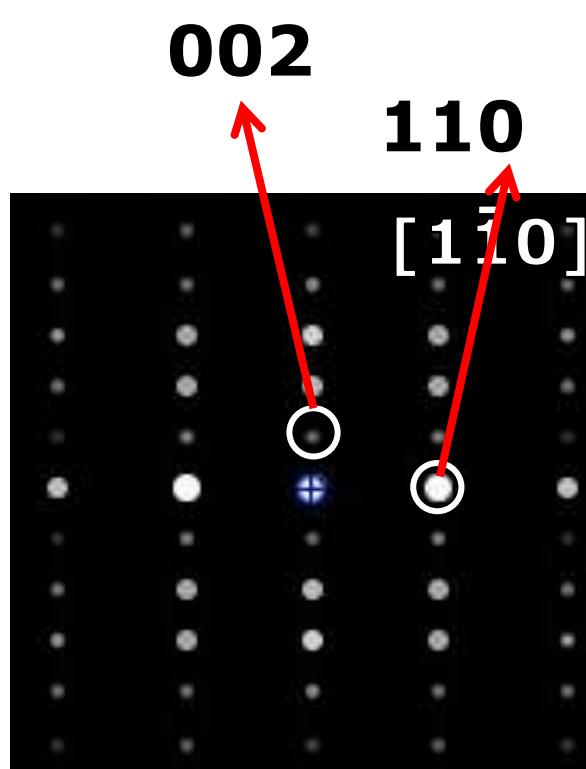
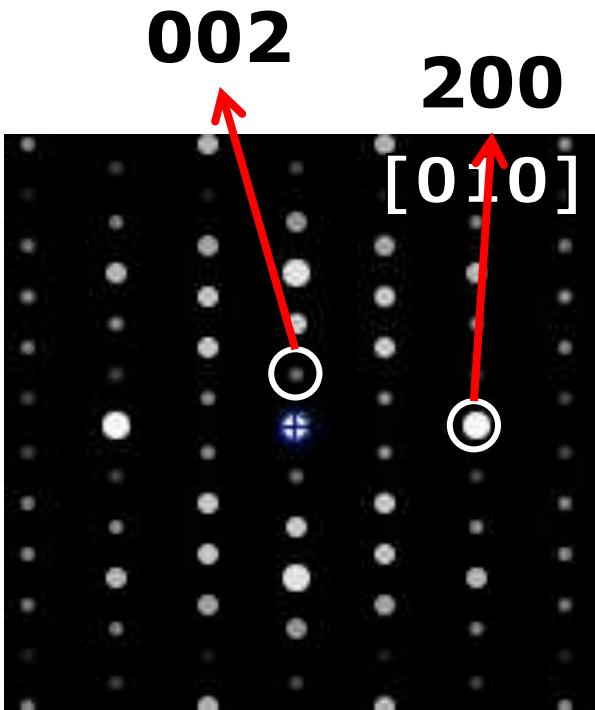
$hk0: h+k=2n$

$h = 2n$

$k = 2n$



Exercise 1-2: IMS: determine the reflection condition $hk0$

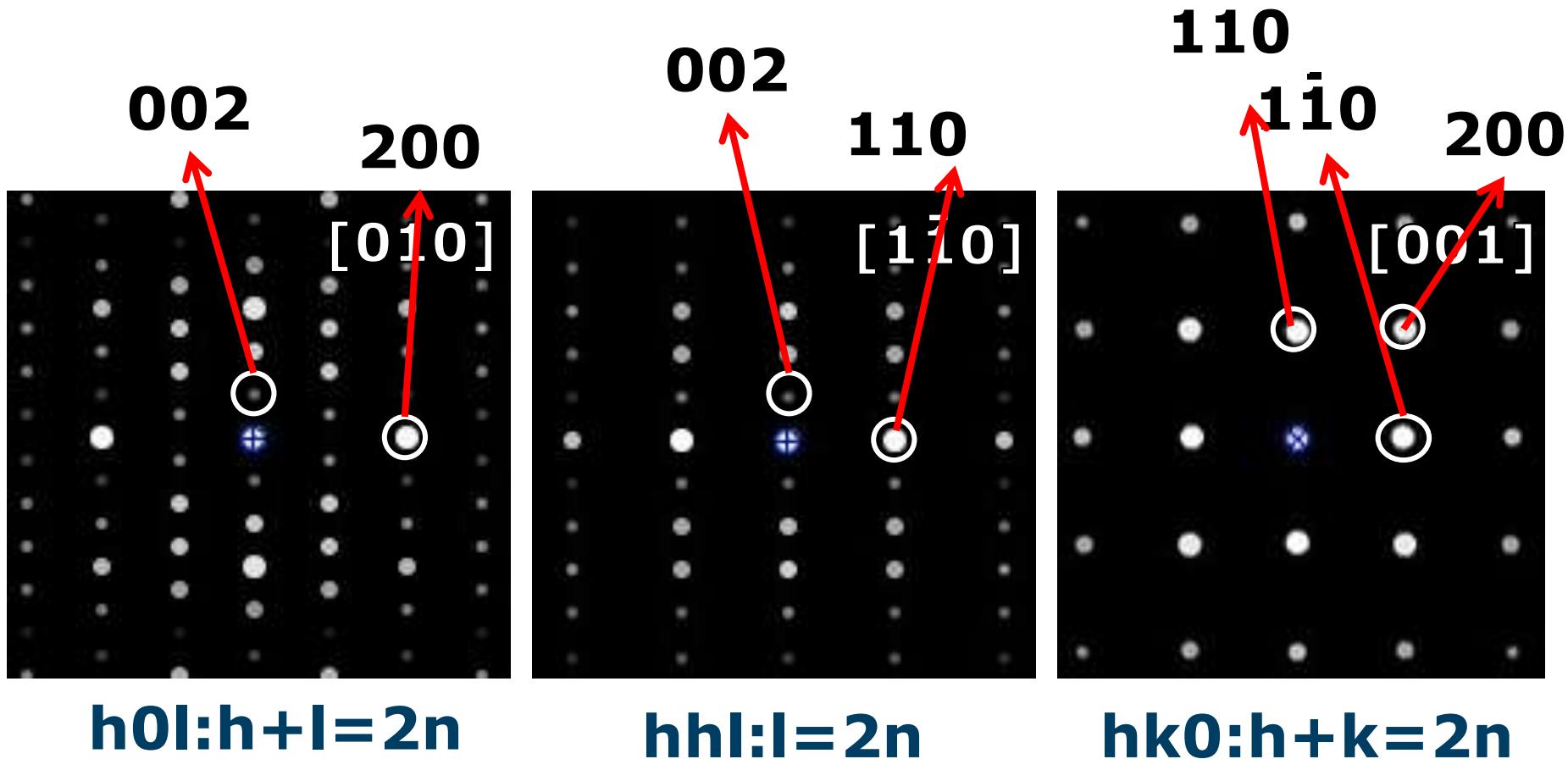


h0l: $h+l=2n$

hh_l: $l = 2n$

hk0: $h+k=2n$

Solved reflection conditions



Also (from rest of the zones) $hkl: h+k+l=2n$.

Determine space group

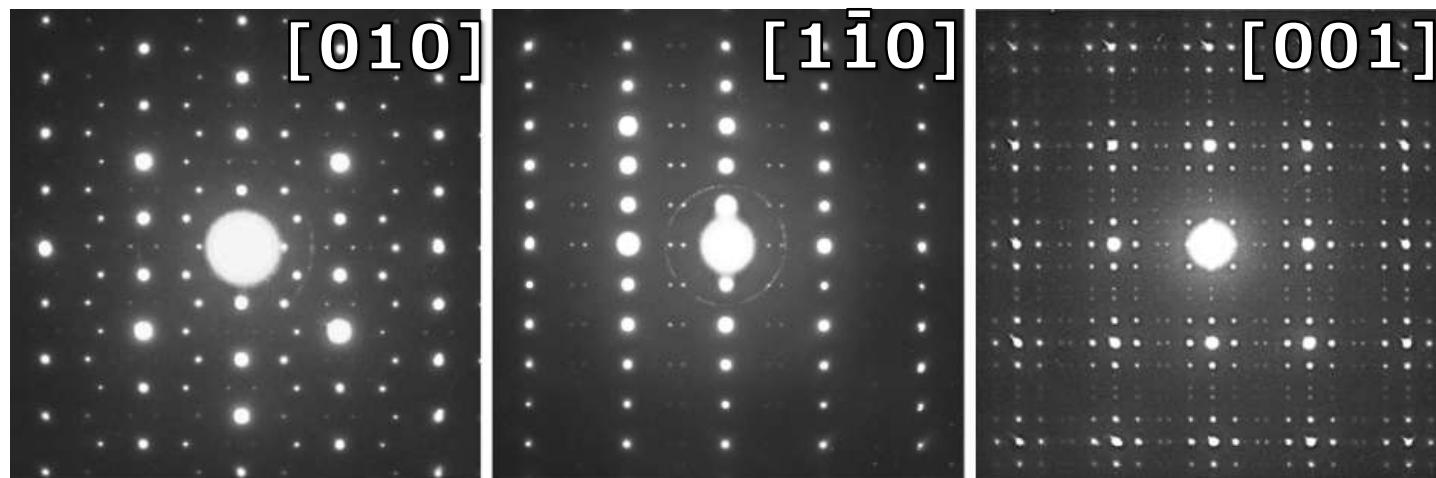
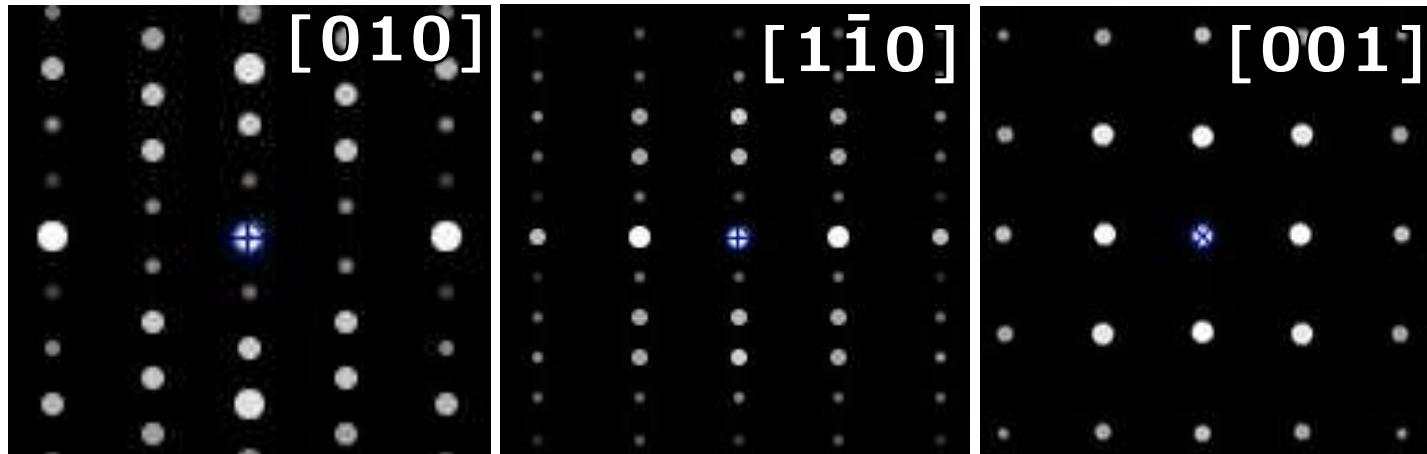
TETRAGONAL, Laue classes $4/m$ and $4/mmm$ (cont.)

Reflection conditions							Laue class								
hkl	$hk0$	$0kl$	hhl	$00l$	$0k0$	$hh0$	Extinction symbol	$4/m$		$4/mmm$ ($4/m$ $2/m$ $2/m$)					
								4	$\bar{4}$	$4/m$	422	$4nm$	$\bar{4}2m$	$\bar{4}m2$	$4/mmm$
	$h+k$	k				k	Pnb -								$P4/nbm$ (125)
	$h+k$	k	l	l		k	$Pnbc$								$P4_2/nbc$ (133)
	$h+k$	l		l		k	Pnc -								$P4_2/ncm$ (138)
	$h+k$	l	l	l		k	$Pncc$								$P4/ncc$ (130)
	$h+k$	$k+l$		l		k	Pnn -								$P4_2/nmm$ (134)
	$h+k$	$k+l$	l	l		k	$Pnnc$								$P4/nnc$ (126)
$h+k+l$	$h+k$	$k+l$	l	l		k	$I---$	$I4$ (79)	$I\bar{4}$ (82)	$I4/m$ (87)	$I422$ (97)	$I4nm$ (107)	$I\bar{4}2m$ (121)	$I\bar{4}m2$ (119)	$I4/mmm$ (139)
$h+k+l$	$h+k$	$k+l$	l	$l=4n$	k		$I4_1---$	$I4_1$ (80)			$I4_122$ (98)				
$h+k+l$	$h+k$	$k+l$	$\frac{1}{2}$	$l=4n$	k	h	$I--d$					$I4_1md$ (109)	$I\bar{4}2d$ (122)		
$h+k+l$	$h+k$	k, l	t	l	k		$I-c-$					$I4cm$ (108)	$I\bar{4}c2$ (120)	$I4/mcm$ (140)	
$h+k+l$	$h+k$	k, l	$\frac{1}{2}$	$l=4n$	k	h	$I-cd$					$I4_1cd$ (110)			
$h+k+l$	h, k	$k+l$	t	$l=4n$	k		$I4_1/a$ (88)								$I4_1/and$ (141)
$h+k+l$	h, k	$k+l$	$\frac{1}{2}$	$l=4n$	k	h	$Ia-d$								
$h+k+l$	$h+k$	$k+l$	l	l		k									$I4/mmm$ (139)

- International Tables: I---
- Most symmetrical $I4/mmm$



Incommensurate vs. basic cell



Hadermann et al., Journal of Materials Chemistry, 17, 22, 2007, 2344-2350

$\text{LaSrCuO}_{3.52}$

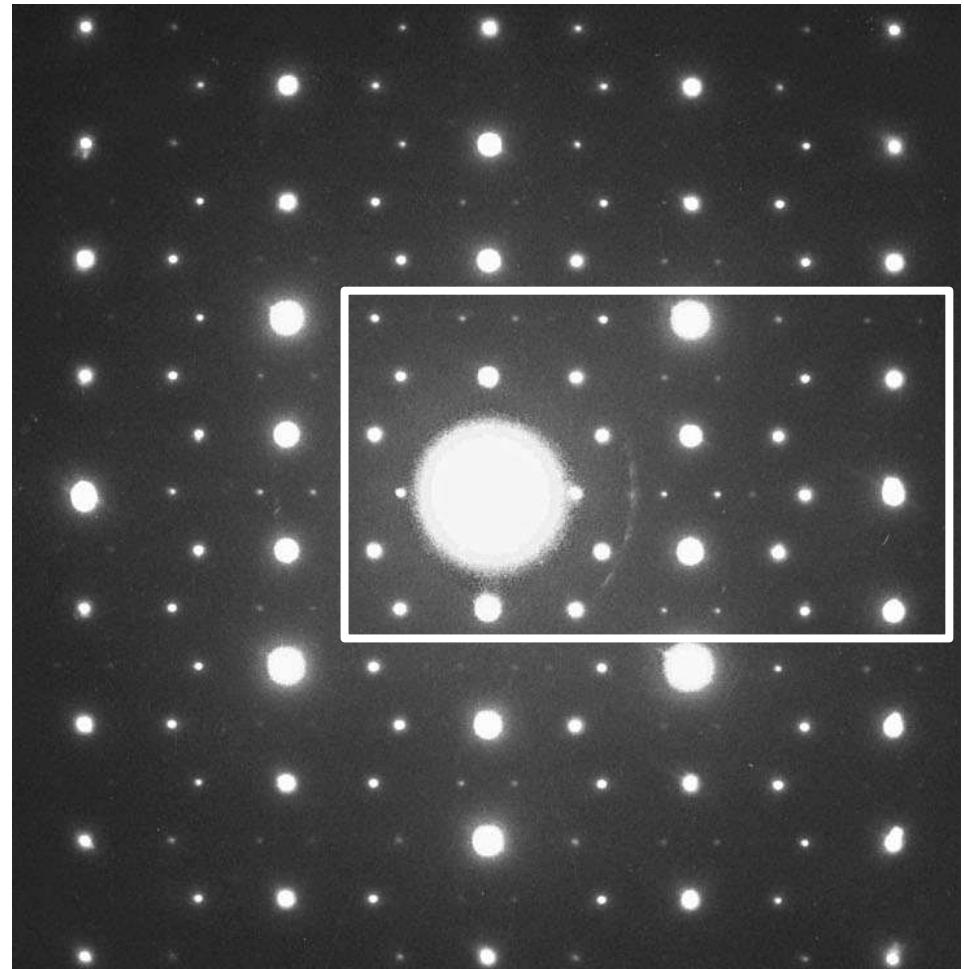
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Exercise 1-3: index the IMS

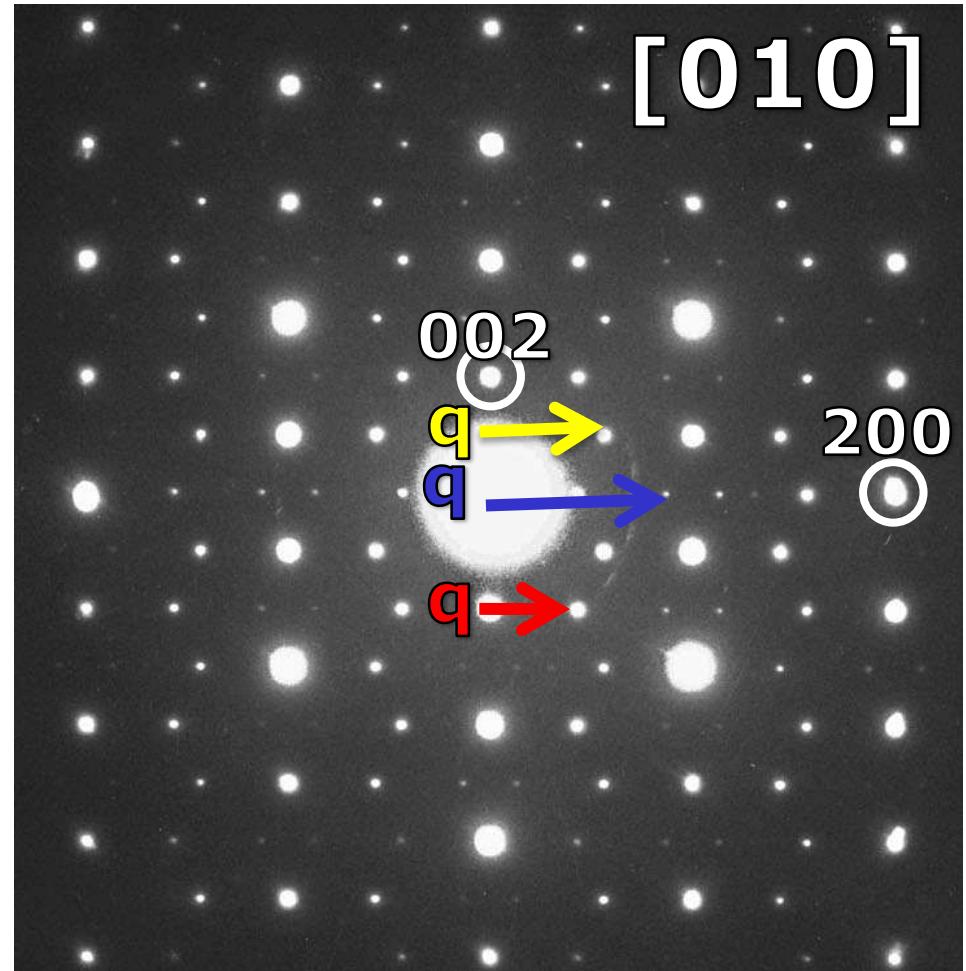
Identify and
index the subcell
reflections.



LaSrCuO_{3.52}

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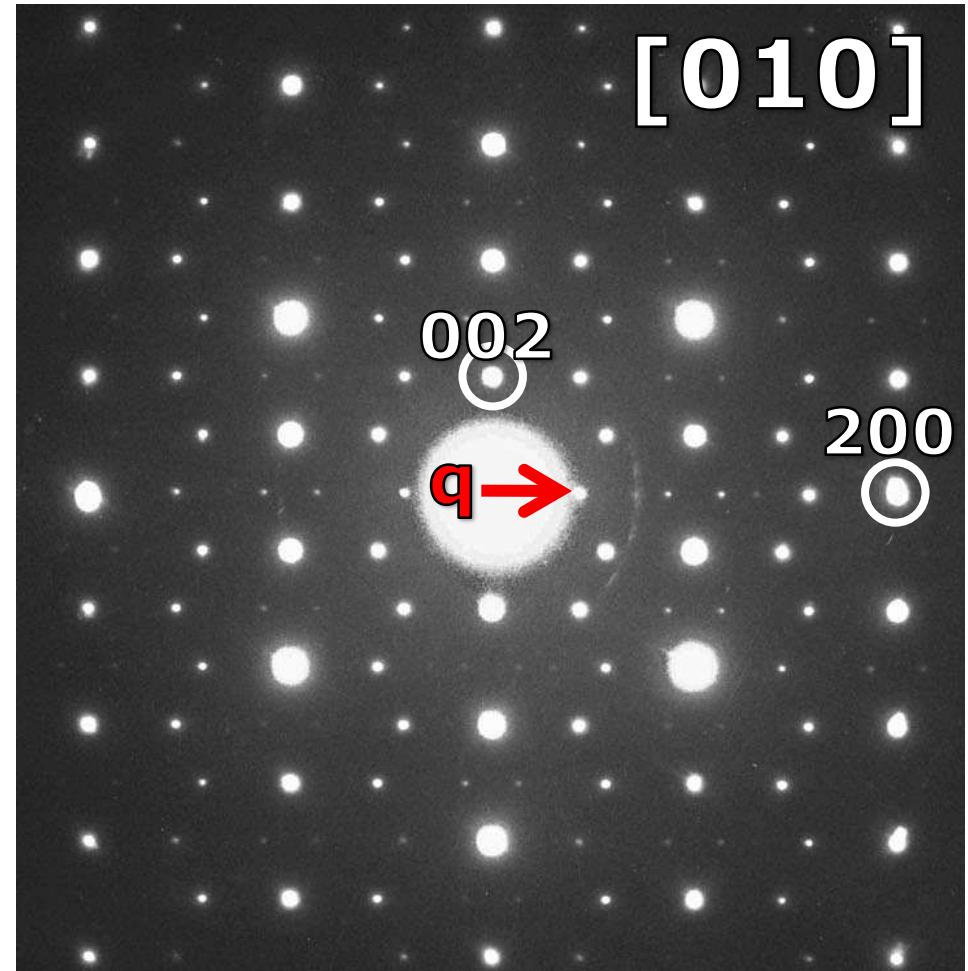
Propose a
modulation vector.



LaSrCuO_{3.52}

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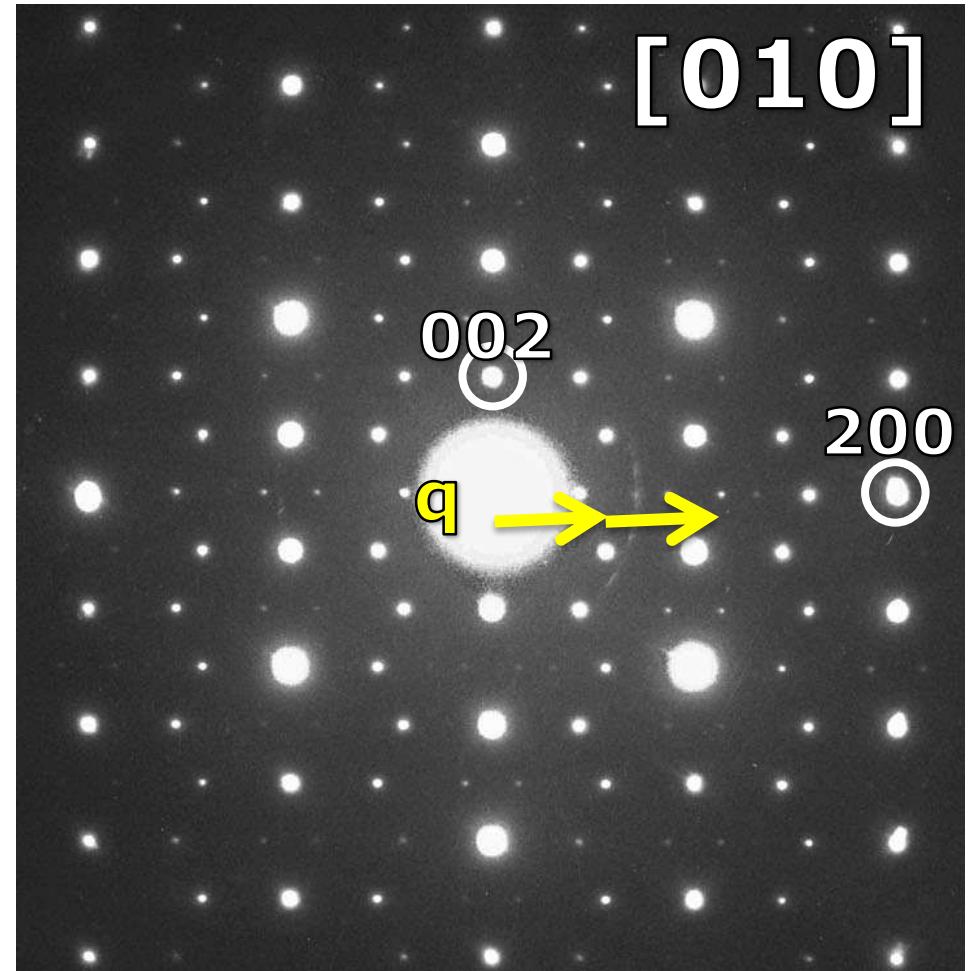
- $q = \alpha a^*$
- $\alpha < 0.5$



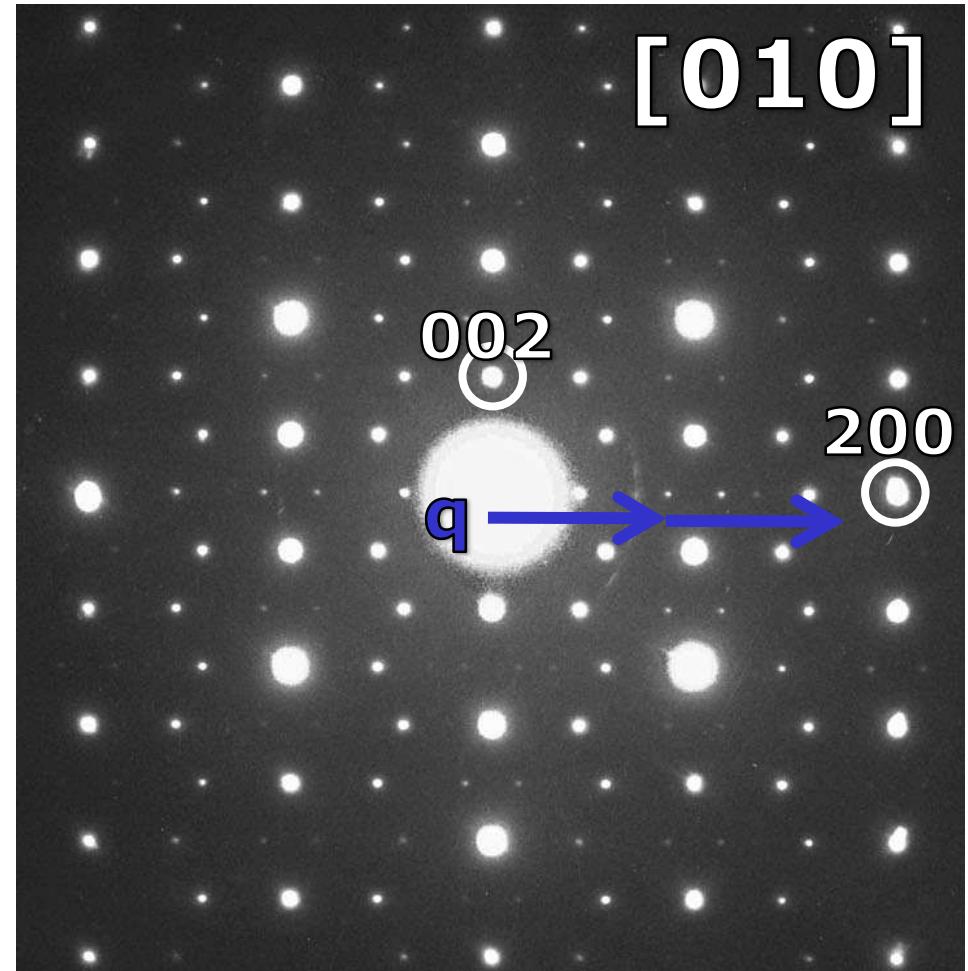
LaSrCuO_{3.52}

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- $q = \alpha a^*$
- $\alpha < 0.5$



- $q = \alpha a^*$
- $\alpha < 0.5$

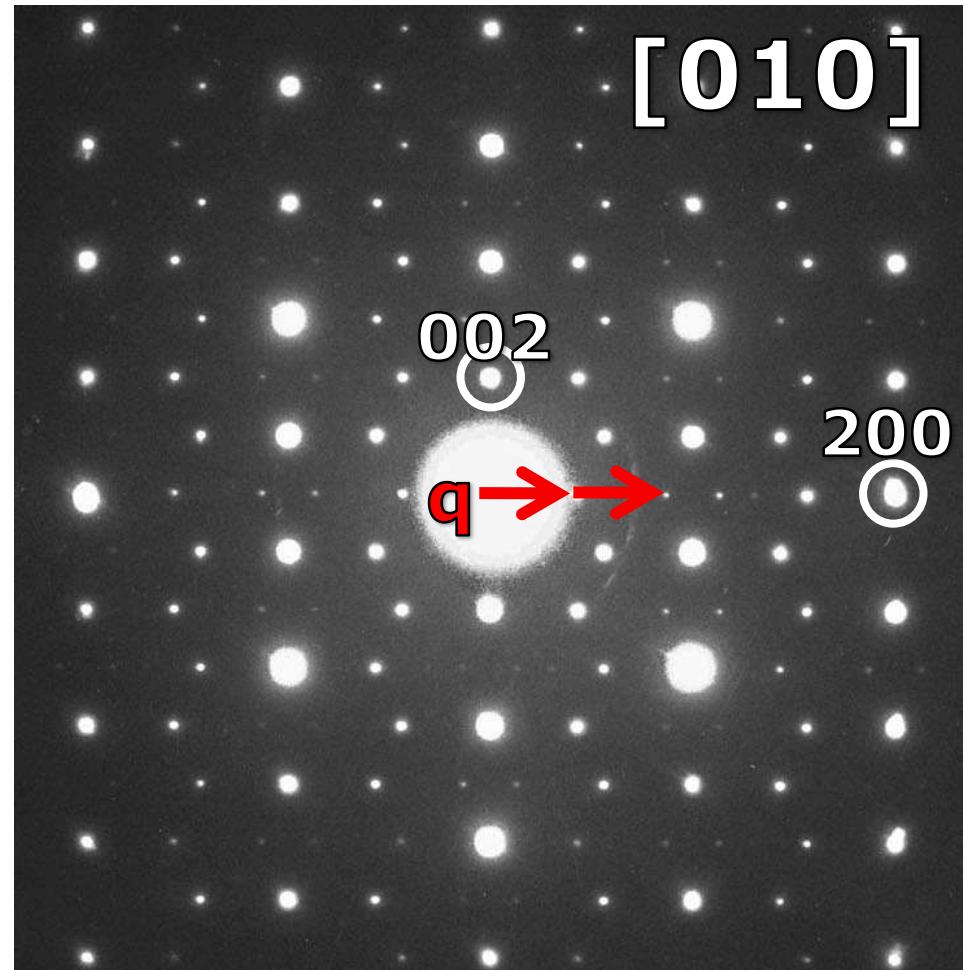


LaSrCuO_{3.52}

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Possible solution

- $q = \alpha a^*$
- $\alpha < 0.5$

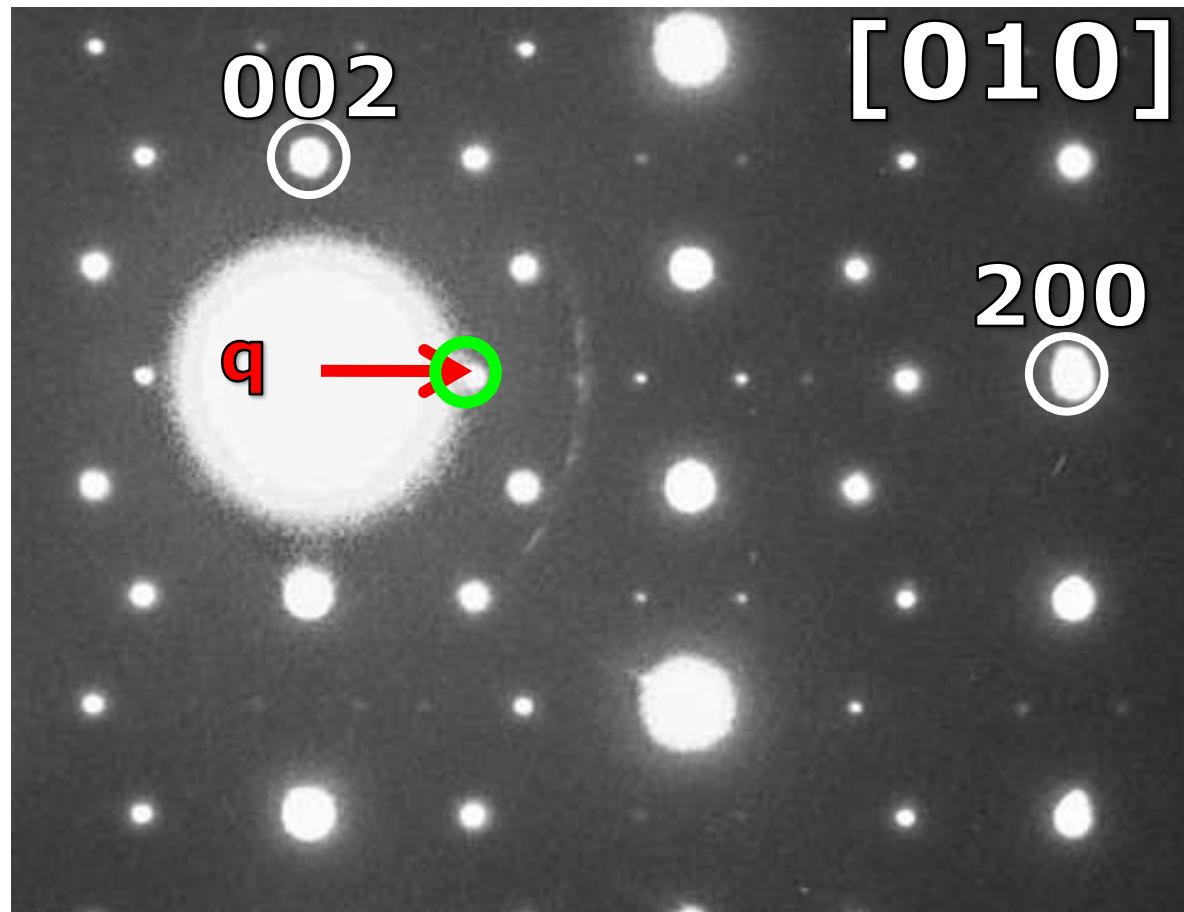


LaSrCuO_{3.52}

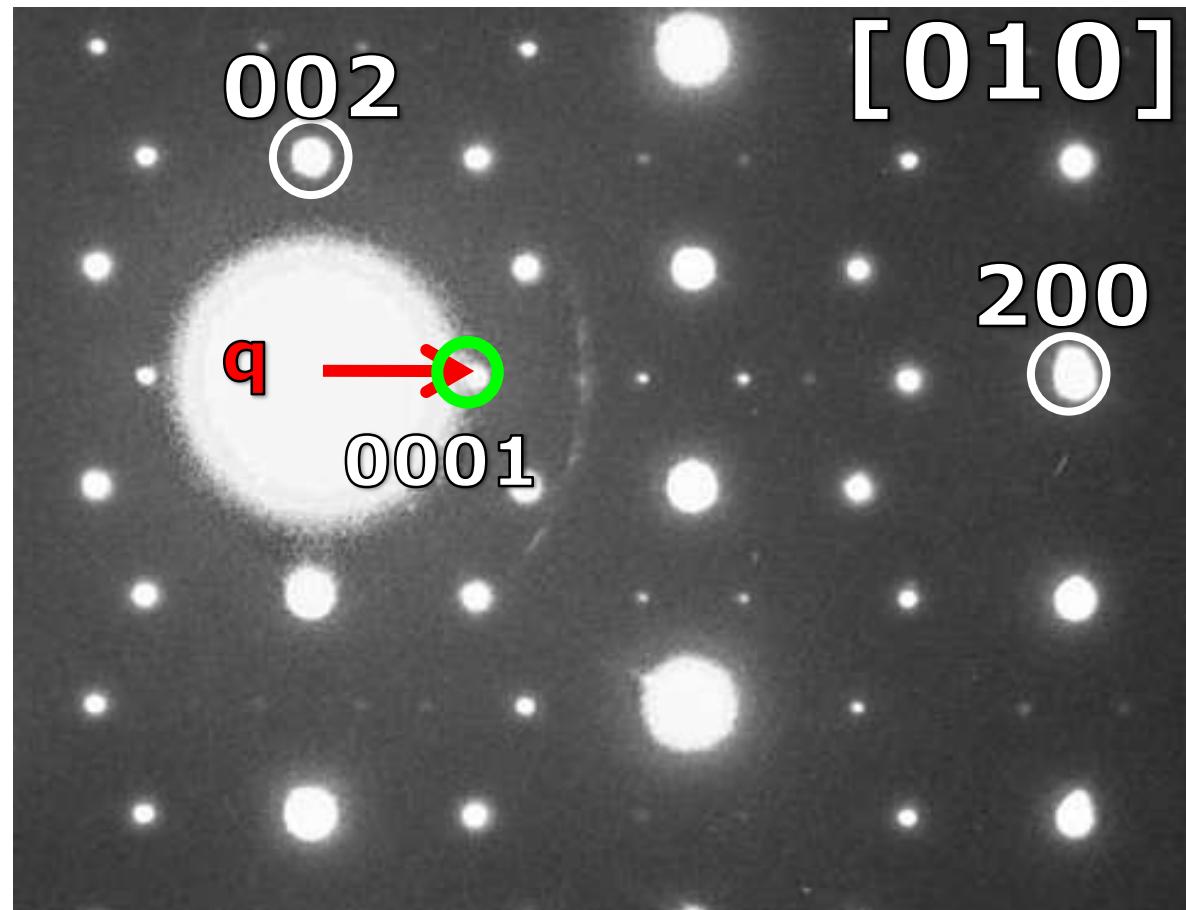
Universiteit Antwerpen

Index the satellite indicated in green

- 001
- 0001
- 100



Index the satellite indicated in green

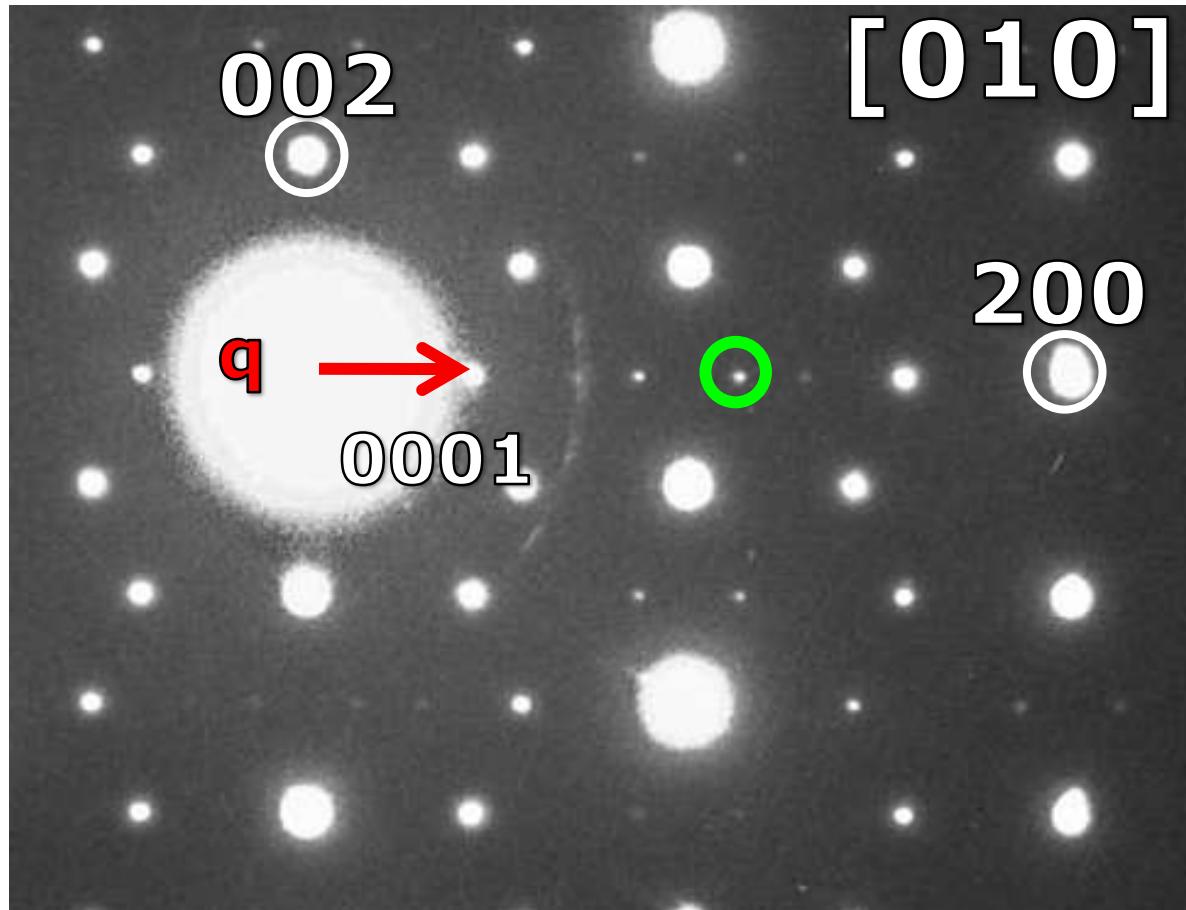


LaSrCuO_{3.52}

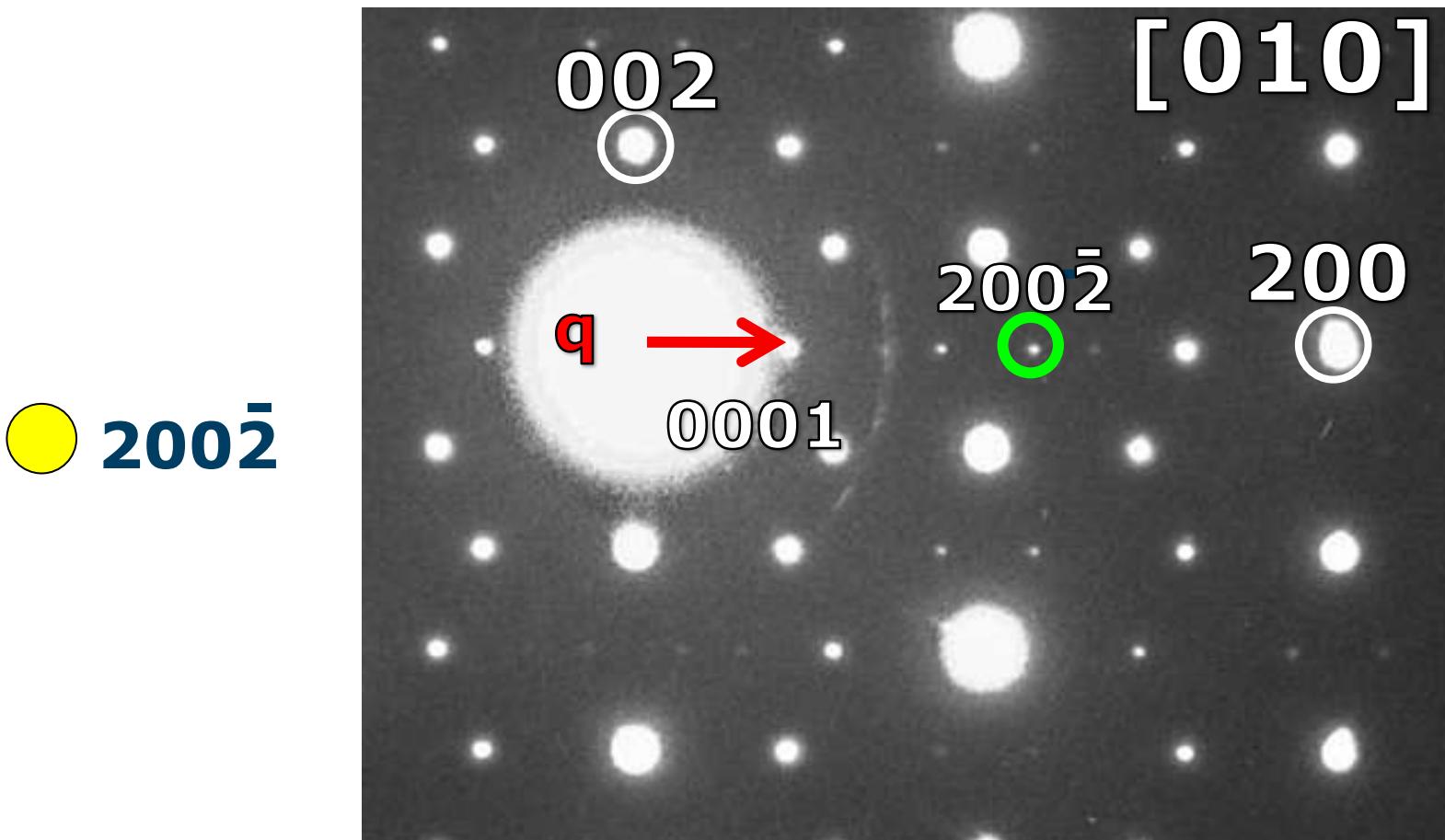
Universiteit Antwerpen

Index the next indicated satellite

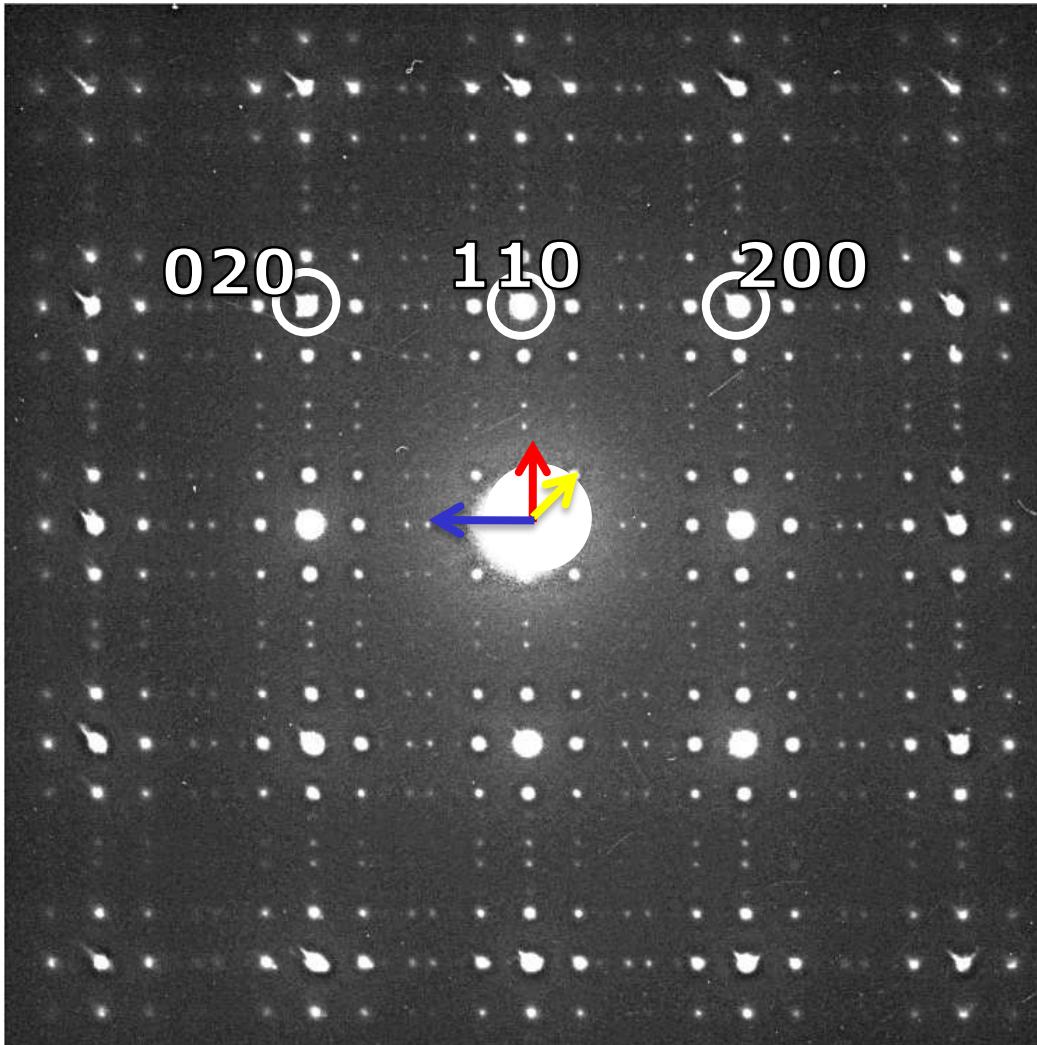
- 0002
- 200 $\bar{1}$
- 200 $\bar{2}$



Index the next indicated satellite

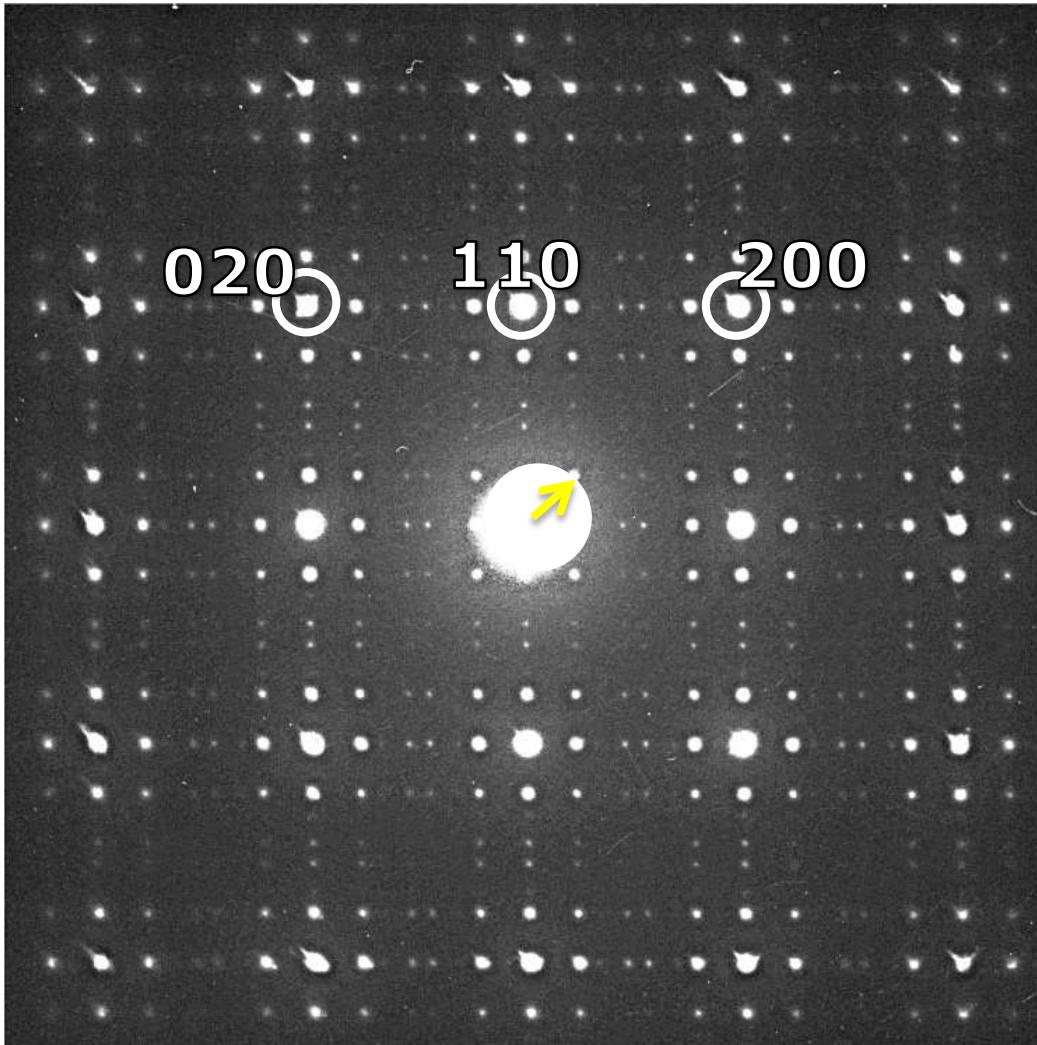


Exercise 1-4: IMS with more complete data

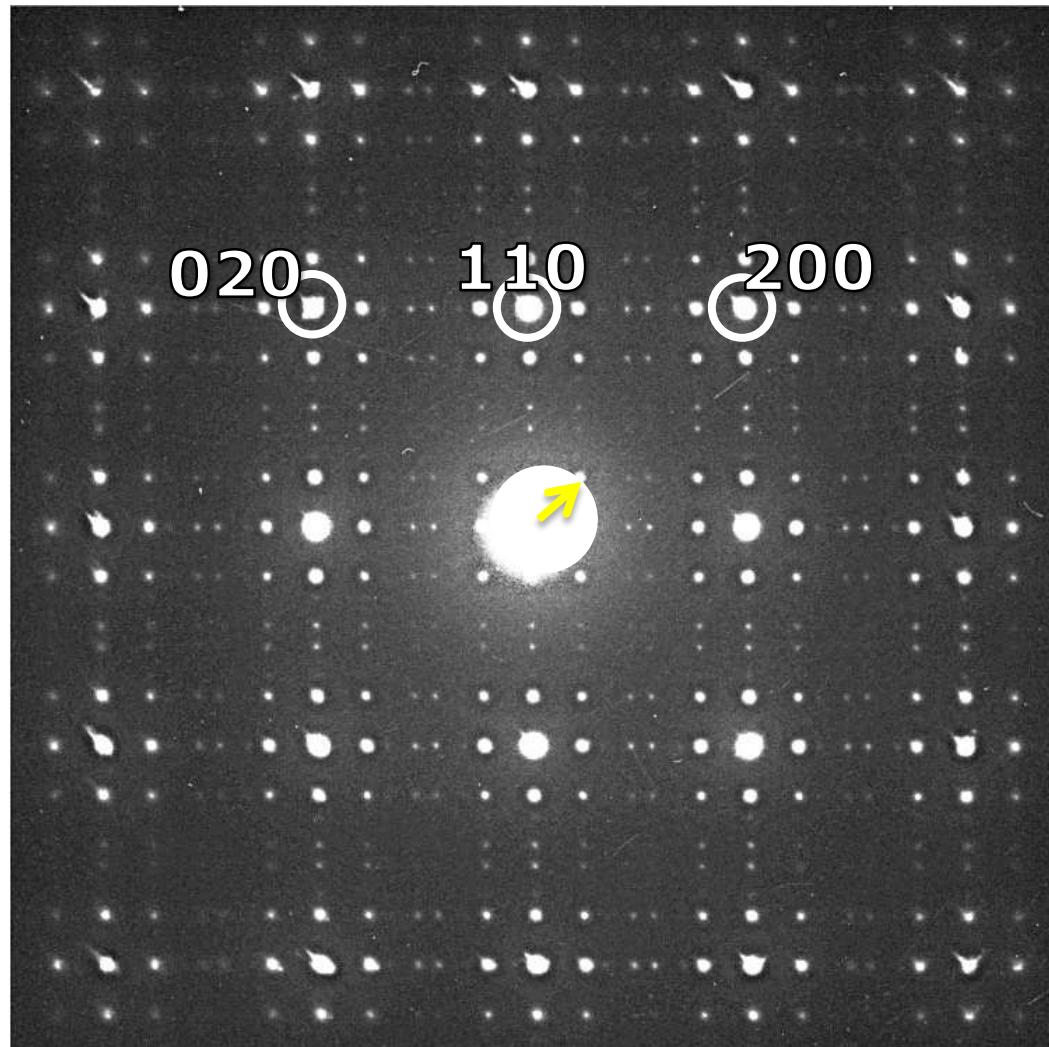


- Which of the indicated vectors corresponds to the modulation vector chosen on the previous slides?

Exercise 1-4: IMS with more complete data



- Which of the indicated vectors corresponds to the modulation vector chosen on the previous slides?



Is the proposed
vector still valid?

- yes
- no

Indicate the correct modulation vector(s) on the central white area.

You need



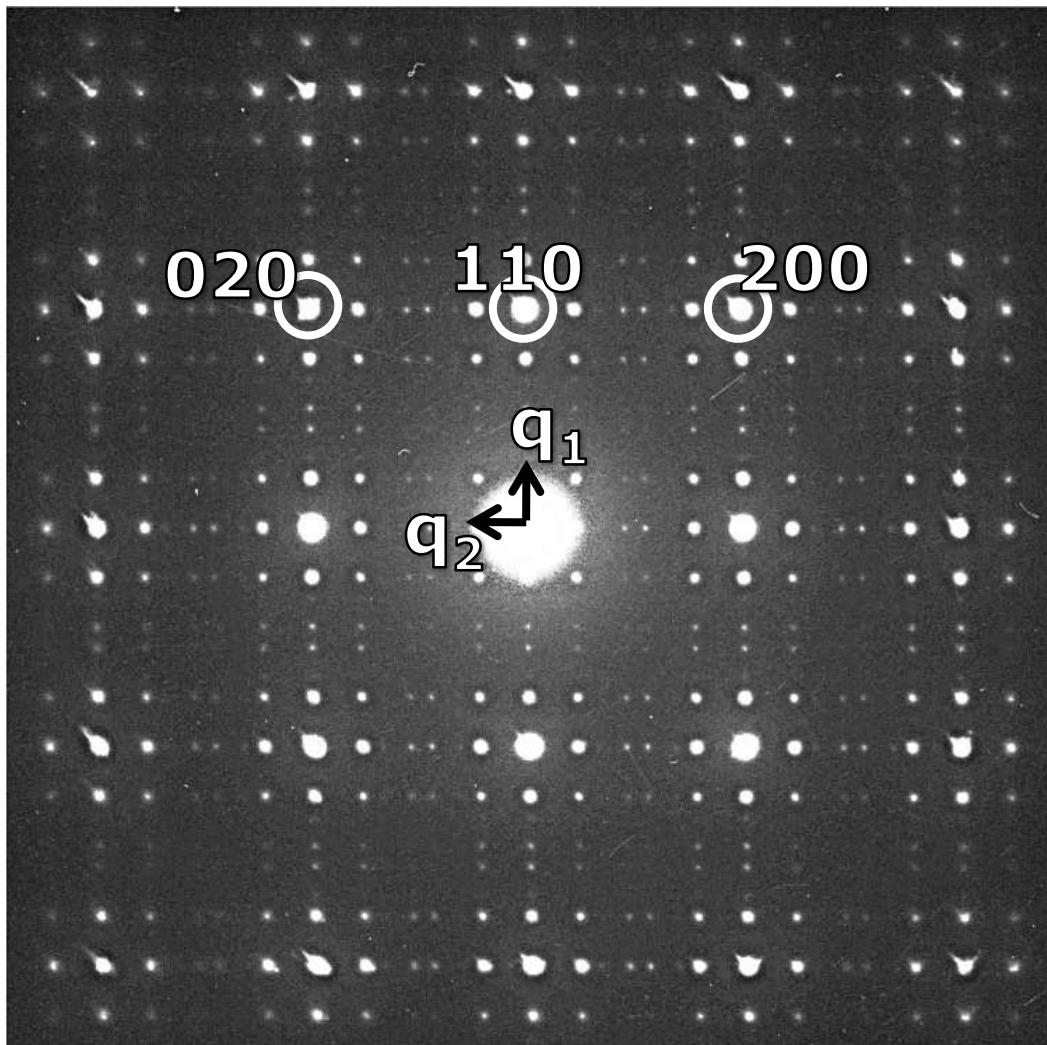
q_1 ànd q_2

$$q_1 = \alpha a^* + \beta b^*$$

$$q_2 = -\alpha a^* + \beta b^*$$

$$\alpha = \beta < 0.25$$

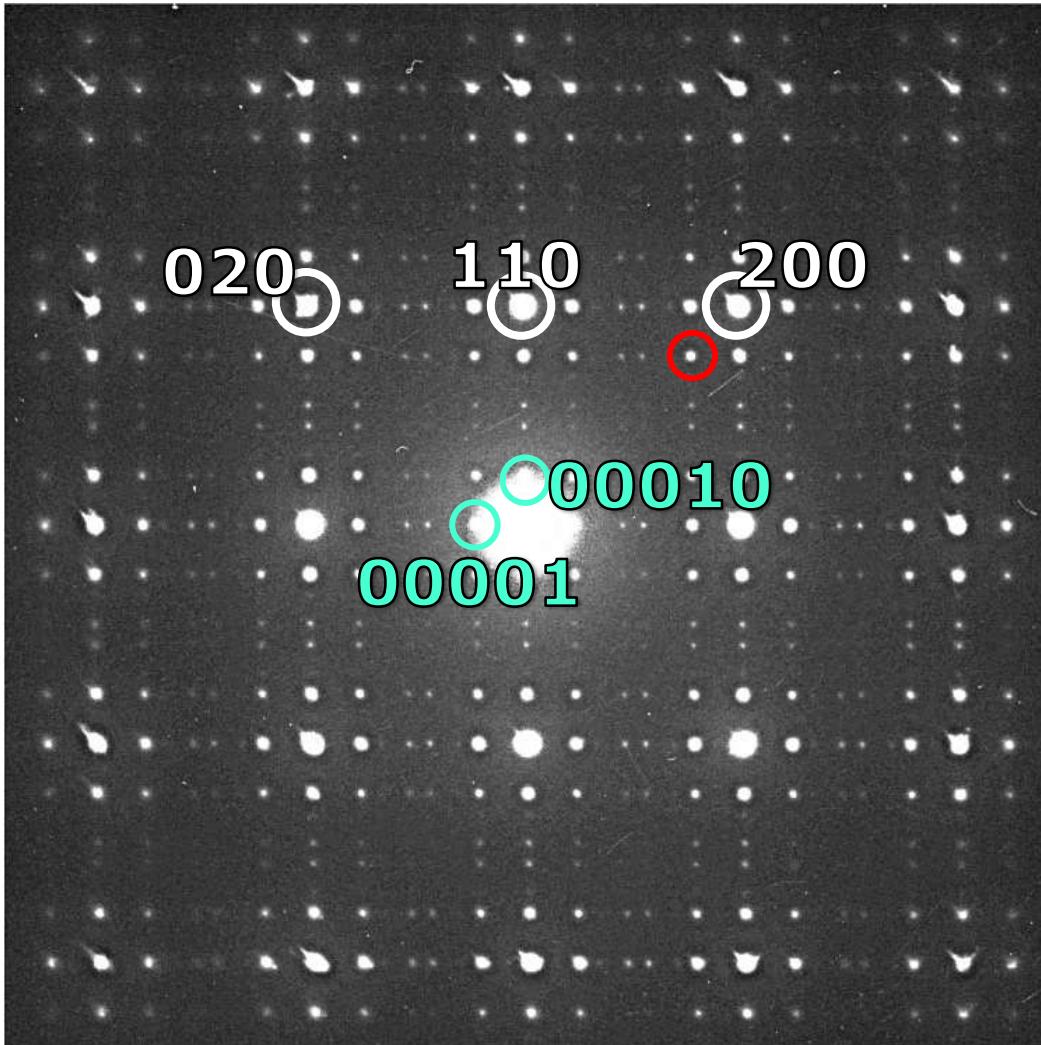
hklmn



LaSrCuO_{3.52}

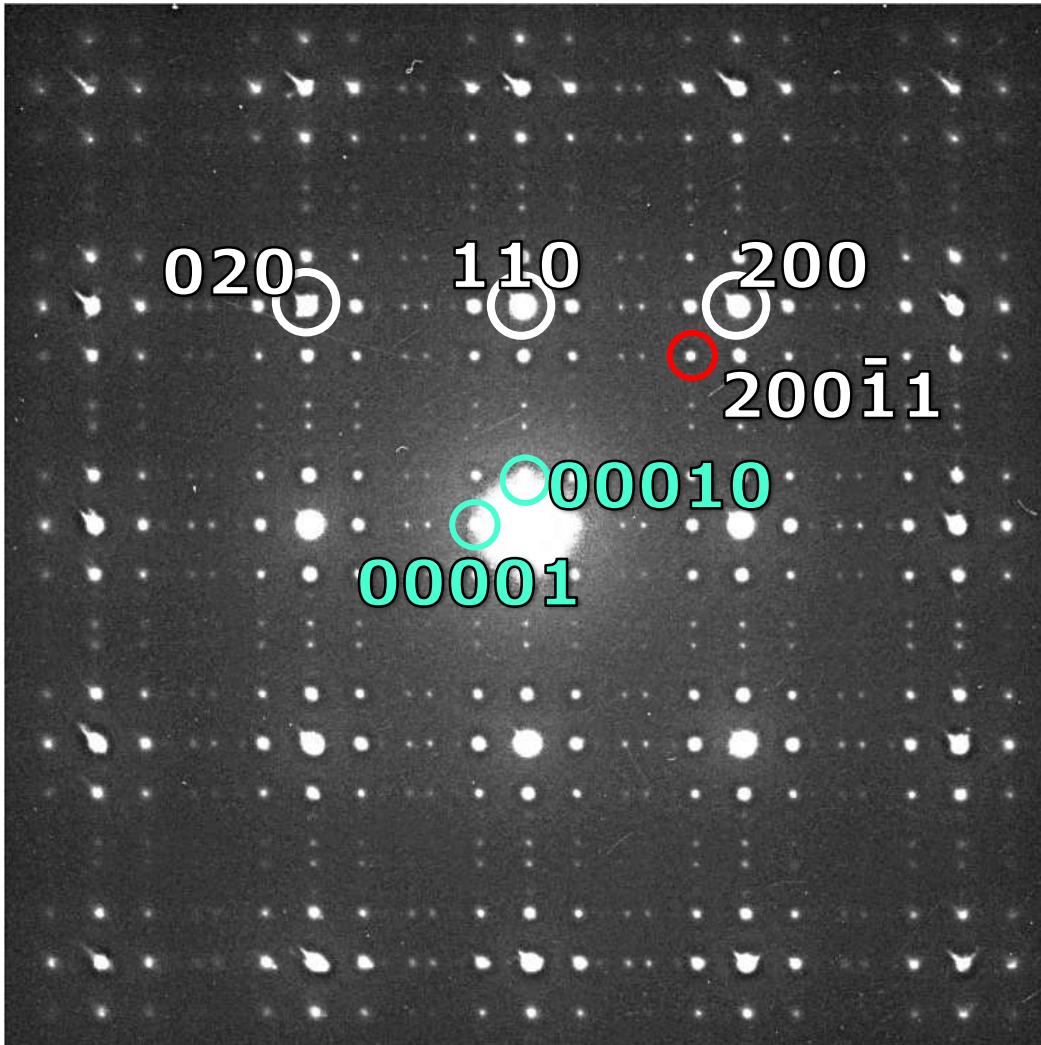
Universiteit Antwerpen

Index the reflection indicated in red



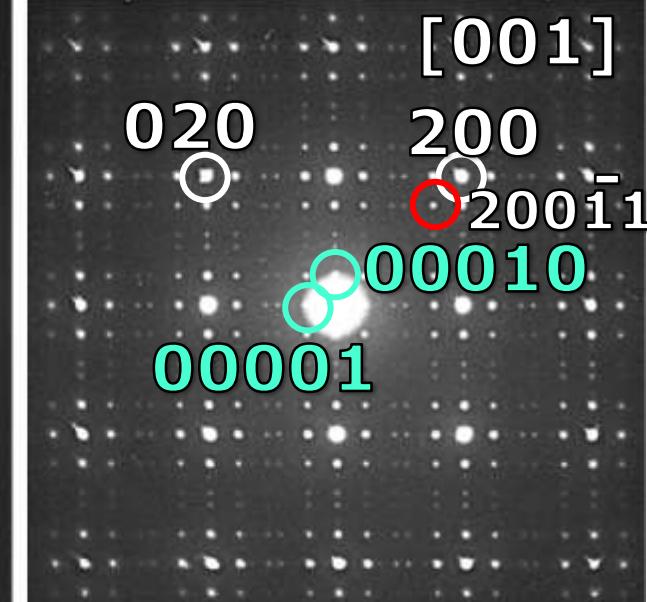
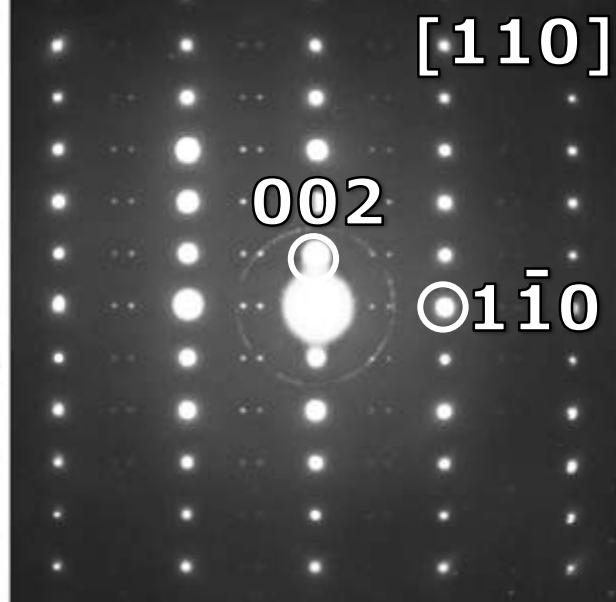
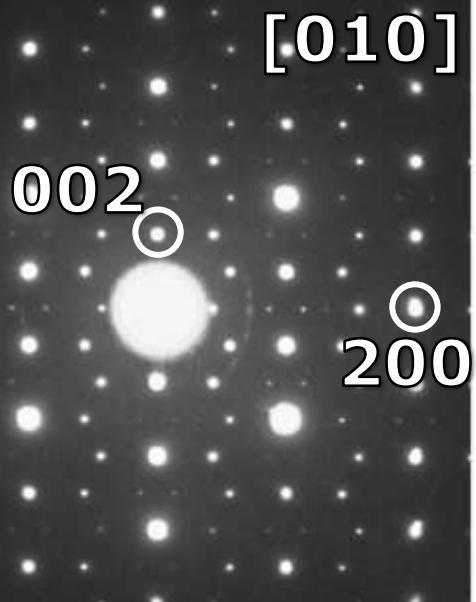
- 20011
- 200 $\bar{1}$ 1
- 2001 $\bar{1}$

Index the reflection indicated in red

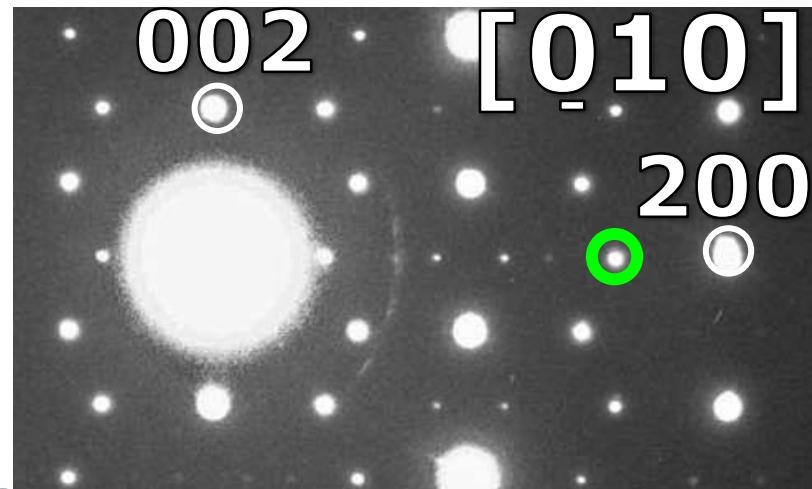
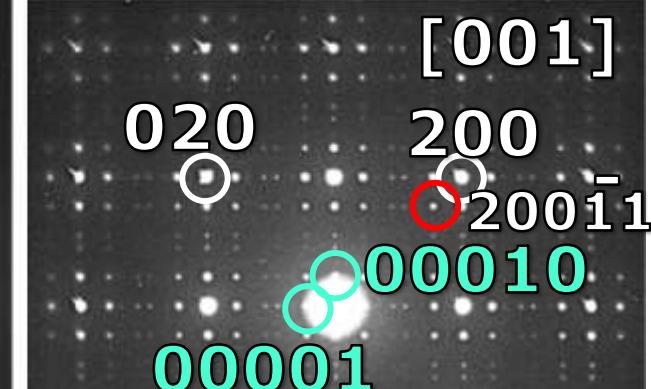
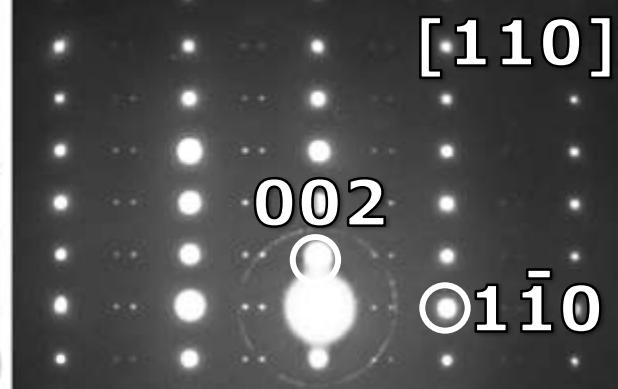
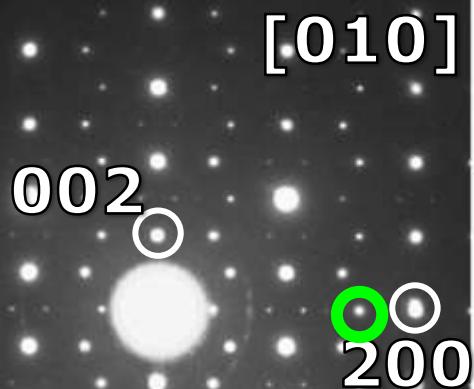


200 $\bar{1}$ 1

Index the others patterns consistently with this new choice.

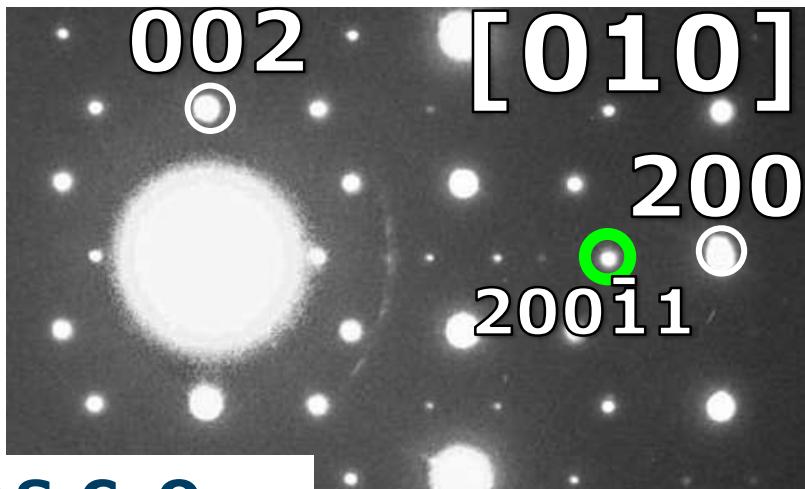
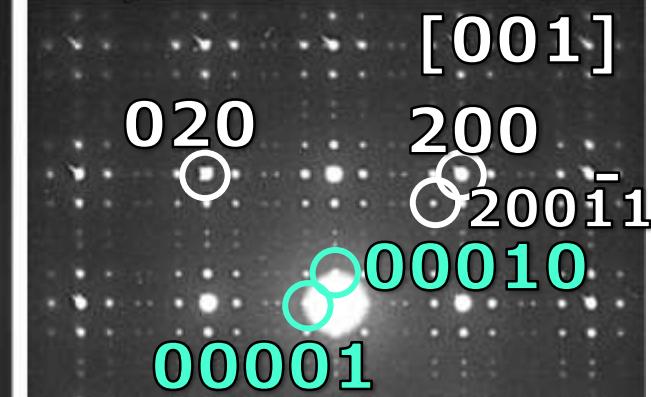
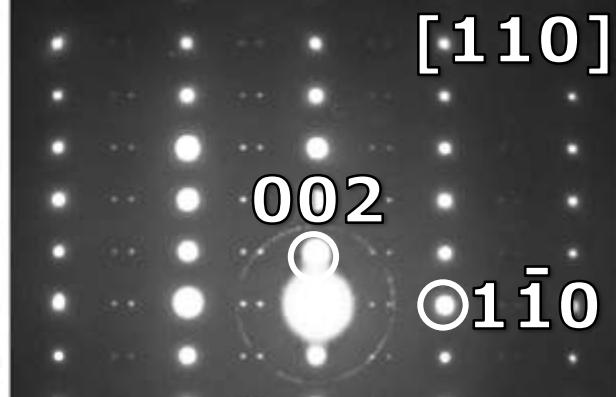
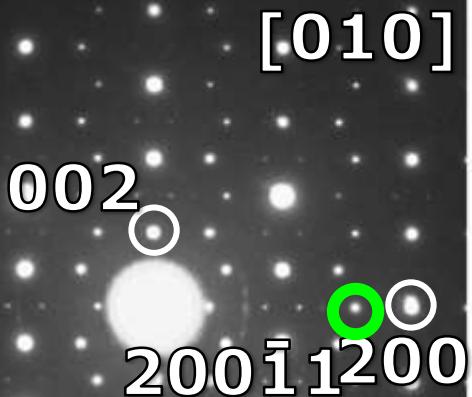


Index the reflection indicated in green



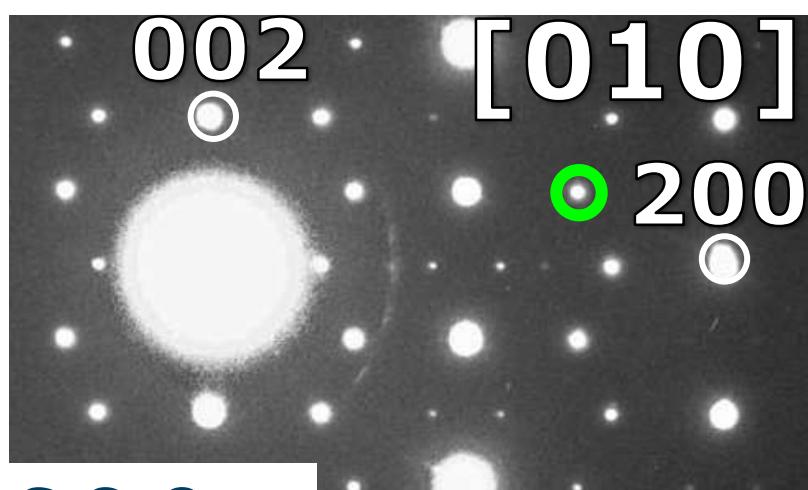
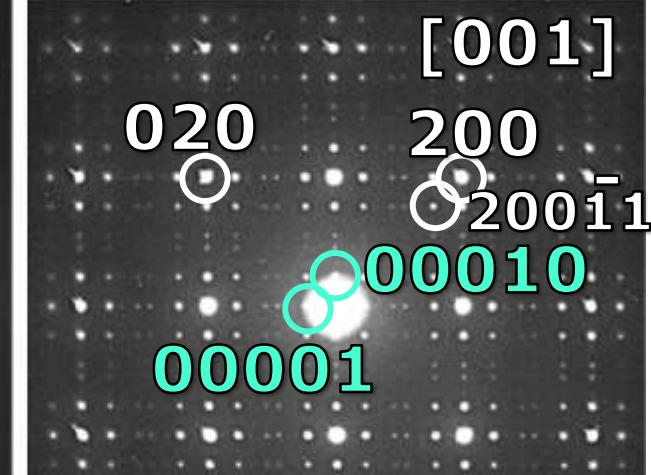
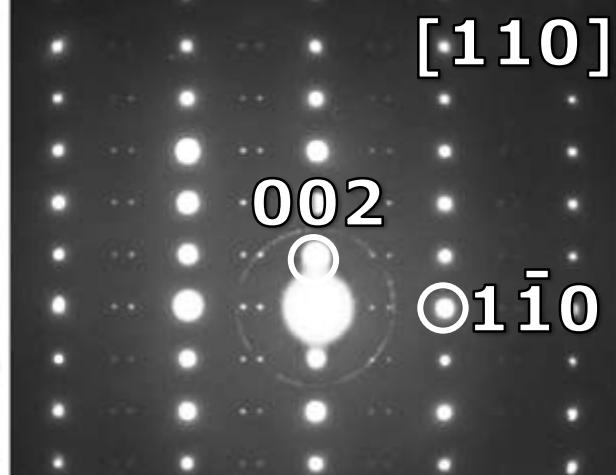
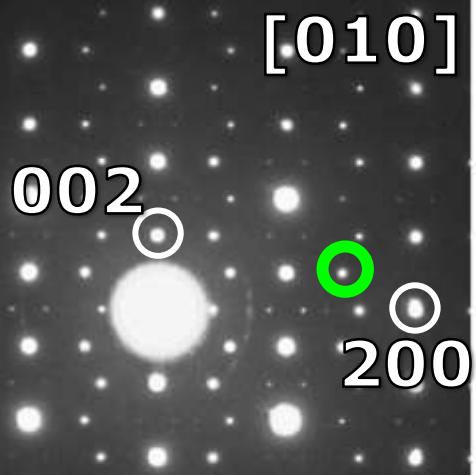
- 20011
- 200̄11
- 2001̄1

Index the reflection indicated in green



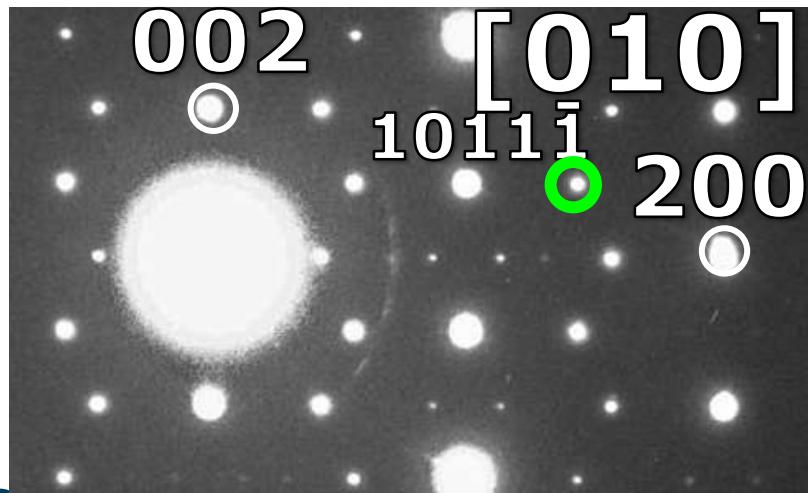
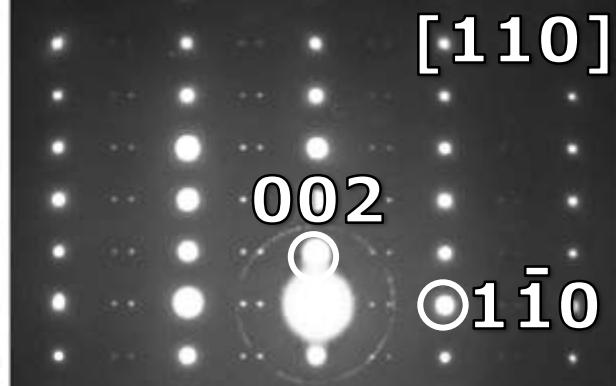
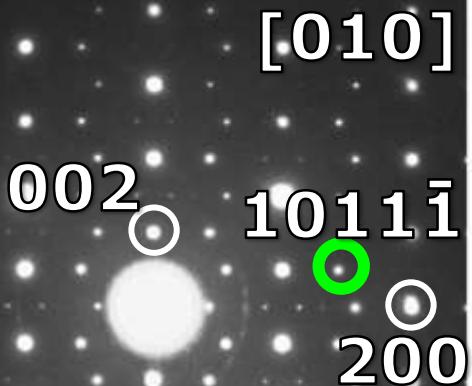
200̄11

Index the reflection indicated in green



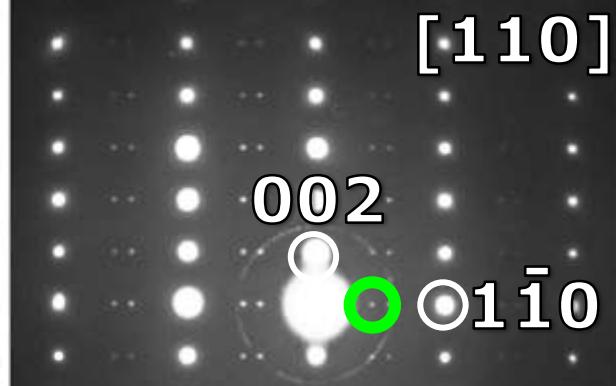
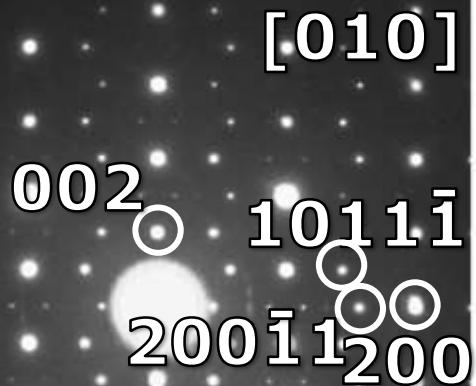
● 10110
● 10111
● 1011 $\bar{1}$

Index the reflection indicated in green



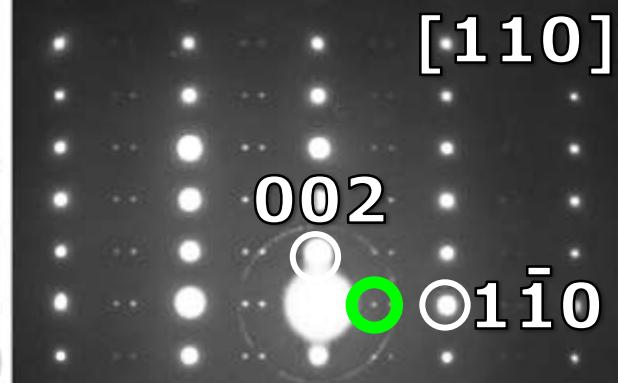
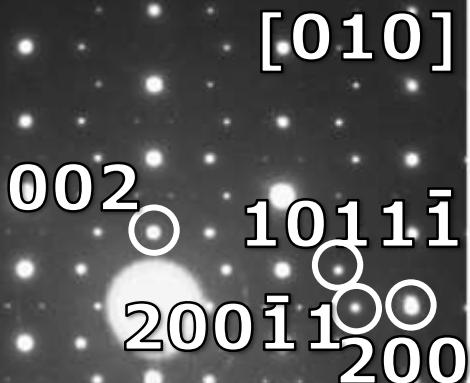
1011̄̄̄

Index the reflection indicated in green



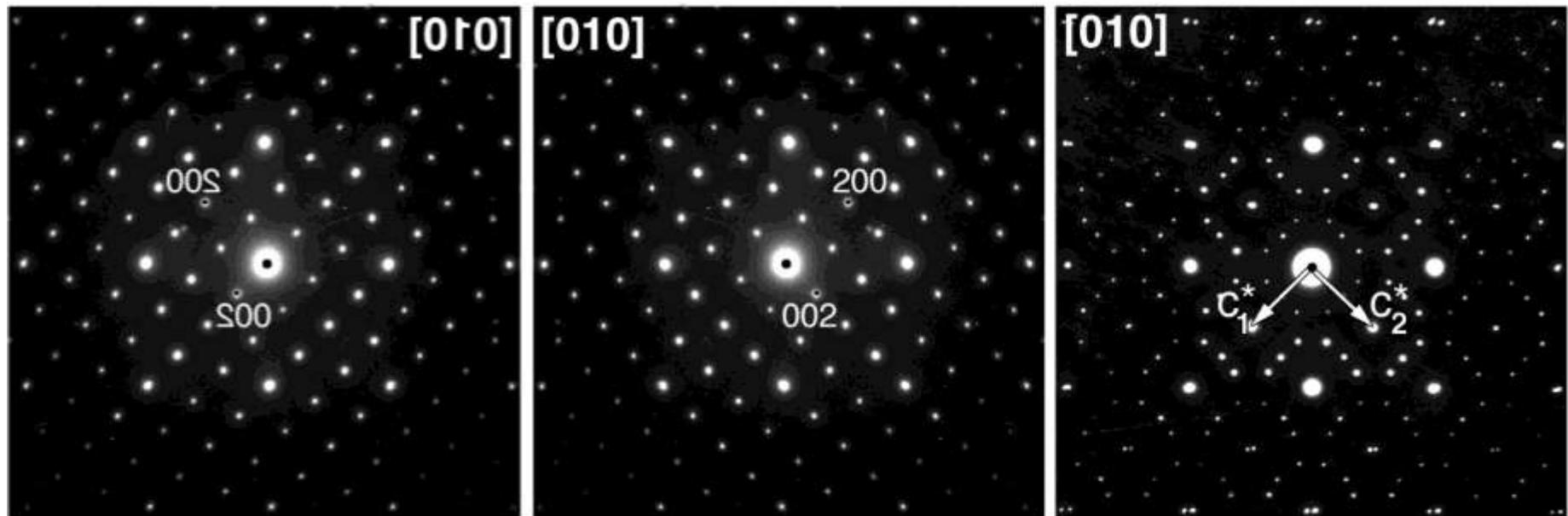
- 00002
- 0000̄2
- 0000̄1

Index the reflection indicated in green



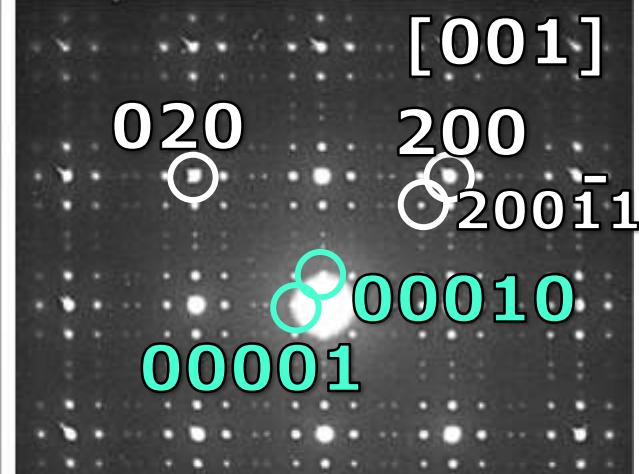
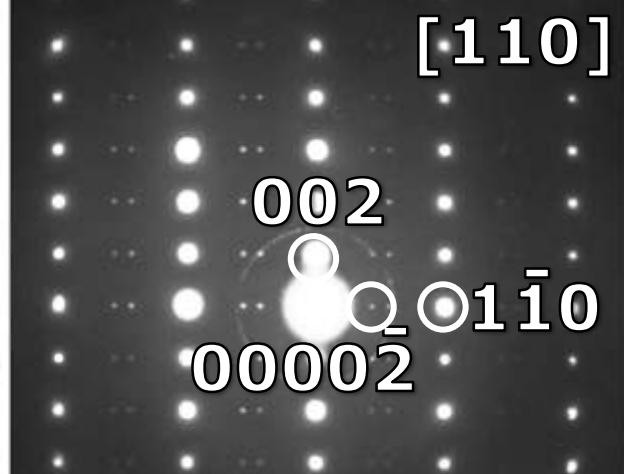
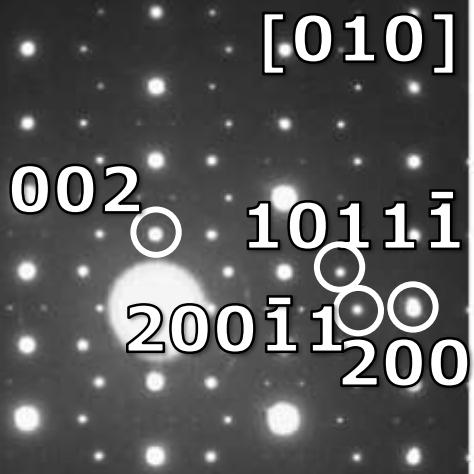
0000̄2

Look out for twinning!



Gillie et al., Journal of Physics and Chemistry of Solids, 65, 1 (2004) 87-93

Exercise 1-5: twinning?

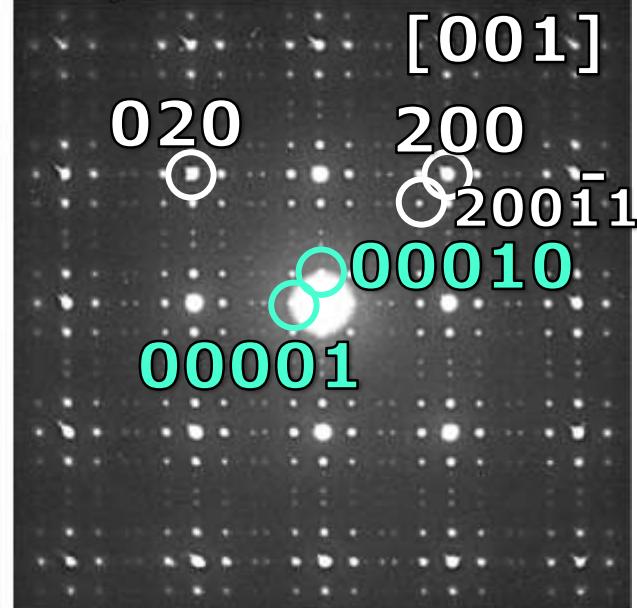
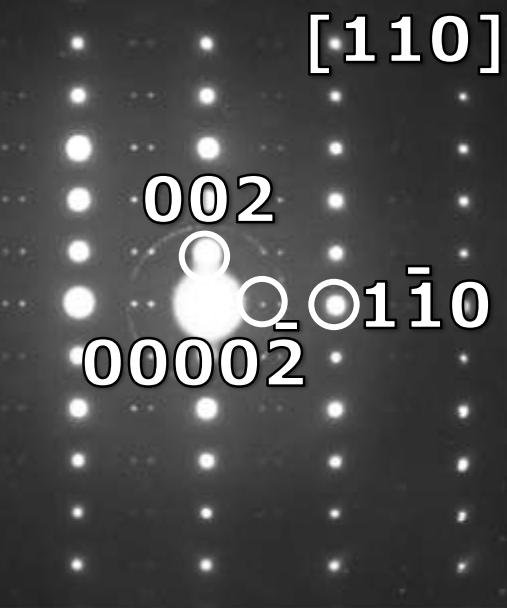
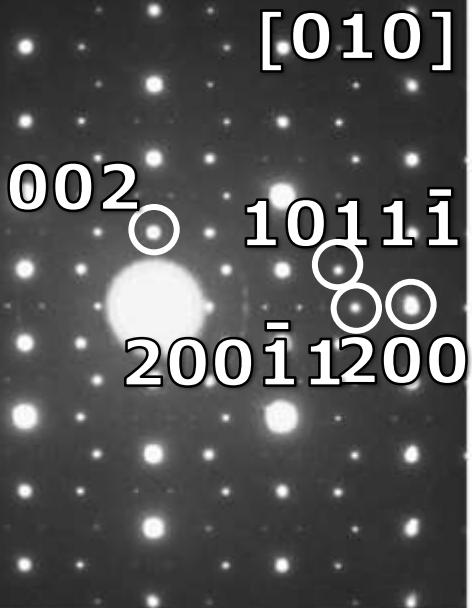


Could it be twinning and thus sufficient to use only one q-vector?

 yes

 no

Exercise 1-6: derive the reflection conditions

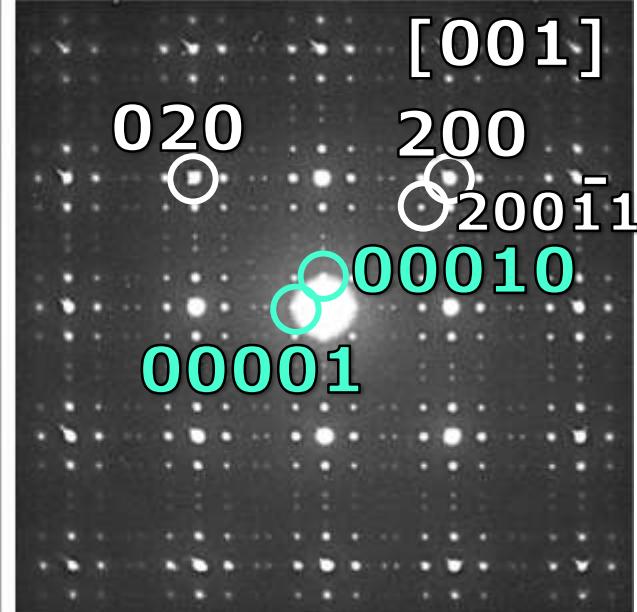
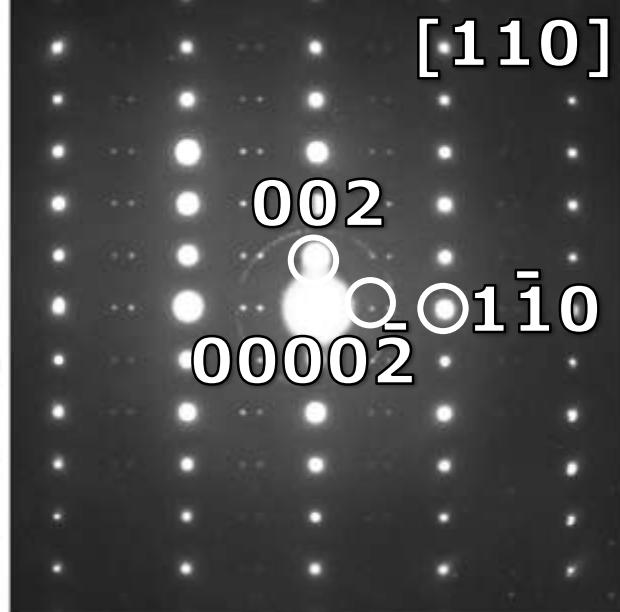
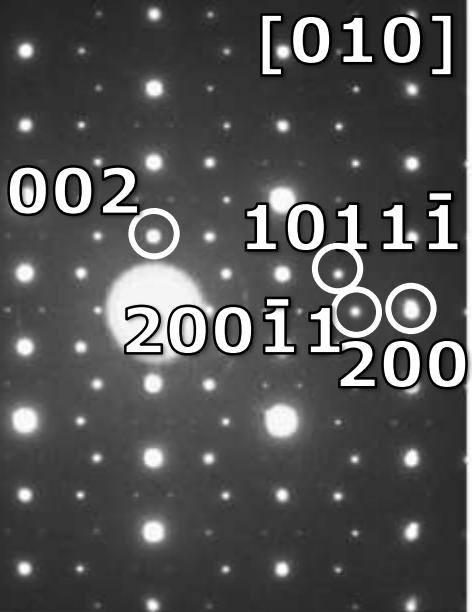


$$\left. \begin{array}{l} hkl : h+k+l=2n \\ hhl : l=2n \end{array} \right\}$$

hklmn:

- $h+k+l+m+n=2i$
- $h+k+l=2i$
- $m+n=2i$

Exercise 1-6: derive the reflection conditions

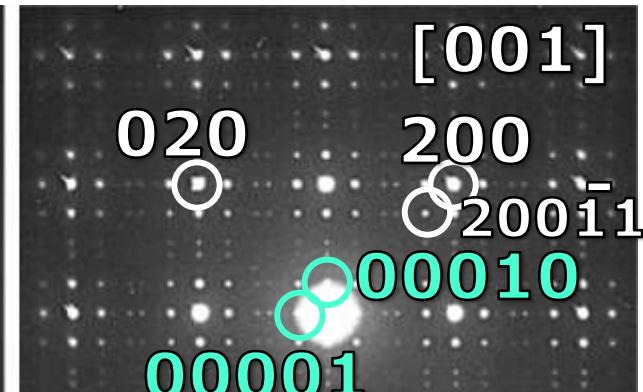
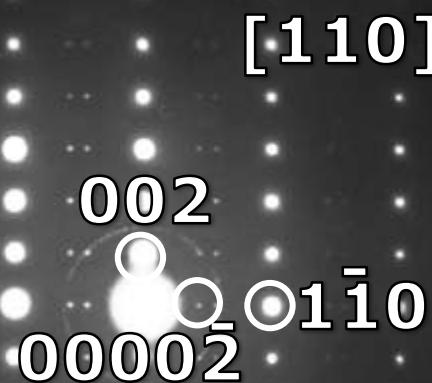
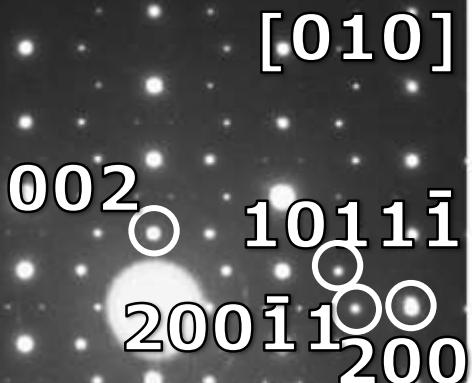


$$\left. \begin{array}{l} hkl : h+k+l=2n \\ hhl : l=2n \end{array} \right\}$$

hklmn:

h+k+l=2i

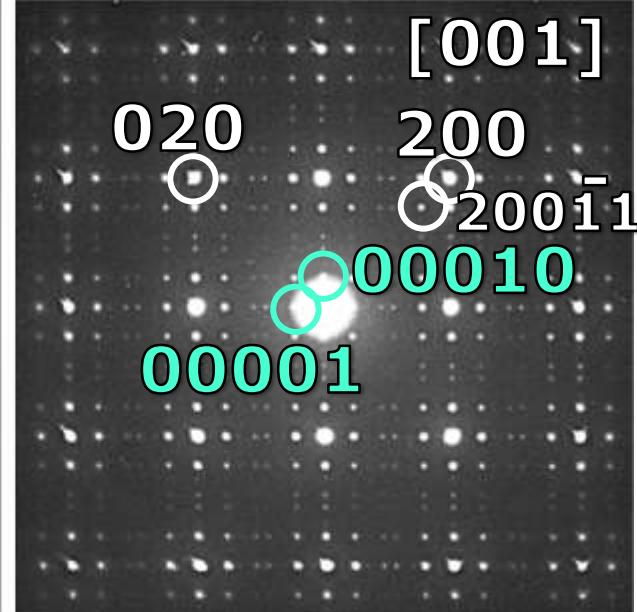
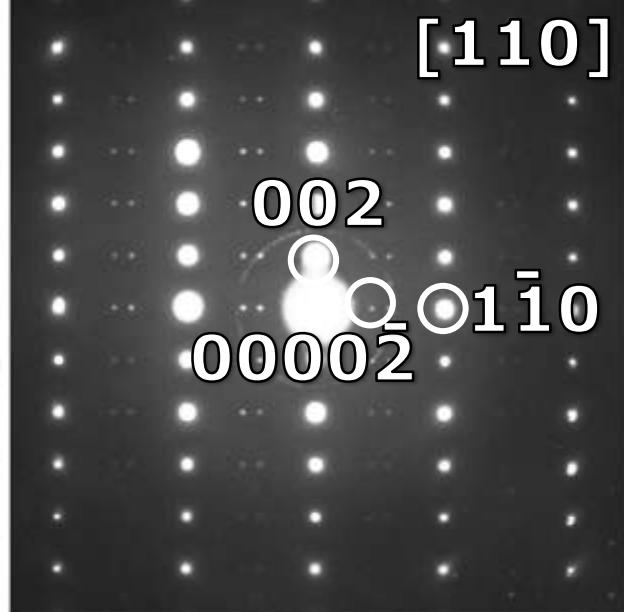
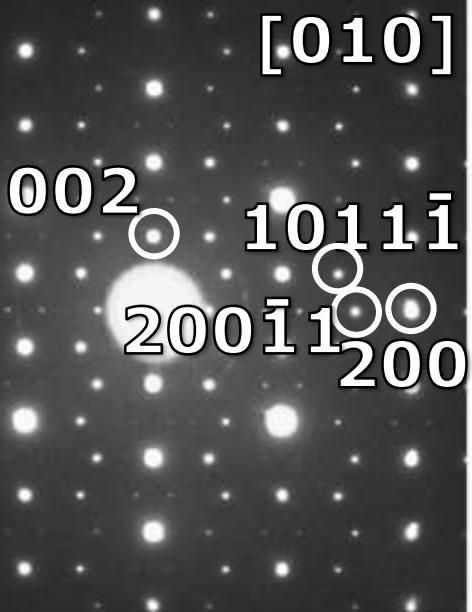
Exercise 1-6: derive the reflection conditions



$$\left. \begin{array}{l} \text{hkl: } h+k+l=2n \\ \text{hh̄l: } l=2n \\ \text{hhlm̄0: } \end{array} \right\}$$

- $l=2i$
- $h=2i$
- $l,m=2i$

Exercise 1-6: derive the reflection conditions

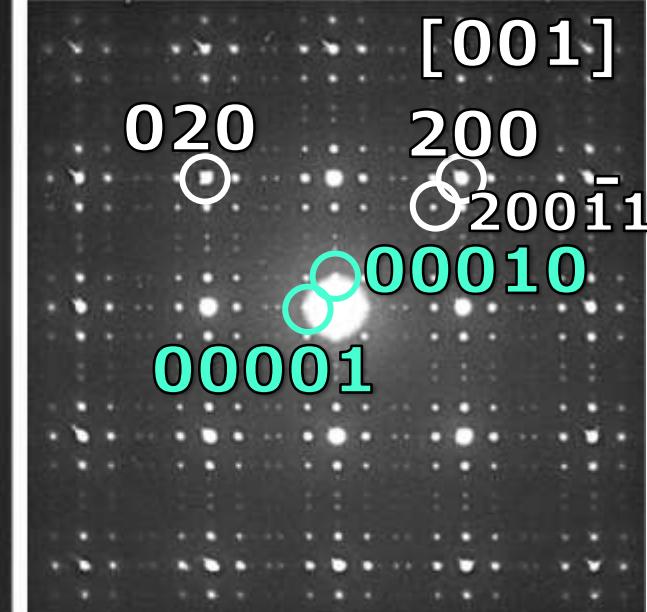
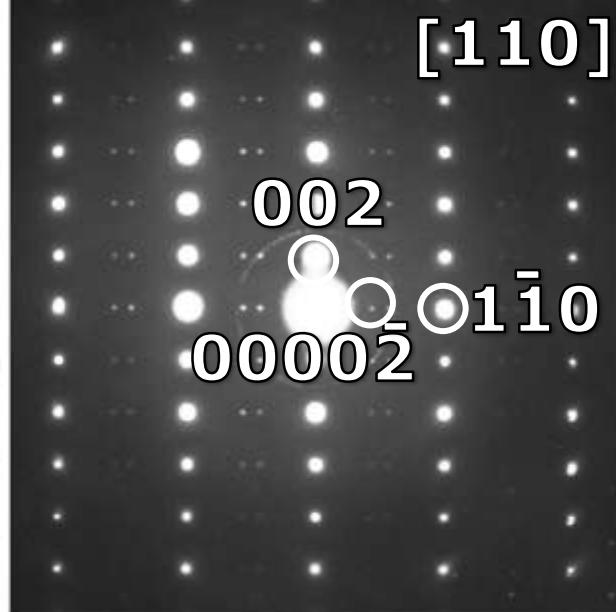
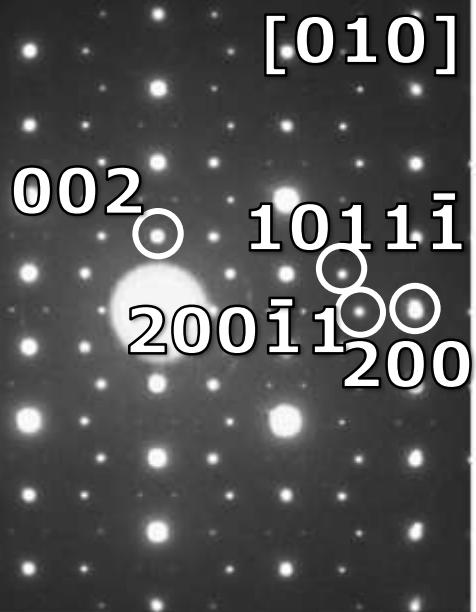


$$\left. \begin{array}{l} hkl : h+k+l = 2n \\ hhl : l = 2n \end{array} \right\}$$

hlm0:

but
 $hhlm0 : m = 2i$
is sufficient

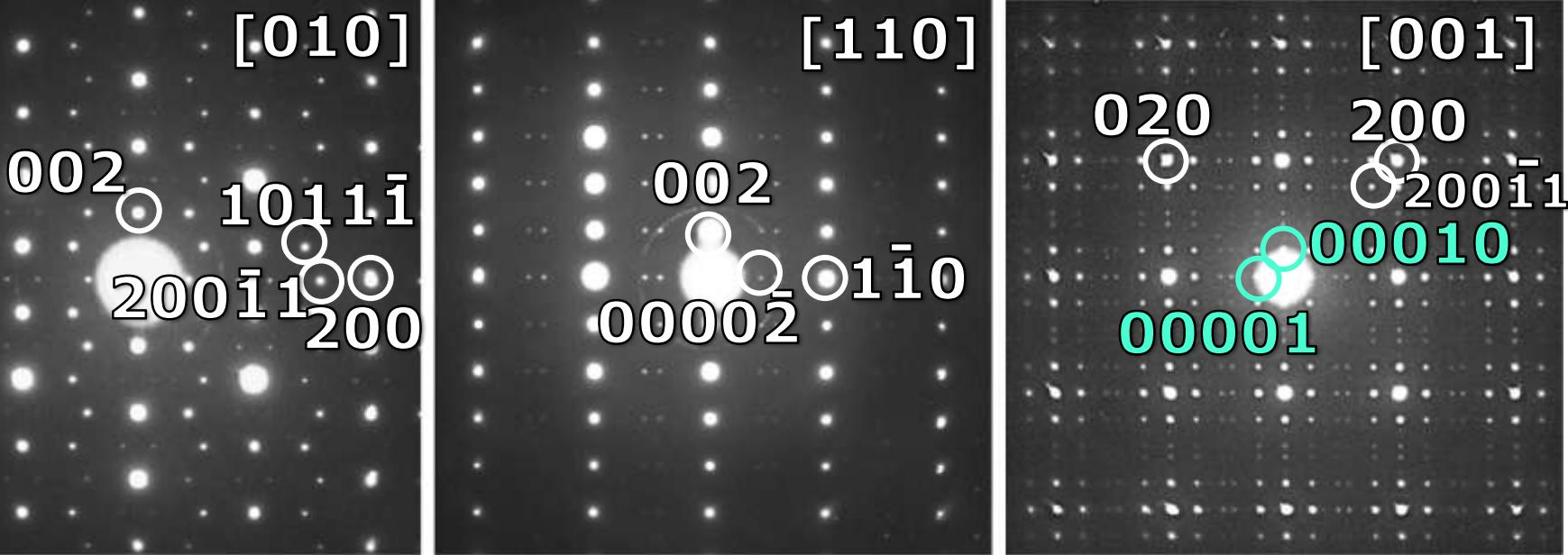
Exercise 1-6: derive the reflection conditions



$$\left. \begin{array}{l} \text{hkl: } h+k+l=2n \\ \text{hh̄l: } l=2n \end{array} \right\}$$

- $\text{hklmn: } h+k+l=2i$
- $\text{hh̄lm0: } m=2i$
($-\text{hh̄l}0n: n=2i$)

Determining the superspace group



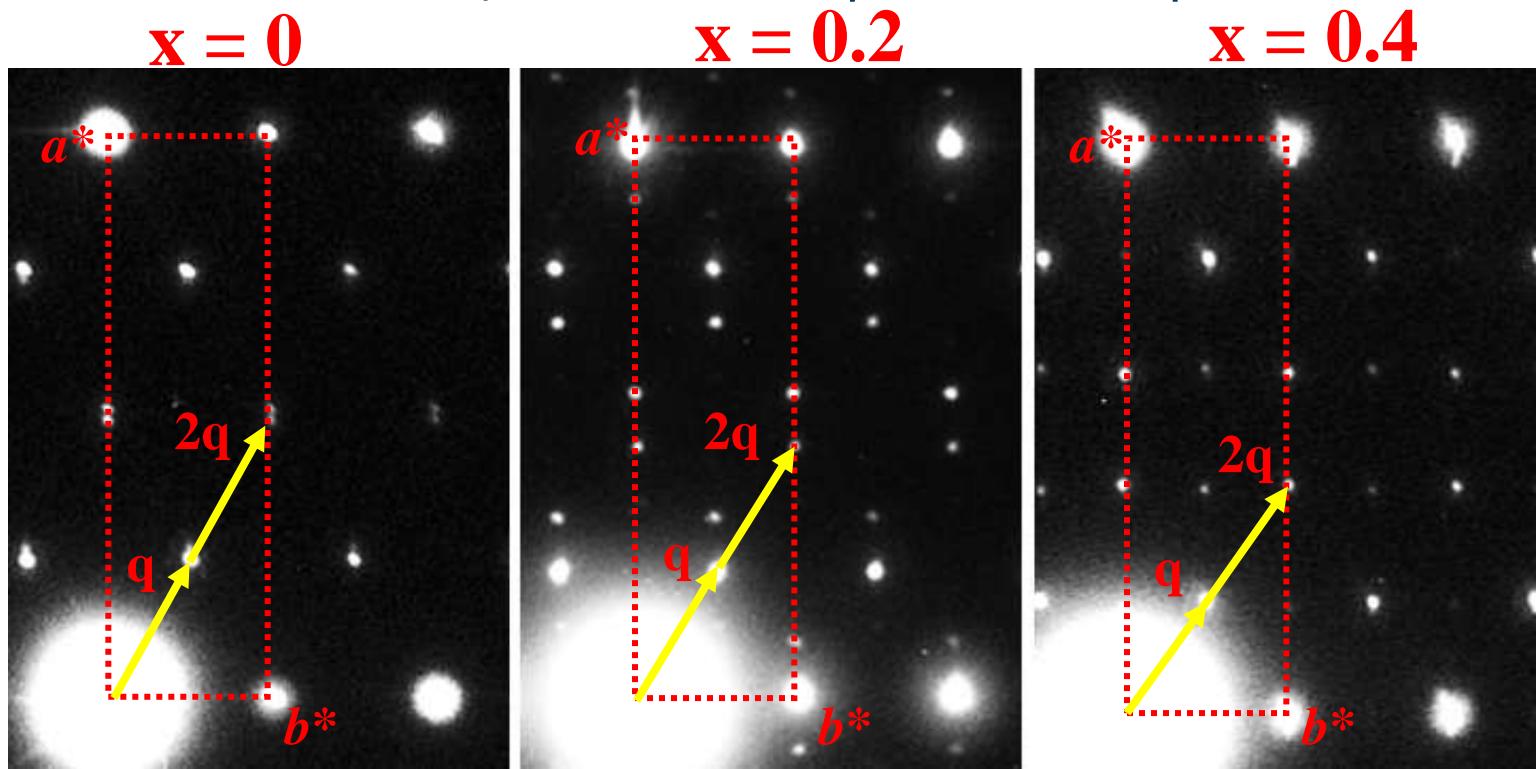
- Online tables Yamamoto (Acta Cryst.A 1996, A52, 509)
- I4/mmm(α α 0, - α α 0)00mg

Use of doing ED

- Deriving approximate cell parameters, modulation vector and superspace group
- Not useful for: precise parameters, precise modulation vector
- Excellent if multiphase sample or satellites very weak in XRD/ND
- Be aware of:
 - Twinning
 - Multiple diffraction (for forbidden refls., for using intensities,...)
 - The possibility of missing important sections of reciprocal space

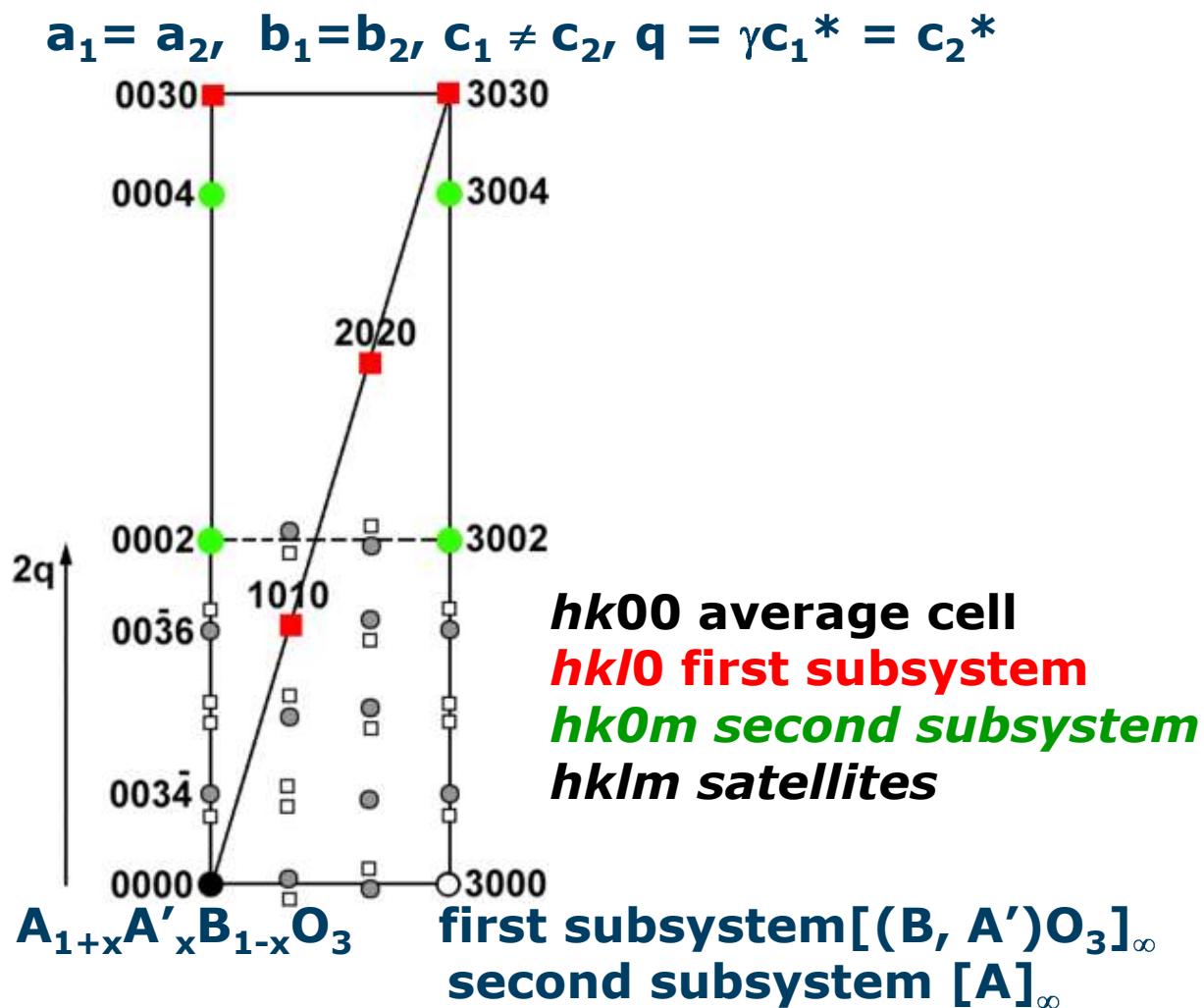
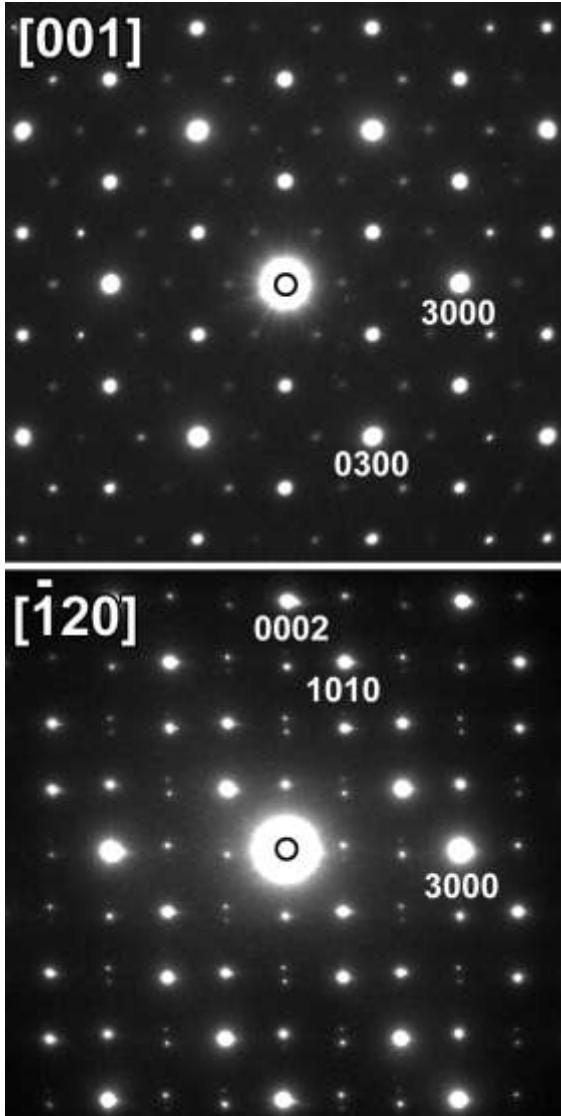
Relation composition – modulation vector

- Most TEM have EDX available for composition determination
- Determination of direct links between composition and modulation vector, also in multiphased samples



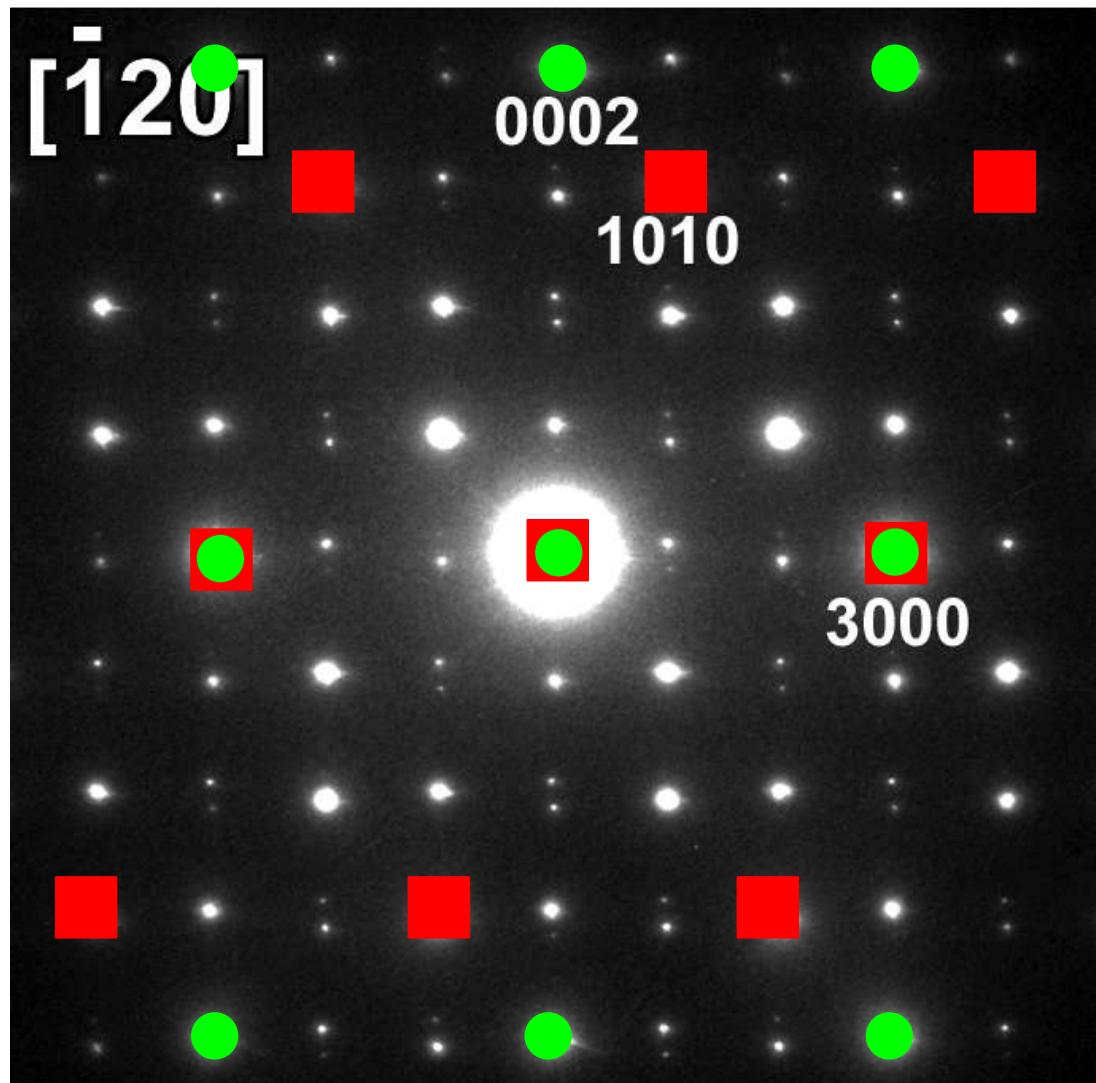
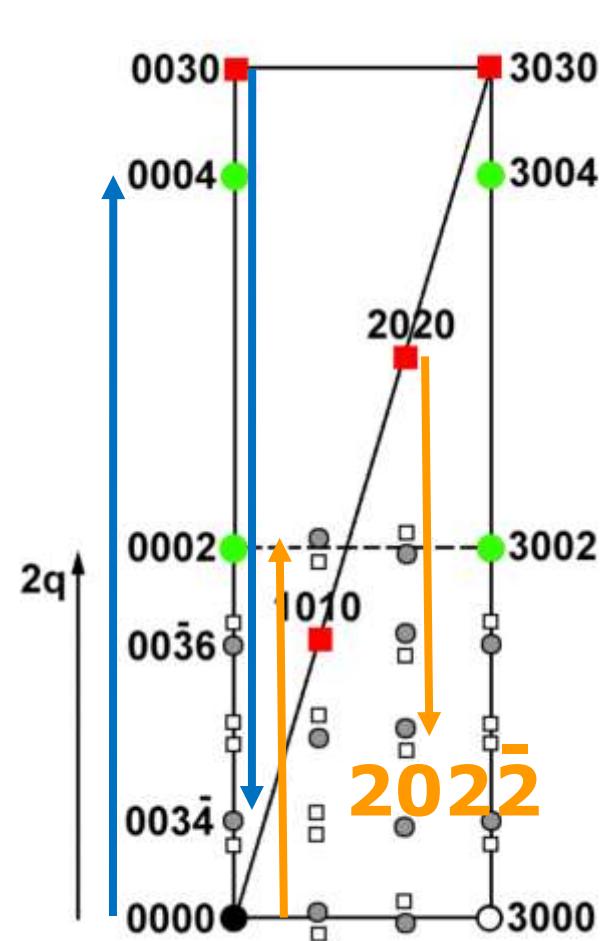
(Abakumov et al., Chem. Mater. 20
(13), pp. 4457-4467)

ED of a composite structure (CS)

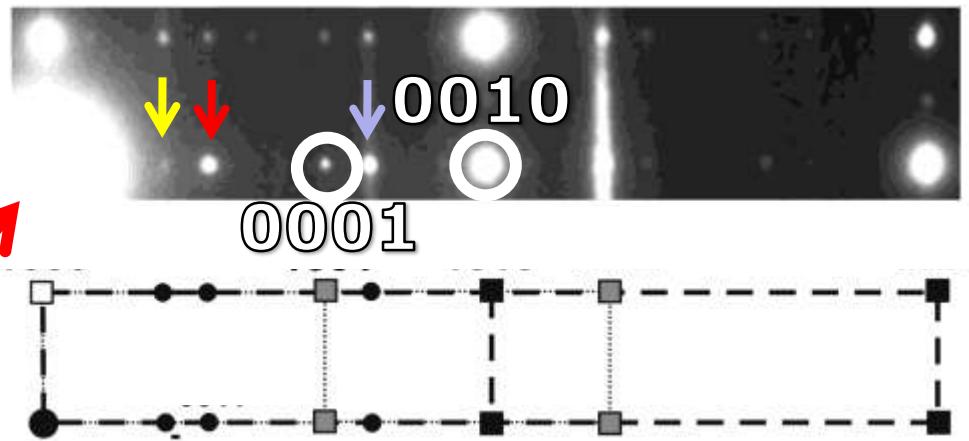
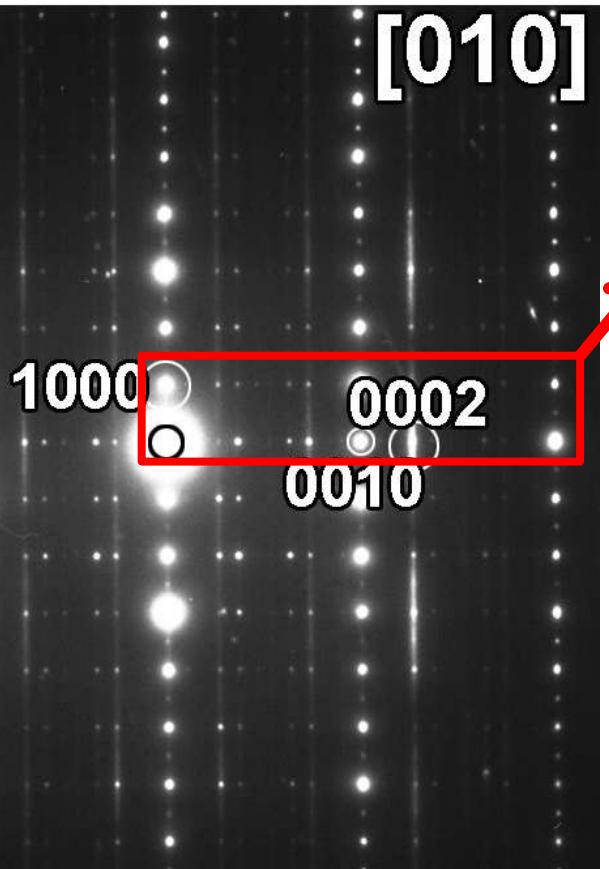


Mandal et al., Chem. Mater., 19, 25 (2007) 6158

ED of a composite structure (CS)



Exercise 2: composite structure: index the pattern

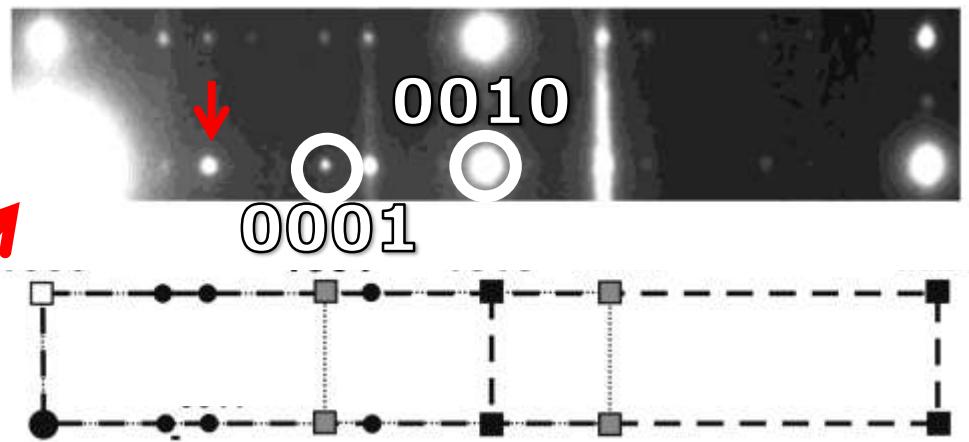
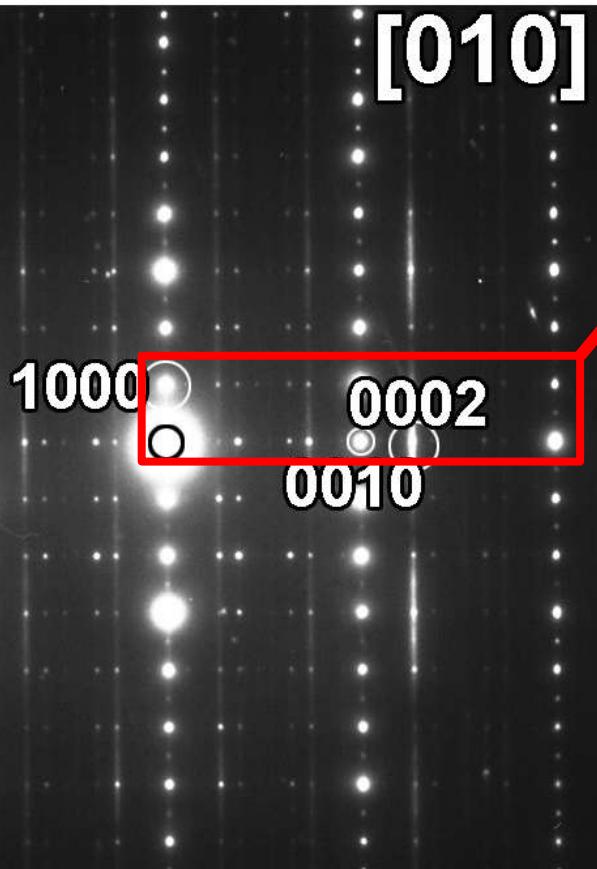


Which reflection is $001\bar{1}$?



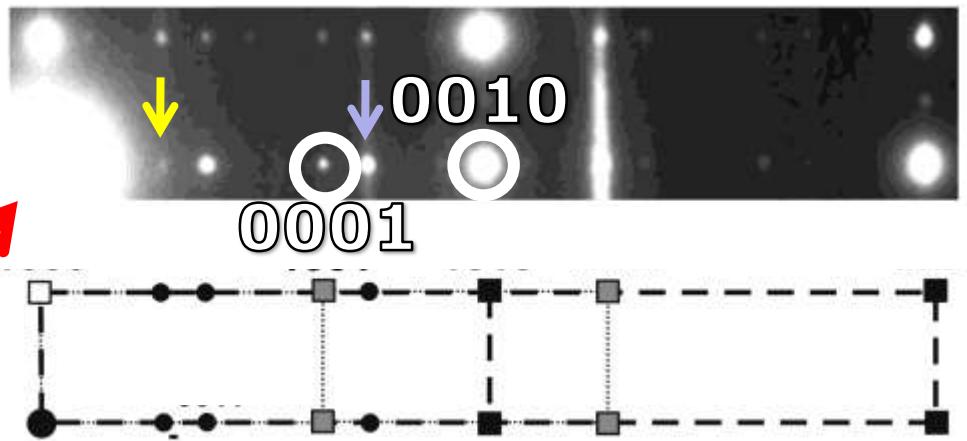
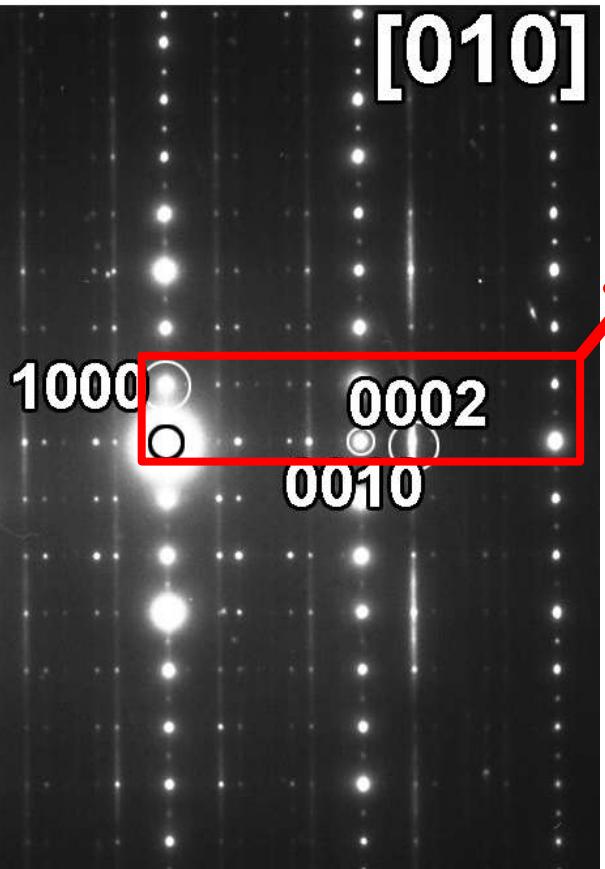
Abakumov et al., Chem.Mater. 19, 5 (2007) 1181-1189

Exercise 2: composite structure: index the pattern



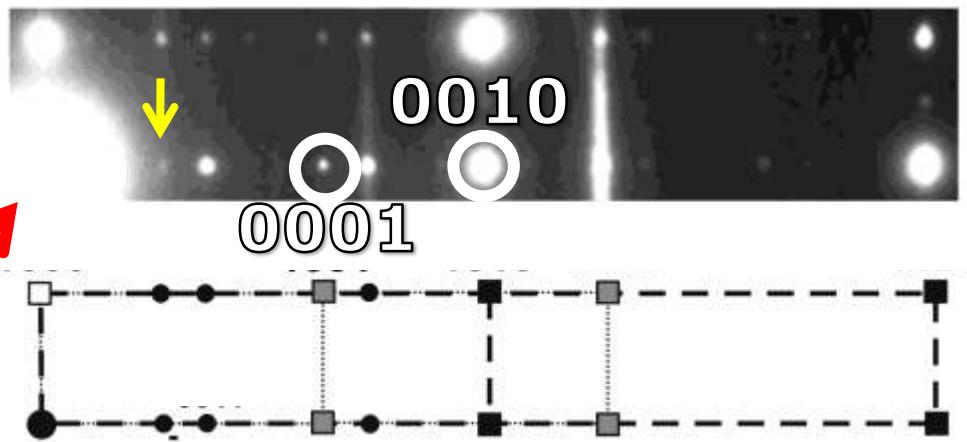
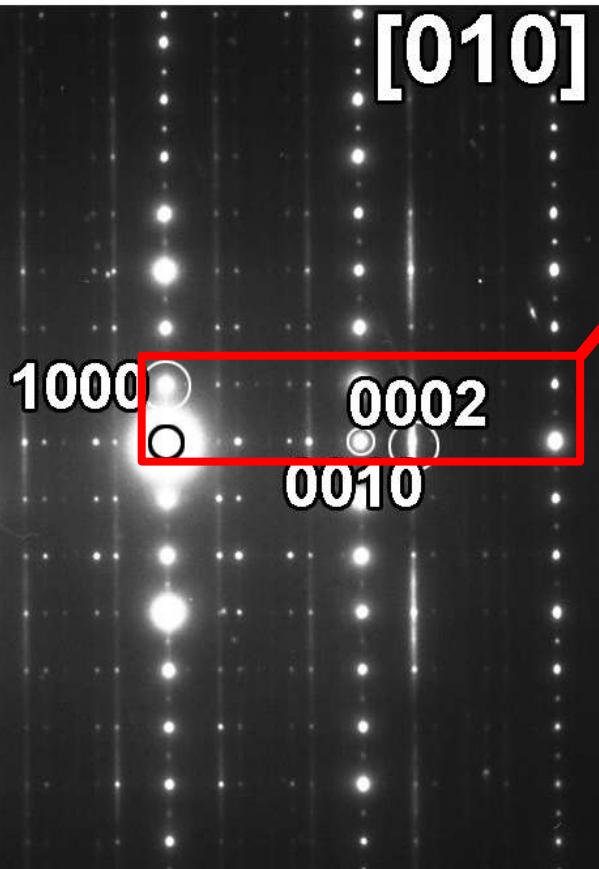
Which reflection is $001\bar{1}$?

Exercise 2: composite structure: index the pattern



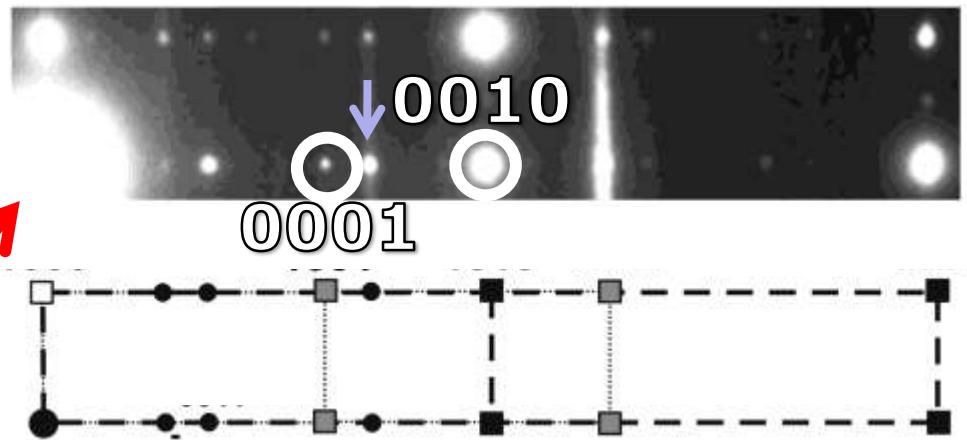
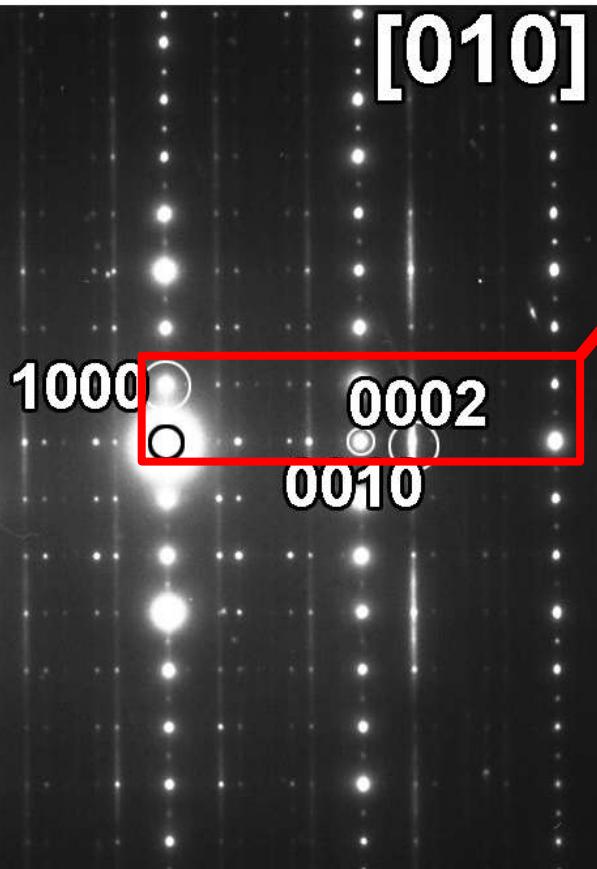
Which reflection is $00\bar{1}2$?

Exercise 2: composite structure: index the pattern



Which reflection is $00\bar{1}2$?

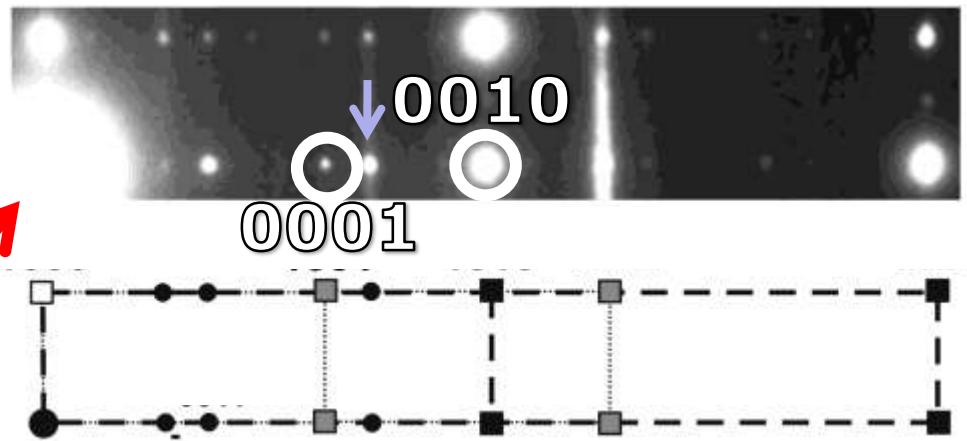
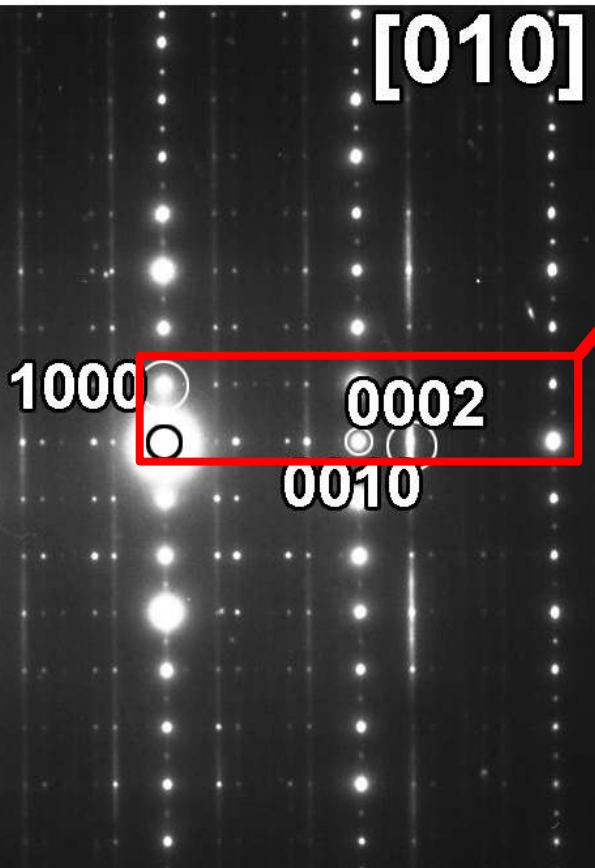
Exercise 2: composite structure: index the pattern



Which reflection does the blue arrow point at?

- 002 $\bar{2}$ ●
- 00 $\bar{2}$ 1 ●
- 001 $\bar{2}$ ●

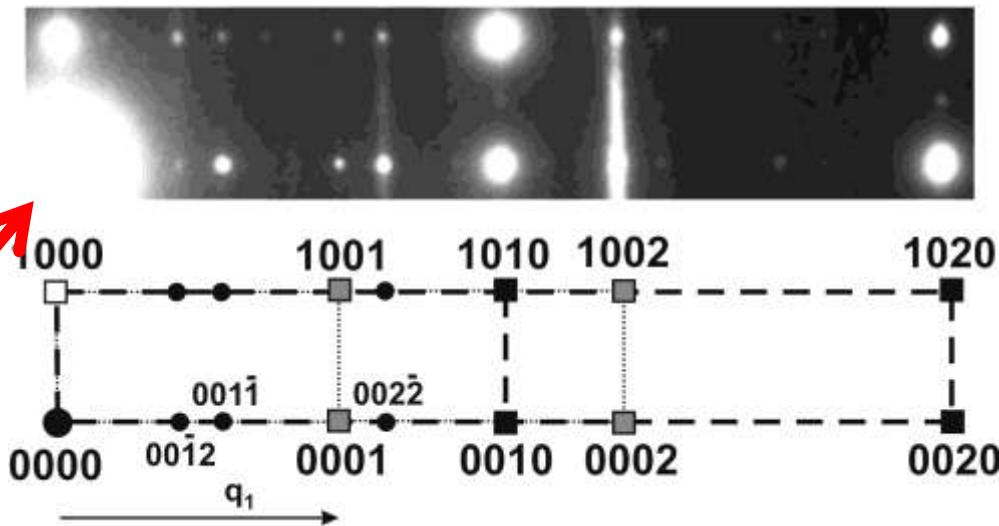
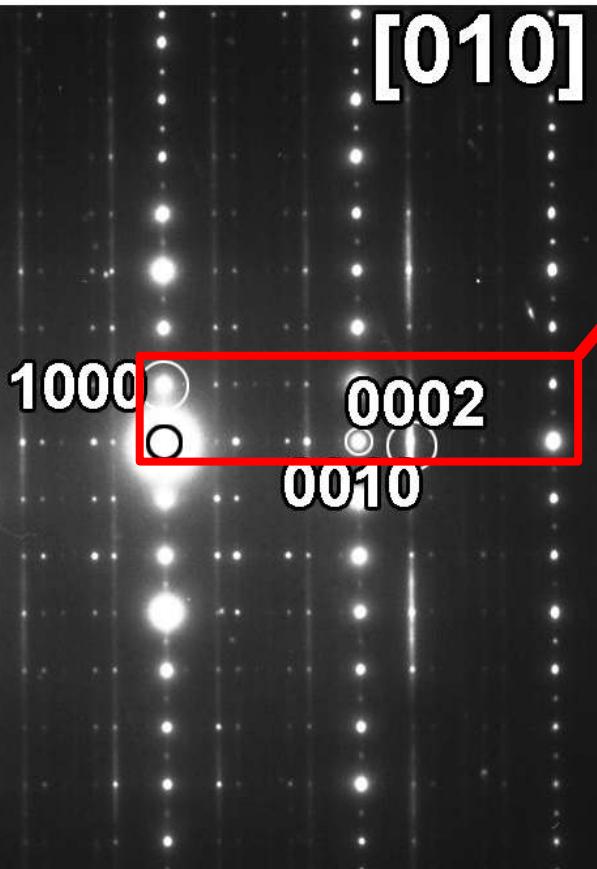
Exercise 2: composite structure: index the pattern



Which reflection does the blue arrow point at?
002 $\bar{2}$

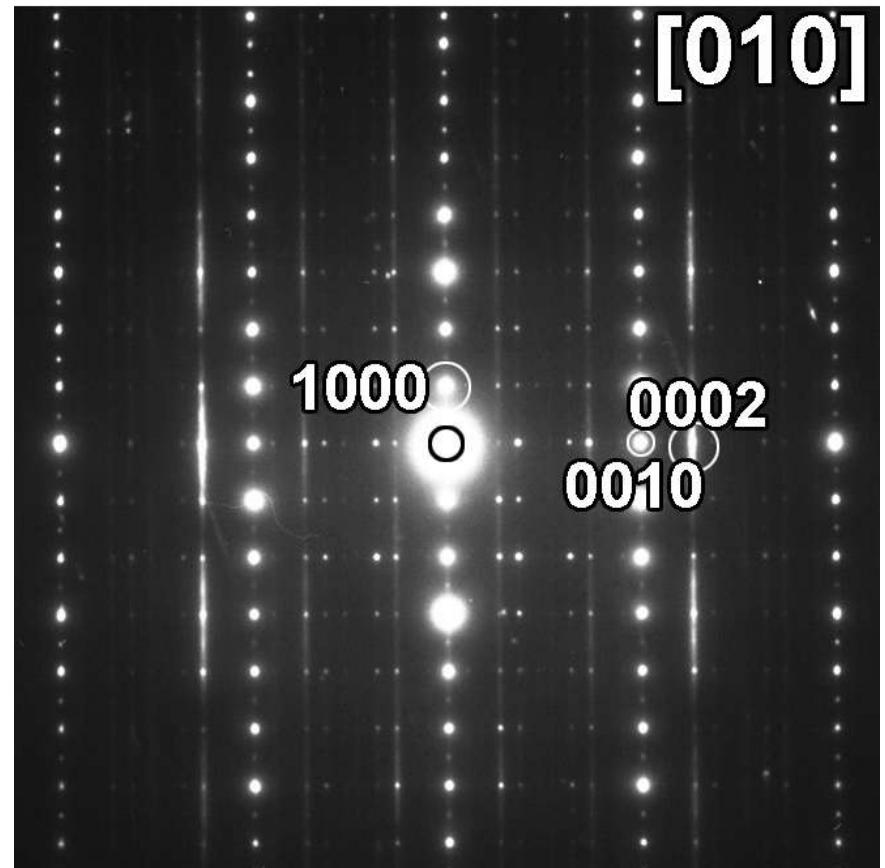


Exercise 2: composite structure: index the pattern - solution

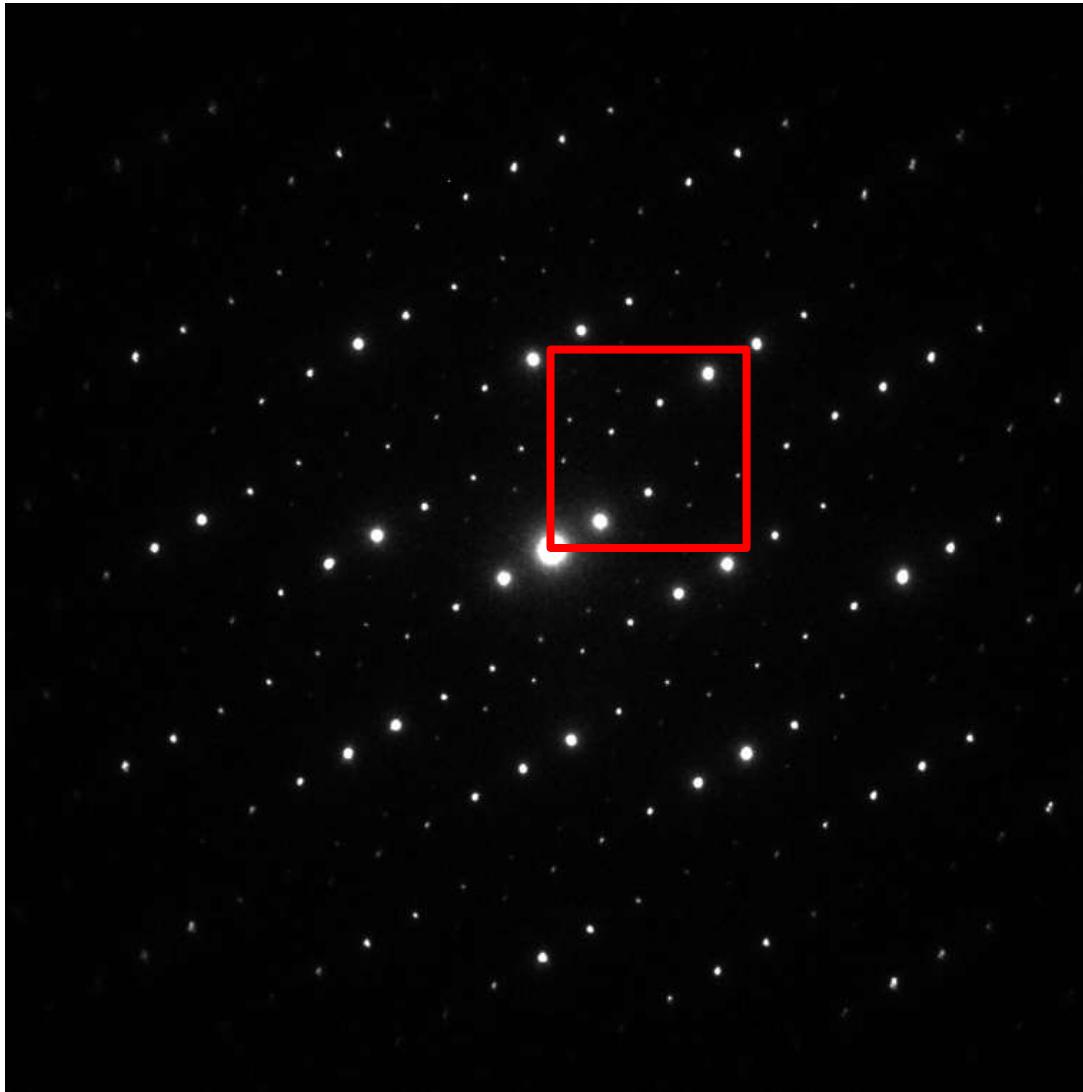


Streaks

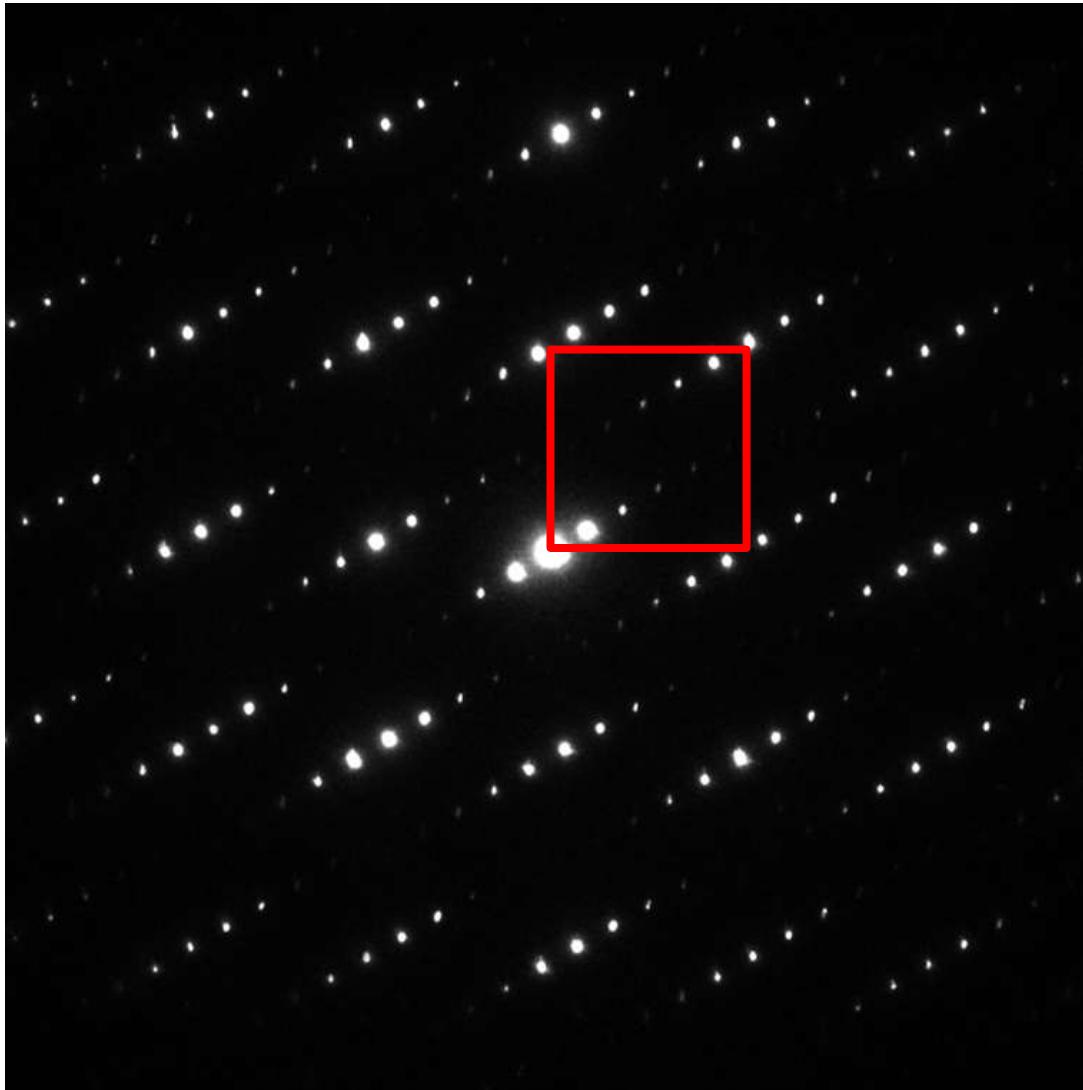
- order in the plane, disorder between these planes: streak on ED pattern
- disorder in two directions: diffuse intensity plane: streak or background intensity depending on the intersection



$Pb_{1-x}Bi_xFeO_{2.5+x/2}$ ($Pb:Bi=7:0.5$)

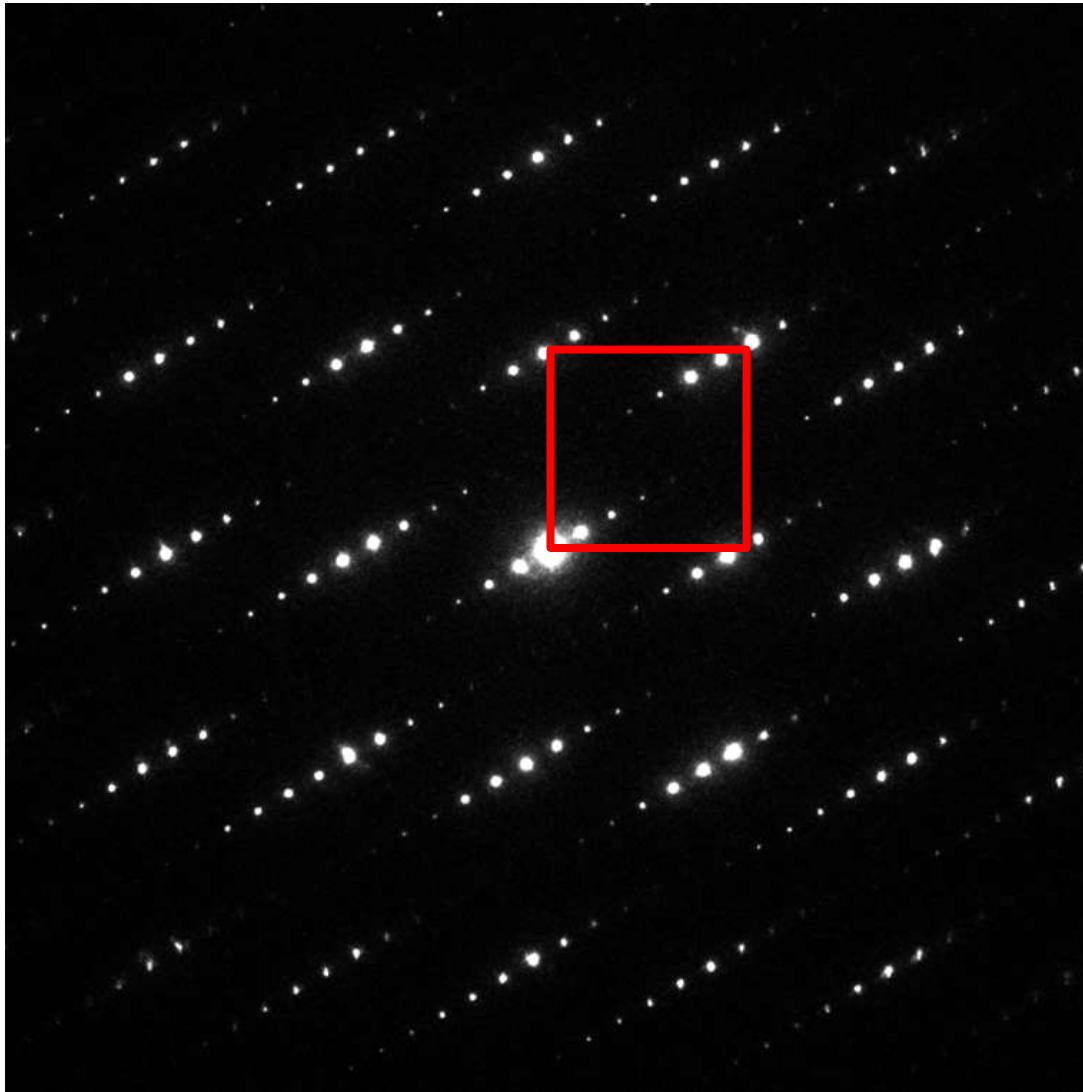


$Pb_{1-x}Bi_xFeO_{2.5+x/2}$ ($Pb:Bi=7:3$)

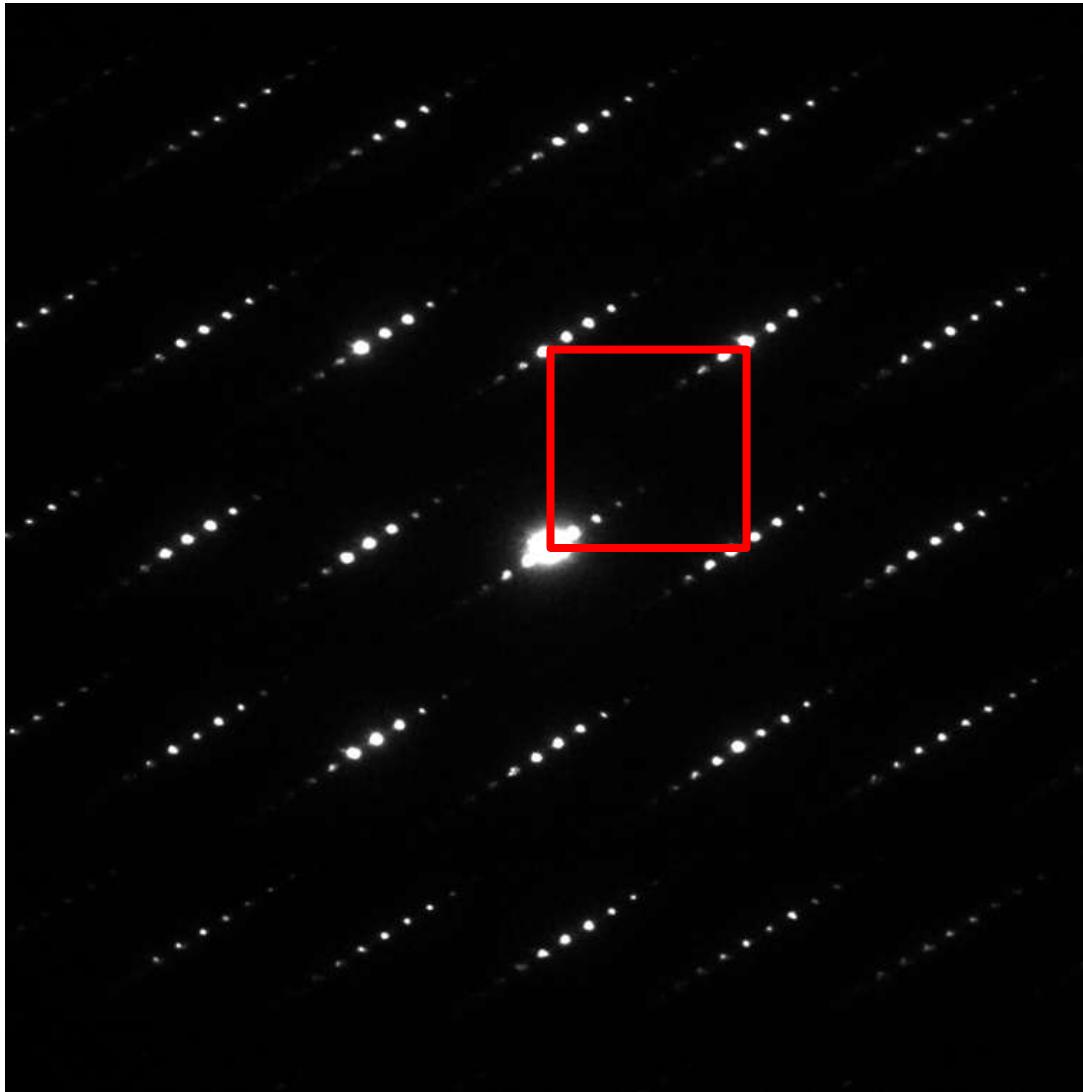


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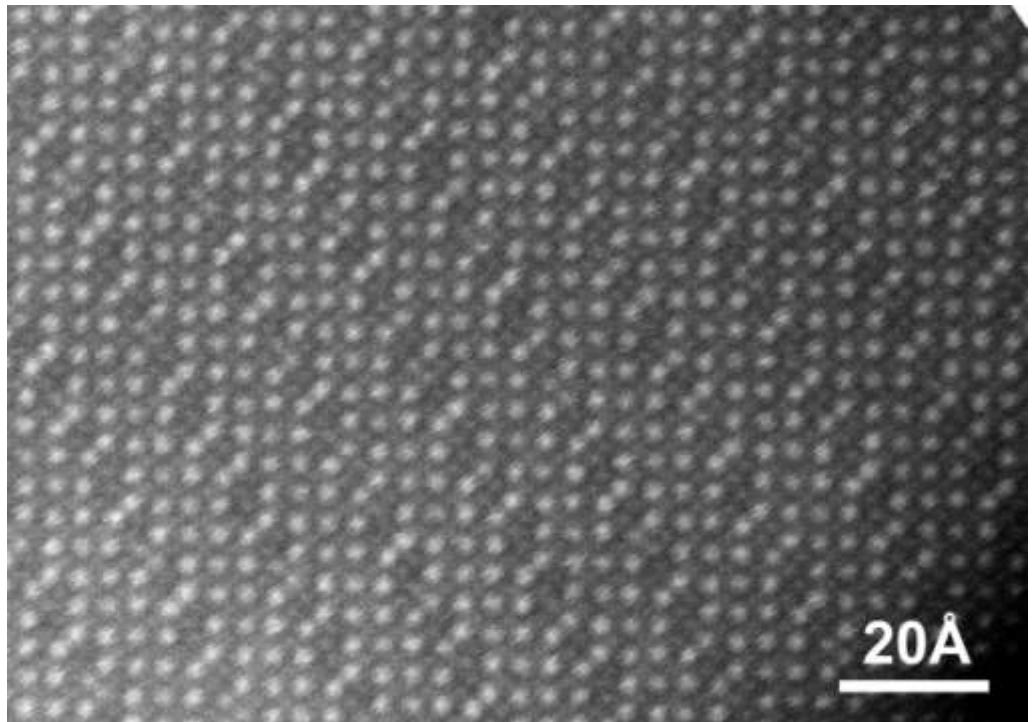
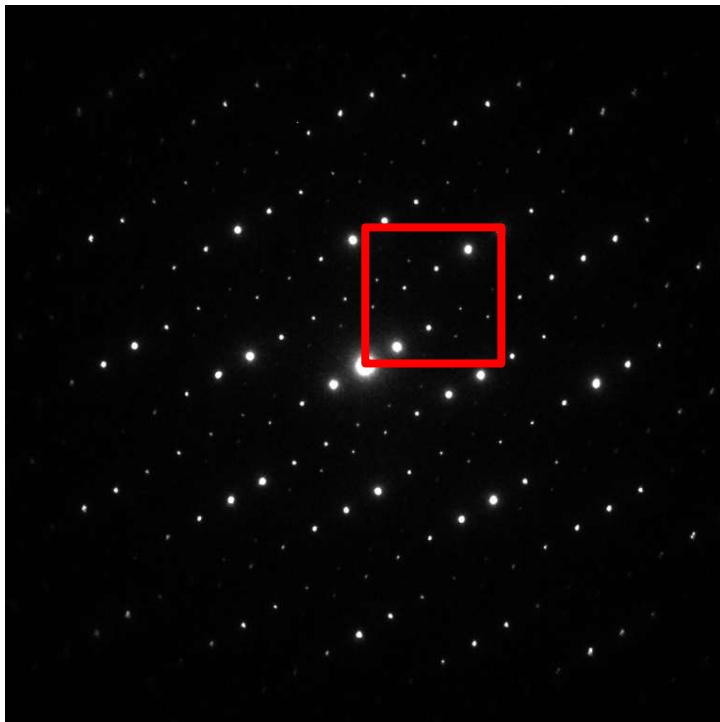
$Pb_{1-x}Bi_xFeO_{2.5+x/2}$ ($Pb:Bi=7:5$)



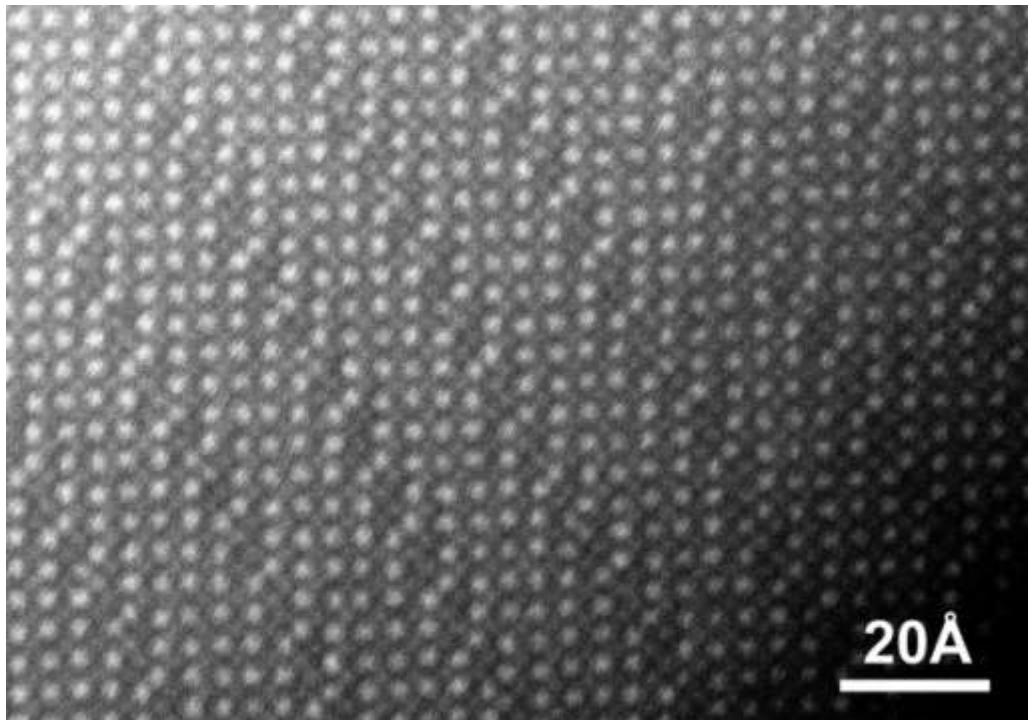
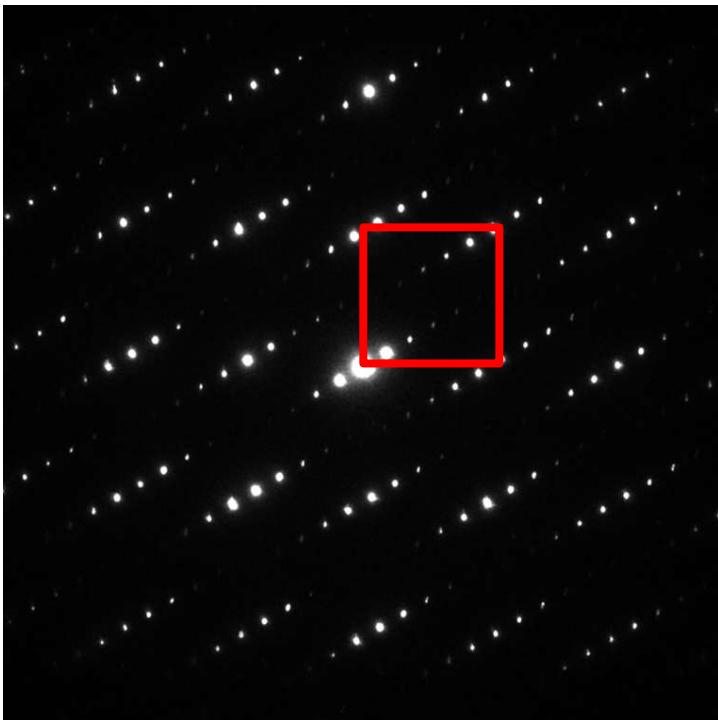
$Pb_{1-x}Bi_xFeO_{2.5+x/2}$ (Pb:Bi=7:8)



$Pb_{1-x}Bi_xFeO_{2.5+x/2}$ (Pb:Bi=7:0.5)



$\text{Pb}_{1-x}\text{Bi}_x\text{FeO}_{2.5+x/2}$ ($\text{Pb}:\text{Bi}=7:3$)

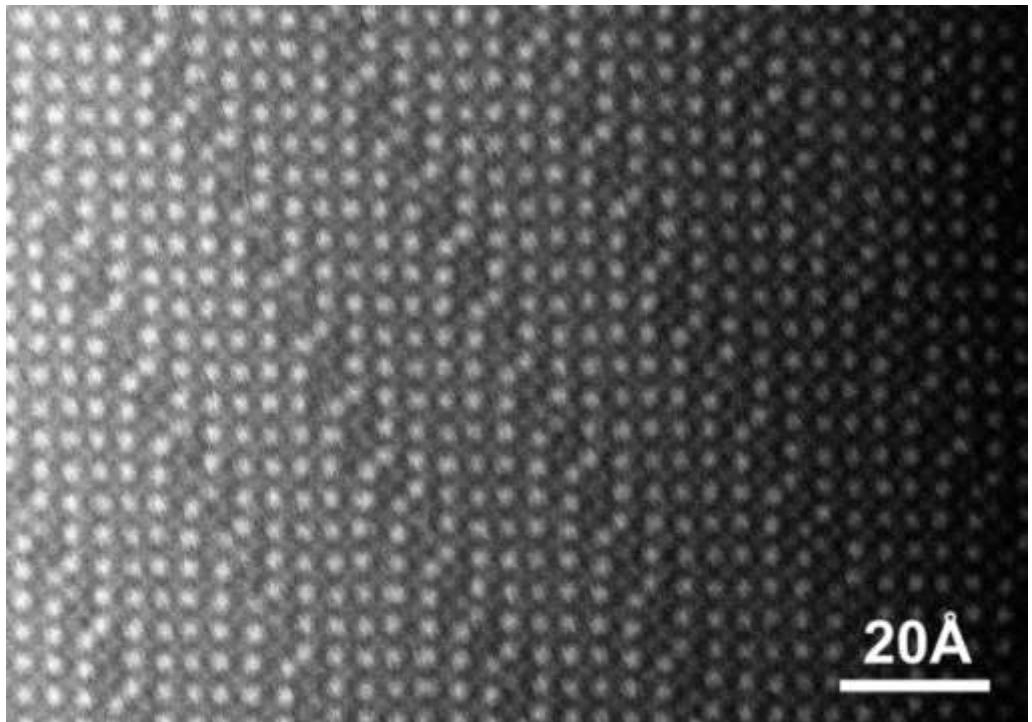
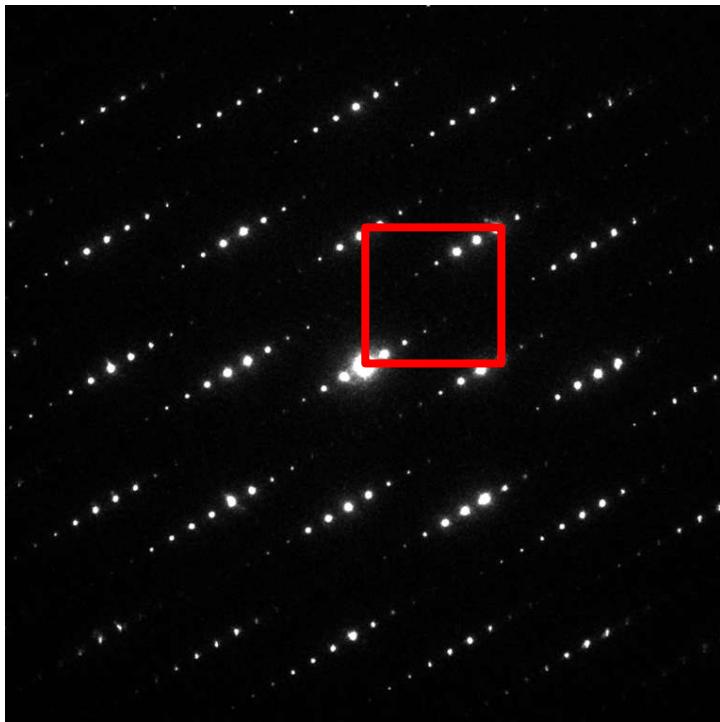


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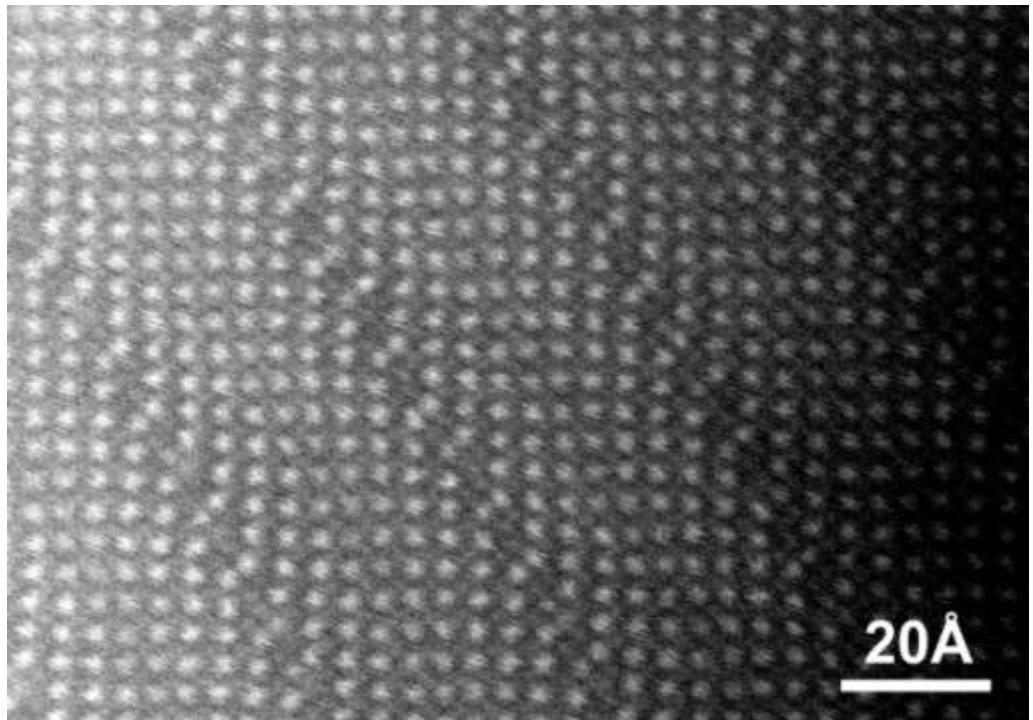
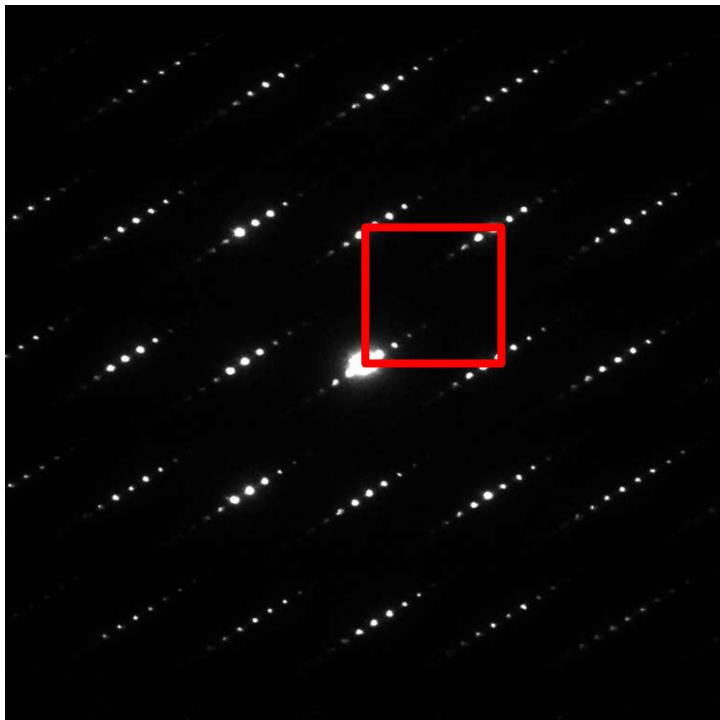
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$Pb_{1-x}Bi_xFeO_{2.5+x/2}$ (Pb:Bi=7:5)



$Pb_{1-x}Bi_xFeO_{2.5+x/2}$ (Pb:Bi=7:8)



HRTEM and HAADFSTEM-images

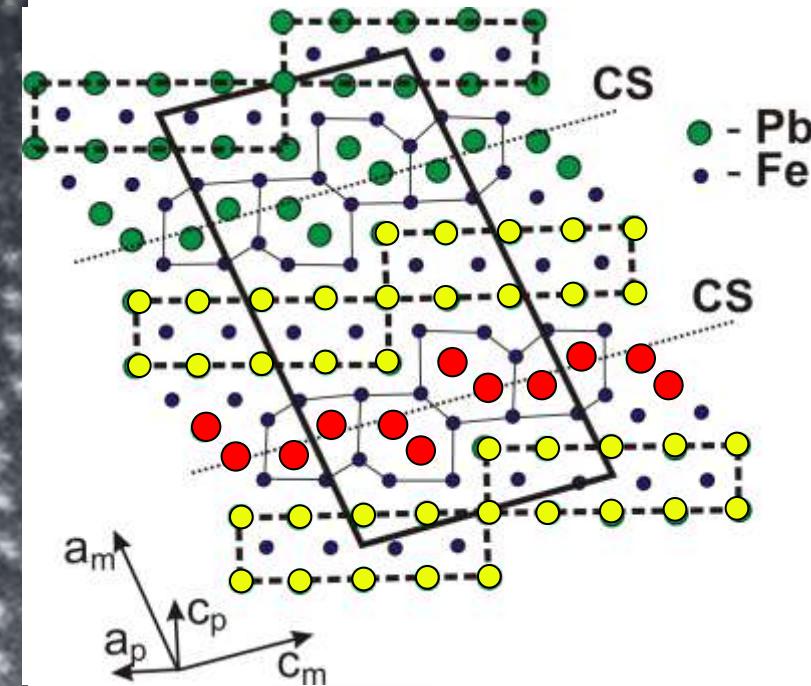
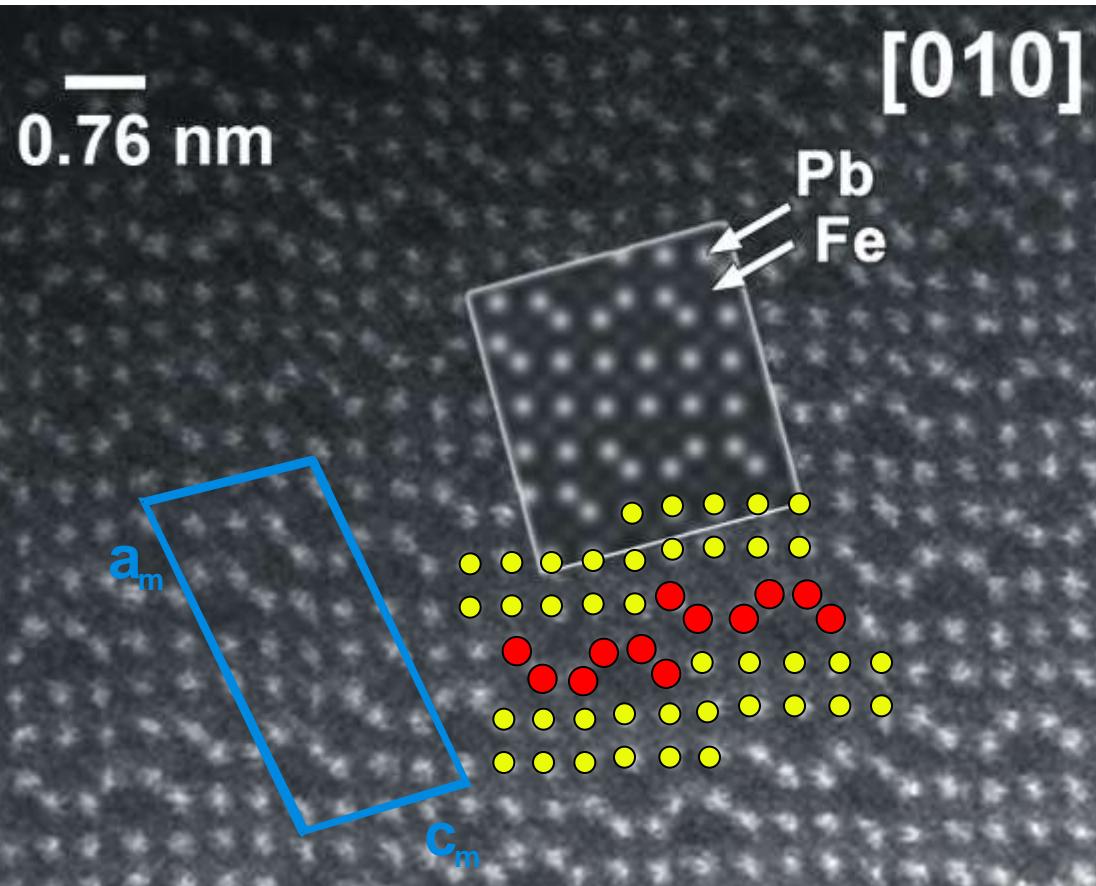
HAADF-STEM

- contrast related to Z^n
- direct information about the heavy atom positions
- brighter with thickness also
- correct orientation very important!
- scanned image: positions imprecise

HRTEM

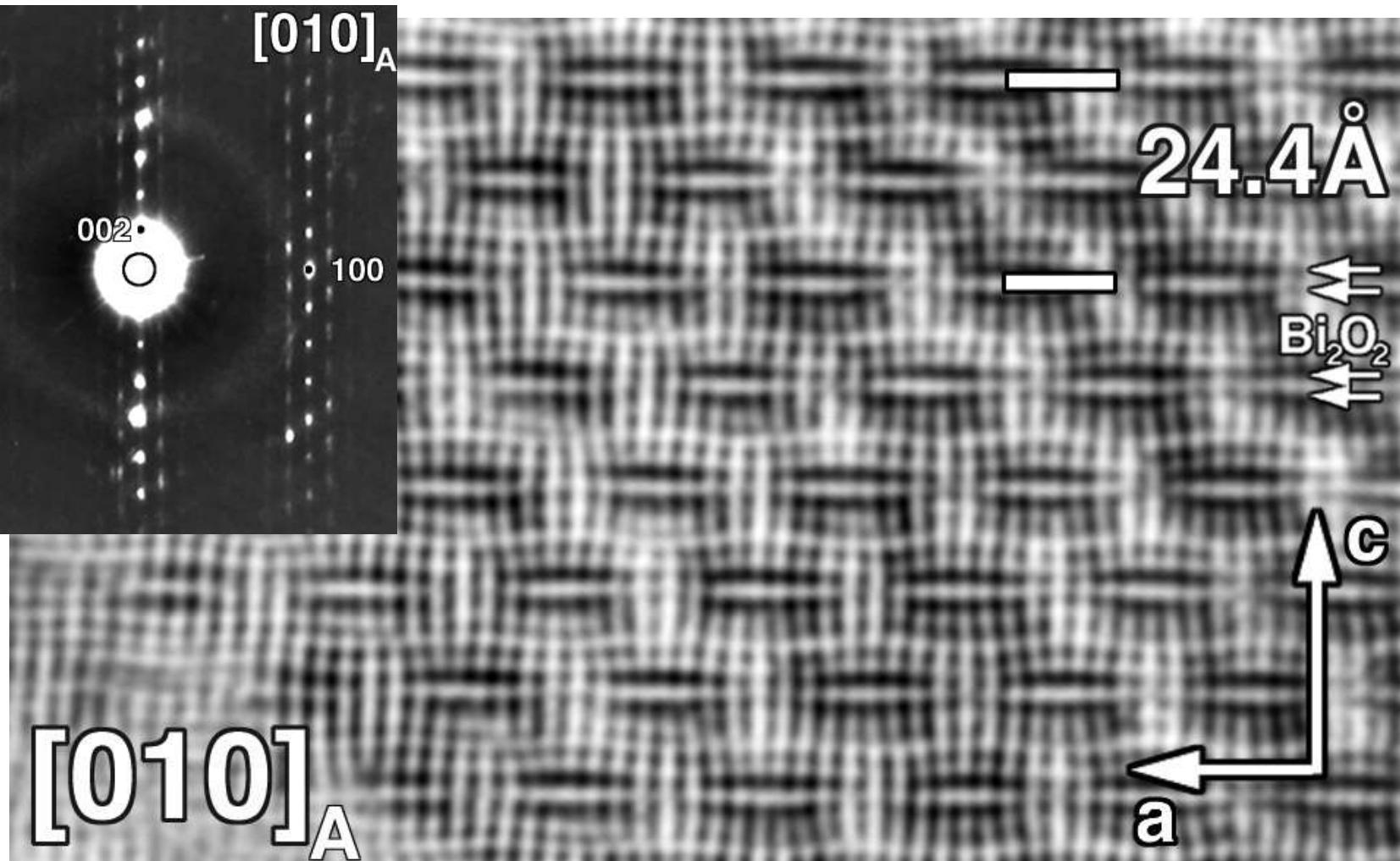
- complicated phase transfer function
- no direct interpretation possible
- contrast changes with thickness and defocus
- correct orientation very important!
- one-shot image: position of the contrast objects reliable

Heavy atoms from HAADF-STEM

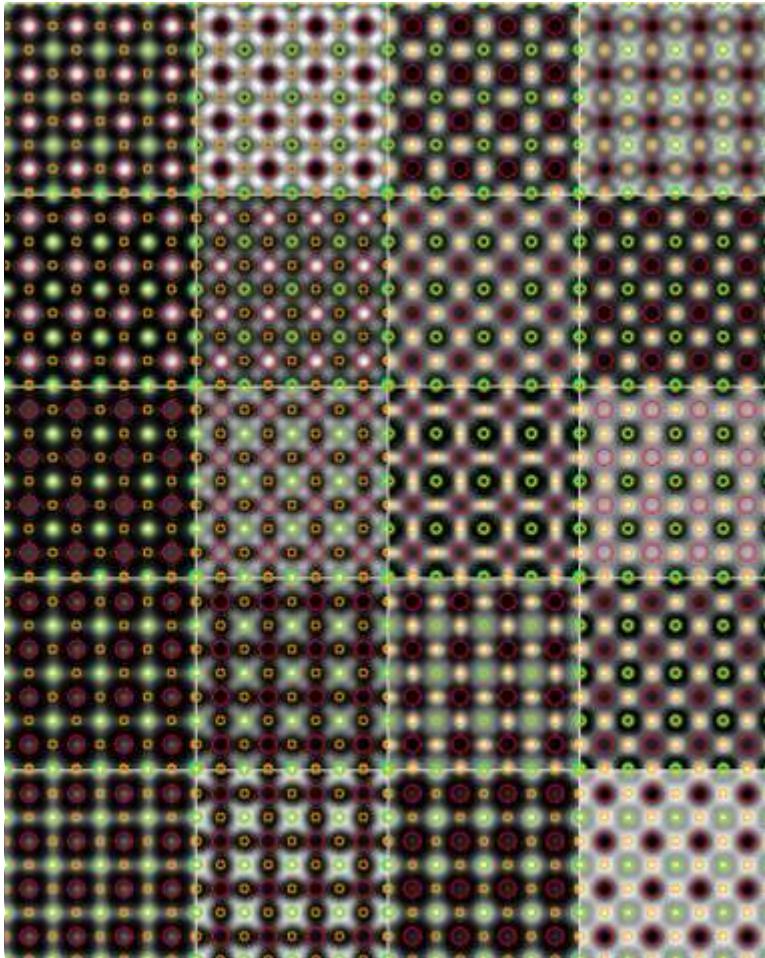


$\text{Pb}_2\text{Fe}_2\text{O}_5$ Abakumov et al., Ang.Chemie-Int.Ed., 45, 40
(2006) 6697-6700

High resolution transmission electron microscopy (HREM)



Relation structure – image in HRTEM

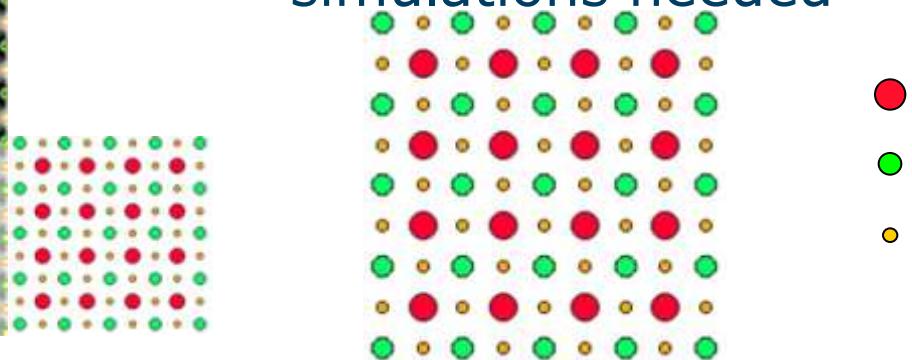


 **simple perovskite**

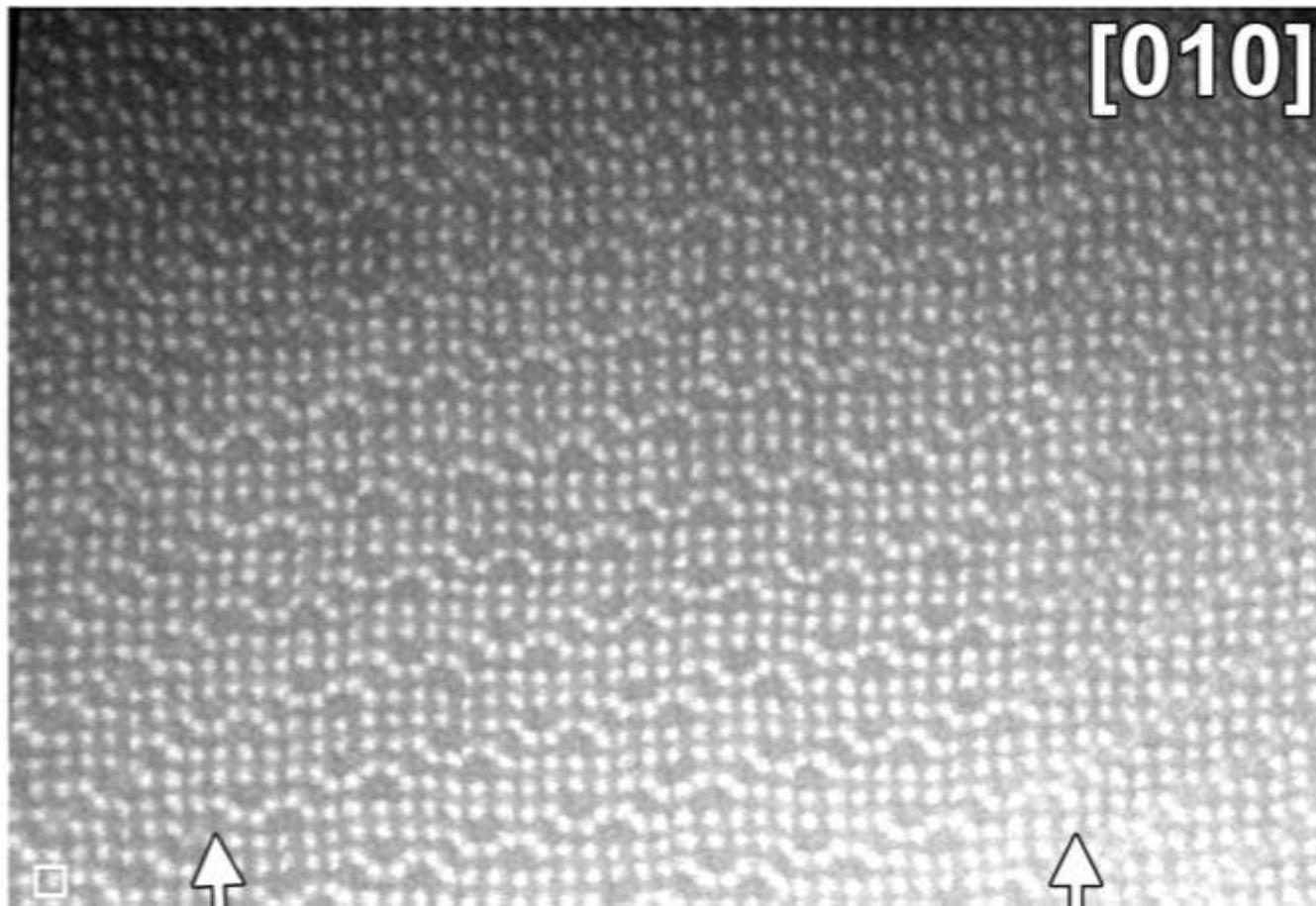
image influenced by contrast
transfer function of the
microscope

no direct correspondence

visual comparison with
simulations needed



Exercise 3: predict what the ED pattern will roughly look like



Hadermann et al., Solid State Sciences, 10; 4 (2008) 382-389



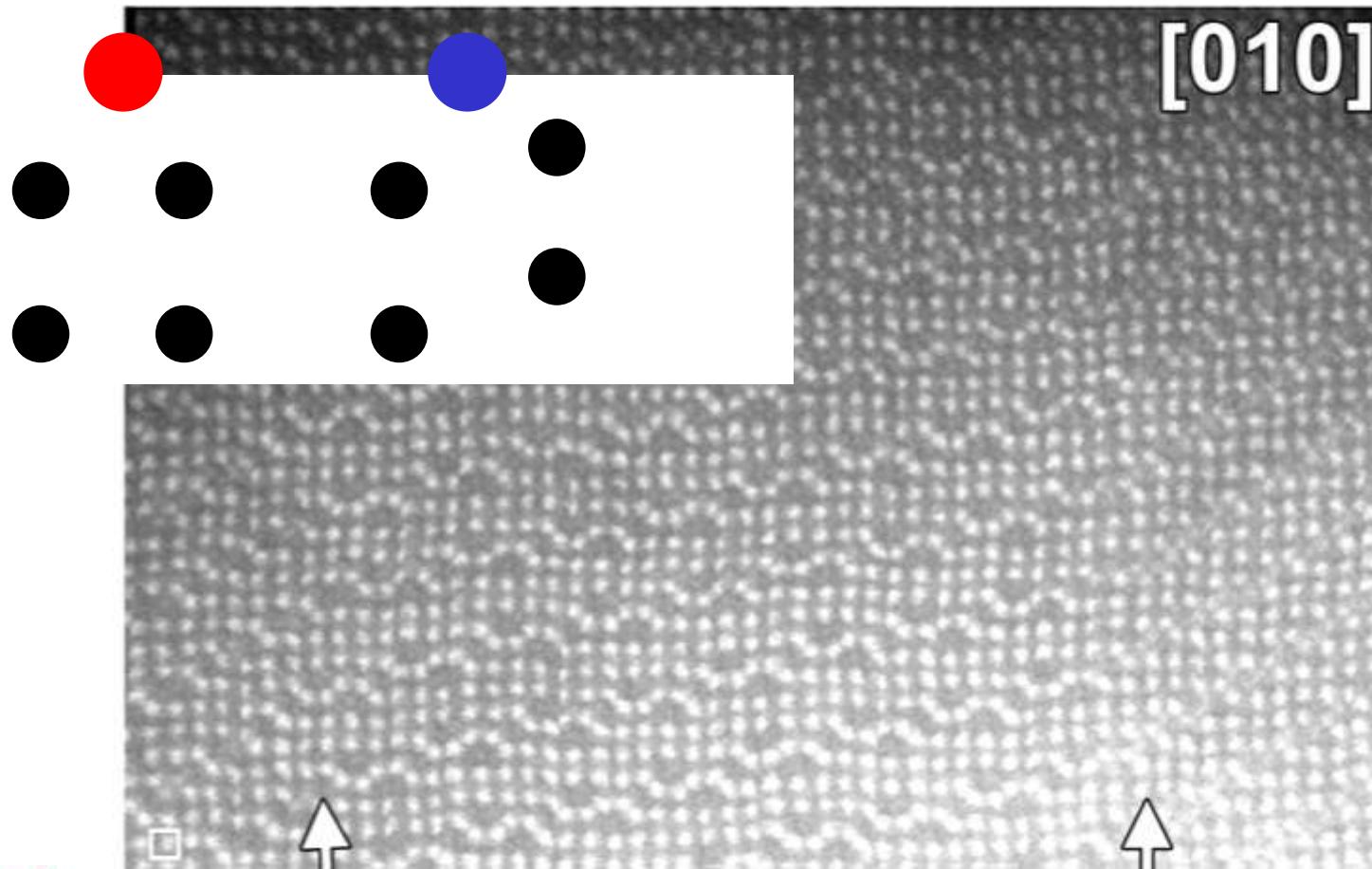
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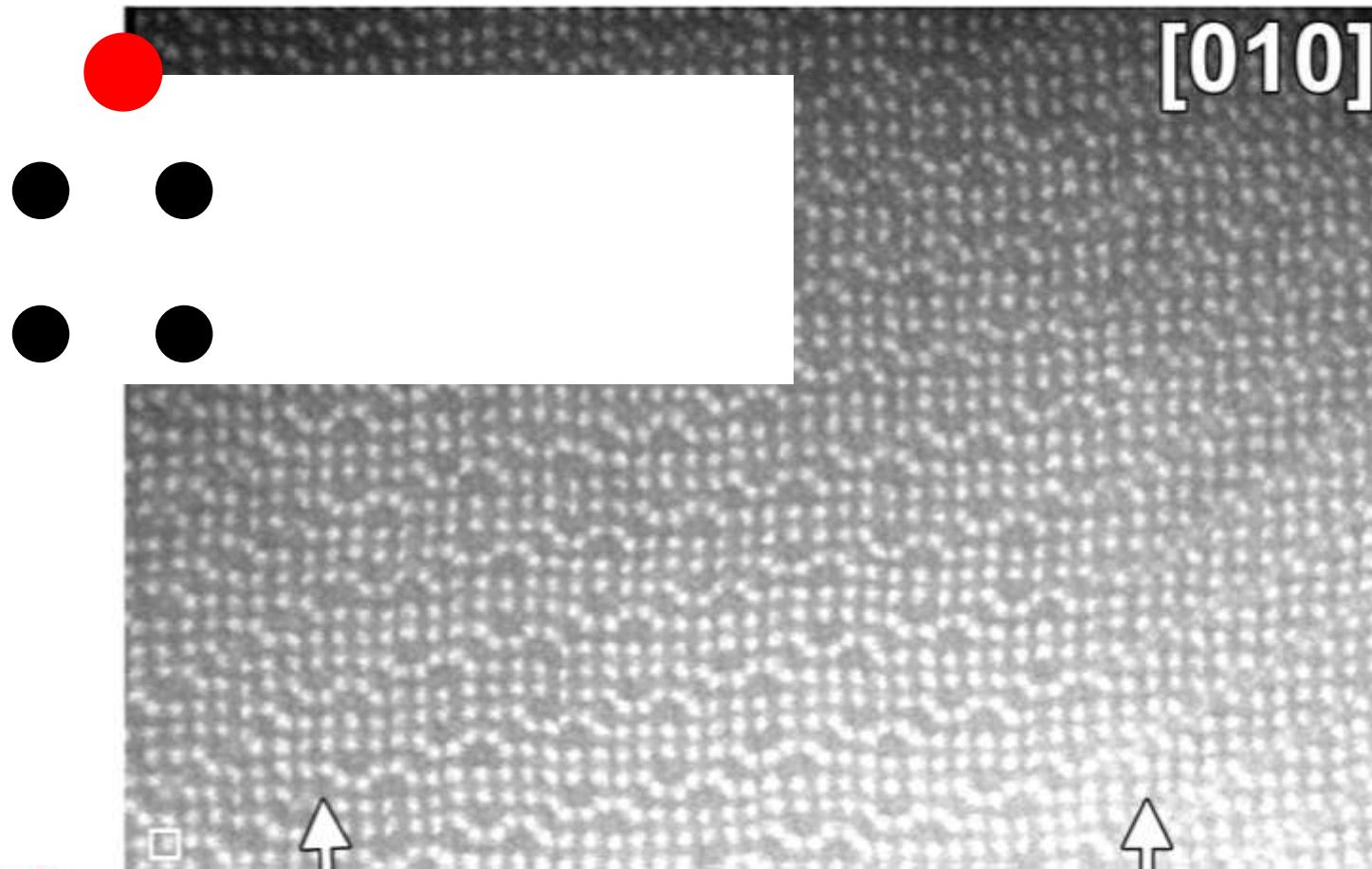
Exercise 3: predict what the ED pattern will roughly look like

What will be the subcell reflections?

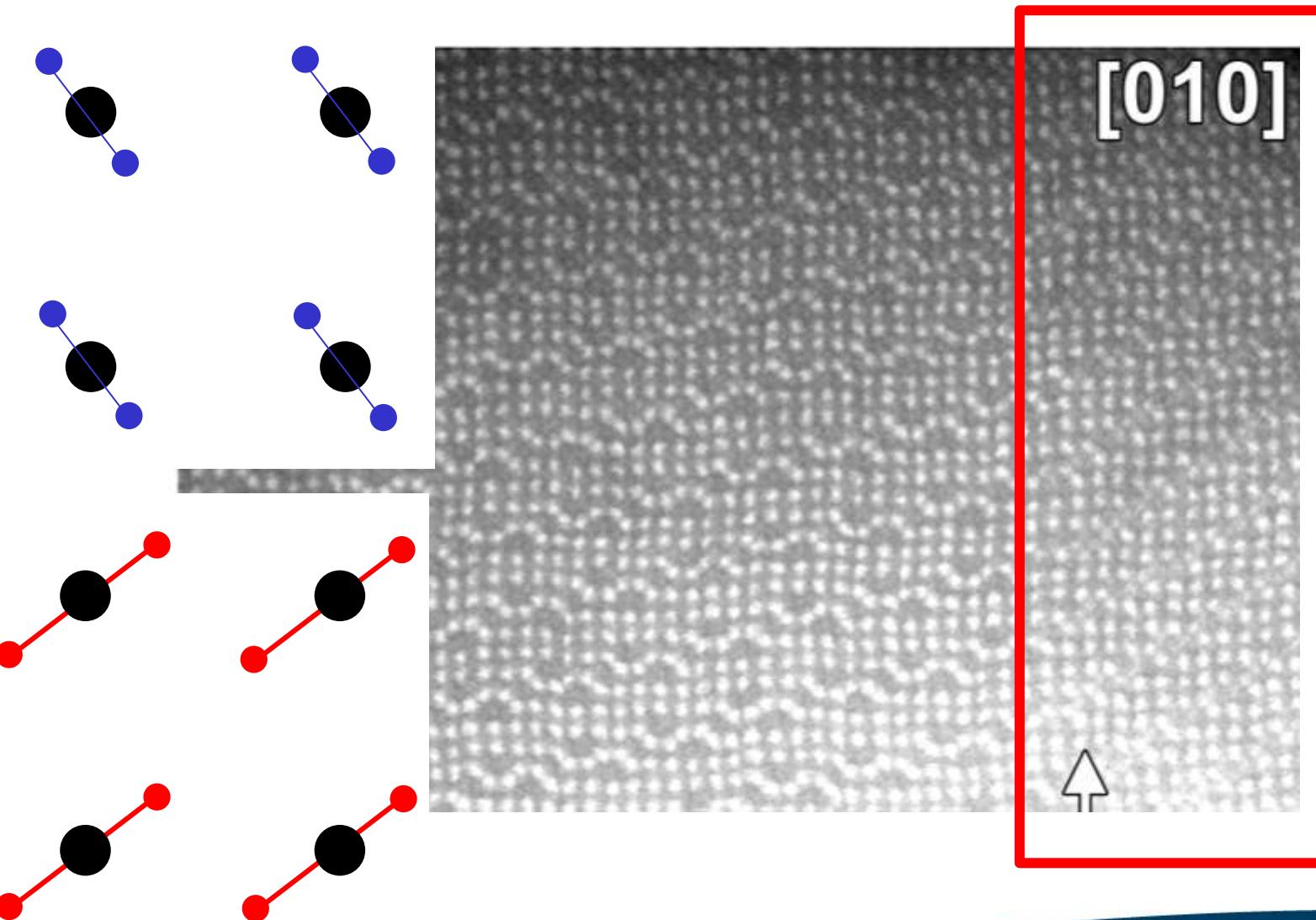


Exercise 3: predict what the ED pattern will roughly look like

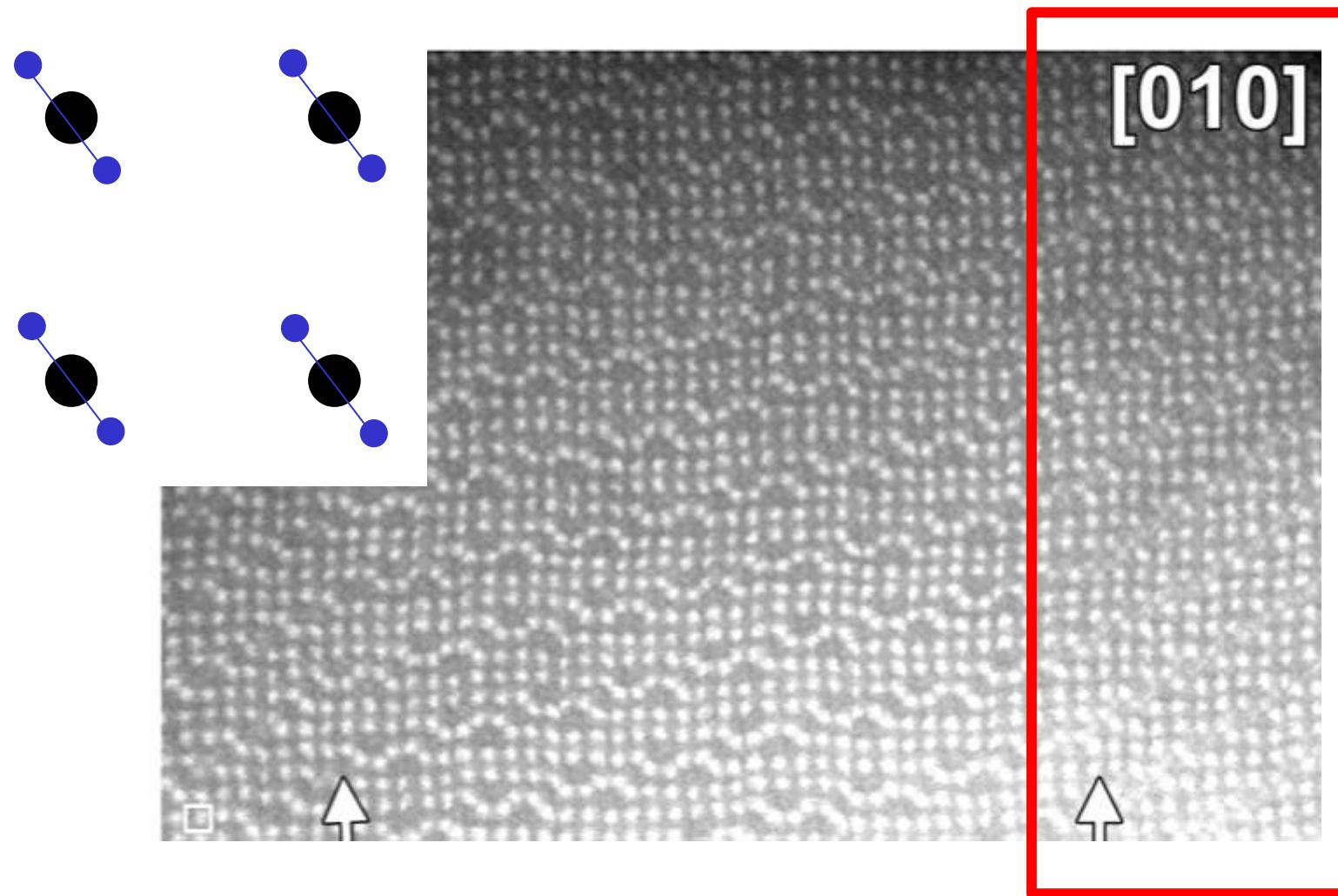
What will be the subcell reflections?



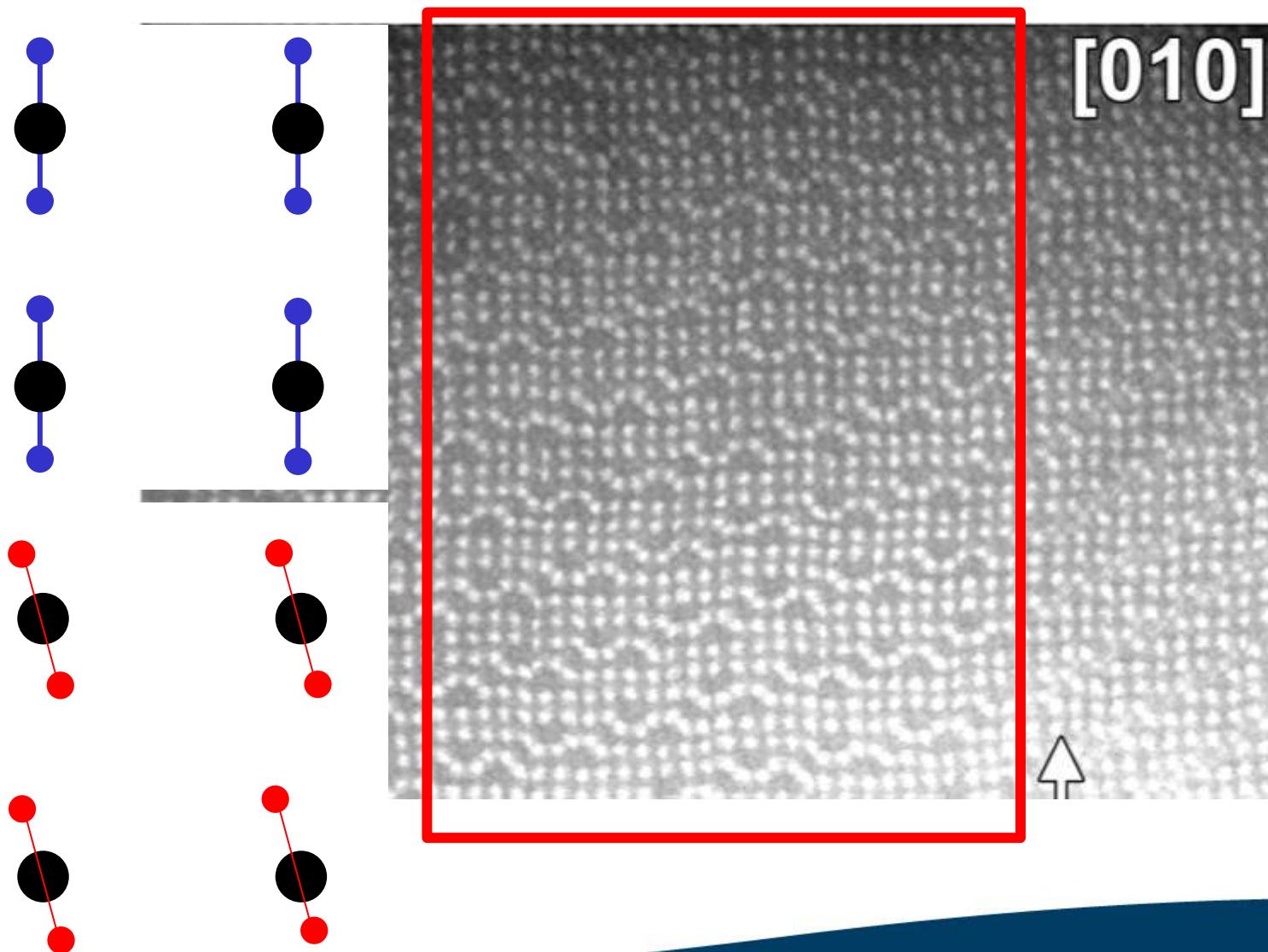
Exercise 3: predict what the ED pattern will roughly look like



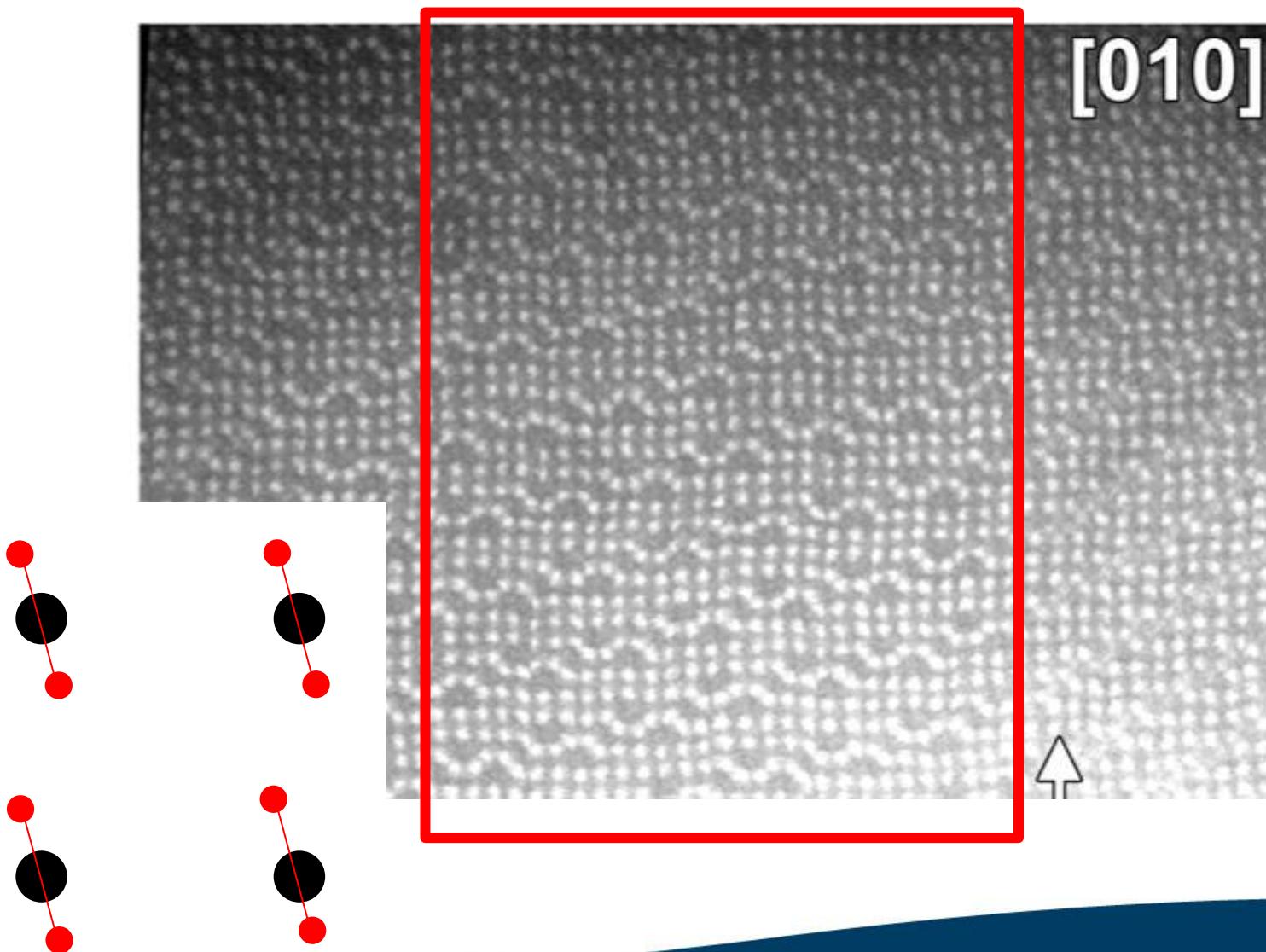
Exercise 3: predict what the ED pattern will roughly look like

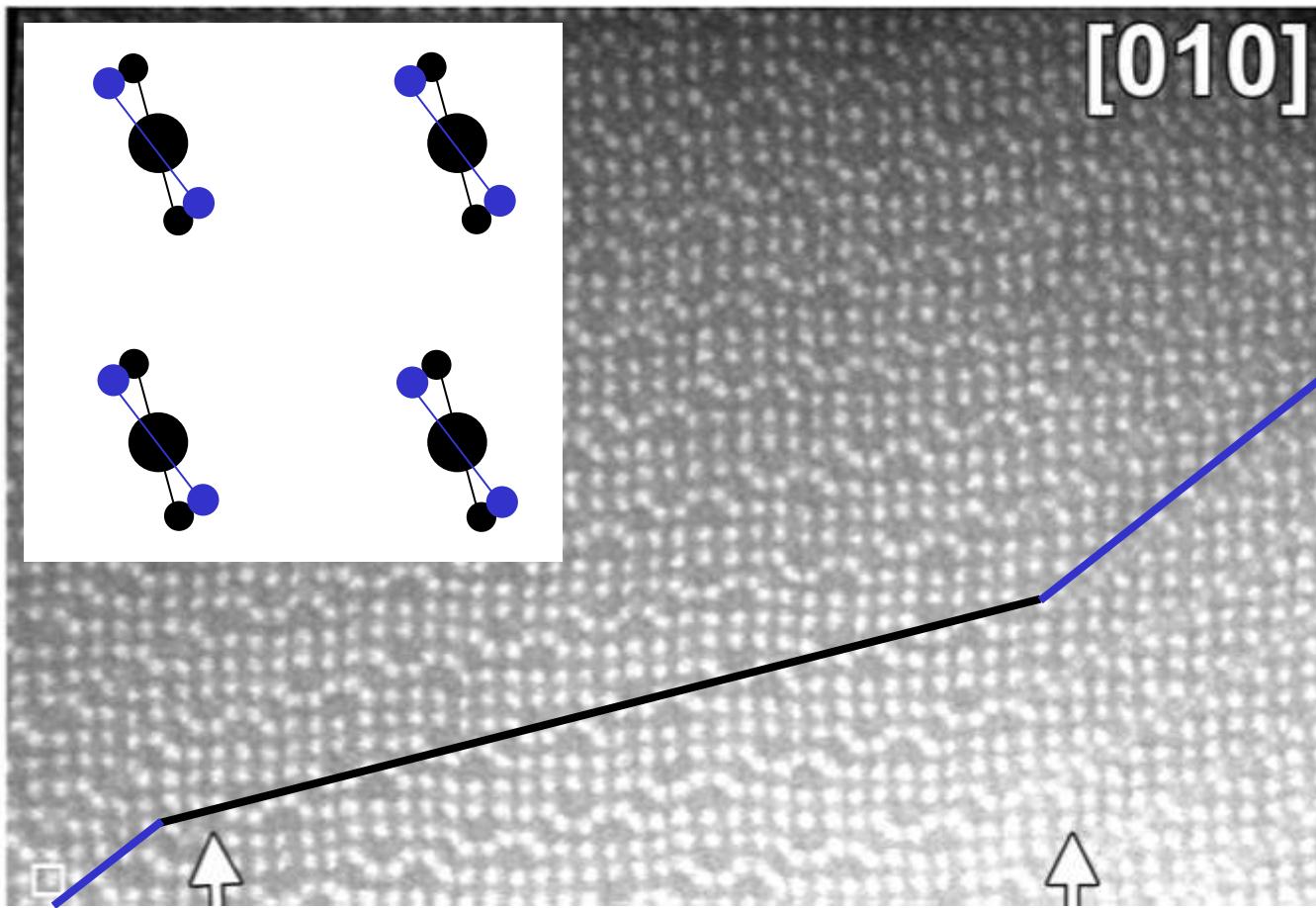


Exercise 3: predict what the ED pattern will roughly look like

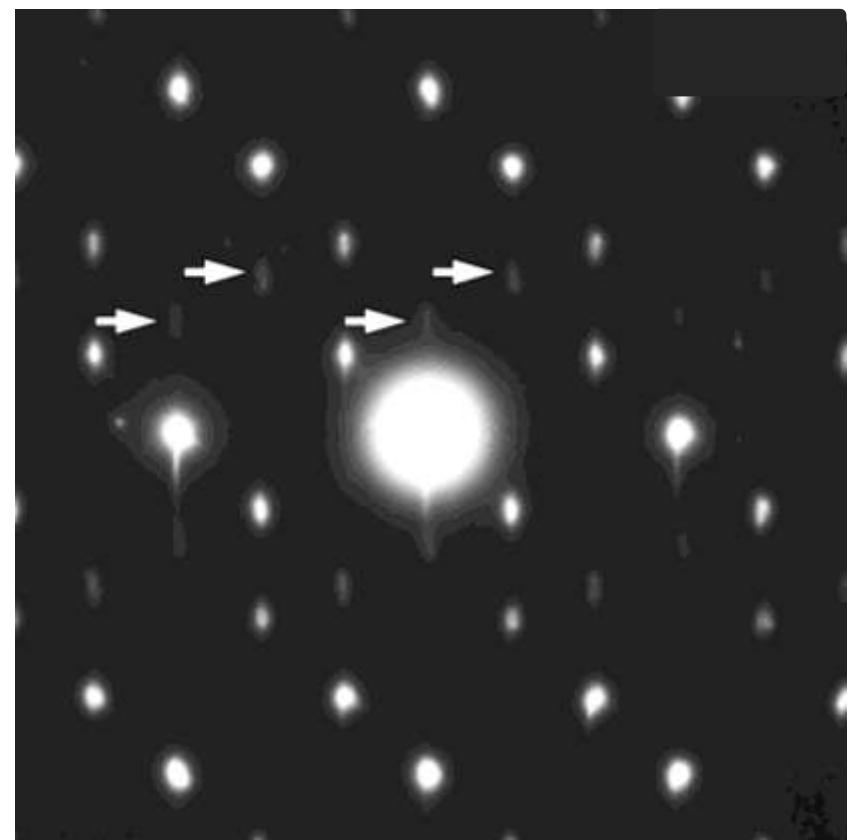
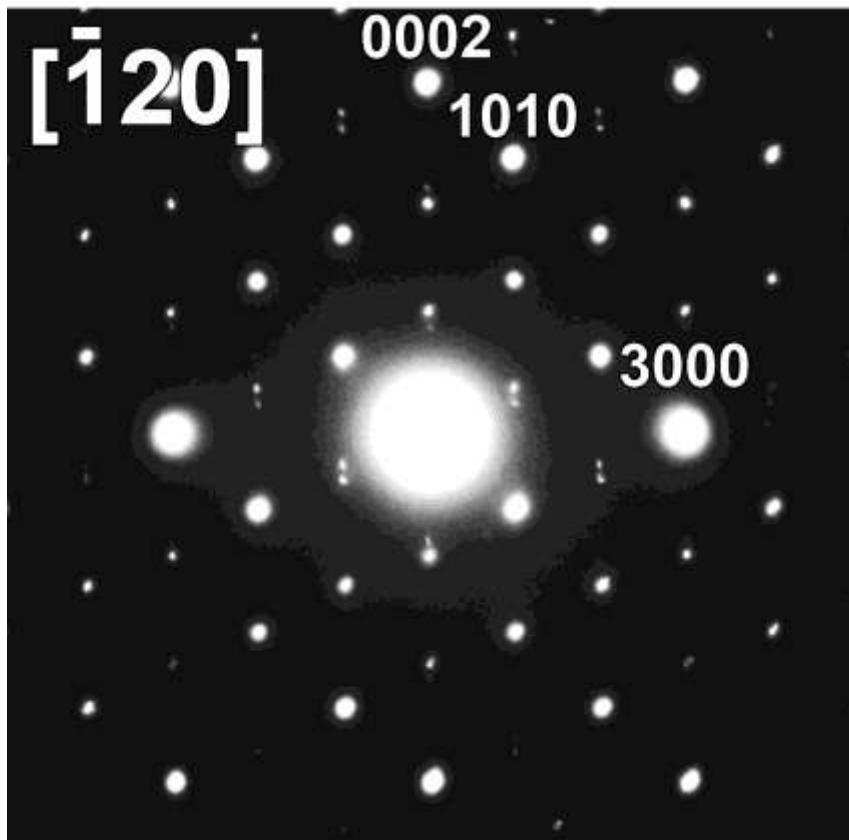


Exercise 3: predict what the ED pattern will roughly look like





Domains with varying q



Mandal et al., Chem. Mater., 19, 25 (2007) 6158

Comments to the applicability of conclusions from TEM



- If the satellites cannot be seen on the XRPD-NPD patterns it is no use to try to refine the structure using the ED cell.
- Precision of parameters determined from ED is not high. ED is good for determining parameters (cell parameters and direction and length of q) qualitatively.
- Good for determining symmetry.
- Can give direct space information and compositional information also.



Purpose of this lecture

At the end of this lecture you should be able to:

- Understand the TEM paragraph in papers about IMS and CS
- Be able to make solid comments about conclusions claimed from TEM by different sources (collaborators, papers,...) by knowing some possible pitfalls
- Decide whether it would be useful to do TEM on your own IMS or CS
- Make basic interpretations of TEM data on your own materials by yourself

