



26 September - 2 October 2010, Carqueiranne, France

Incommensurate composite crystals

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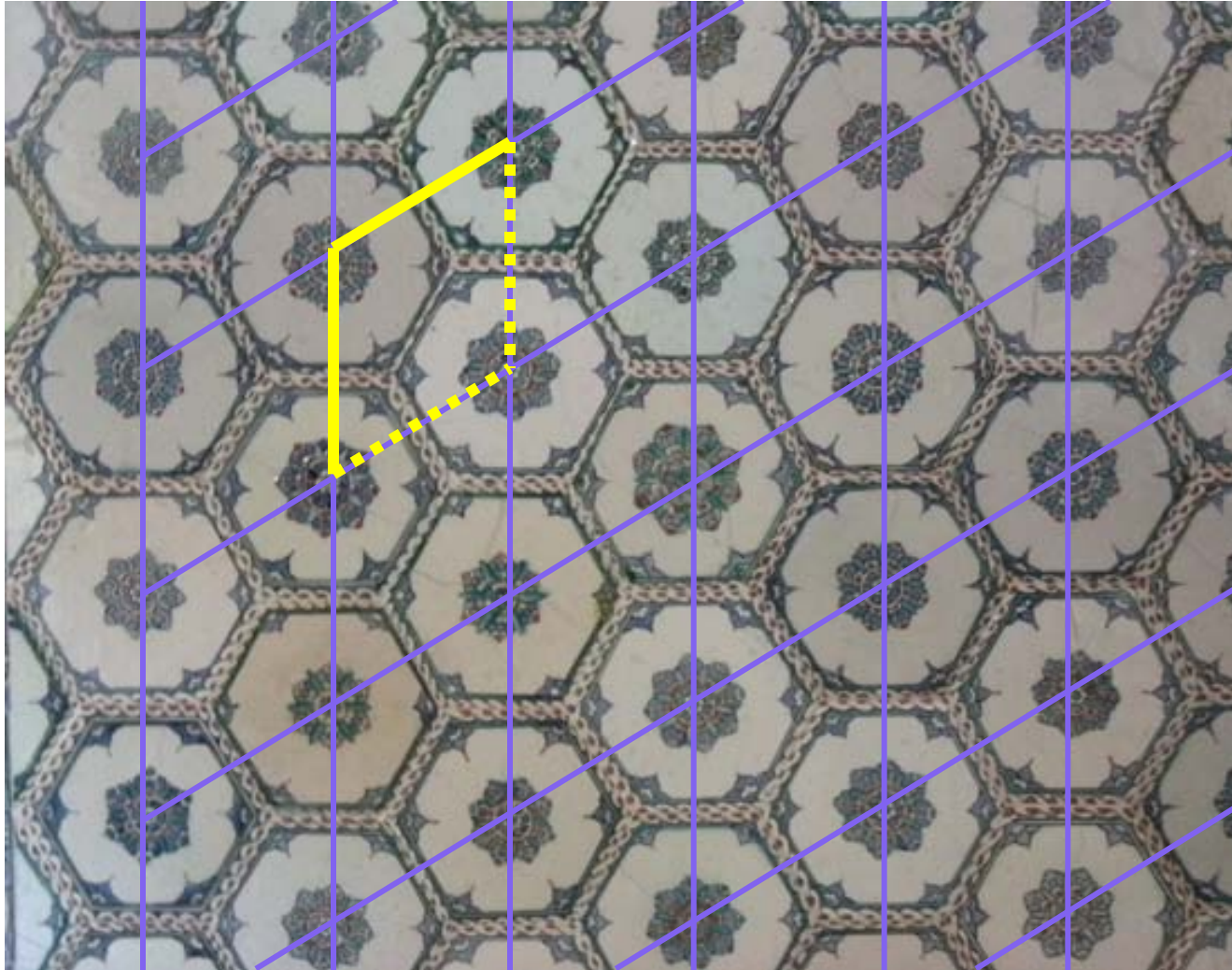
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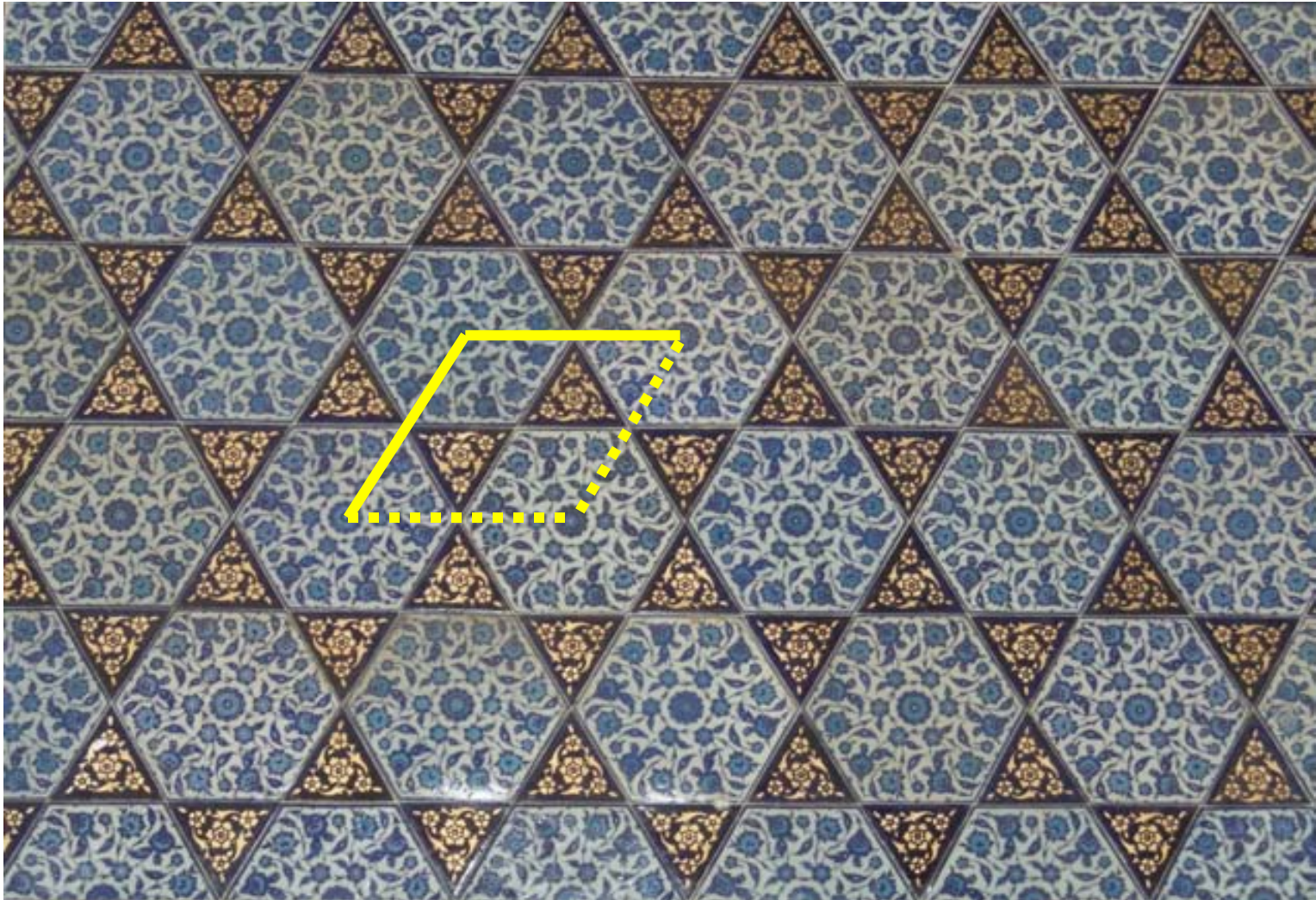
Honeycomb lattice hexagonal lattice with one tile per unit cell



ECM25, Istanbul (2009) harem at topkapi palace

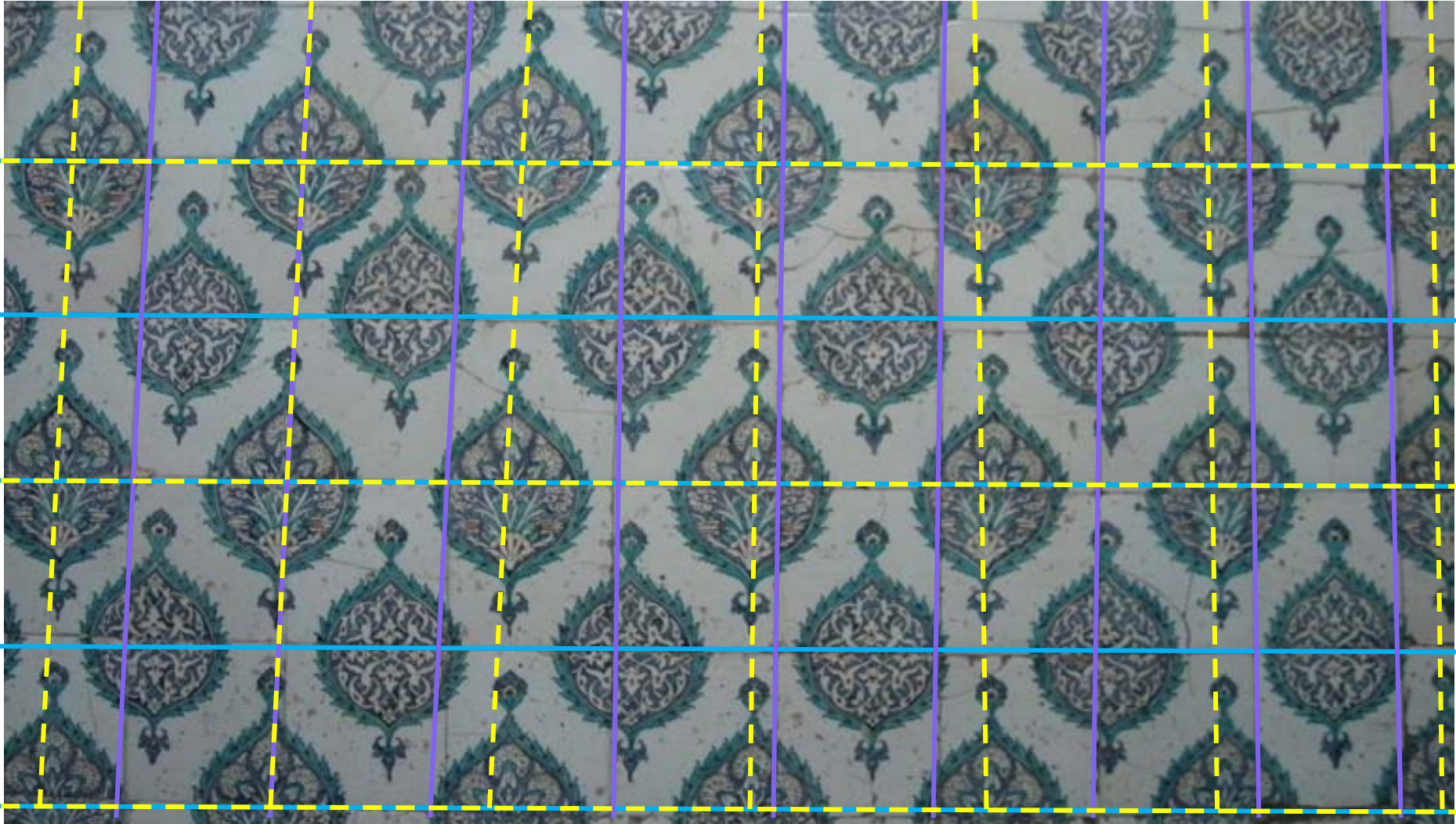
Kagome lattice

One hexagon and two triangles per unit cell



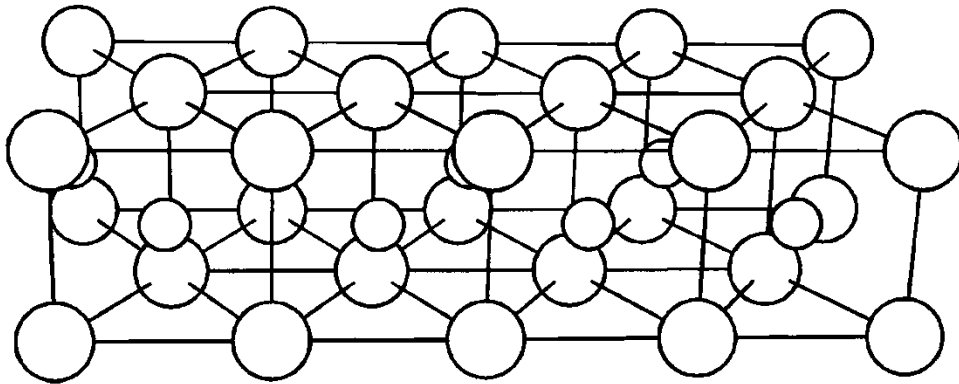
ECM25, Istanbul (2009) harem at topkapi palace

Mutually incommensurate periodicities for tiles
and design layers; $2a \times 1.37 \dots b$ supercell



ECM25, Istanbul (2009) harem at topkapi palace

Misfit layer sulfide $[\text{LaS}]_{1.13}[\text{TaS}_2]$

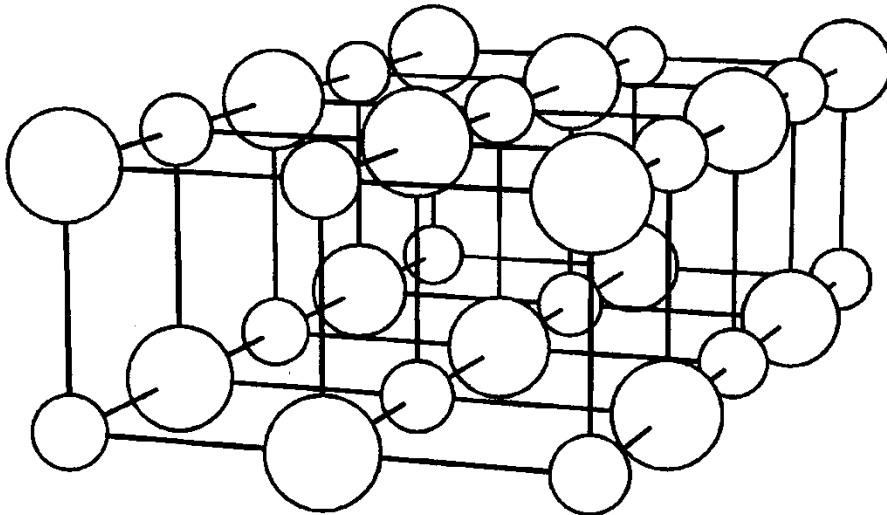


Subsystem $\nu = 2$: TaS_2

$$a_2 = 3.295 \text{ \AA}$$

$$b = 5.775 \text{ \AA}$$

$$c = 23.06 \text{ \AA}$$



Subsystem $\nu = 1$: LaS

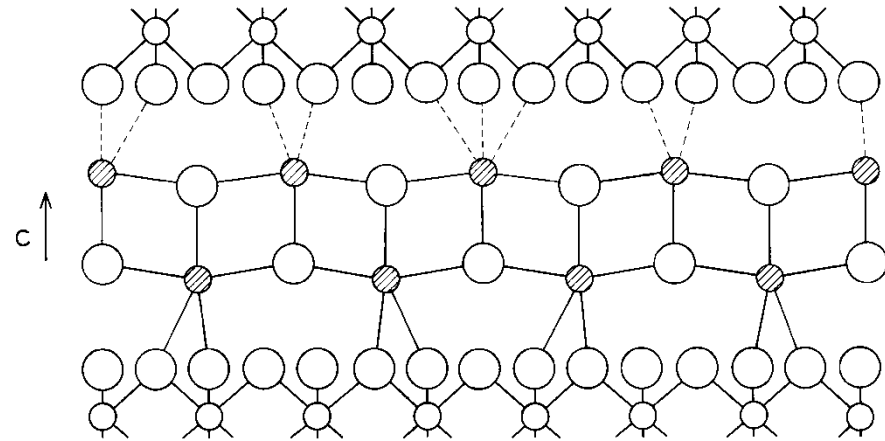
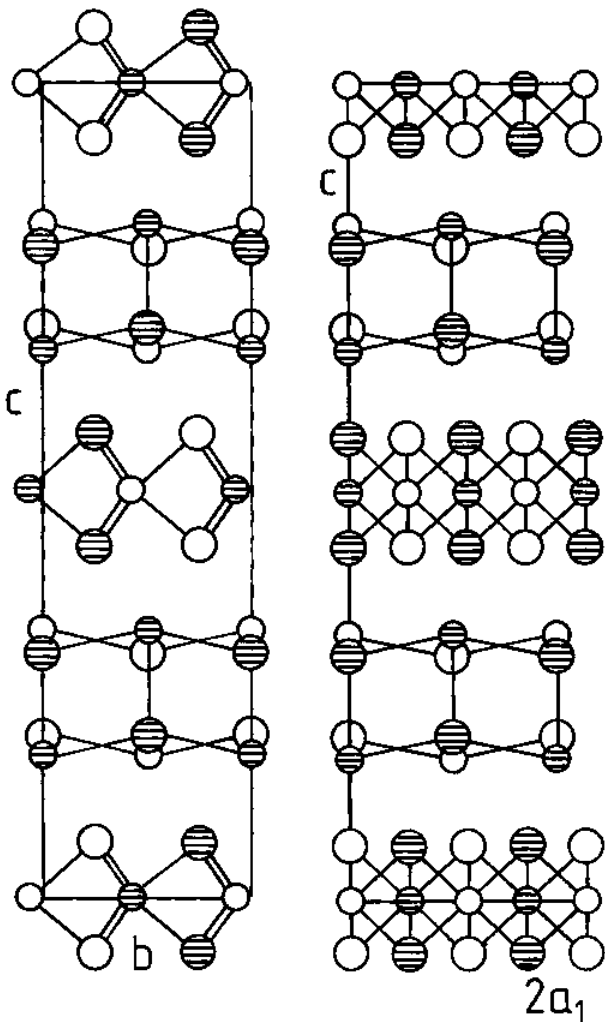
$$a_1 = 5.813 \text{ \AA}$$

$$b = 5.775 \text{ \AA}$$

$$c = 23.06 \text{ \AA}$$

$$\alpha = a_1/a_2 = 0.5668\dots$$

Packing principles of composite crystals

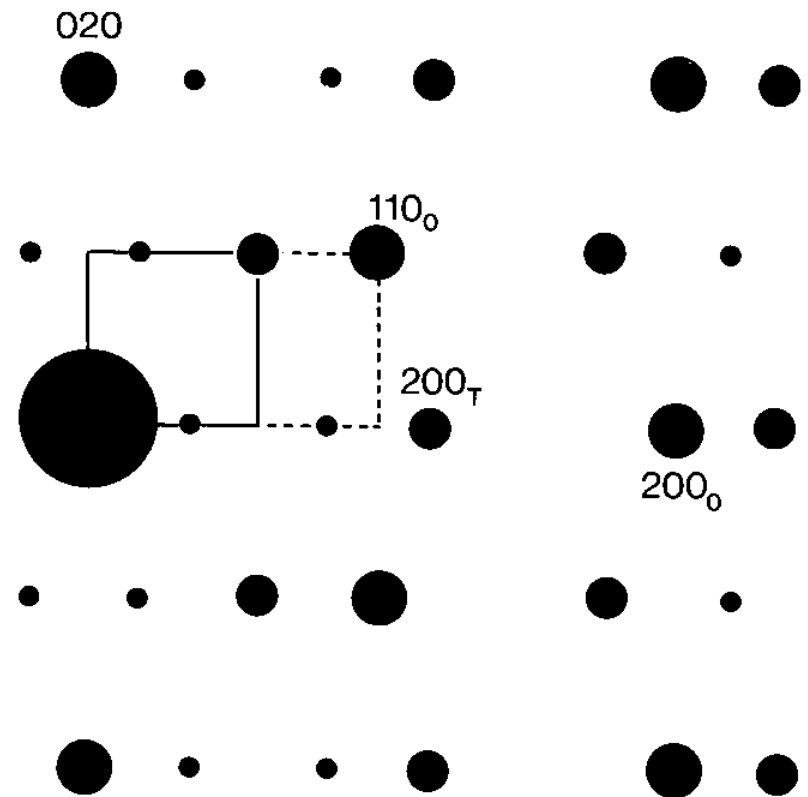
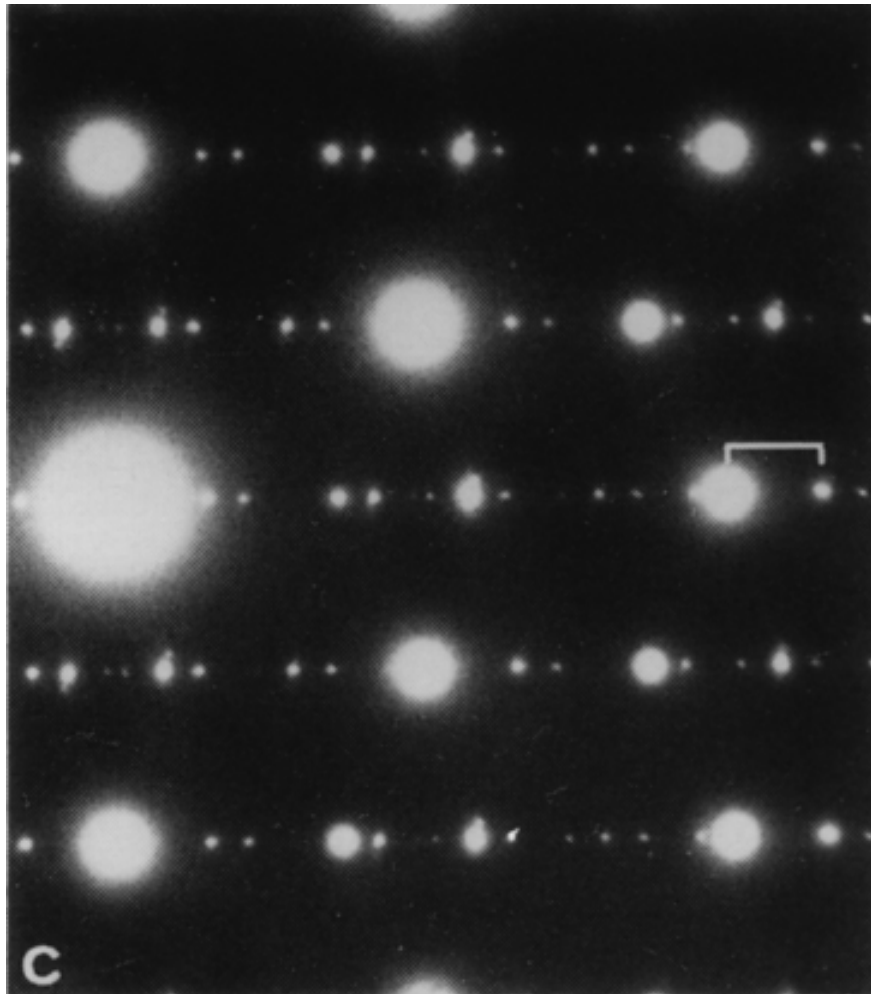


Two unit cells with common \mathbf{b}^* and \mathbf{c}^* axes

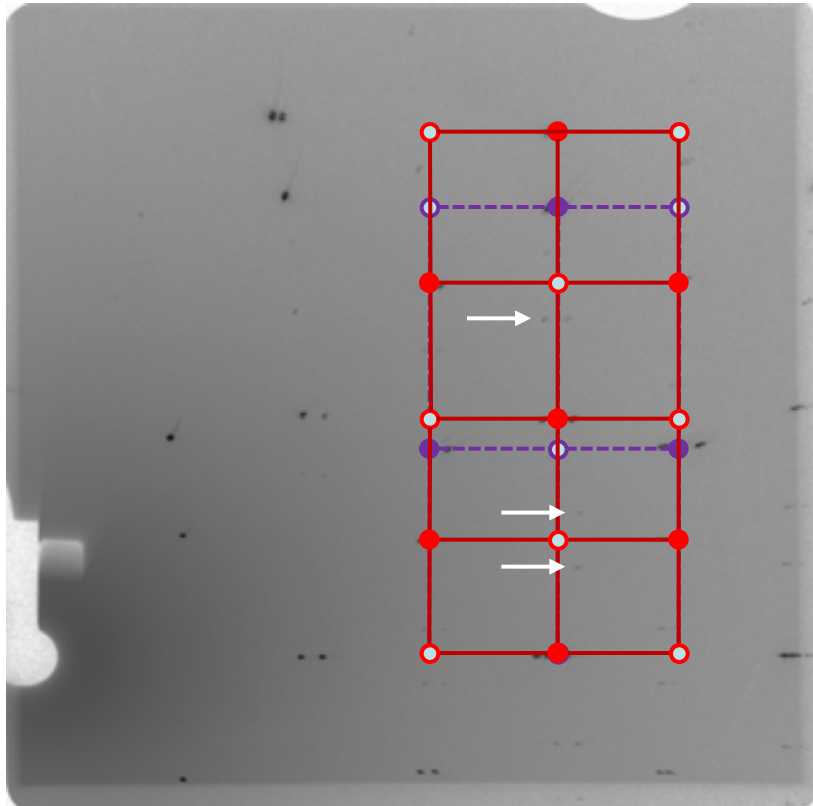
Incommensurate structure in
(3+1)D Superspace

$$a_2 = 1.761 a_1$$

Electron Diffraction by $[\text{SnS}]_{1.17}[\text{NbS}_2]$



Diffraction by composite crystals



Rotation Photograph (20 deg.)
of $[\text{LaS}]_{1.13}[\text{TaS}_2]$

$$h_1 = 2$$

$$h_2 = 3$$

$$h_2 = 2$$

$$h_1 = 1$$

$$h_2 = 1$$

$$h = 0$$

Scattering vector:

$$h \mathbf{a}_1^* + k \mathbf{b}^* + l \mathbf{c}^* + m \mathbf{a}_2^*$$

$$v=1 \text{ LaS: } (h, k, l, 0)$$

$$v=2 \text{ TaS}_2: (0, k, l, m)$$

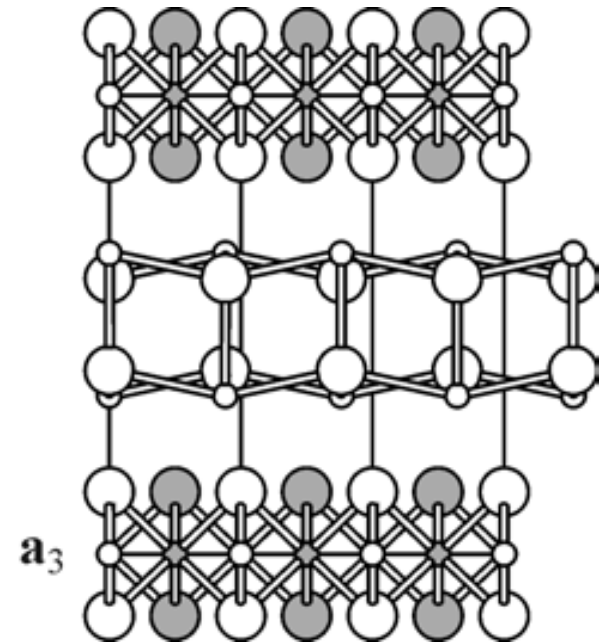
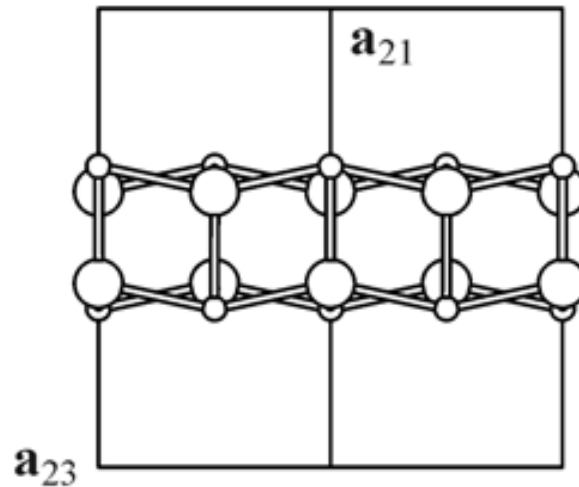
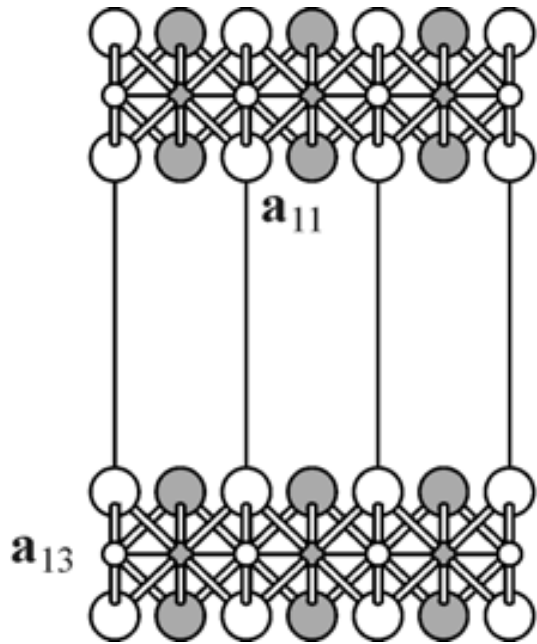
$$\text{Common: } (0, k, l, 0)$$

$$\text{Satellites: } (h, k, l, m)$$

with $h \neq 0$ & $m \neq 0$

Order of satellites:

Minimum $[h, m]$



$$\mathbf{c}^* = \mathbf{a}_{13}^* = \mathbf{a}_{23}^*$$

Perpendicular to layers

$$\mathbf{b}^* = \mathbf{a}_{12}^* = \mathbf{a}_{22}^*$$

Interactions between layers

$$\mathbf{a}_2^* = \mathbf{a}_{21}^* = (a_{11}/a_{21}) \mathbf{a}_{11}^* = \alpha \mathbf{a}_1^* \quad \text{Incommensurate}$$

The W^ν matrix defines subsystem ν

$$\begin{pmatrix} \mathbf{a}_{\nu 1}^* \\ \mathbf{a}_{\nu 2}^* \\ \mathbf{a}_{\nu 3}^* \\ \mathbf{q}^\nu \end{pmatrix} = \begin{pmatrix} W_{11}^\nu & W_{12}^\nu & W_{13}^\nu & W_{14}^\nu \\ W_{21}^\nu & W_{22}^\nu & W_{23}^\nu & W_{24}^\nu \\ W_{31}^\nu & W_{32}^\nu & W_{33}^\nu & W_{34}^\nu \\ W_{41}^\nu & W_{42}^\nu & W_{43}^\nu & W_{44}^\nu \end{pmatrix} \begin{pmatrix} \mathbf{a}_1^* \\ \mathbf{a}_2^* \\ \mathbf{a}_3^* \\ \mathbf{a}_4^* \end{pmatrix}$$

$$W^{\nu=1} = \begin{pmatrix} 1 & 0 & 0 & 0 \\ 0 & 1 & 0 & 0 \\ 0 & 0 & 1 & 0 \\ 0 & 0 & 0 & 1 \end{pmatrix} \quad W^{\nu=2} = \begin{pmatrix} 0 & 0 & 0 & 1 \\ 0 & 1 & 0 & 0 \\ 0 & 0 & 1 & 0 \\ 1 & 0 & 0 & 0 \end{pmatrix}$$

The subsystem superspace group

W^ν is coordinate transformation in superspace

$$\{R_s | \mathbf{v}_s\} \in G_s \quad \begin{cases} R_s^\nu = W^\nu R_s (W^\nu)^{-1} \\ \mathbf{v}_{\nu s} = W^\nu \mathbf{v}_s \end{cases}$$

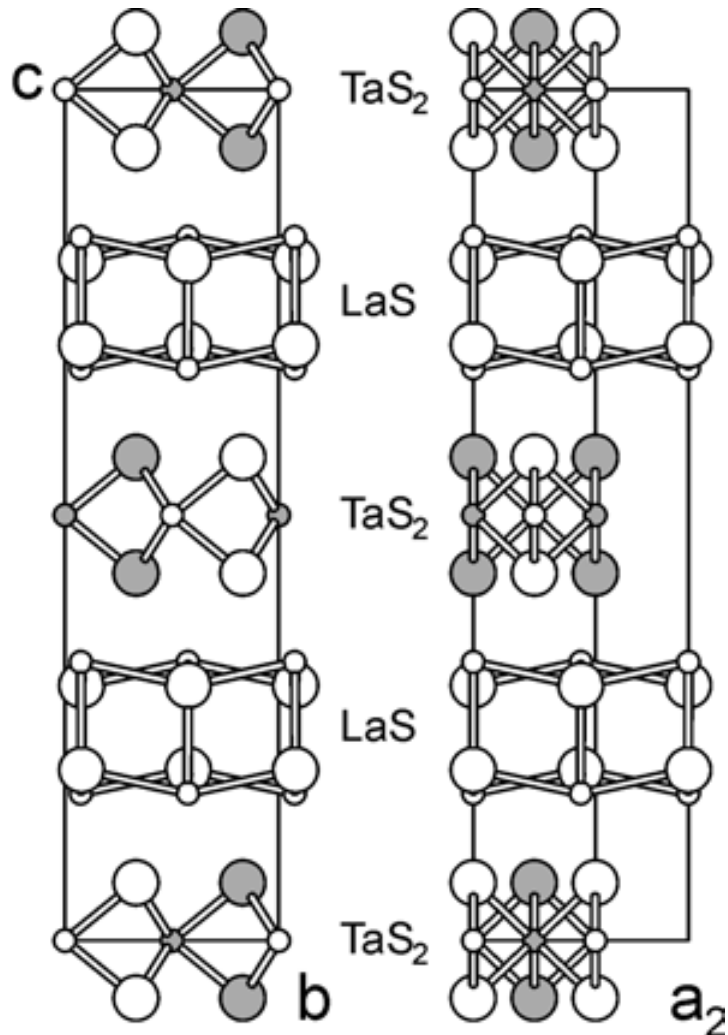
$\{R_s^\nu | \mathbf{v}_{\nu s}\} \in G_s^\nu$ subsystem superspace group

Subsystem superspace group gives symmetry of subsystem ν

Symmetry of the periodic basic structure of subsystem ν by

$\{R^\nu | \mathbf{v}_{\nu,1}, \mathbf{v}_{\nu,2}, \mathbf{v}_{\nu,3}\} \in G^\nu$ subsystem spacegroup

Unit cells of subsystems of $[\text{LaS}]_{1.13}[\text{TaS}_2]$



Subsystem $\nu = 1$: TaS_2

F-centered

$$a_{13} = 23.06 \text{ \AA}$$

Subsystem $\nu = 2$: LaS

C-centered

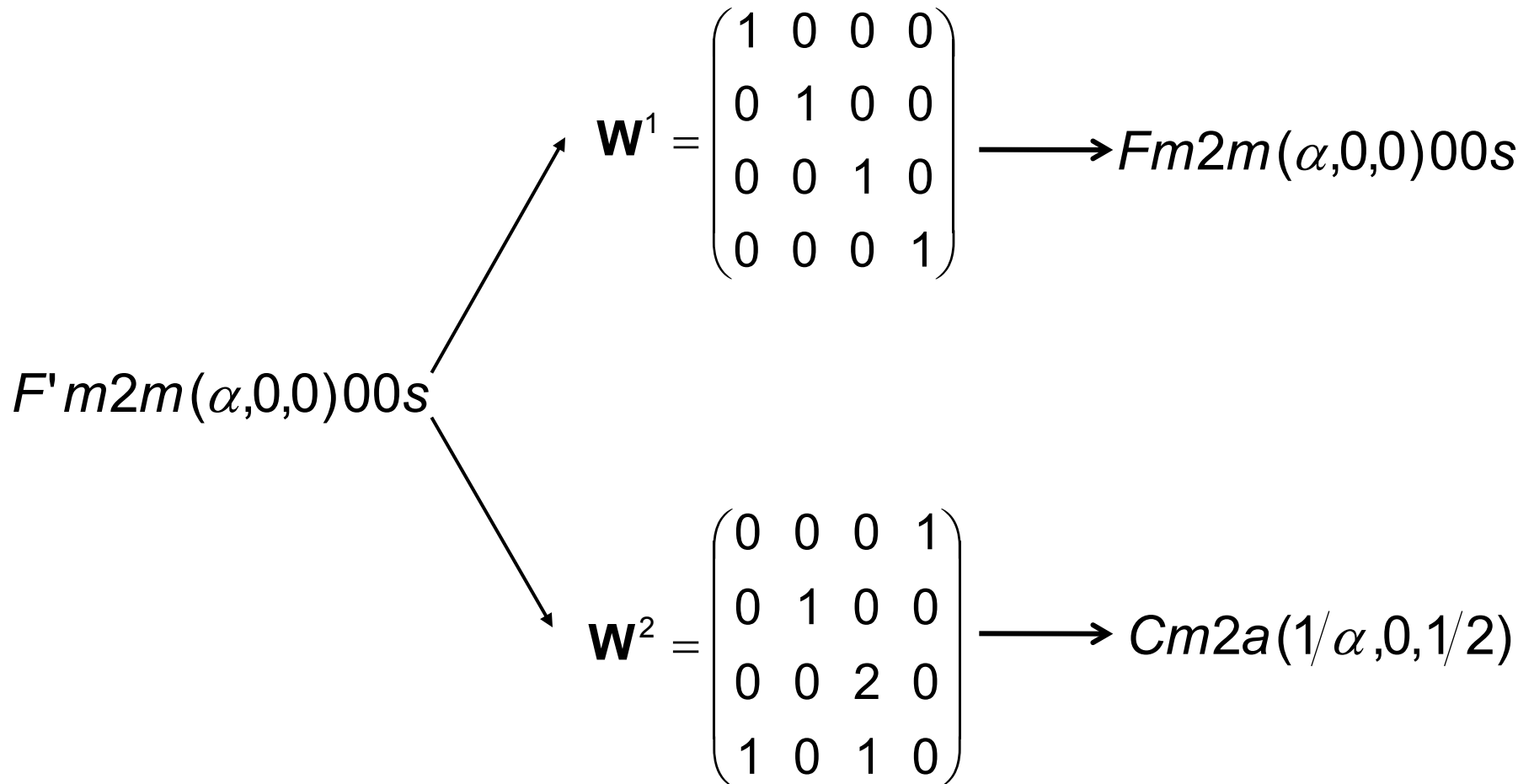
$$a_{23} = (1/2) 23.06 = 11.58 \text{ \AA}$$

$$(0, k_1, l_1, 0) : k_1, l_1 \text{ even}$$

$$(0, k_2, l_2, 0) : k_2 \text{ even AND}$$

$$l_1 = 2l_2 \text{ due to } a_{23}^* = 2a_{13}^*$$

Non-equivalent subsystem superspace groups



Exercise: Subsystem superspace groups of monoclinic $[\text{PbS}]_{1.18}[\text{TiS}_2]$

SSG: $C_c 2/m(\alpha 0 0)s_0$

$\alpha = a_{11}/a_{21} = 0.5878$ origin at i

$C_c = (1/2, 1/2, 0, 1/2)$

Subsystem TiS_2 ($\nu = 1$)

Subsystem PbS ($\nu = 2$)

$a_{11} = 3.409 \text{ \AA}$

$a_{21} = 5.800 \text{ \AA}$

$a_{12} = 5.880 \text{ \AA}$

$a_{22} = 5.881 \text{ \AA}$

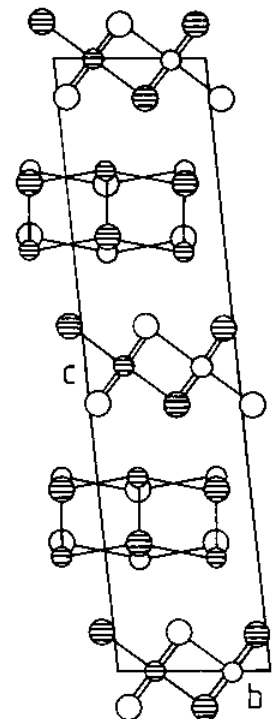
$a_{13} = 11.760 \text{ \AA}$

$a_{23} = 11.759 \text{ \AA}$

$\alpha_1 = 95.29^\circ$

$\alpha_2 = 95.27^\circ$

$$\mathbf{W}^1 = \begin{pmatrix} 1 & 0 & 0 & 0 \\ 0 & 1 & 0 & 0 \\ 0 & 0 & 1 & 0 \\ 0 & 0 & 0 & 1 \end{pmatrix} \begin{cases} R_s^\nu = \mathbf{W}^\nu R_s (\mathbf{W}^\nu)^{-1} \\ \mathbf{v}_{\nu s} = \mathbf{W}^\nu \mathbf{v}_s \end{cases} \quad \mathbf{W}^2 = \begin{pmatrix} 0 & 0 & 0 & 1 \\ 0 & 1 & 0 & 0 \\ 0 & 0 & 1 & 0 \\ 1 & 0 & 0 & 0 \end{pmatrix}$$



Solution: Subsystem TiS_2 ($\nu = 1$)

$G_s^1 = G_s = C_2/m(\alpha 0 0)s_0$ because $W^1 =$ identity matrix
 mirror plane has an origin-dependent translation

Subsystem space group

$$G_2 = C_2/m$$

$$\begin{cases} R_s^\nu = W^\nu R_s (W^\nu)^{-1} \\ \mathbf{v}_{\nu s} = W^\nu \mathbf{v}_s \end{cases}$$

$$\left\{ \begin{array}{ll} (E, 0) & \{E, 1 | l_1, l_2, l_3, l_4\} \\ \mathbf{ct} & \{E, 1 | 1/2, 1/2, 0, 1/2\} \\ (2, s) & \{2^x, 1 | 0, 0, 0, 1/2\} \\ (i, \bar{1}) & \{i, \bar{1} | 0, 0, 0, 0\} \\ (m, \bar{1}) & \{m_x, \bar{1} | 0, 0, 0, 1/2\} \end{array} \right.$$

Solution: Subsystem PbS ($\nu = 2$)

$G_s = C_{2/m}(\alpha \ 0 \ 0)_{s0}$. Rotation matrices of all four operators are diagonal matrices, so:

$$R_s^\nu = W^\nu R_s (W^\nu)^{-1} = R_s W^\nu (W^\nu)^{-1} = R_s$$

$$W^2 = \begin{pmatrix} 0 & 0 & 0 & 1 \\ 0 & 1 & 0 & 0 \\ 0 & 0 & 1 & 0 \\ 1 & 0 & 0 & 0 \end{pmatrix}$$

W^2 applied to \mathbf{v}_s interchanges

the 1st and 4th coordinates:

$$\mathbf{v}_{2s} = W^2 \mathbf{v}_s = (v_{s4}, v_{s2}, v_{s3}, v_{s1})$$

$$\mathbf{q}^2 = \mathbf{a}_{11}^* = (1/\alpha) \mathbf{a}_{21}^*$$

Subsystem superspace group

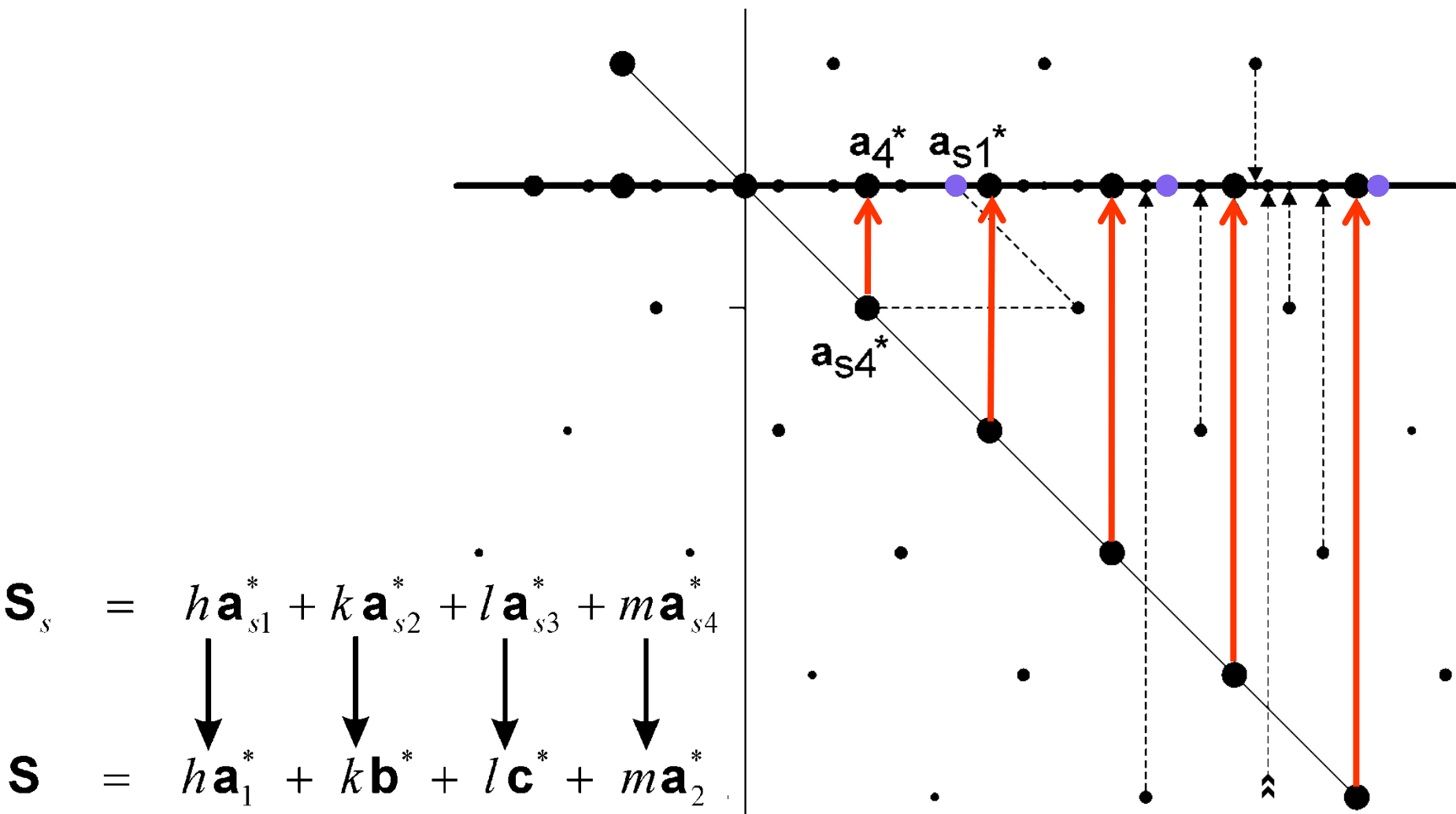
$$G_s^2 = C_{2_1/m}(\alpha' \ 0 \ 0)_{00} \text{ with } \alpha' = 1/\alpha$$

Subsystem space group

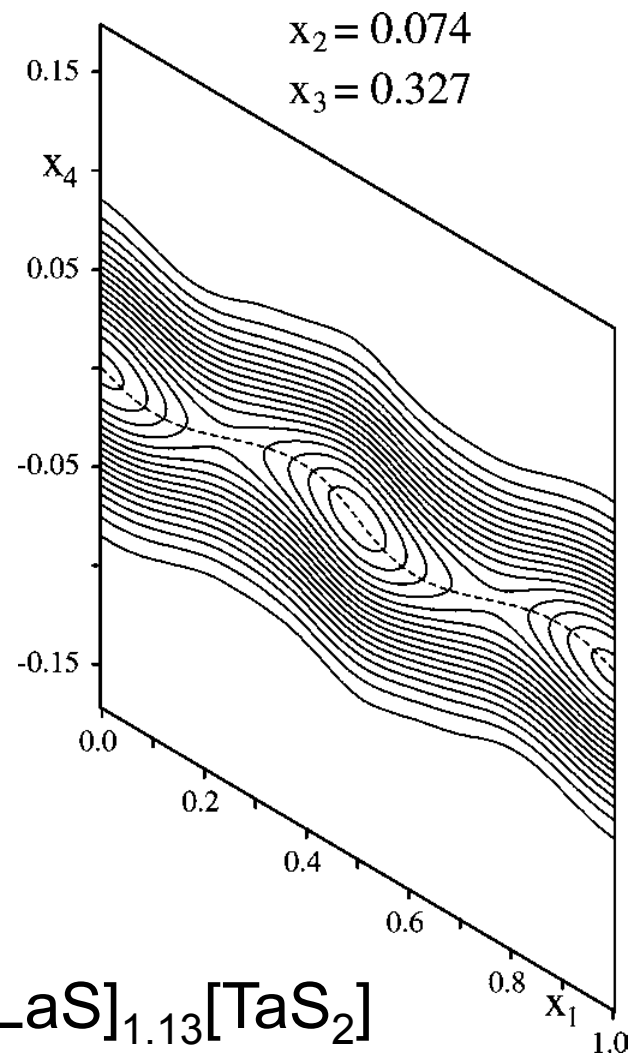
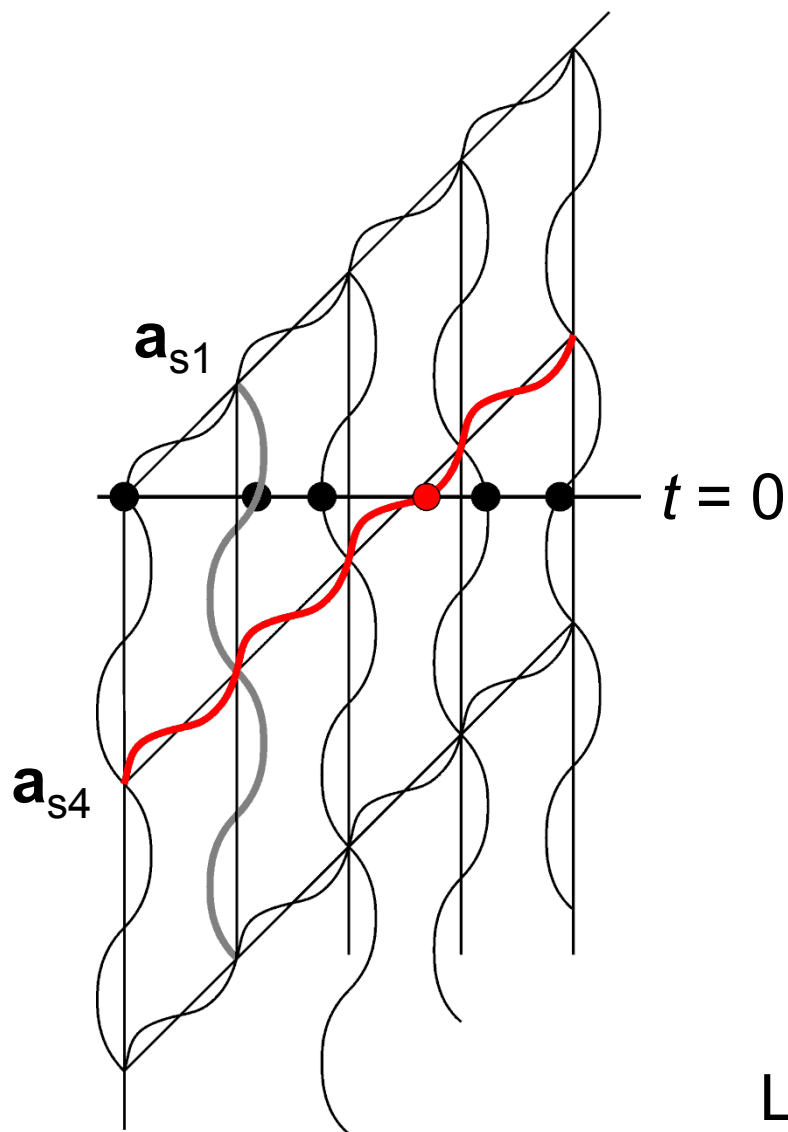
$$G_2 = C_{2_1/m}$$

$$\left\{ \begin{array}{ll} (E, 0) & \{E, 1 | l_1, l_2, l_3, l_4\} \\ \mathbf{ct} & \{E, 1 | 1/2, 1/2, 0, 1/2\} \\ (2_1, 0) & \{2^x, 1 | 1/2, 0, 0, 0\} \\ (i, \bar{1}) & \{i, \bar{1} | 0, 0, 0, 0\} \\ (m, \bar{1}) & \{m_x, \bar{1} | 1/2, 0, 0, 0\} \end{array} \right.$$

Reciprocal superspace of composite crystals



Modulation functions in superspace



La atom of $[\text{LaS}]_{1.13}[\text{TaS}_2]$

Structural parameters for composite crystals

Coordinates are with respect to the subsystem lattices

$$\mathbf{x}_{vi}^{\mu} = l_{vi} + \mathbf{x}_{vi}^{\mu 0} + u_{vi}^{\mu} [t_v + \mathbf{q}^v \cdot (\mathbf{L}_v + \mathbf{x}_v^{\mu 0})]$$

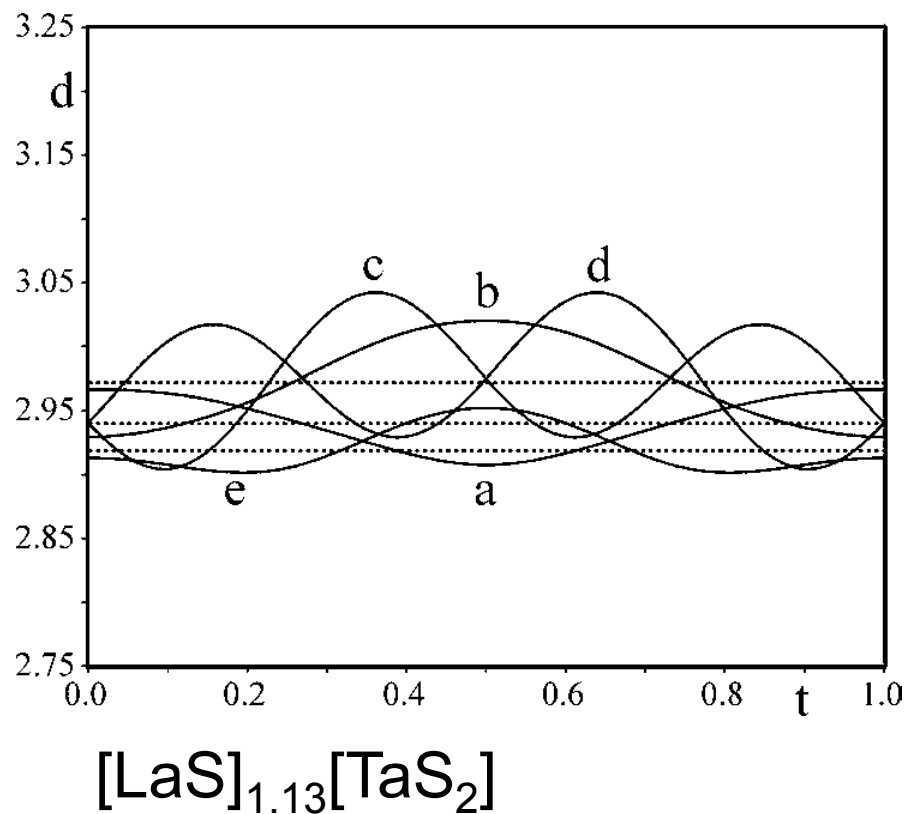
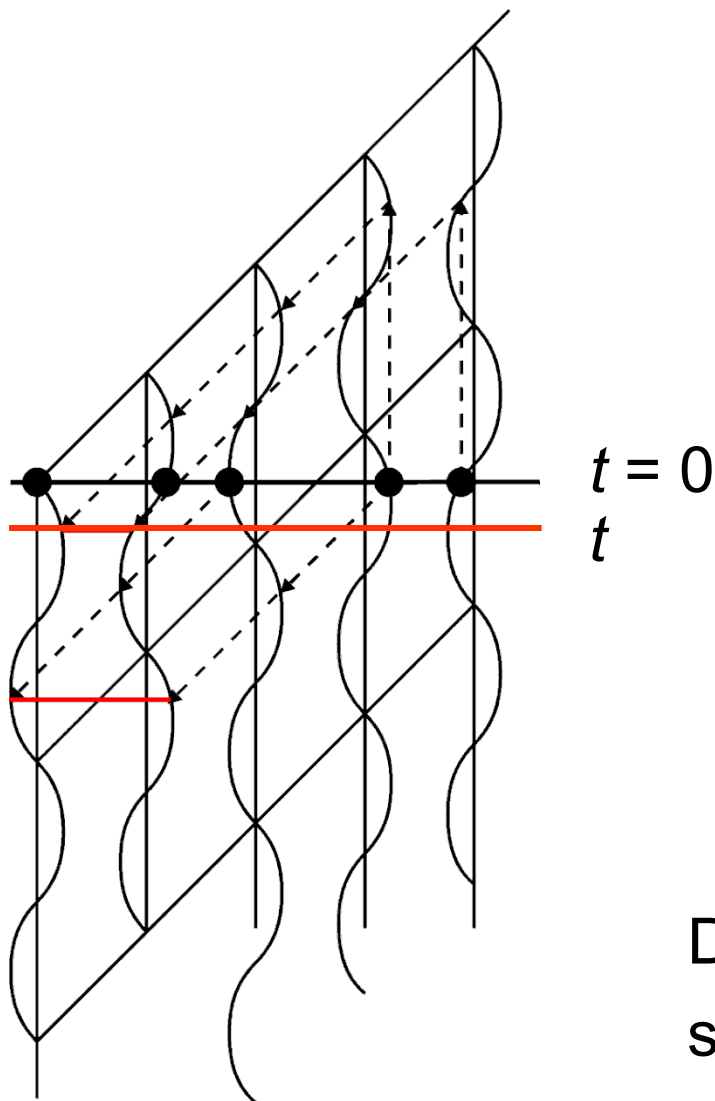
$$u_{vi}^{\mu}(\bar{x}_{vs4}) = \sum_{n=1}^{\infty} A_{vni}^{\mu} \sin(2\pi n \bar{x}_{vs4}) + B_{vni}^{\mu} \cos(2\pi n \bar{x}_{vs4})$$

$$\bar{x}_{vs4} = t_v + \mathbf{q}^v \cdot (\mathbf{L}_v + \mathbf{x}_v^{\mu 0})$$

$$\mathbf{q}^{v=1} = \mathbf{a}_2^* = 0.5668 \mathbf{a}_1^* \quad \mathbf{q}^{v=2} = \mathbf{a}_1^* = 1.764 \mathbf{a}_2^*$$

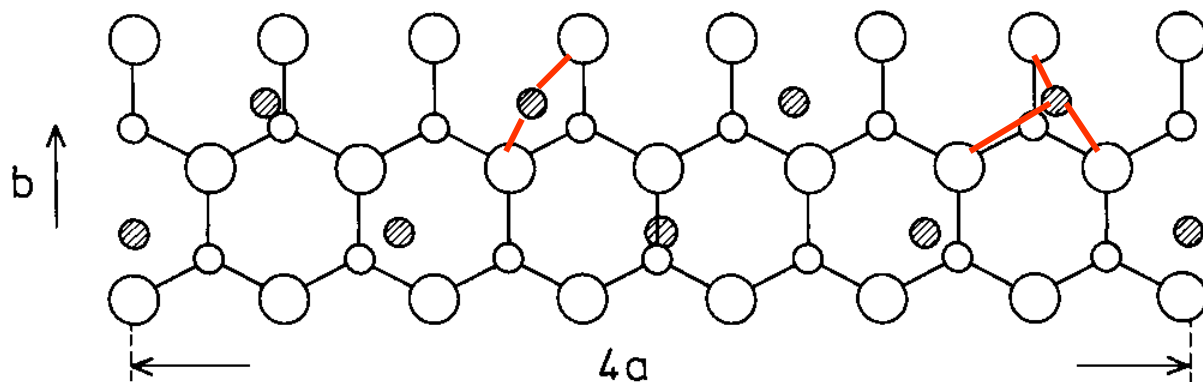
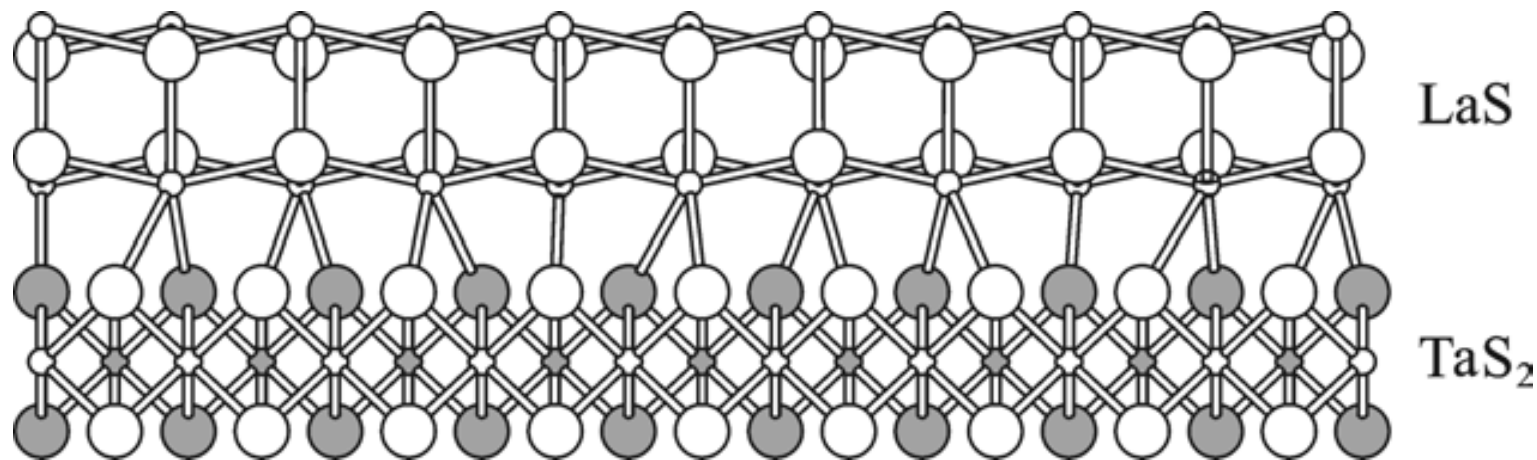
Relations between subsystems via \mathbf{W}^v -matrices allows computation of structure factor, distances and others

Superspace and interatomic distances: t plots

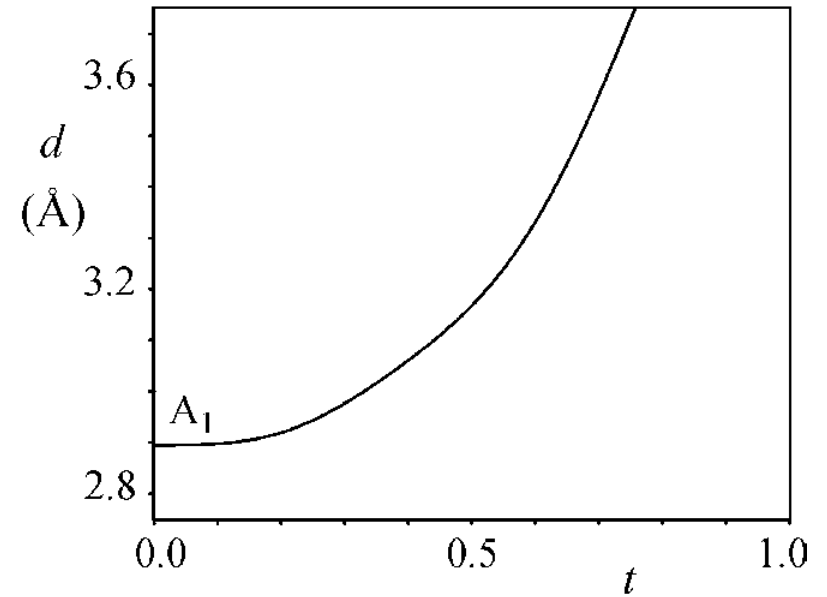
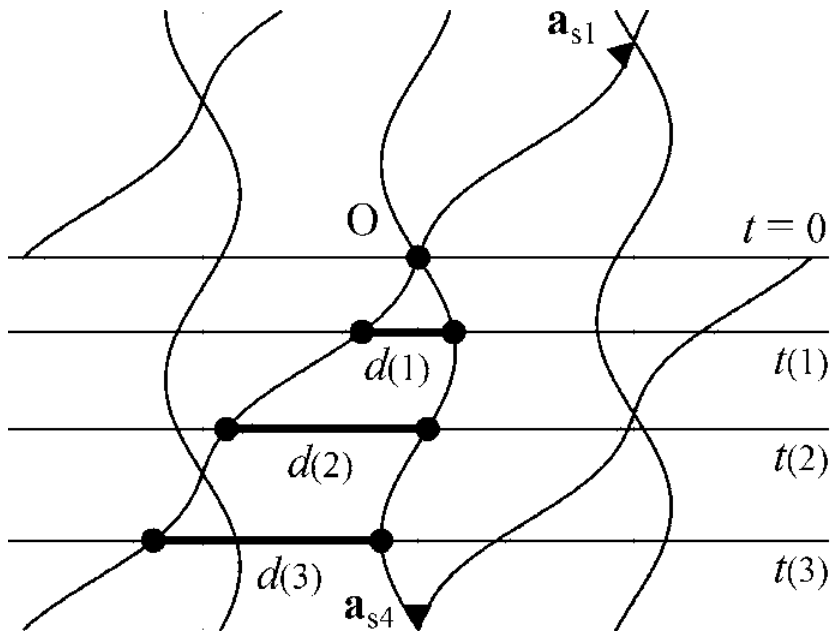


Distances of La towards the five surrounding S of the LaS subsystem

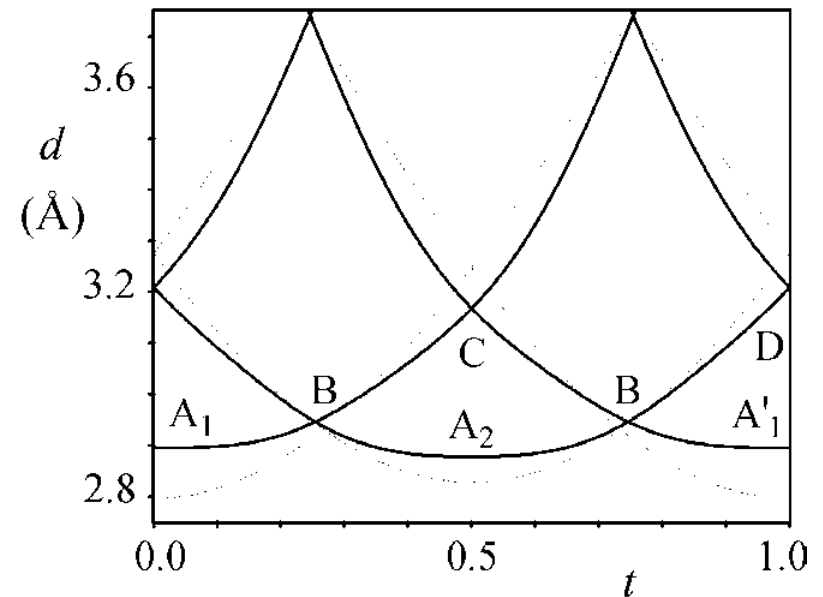
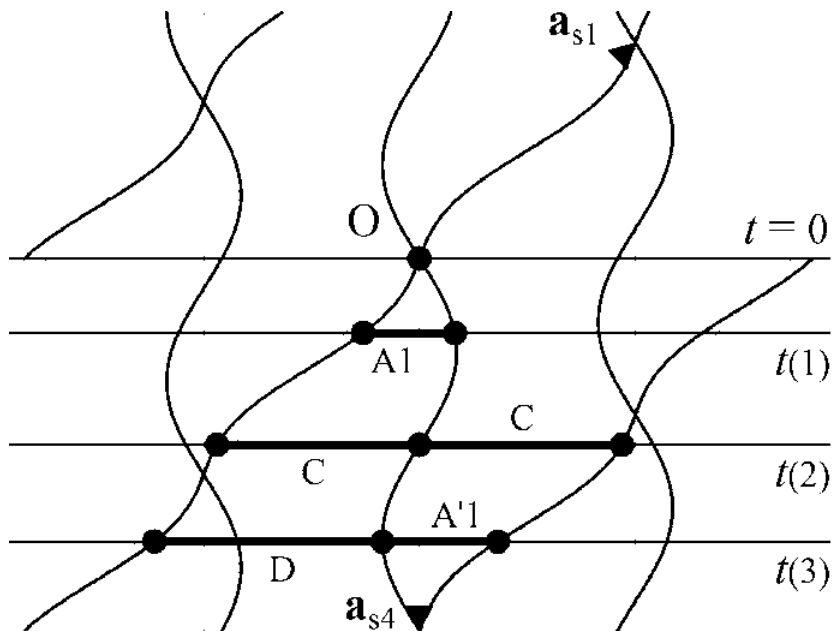
Chemical bonding across the incommensurate gap



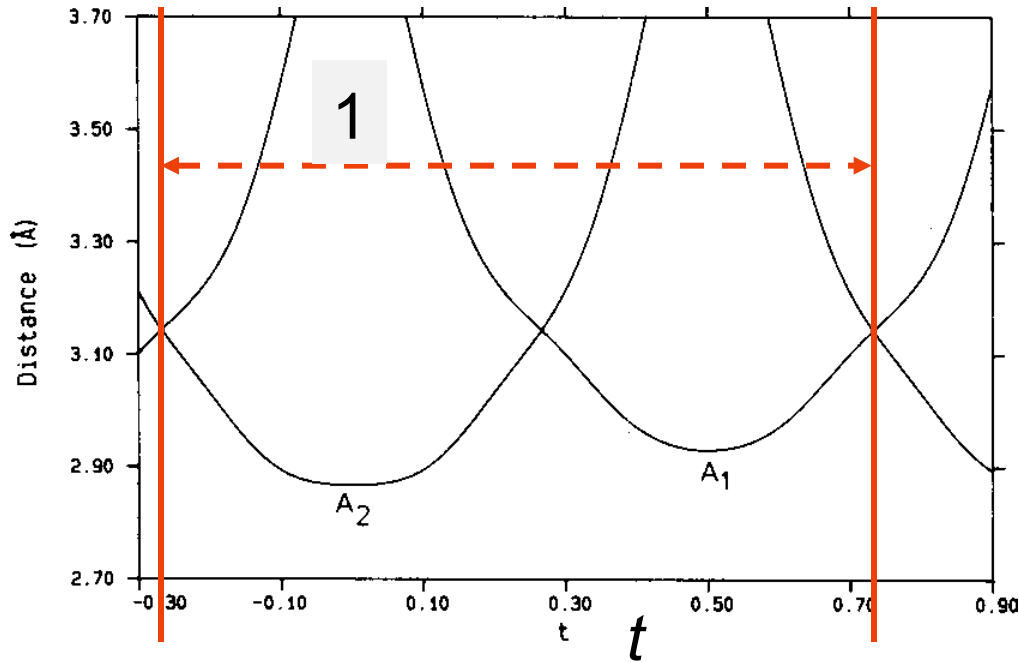
Interatomic distance between
one atom of $\nu = 1$ and one atom of $\nu = 2$.



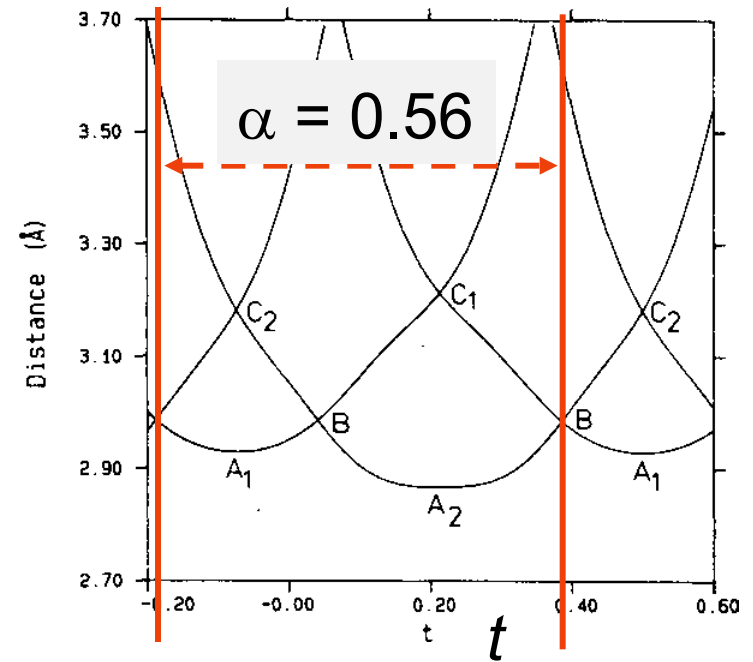
La of $[\text{LaS}]_{1.13}[\text{TaS}_2]$ as central atom
 Distances to all S atoms of $\nu = 2$
 gives periodic function in t



[LaS]_{1.14}[NbS₂]: distances S1 ($\nu=1$) – La ($\nu=2$)

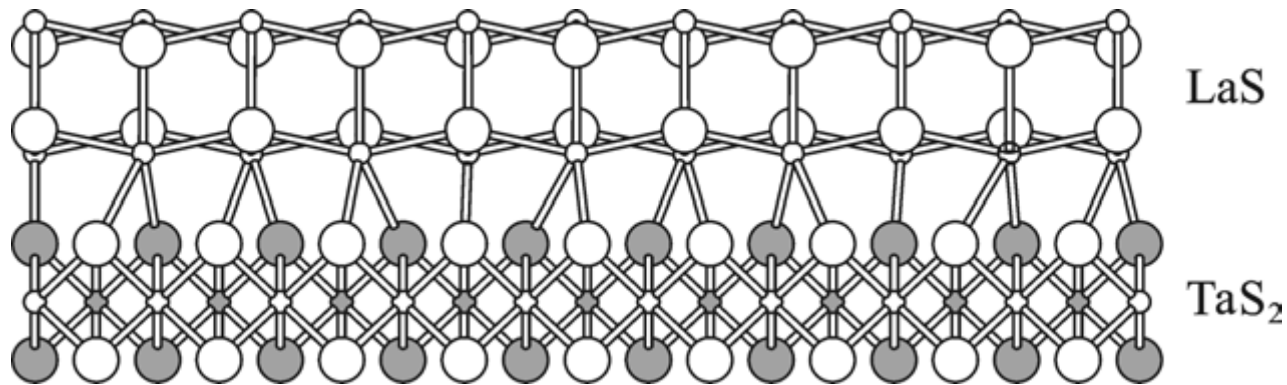
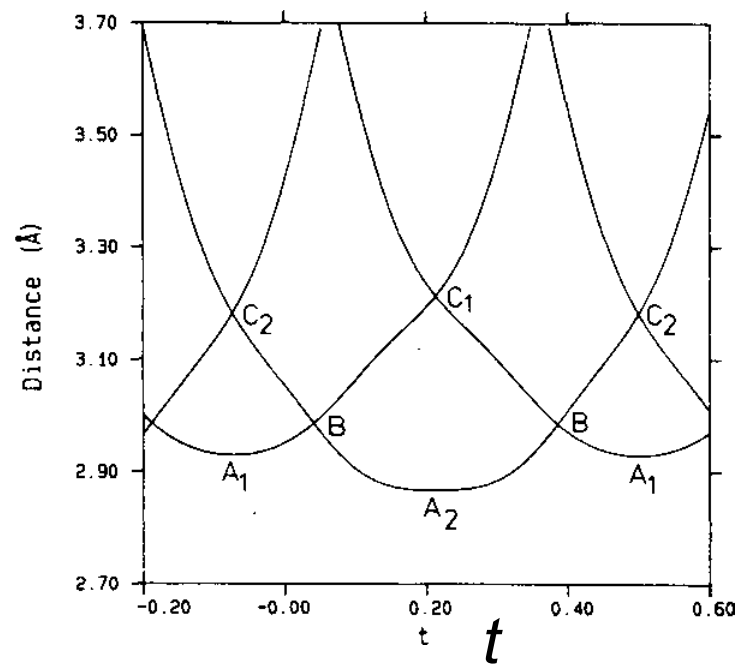
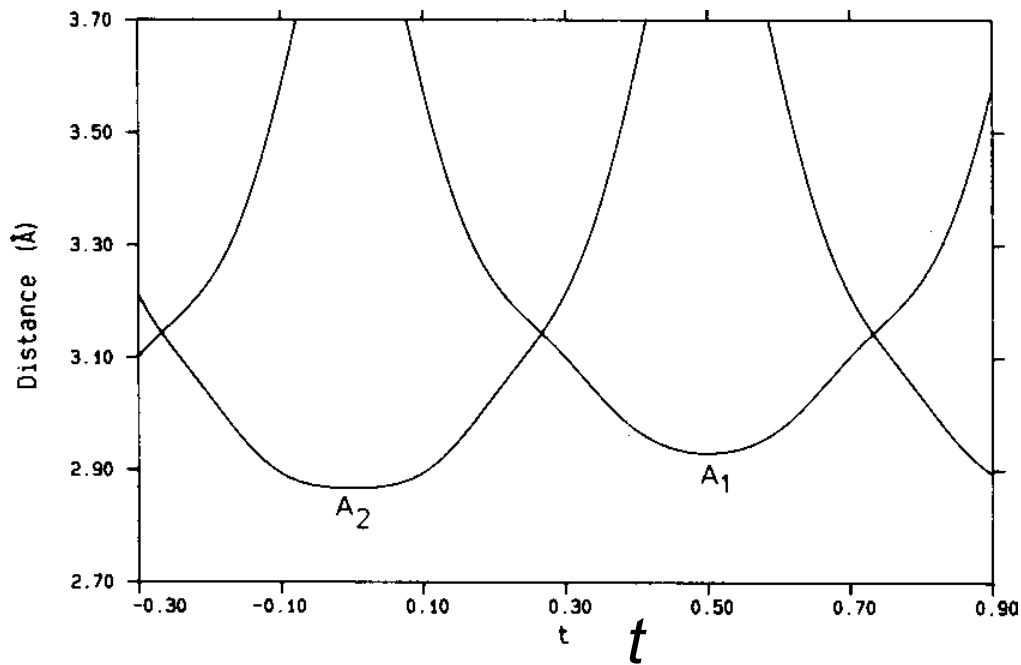


Central atom S1 ($\nu = 1$)
 Periodicity 1 in $t_1 = t$



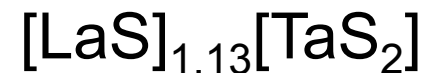
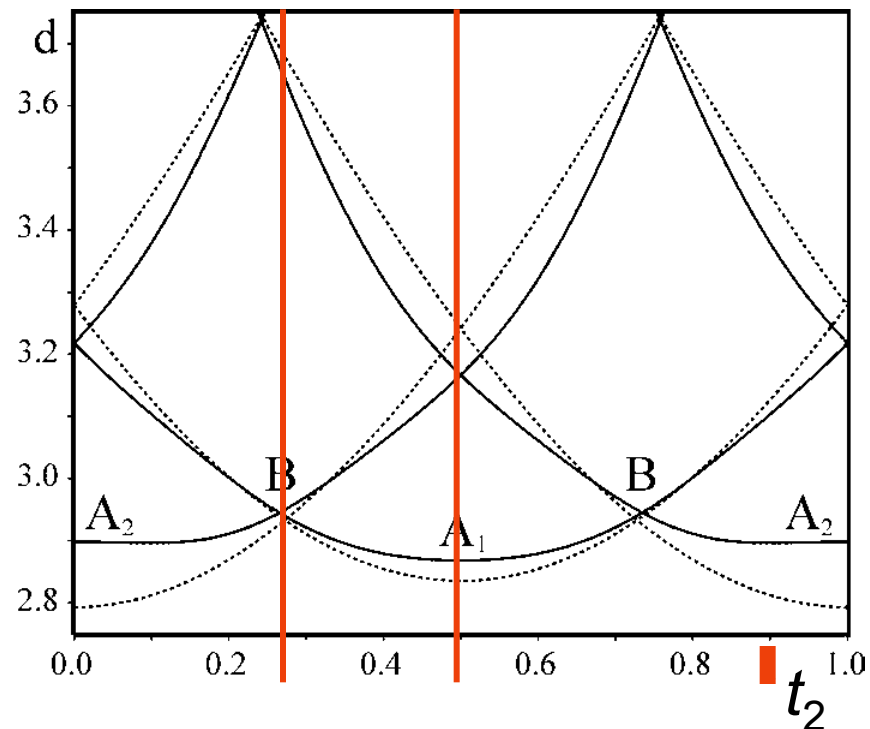
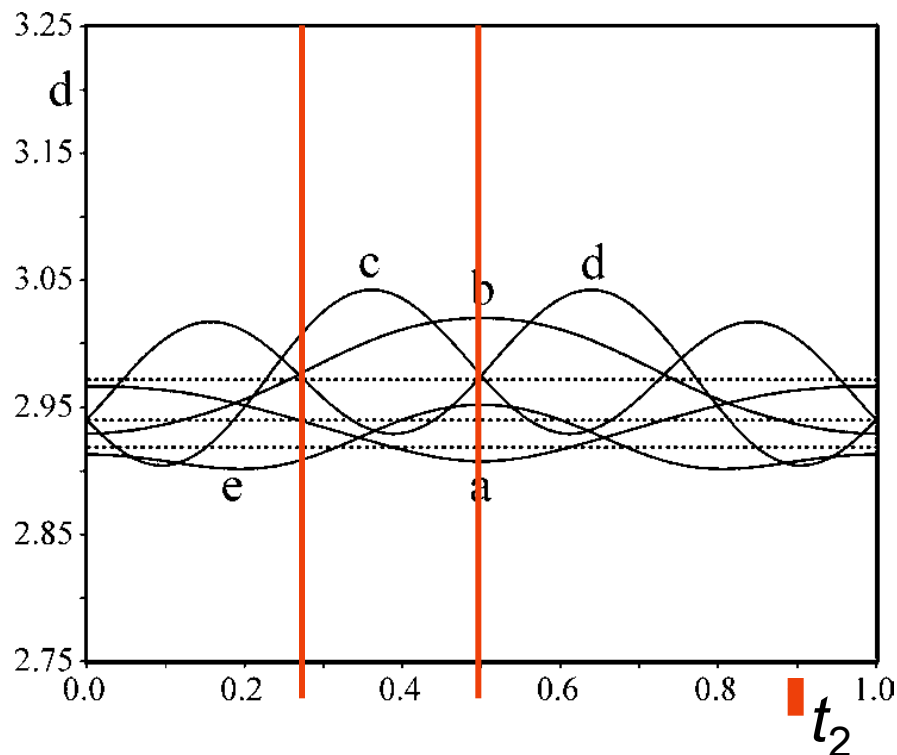
Central atom La ($\nu = 2$)
 Periodicity 1 in $t_2 = t/\alpha$
 Periodicity α in $t = \alpha t_2$

Distance S1 ($\nu=1$)–La ($\nu=2$) in physical space

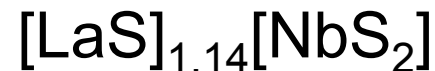
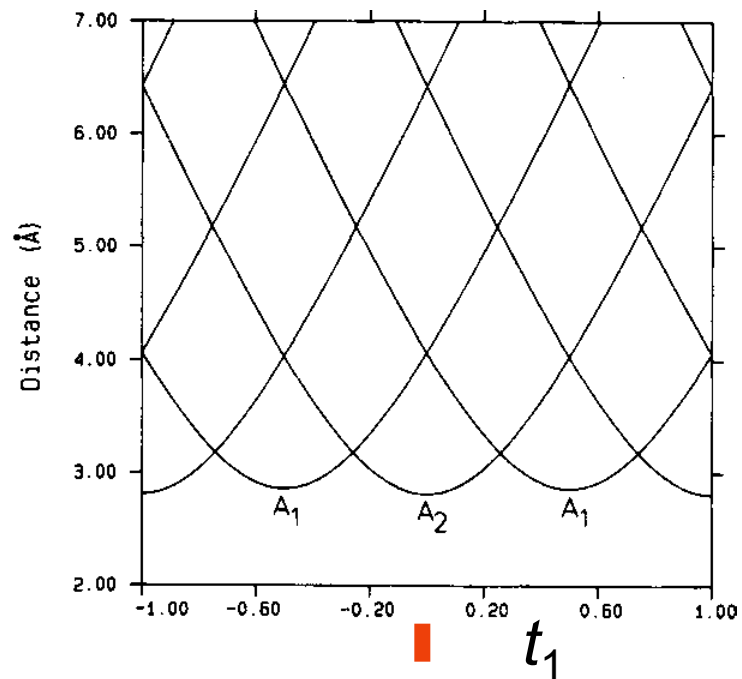
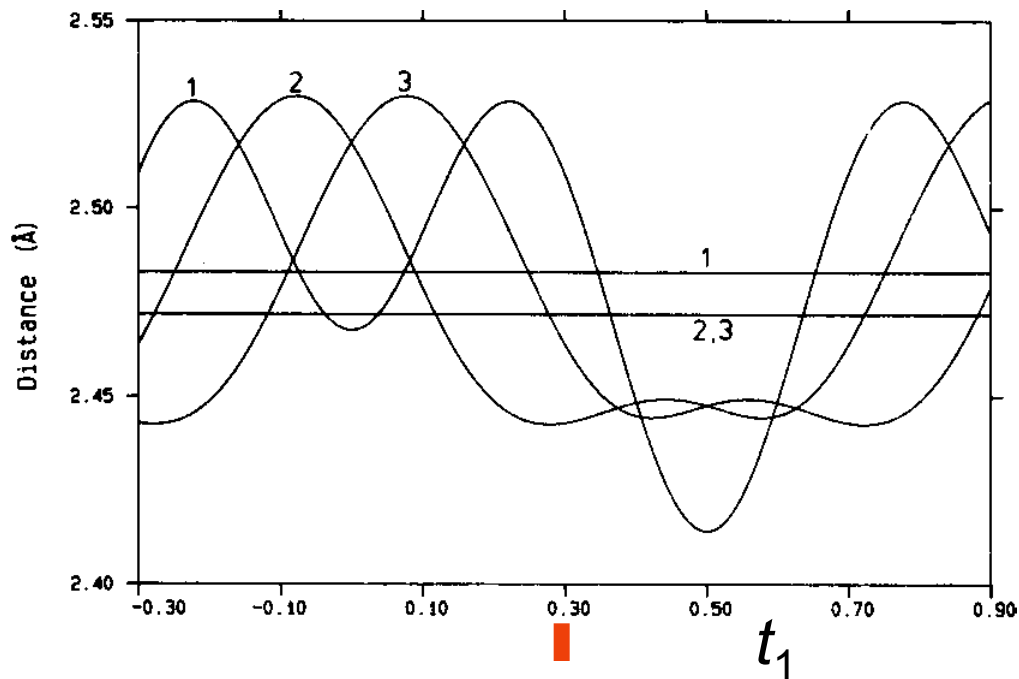


Central atom La of subsystem $\nu = 2$: LaS

Distances to atoms S1 ($\nu = 1$) and S2 ($\nu = 2$)



Central atom S1 of subsystem $\nu = 1$: TaS₂ Distances to atoms La ($\nu = 2$) and Nb ($\nu = 1$)



Summary

Composite crystals are a single thermodynamic phase

Comprise modulated subsystems $\nu = 1, 2, \dots$

Subsystem superspace groups follow by W^ν matrices from the single $(3+d)D$ superspace group

t plots of distances and other structural parameters



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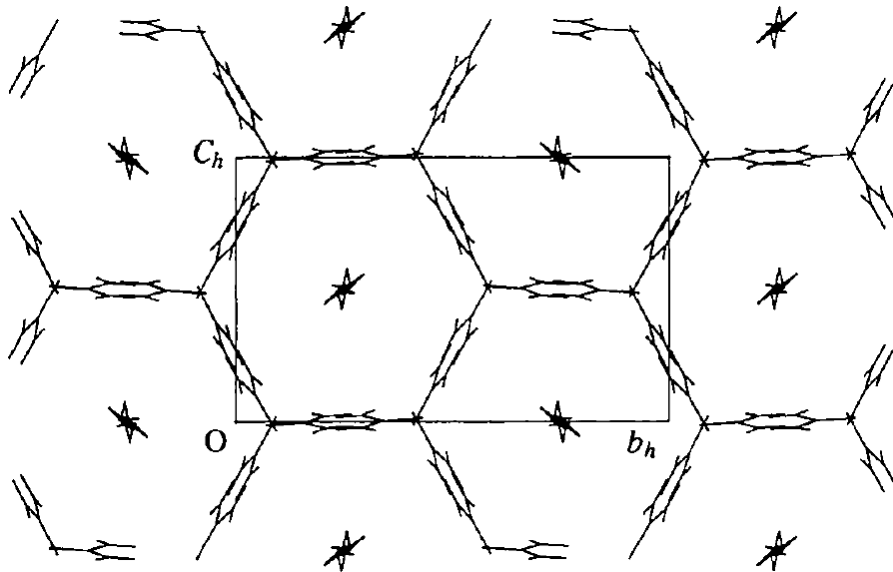
Incommensurate composite crystals materials and properties

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Laboratory of Crystallography

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Channel-type incommensurate composite crystals—urea/alkane inclusion compounds



(3+1)D superspace group
 $P6_122(00\gamma)000$ — orthorhombic
distortion for ordered guest

Subsystem 1 urea

$P6_122$

$a = 8.24 \text{ \AA}$

$c = 11.05 \text{ \AA}$

Subsystem 2 n-alkane

"disordered"

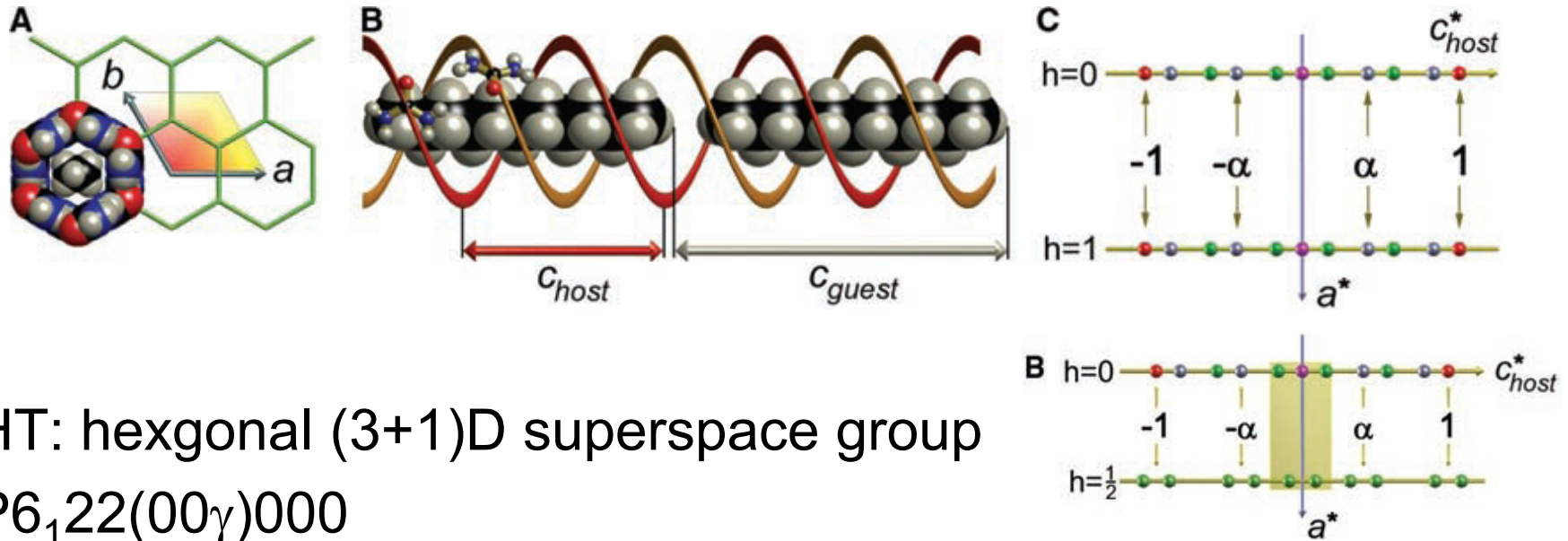
common $(\mathbf{a}^*, \mathbf{b}^*)$ -plane

collinear \mathbf{c} -axes

incommensurability

$c_{\text{guest}}/c_{\text{host}} = \text{irrational}$

Urea/alkane inclusion compounds



HT: hexagonal (3+1)D superspace group

$P6_122(00\gamma)000$

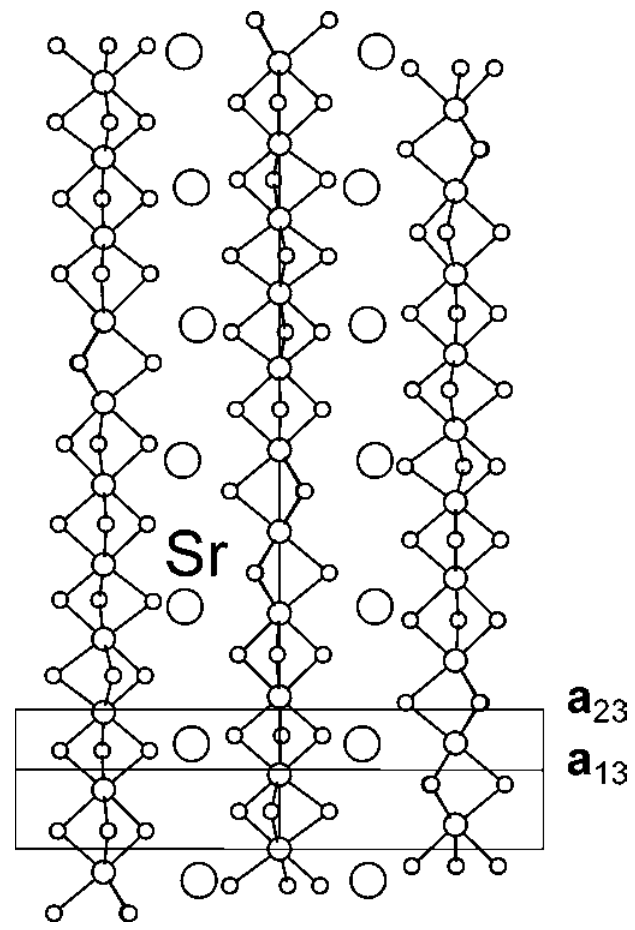
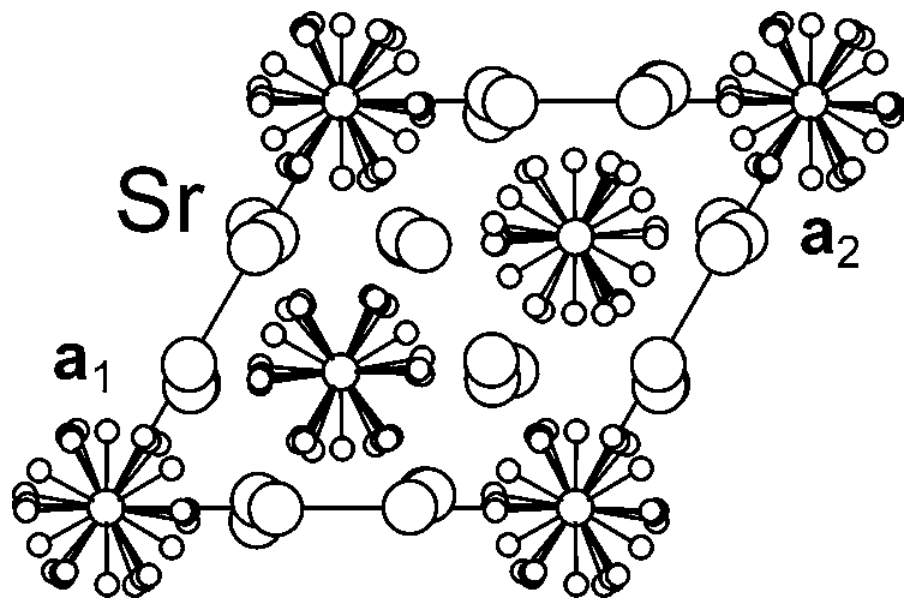
$$\mathbf{H} = h \mathbf{a}^* + k \mathbf{b}^* + l \mathbf{c}_{host}^* + m \mathbf{c}_{guest}^*$$

LT: orthohexagonal supercell $C222_1$

$$\mathbf{H} = h \mathbf{a}^* + k \mathbf{b}^* + l \mathbf{c}_{host}^* + m_1 \mathbf{c}_{guest}^* + m_2 \mathbf{q}^2$$

$$\mathbf{q}^2 = (1, 0, \gamma')$$

Columnar type composite crystals— $[\text{Sr}]_x[\text{TiS}_3]$

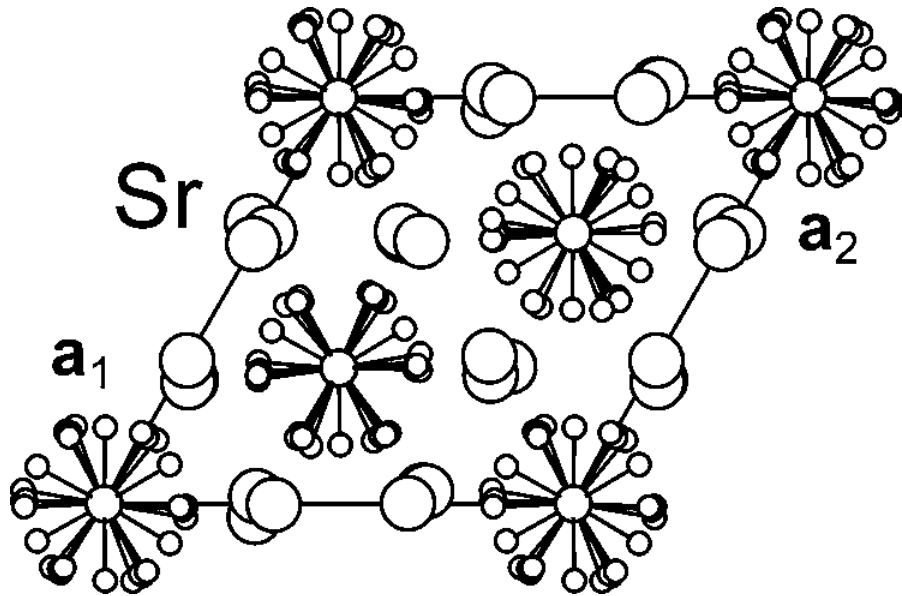


(3+1)D superspace group

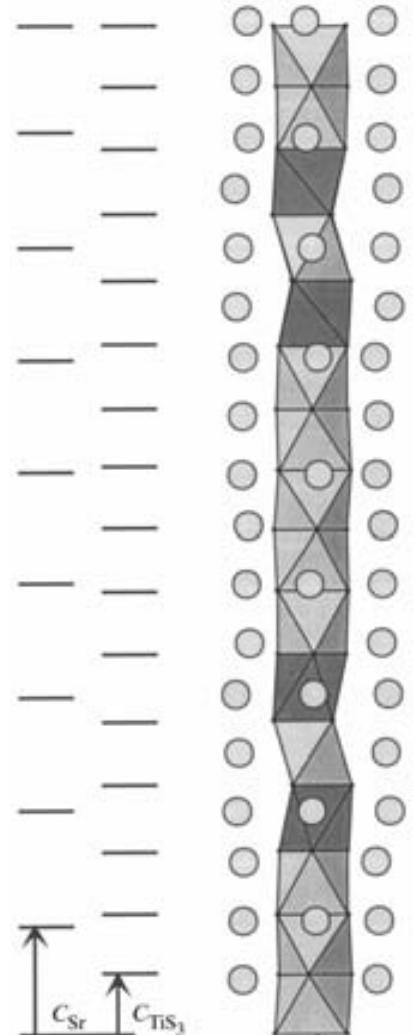
$R3m(00\gamma)0s$

$$\mathbf{H} = h \mathbf{a}^* + k \mathbf{b}^* + l \mathbf{c}_{\text{host}}^* + m \mathbf{c}_{\text{guest}}^*$$

Atomic modulations in $[\text{Sr}]_x[\text{TiS}_3]$ ($x \approx 1.12$)



Modulation of Sr atomic positions.
Aperiodic sequence of octahedral (short)
and trigonal prismatic (long), face-sharing
TiS₆ groups



Rhombohedral / trigonal composite crystal

$[\text{Sr}]_x[\text{TiS}_3]$ ($x \approx 1.12$)

$$G_s = R\bar{3}m(00\gamma)0s$$

$$(h k l m): -h + k + l = 3n$$

$$\begin{cases} \mathbf{ct}_1 & \{E, 1 | 2/3, 1/3, 1/3, 0\} \\ \mathbf{ct}_2 & \{E, 1 | 1/3, 2/3, 2/3, 0\} \end{cases}$$

$$(h_2 k_2 l_2 m_2): -h_2 + k_2 + m_2 = 3n$$

$$\begin{cases} \mathbf{ct}_1 & \{E, 1 | 2/3, 1/3, 0, 1/3\} \\ \mathbf{ct}_2 & \{E, 1 | 1/3, 2/3, 0, 2/3\} \end{cases}$$

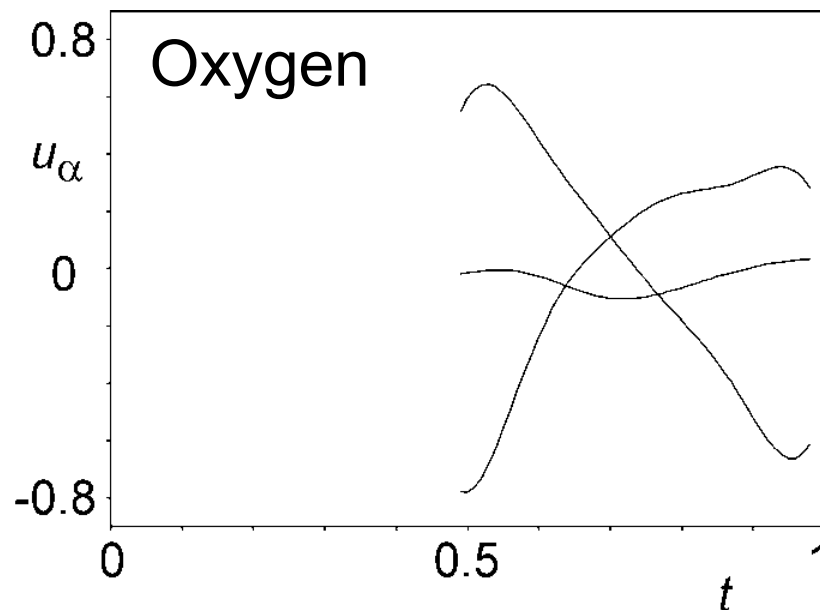
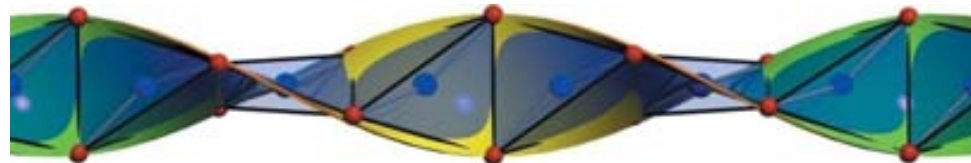
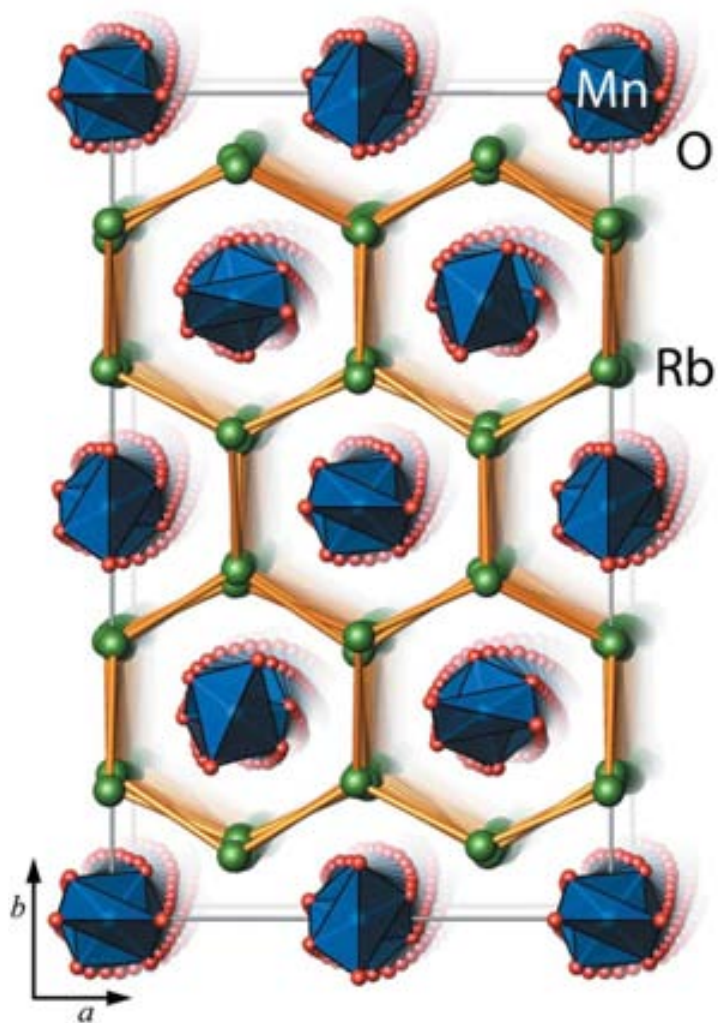
$$\begin{cases} R_s^\nu = W^\nu R_s (W^\nu)^{-1} \\ \mathbf{v}_{\nu s} = W^\nu \mathbf{v}_s \end{cases}$$

$$W^2 = \begin{pmatrix} 1 & 0 & 0 & 0 \\ 0 & 1 & 0 & 0 \\ 0 & 0 & 0 & 1 \\ 0 & 0 & 1 & 0 \end{pmatrix}$$

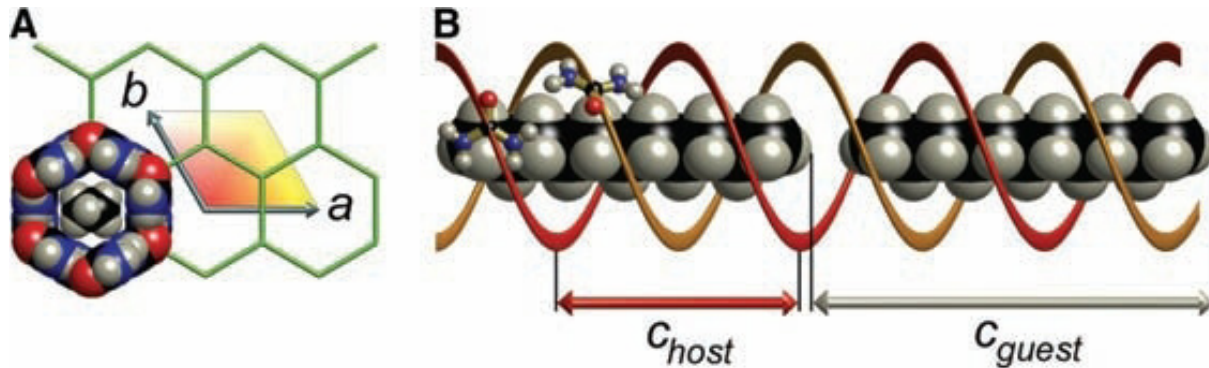
$$G_s^2 = P\bar{3}1c(00\gamma')000$$

$$G_s^2 = H'\bar{3}c1(00\gamma')000$$

Very large modulations in $[\text{Rb}]_x[\text{MnO}_2]$



Sliding mode and phasons



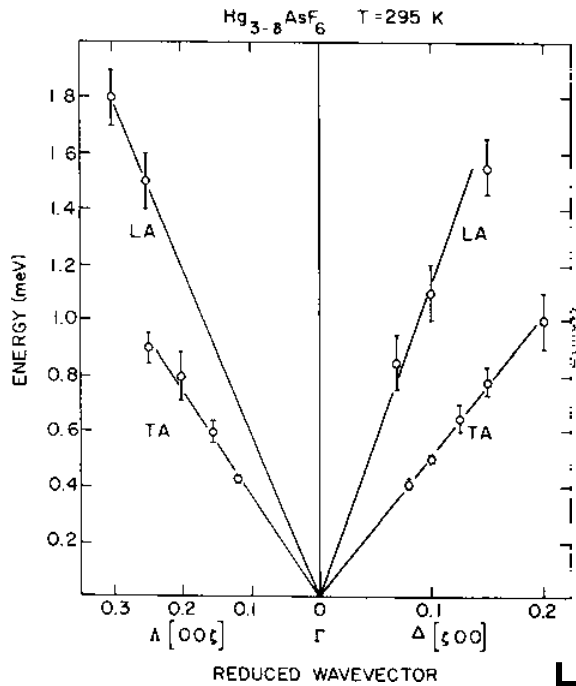
Relative shift of subsystems \Leftrightarrow phase shift of modulation (t)

All states have equal energy \Rightarrow sliding or phason mode

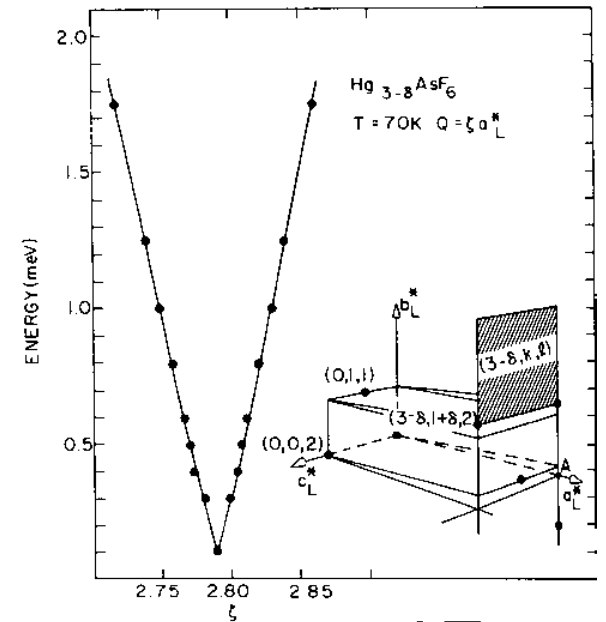
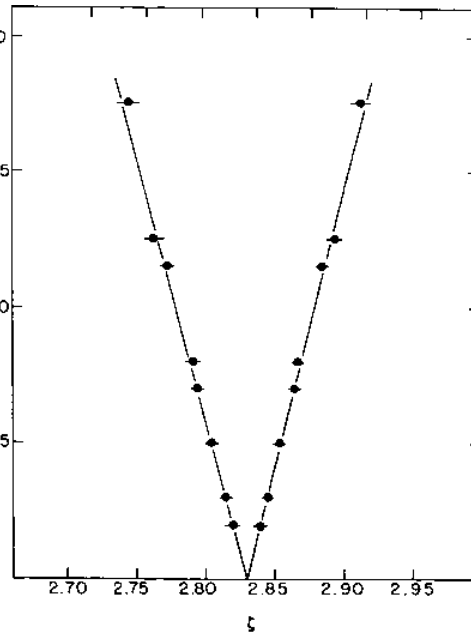
But: Pinning to surfaces and impurities

And: finite damping \Rightarrow low-energy modes are overdamped

Sliding mode in $[\text{Hg}]_{3-\delta}[\text{AsF}_6]$ observed by inelastic neutron scattering

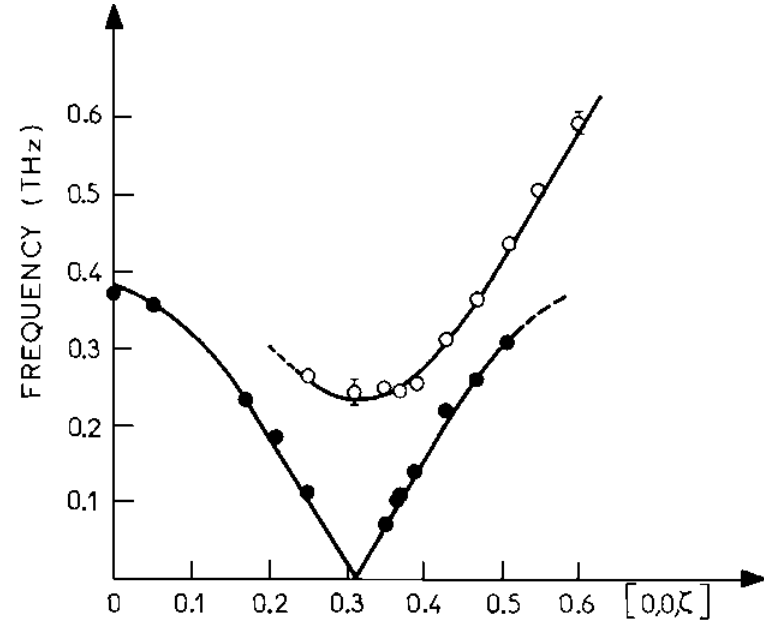
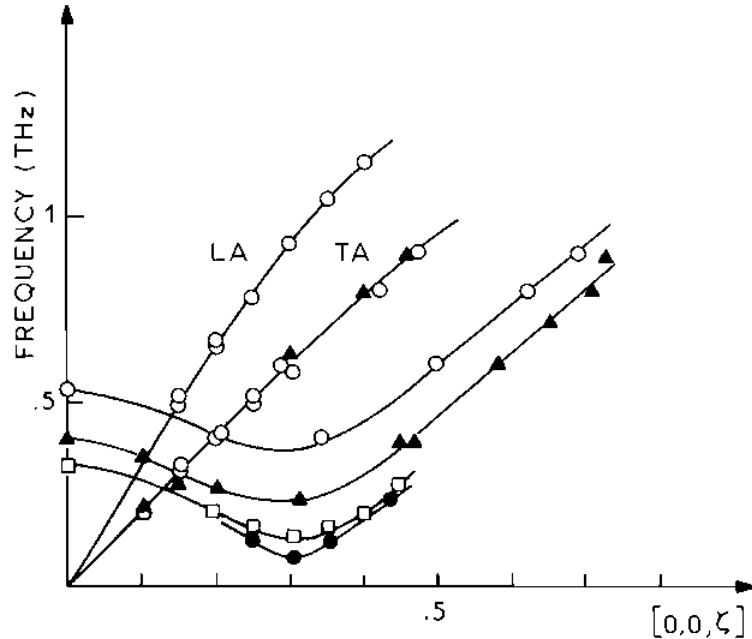


HT



LT

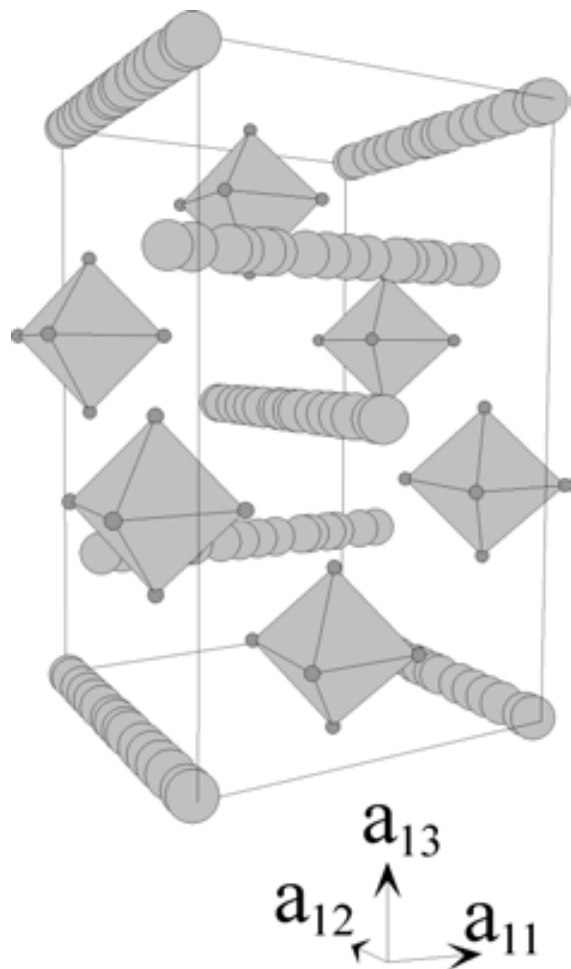
Phason mode in ThBr₄



$$\omega_{\phi}^2 = \eta^2 + v_{\phi}^2 (\delta q)^2 \quad \eta = 0 \quad \text{phason}$$

$$\omega_A^2 = \omega_0^2 + v_{\phi}^2 (\delta q)^2 \quad \text{amplitudon}$$

Composite crystal $[\text{Hg}]_{3-\delta}[\text{AsF}_6]_6$ ($\delta = 0.18$)



$T > 120 \text{ K}$ $I4_1/amd$

$a = 7.5 \text{ \AA}$ $c = 12.4 \text{ \AA}$

(2+1)D-order of the Hg chains

$3 \times d(\text{Hg-Hg}) = 7.9 \text{ \AA}$

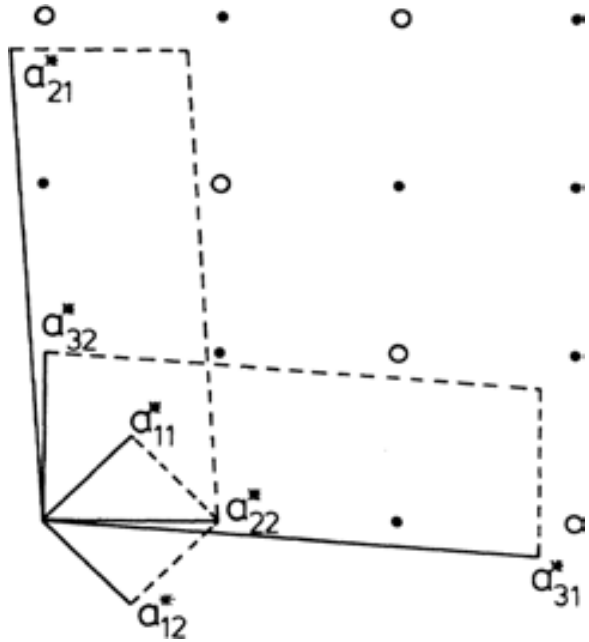
$T < 120 \text{ K}$

AsF_6 $Fddd$

$\text{Hg} (\nu = 2)$ $I2/m$

$\text{Hg} (\nu = 3)$ $I2/m$

[Hg]_{3-δ}[AsF₆] in superspace—*Fddd*(α 0 0)00s



$$W^1 = \begin{pmatrix} 1 & 0 & 0 & 0 \\ 0 & 1 & 0 & 0 \\ 0 & 0 & 1 & 0 \\ 0 & 0 & 0 & 1 \end{pmatrix}$$

$$\alpha = -2\delta$$

$$W^2 = \begin{pmatrix} 3 & \bar{3} & 0 & 1 \\ 1 & 1 & 0 & 0 \\ 0 & 0 & 1 & 0 \\ 0 & 2 & 0 & 0 \end{pmatrix}$$

$$W^2 = \begin{pmatrix} 3 & 3 & 0 & 1 \\ 1 & \bar{1} & 0 & 0 \\ 0 & 0 & \bar{1} & 0 \\ 0 & 2 & 0 & 0 \end{pmatrix}$$

$$\mathbf{a}_1^* = \mathbf{a}_{11}^*$$

$$\mathbf{a}_2^* = \mathbf{a}_{12}^*$$

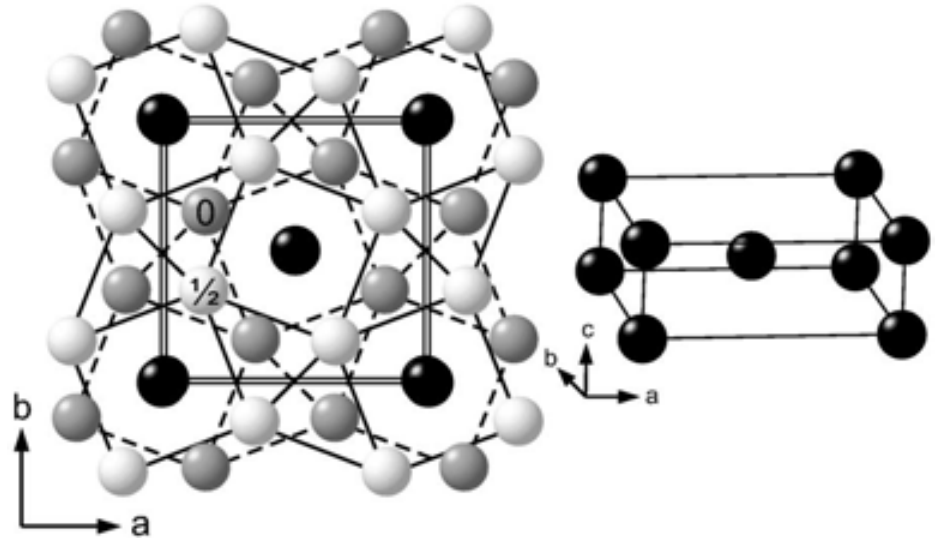
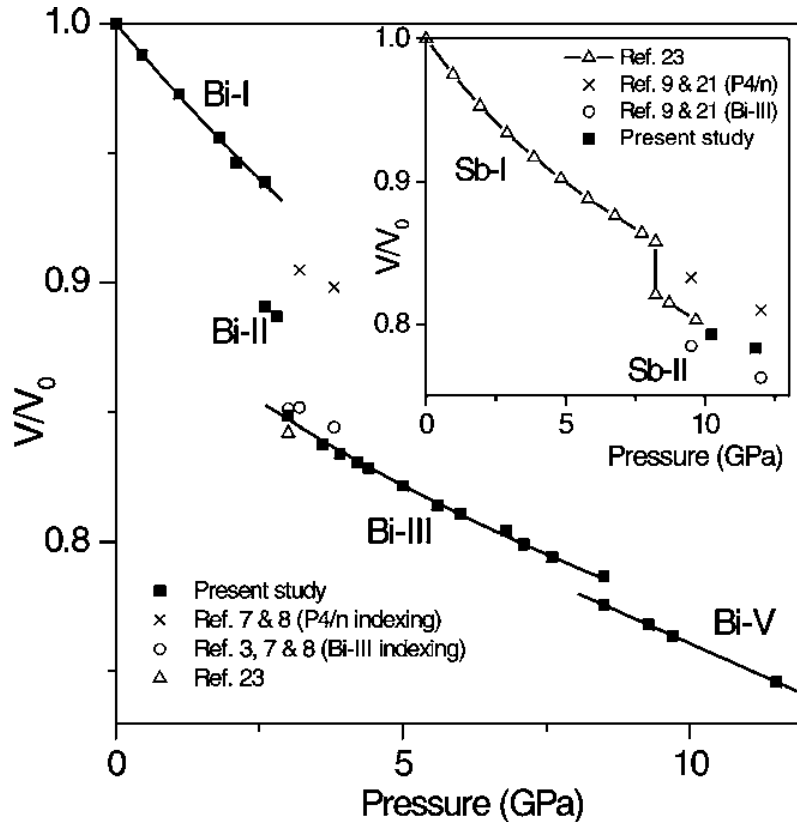
$$\mathbf{a}_3^* = \mathbf{a}_{13}^* = \mathbf{a}_{23}^* = \mathbf{a}_{33}^*$$

$$\mathbf{a}_4^* = -2\delta \mathbf{a}_1^*$$

$$G_s^2 = 12/m(\alpha', 2 - 3\alpha', 0)0s$$

$$\alpha' = 1/(3 - \delta)$$

High-pressure phase III of Bi at $p = 5.5$ GPa



Tetragonal

$$a = 8.56 \text{ \AA}$$

I4/mcc

$$c_1 = 4.18 \text{ \AA}$$

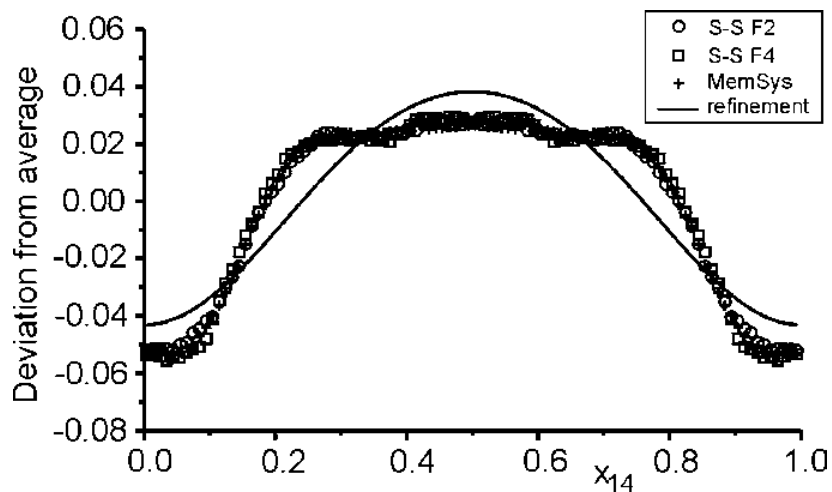
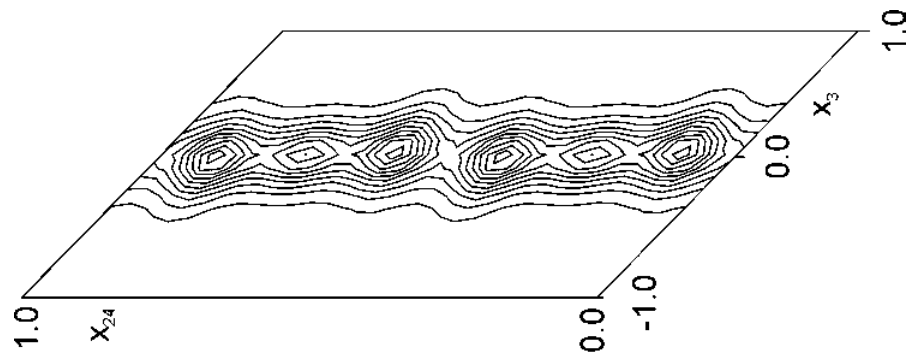
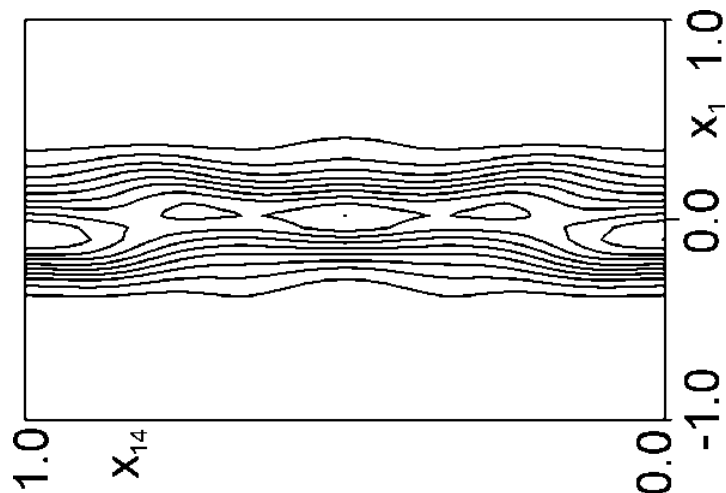
$$c_2 = 3.19 \text{ \AA}$$



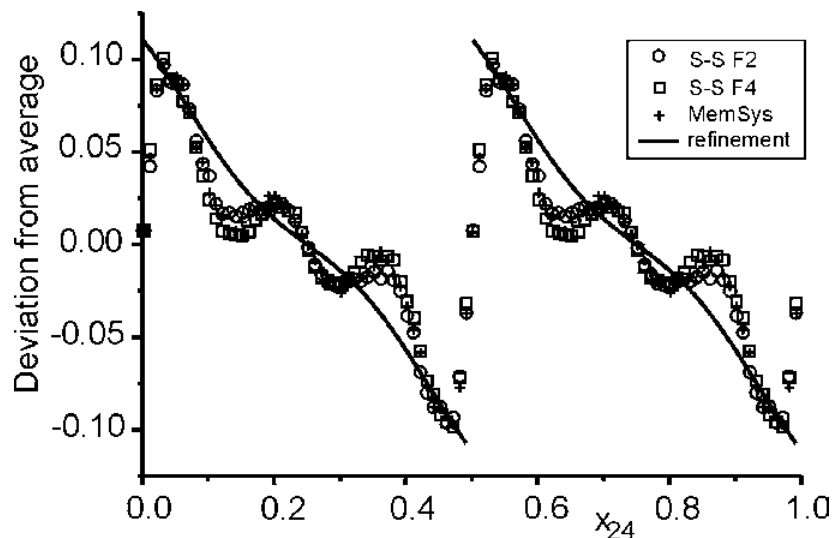
$$x = 4c_2/c_1 = 3.05$$

McMahon *et al.* (2000) PRL **85**, 4896

MEM electron density of Bi-III in superspace

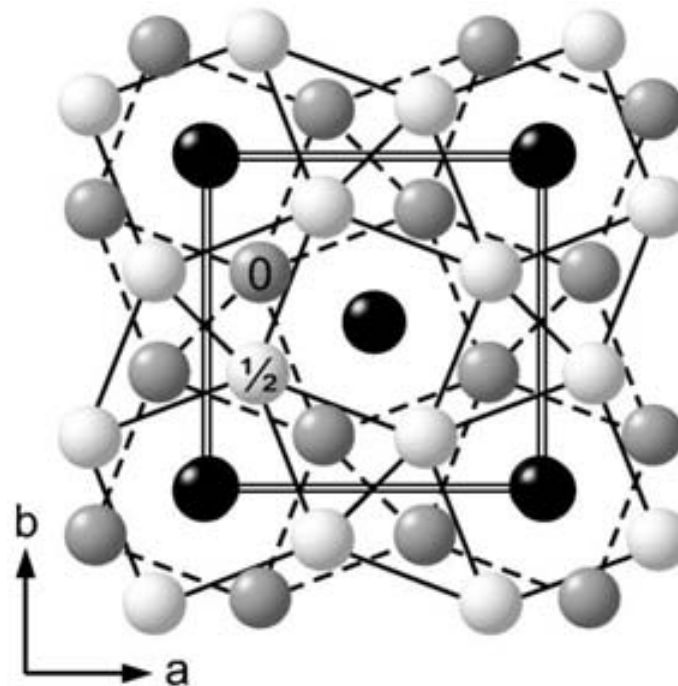
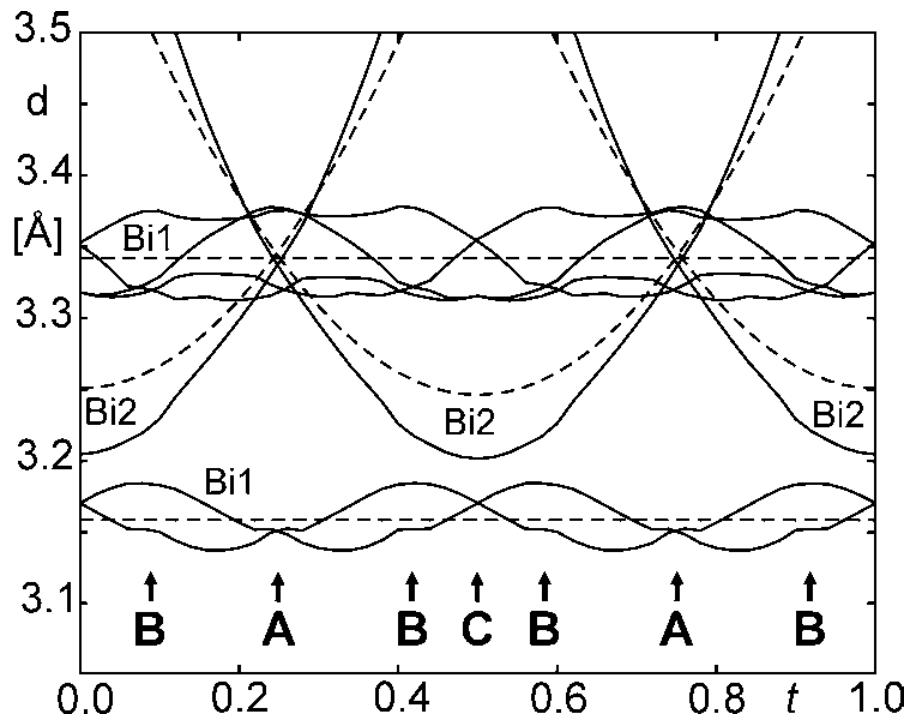


Bi1



Bi2

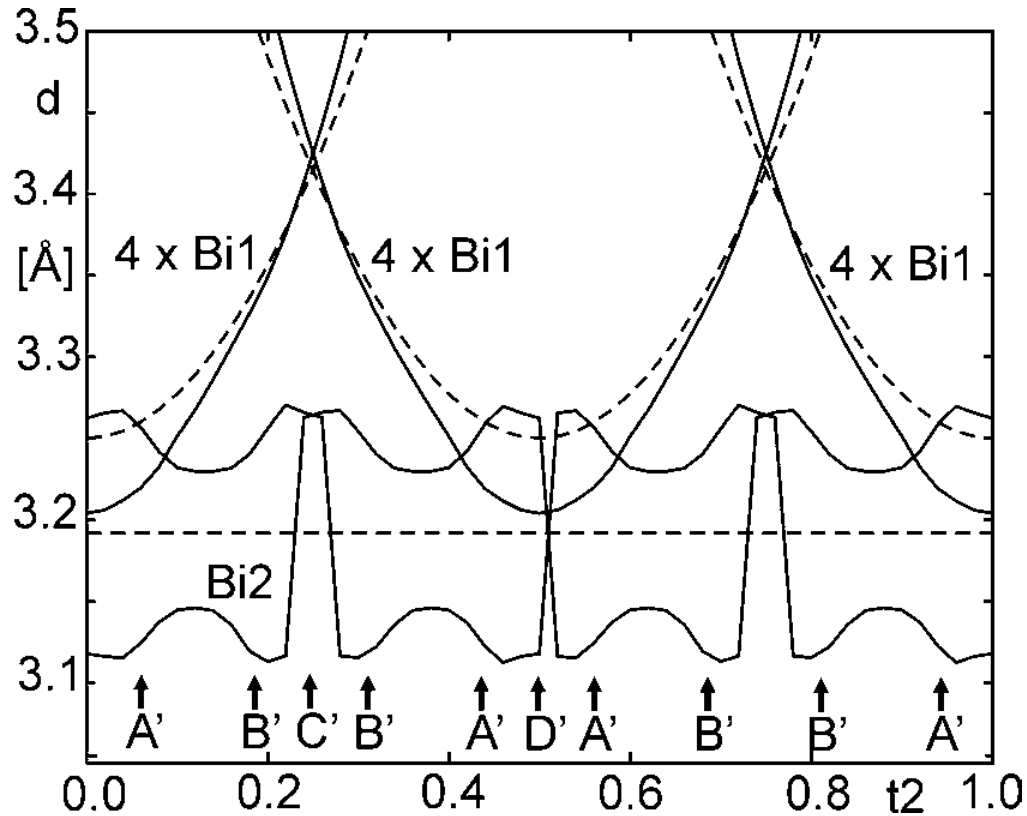
Coordination of Bi1 host atoms in Bi-III



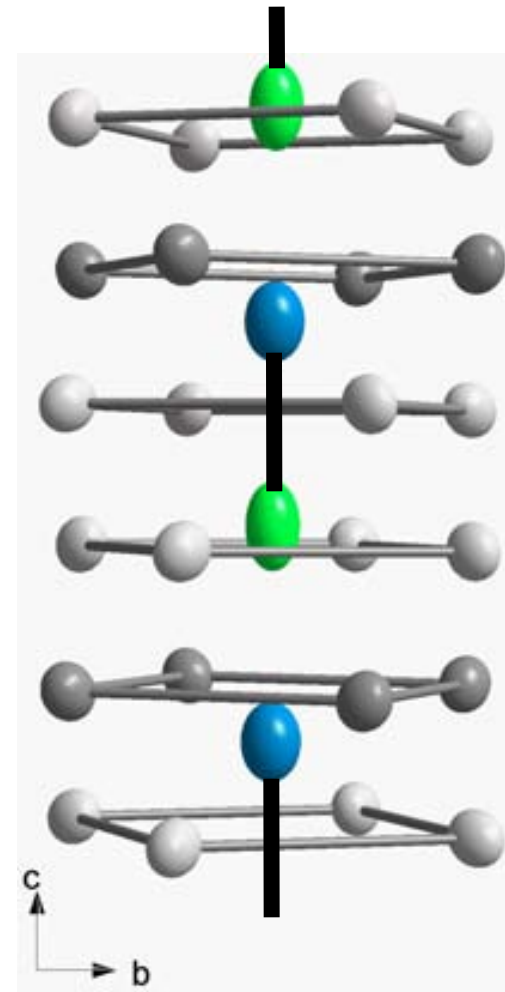
Distances around Bi1

Increased inter-subsystem bonding

Quasi dimers of Bi2 guest atoms in Bi-III

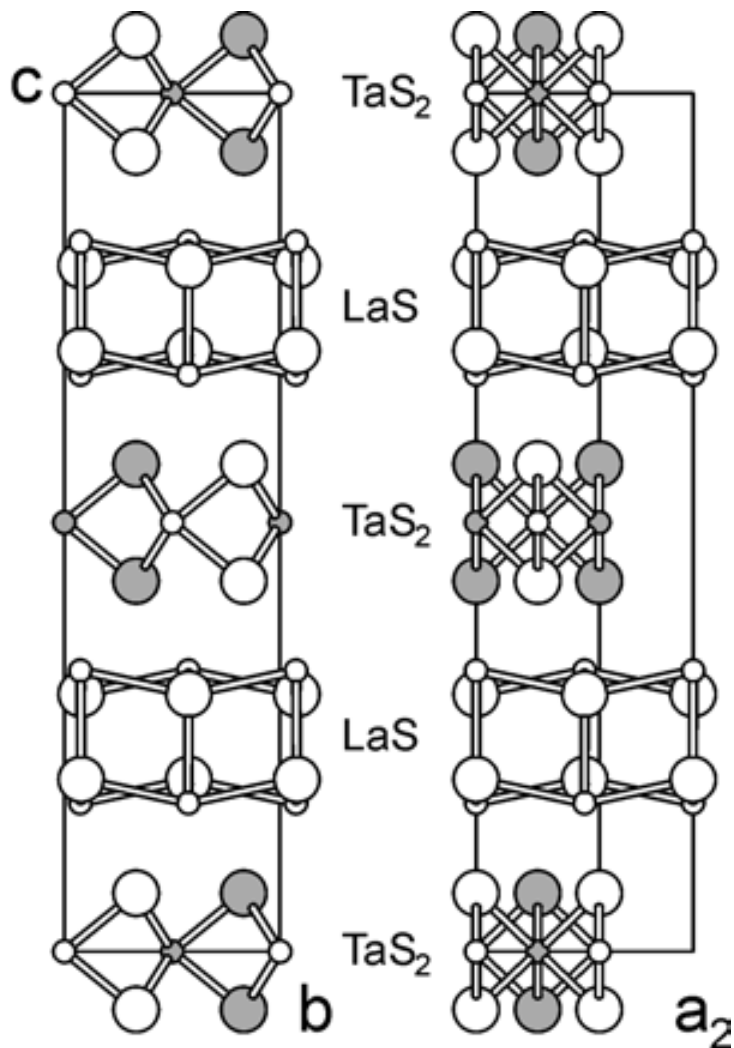


Distances around Bi2



$$4c_g/c_h = 3.05$$

X-ray diffraction data for $[\text{LaS}]_{1.13}[\text{TaS}_2]$



X-ray diffraction ($h k l m$)

Up to second-order satellites

$$[\sin(\theta)/\lambda]_{\max} = 1.0 \text{ \AA}^{-1}$$

$F'm2m(\alpha 0 0)00s$

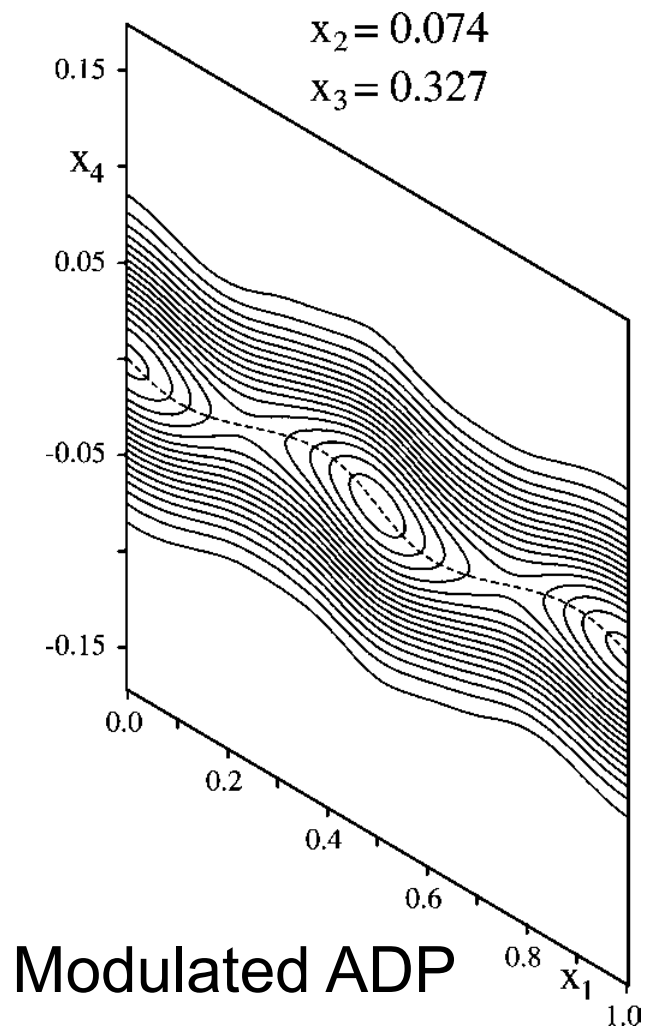
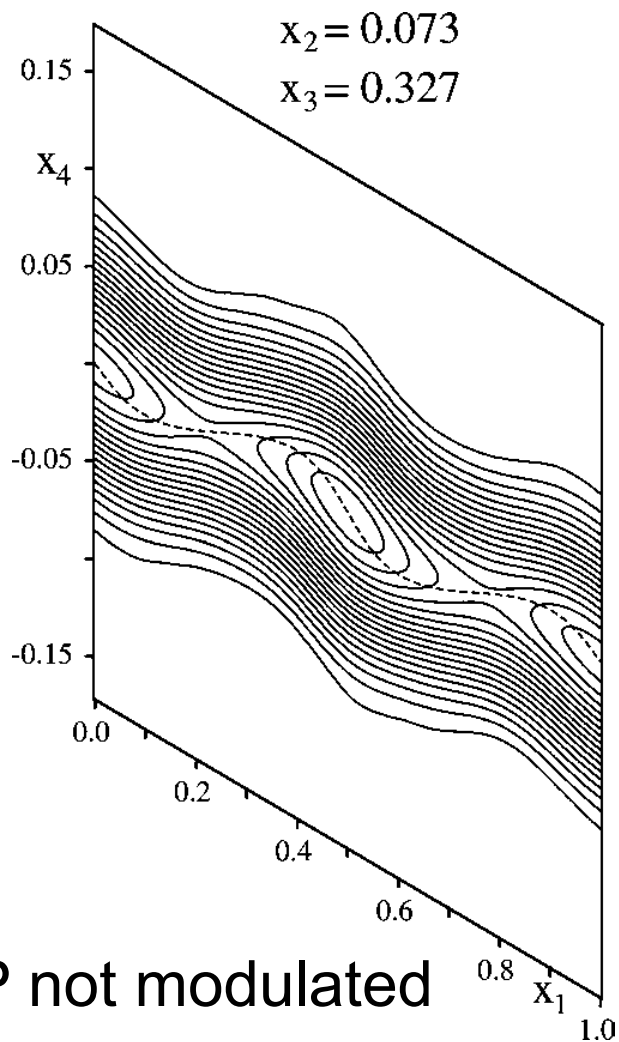
Structure refinements of $[\text{LaS}]_{1.13}[\text{TaS}_2]$

	Displacement modulation	Modulated ADPs
Refl. group	R	R
All	0.062	0.046
Main	0.039	0.034
Sat m=1	0.167	0.103
Sat m=2	0.206	0.128
$(\Delta\rho)_{\text{max}}$	73	12

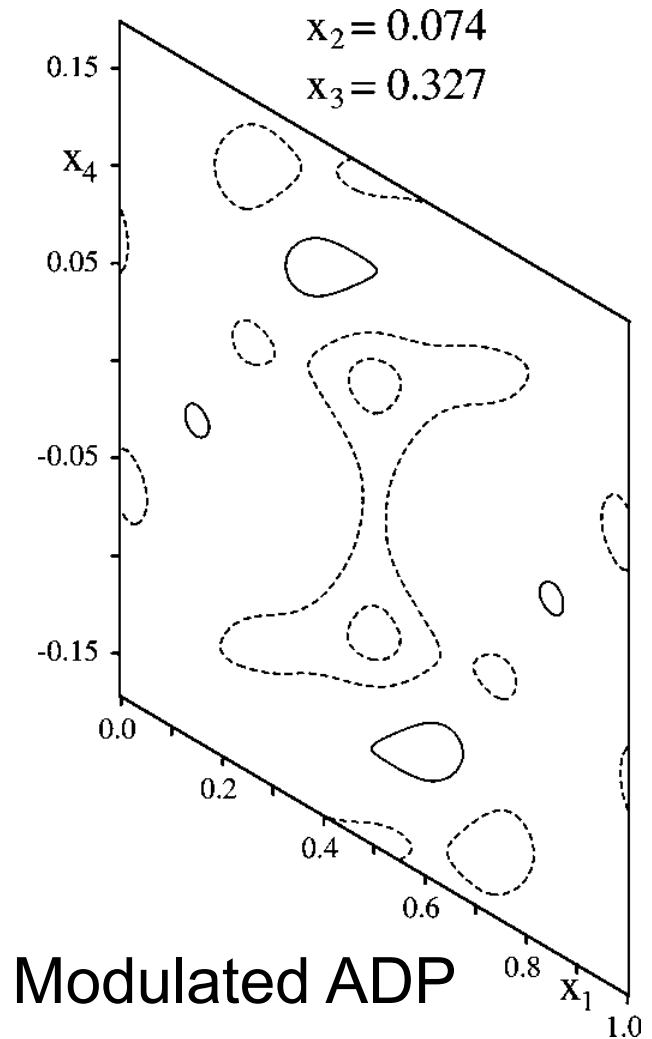
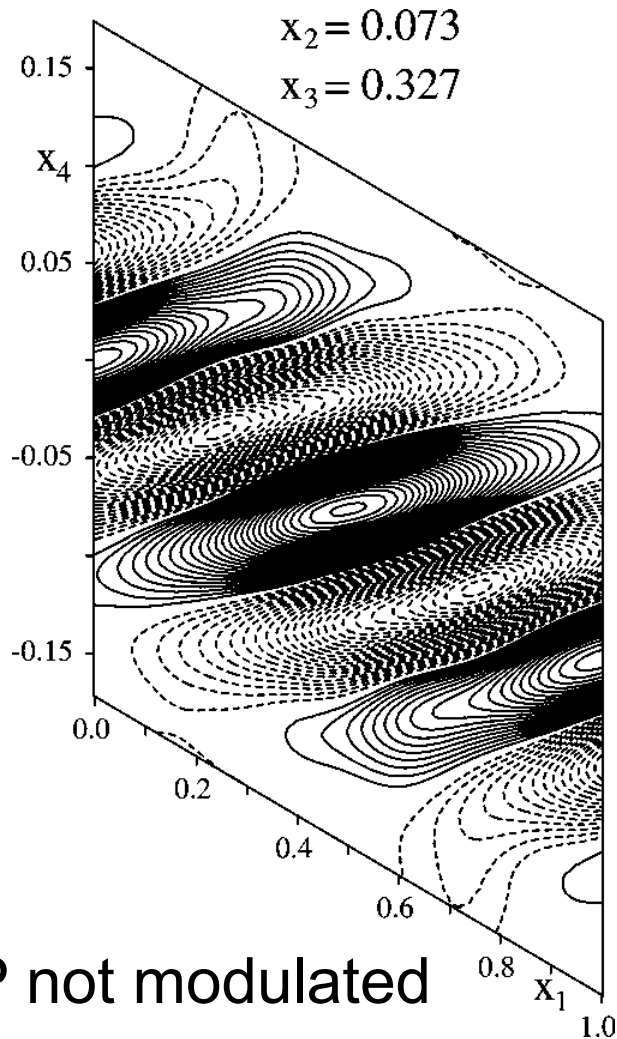
Principal modulation on La: 0.1 Å

Secondary modulation on S1 and S2 : 0.05 Å

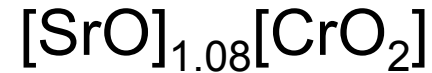
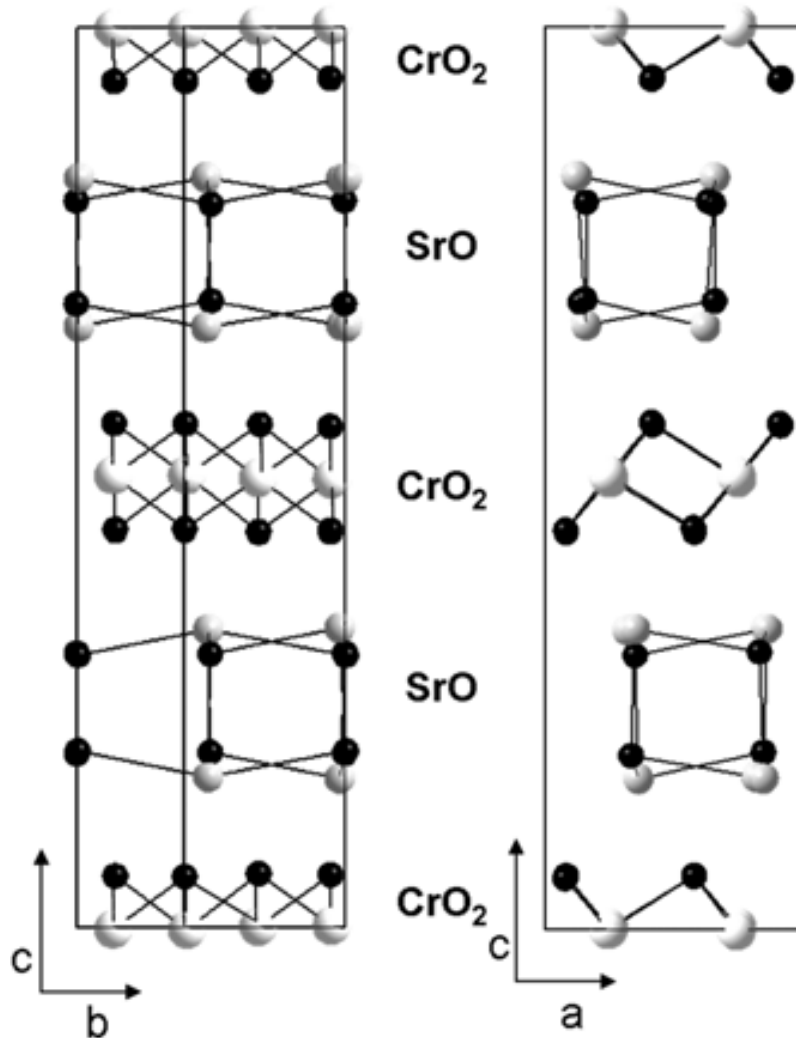
Fourier map of $[\text{LaS}]_{1.13}[\text{TaS}_2]$



Difference Fourier map of $[\text{LaS}]_{1.13}[\text{TaS}_2]$



Misfit layer structure of $[\text{SrO}]_2[\text{CrO}_2]_{1.85}$

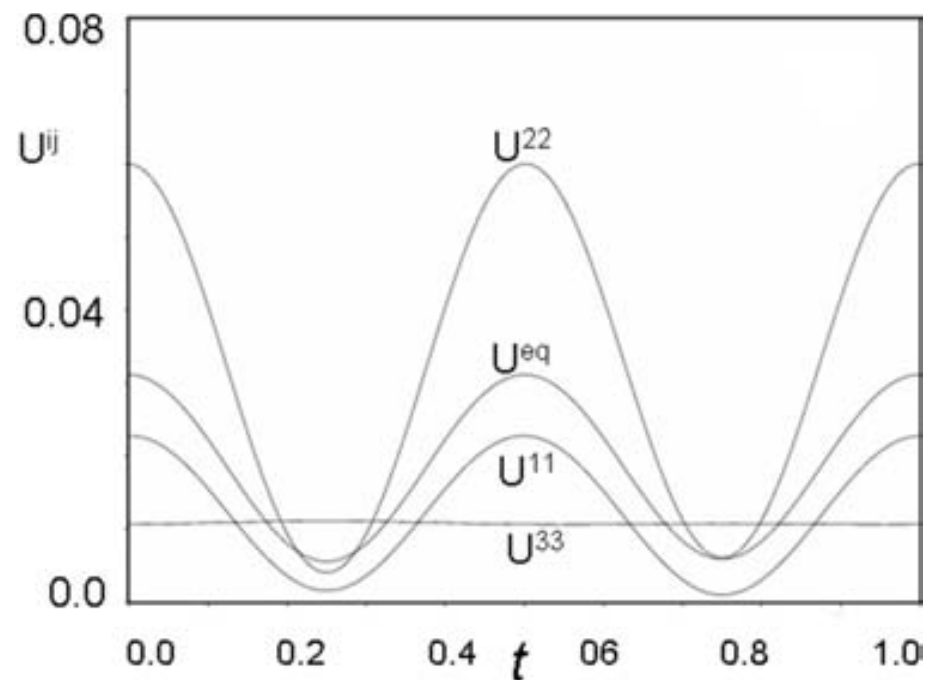


Collaboration with
M.A. Alario-Franco (University
Complutense, Madrid);
PhD-thesis work
E. Castillo-Martinez.

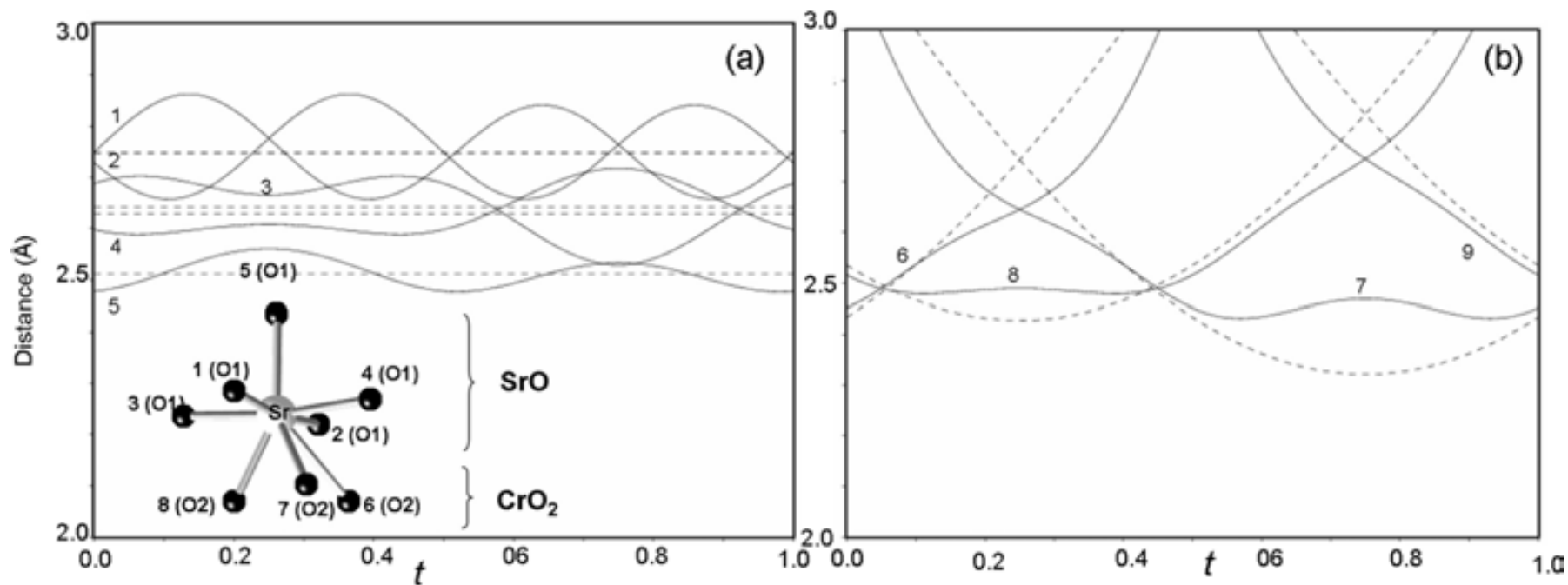
Similar to misfit layer sulfides.
Better crystal quality than
misfit layer cobaltites.

Modulated ADPs of Sr in of $[\text{SrO}]_2[\text{CrO}_2]_{1.85}$

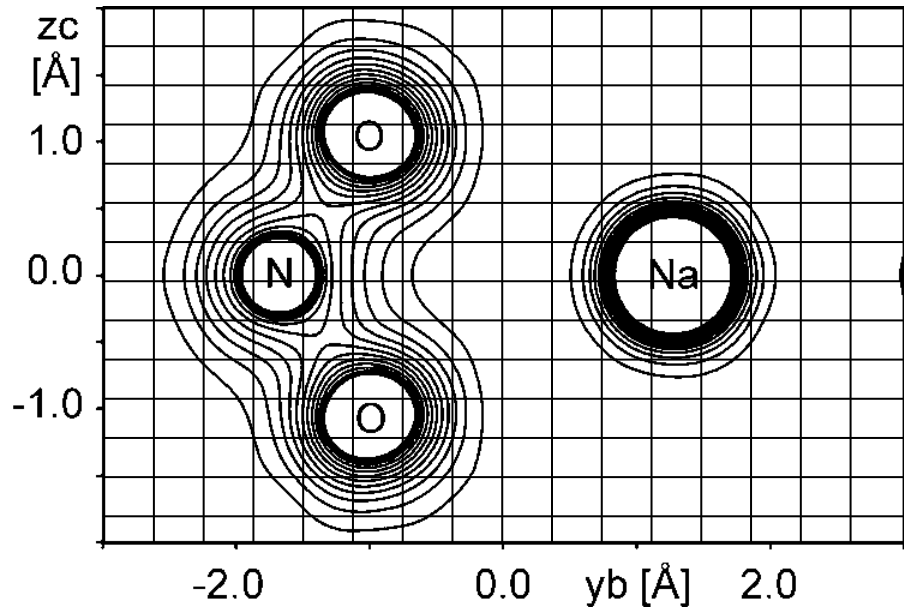
Refl. group	#Refl.	$R_F(\text{obs})$
All	778	0.044
Main	473	0.042
SrO-main	271	0.044
CrO2-main	136	0.048
Common	66	0.030
Sat m=1	248	0.044
Sat m=2	57	0.088



Environment of Sr in of $[\text{SrO}]_2[\text{CrO}_2]_{1.85}$



The Maximum Entropy Method (MEM)



ρ_k Electron density

τ_k Prior density

$N_1 \times N_2 \times N_3$ Grid

Entropy:
$$S = \sum_{k=1}^{N_{pix}} \rho_k \ln(\rho_k / \tau_k)$$

Constraint:
$$\frac{1}{N_F} \sum_{h,k,l} w_{hkl} |F_{MEM}(h,k,l) - F_{obs}(h,k,l)|^2 = 1$$

The computer program BayMEM

Various choices of PRIOR density

Various choices of Constraints

Sakata-Sato and Cambridge algorithms

MEM calculations in $(3+d)$ dimensional superspace

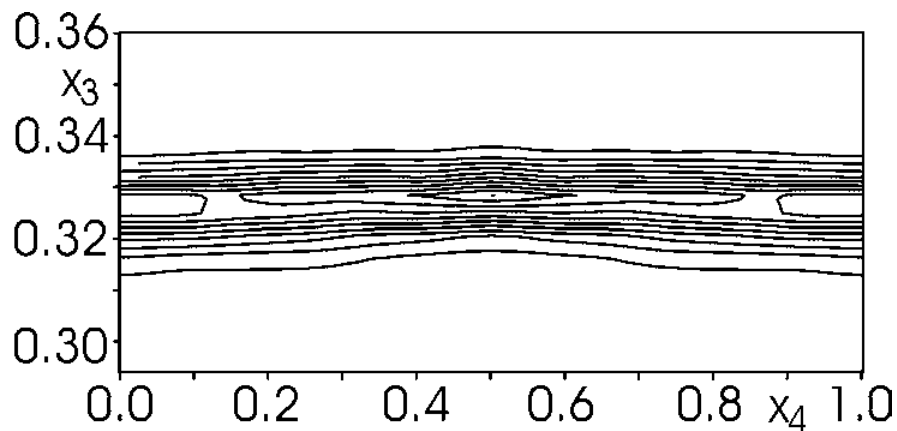
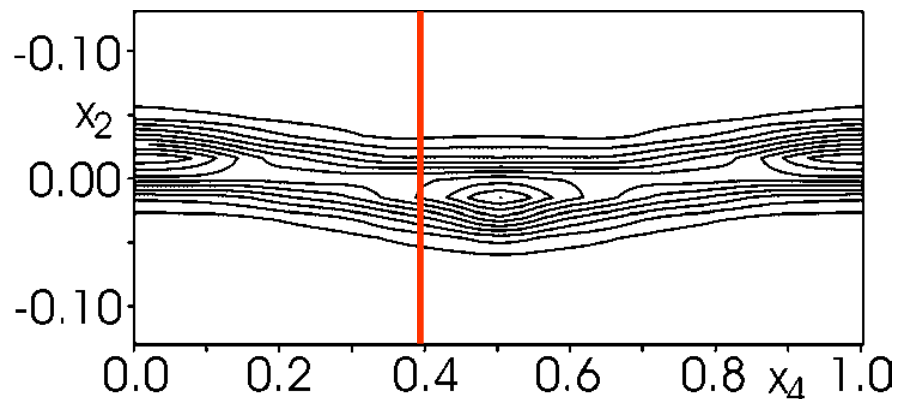
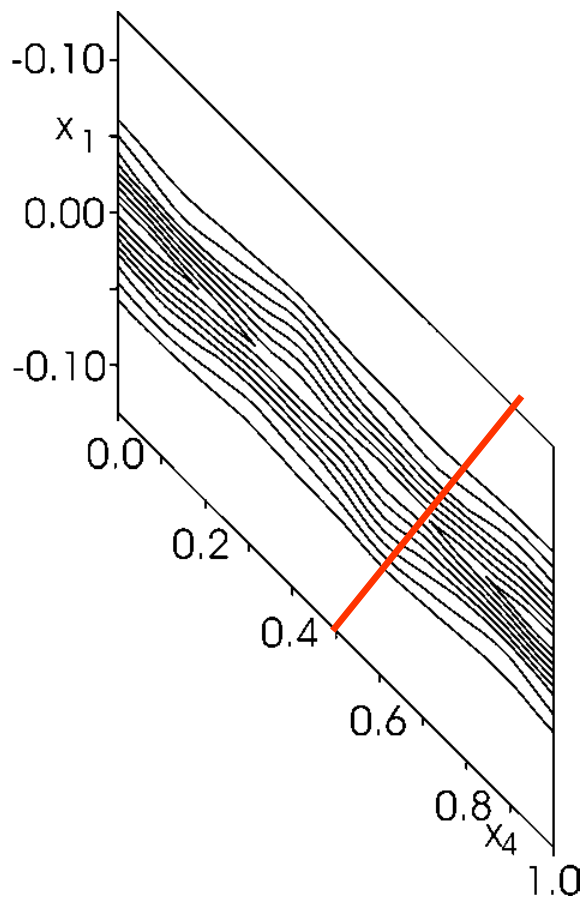
$N_{pix} = N_1 \times \dots \times N_{3+d}$ pixels

Periodic crystals correspond to $d = 0$

Full (super-)space group symmetry

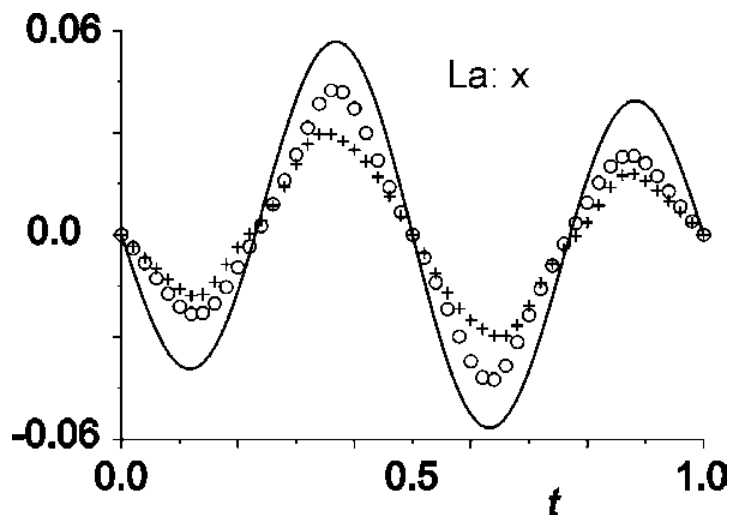
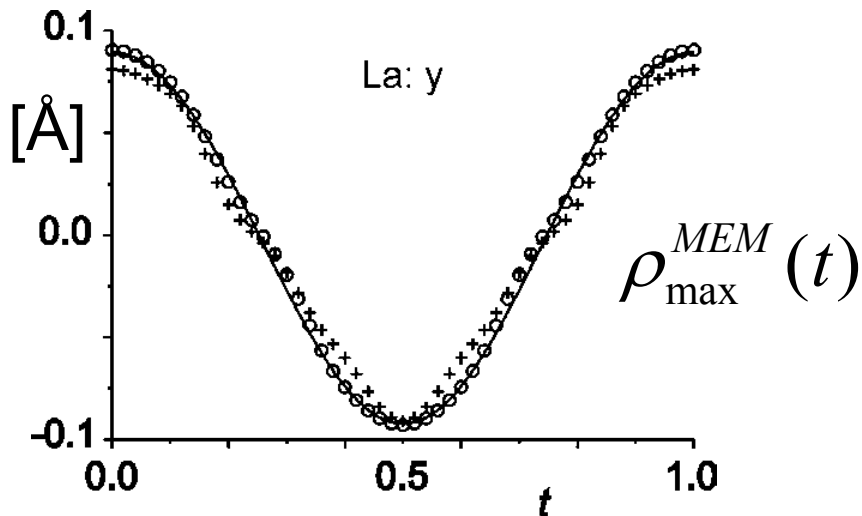
Fast Fourier Transform (FFT) in arbitrary dimensions

Maximum Entropy Method on $[\text{LaS}]_{1.14}[\text{NbS}_2]$



Software BayMEM. Pixels: $32 \times 64 \times 256 \times 32$

Displacement modulation of $[\text{LaS}]_{1.14}[\text{NbS}_2]$



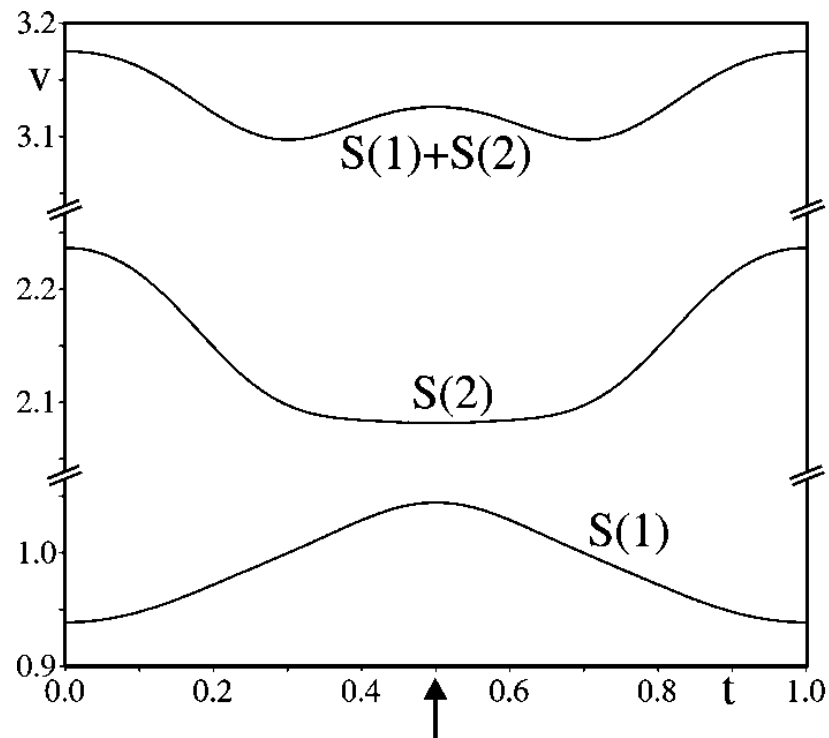
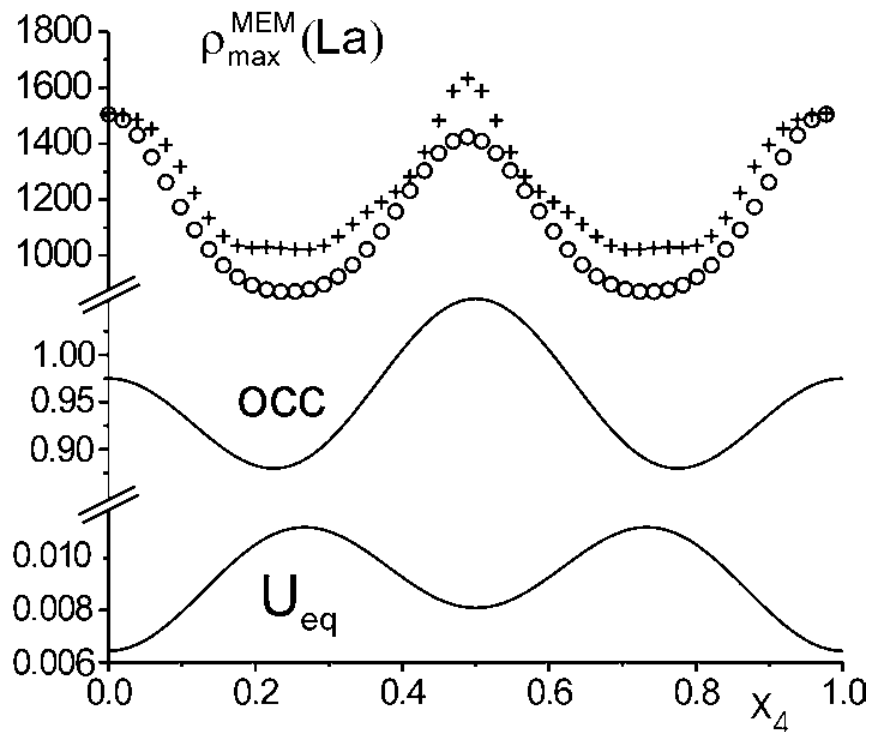
Pixels 32 x 64 x 256 x 32
 0.10 x 0.09 x 0.09 x 0.18 Å⁴

Average difference

$$\Delta U = (u_i - u_i^{\text{MEM}})$$

	ΔU (Å)	ΔU (%)
u_x	0.012	6.7
u_x'	0.003	2.9
u_y	0.002	2.2
u_z	0.001	1.1

Modulation of ADPs of $[\text{LaS}]_{1.14}[\text{NbS}_2]$



Value [$\text{e}/\text{\AA}^3$] of $\rho_{\text{max}}^{\text{MEM}} (t)$

Valence of La

Summary

Incommensurate composite crystals:

layer type

channel type

columnar type

Subsystems superspace groups may not be equivalent

Subsystems equivalent by symmetry in $[\text{Hg}]_{3-\delta}[\text{AsF}_6]$

Phasons & sliding modes due to incommensurability

Modulated ADPs & modulated third-order anharmonic ADPs
are an essential part of the modulation